



# Mechanism of setting reaction in magnesia-phosphate cements

E. Soudée, J. Péra\*

*Institute National des Sciences Appliquées de Lyon, Unité de Recherche Génie Civil—Matériaux, BÂT. 407,  
20 Avenue Albert Einstein, 69621 Villeurbanne Cedex, France*

Received 10 February 1999; accepted 1 December 1999

## Abstract

This paper tries to explain the setting mechanism of magnesia-phosphate cements obtained by mixing equal qualities of magnesia (MgO) and monoammonium dihydrogen phosphate ( $\text{NH}_4\text{H}_2\text{PO}_4$ ) or MAP. As soon as mixing water is added to the mixture, the MAP goes into solution till saturation while magnesia starts to be wetted. The drop in pH observed during this period leads to the dissociation of magnesia through an acid-base reaction. The kinetics of this reaction depends upon the MgO surface structure.  $\text{Mg}^{2+}$  ions present in solution are complexed into  $\text{Mg}(\text{H}_2\text{O})_6^{2+}$  and can substitute to molecules of water and be adsorbed on the MgO surface. These complexes remain attached to the surface and recover it progressively. Ions  $\text{PO}_4^{3-}$  and  $\text{NH}_4^+$  and  $\text{Mg}(\text{H}_2\text{O})_6^{2+}$  complexes can come close to the surface and develop a struvite network, owing to hydrogen bonds. This explanation results from analyses of solutions, X-ray diffractometry, and scanning electron microscopy. © 2000 Elsevier Science Ltd. All rights reserved.

**Keywords:** Hydration; Microstructure; Chemically bonded ceramics; MgO

## 1. Introduction

Chemically bonded ceramics (CBCs) can be processed at low temperatures [1,2]. Magnesia-phosphate cements (MPC) belong to these CBCs and develop strength rapidly. They are attractive for applications such as rapid-repair materials for deteriorated bridge decks, highways, and airport runways, and solidification and stabilization of radioactive waste, because fabrication can be achieved at room temperature or slightly elevated temperatures to minimize processing concerns [3–7].

MPCs involve an acid-base reaction between MgO and some compounds containing ammonium and phosphate ions, like phosphoric acid, monoammonium dihydrogen phosphate ( $\text{NH}_4\text{H}_2\text{PO}_4$  or MAP), or diammonium phosphate [ $(\text{NH}_4)_2\text{HPO}_4$  or DAP]. A number of mechanisms have been proposed during the last several decades to describe interactions between MgO and  $\text{NH}_4\text{H}_2\text{PO}_4$  [8]. The proposed reaction products are:

- A:  $\text{Mg}_3(\text{PO}_4)_2 \cdot x\text{H}_2\text{O}$ , where  $x = 0$  and 4, and  $\text{Mg}_2(\text{HPO}_4)_2$ , [9];
- B:  $\text{NH}_4\text{MgPO}_4 \cdot 6\text{H}_2\text{O}$  (major) and  $\text{Mg}_3(\text{PO}_4)_2 \cdot 4\text{H}_2\text{O}$  (minor), [10–11];

C:  $\text{Mg}(\text{NH}_4)\text{H}_2(\text{PO}_4)_2 \cdot 4\text{H}_2\text{O}$  and  $\text{NH}_4\text{MgPO}_4 \cdot x\text{H}_2\text{O}$ , where  $x = 4$  and 6 [12]; and

D: a gel-like bond and coprecipitation complex products [13].

In coordinating their findings with cited literature, Allan and Asgar [11] proposed the  $\text{NH}_4\text{MgPO}_4 \cdot 6\text{H}_2\text{O}$  (struvite) as the reaction product and surmised that the  $\text{Mg}_3(\text{PO}_4)_2 \cdot 4\text{H}_2\text{O}$  should form, but this has not been experimentally identified. Connaway-Wagner et al. [14] have detected struvite as the sole phosphate reaction product according to X-ray powder diffraction and P magic angle spinning nuclear magnetic resonance, while Suguma and Kukacka [15] also detected  $\text{Mg}_3(\text{PO}_4)_2 \cdot 4\text{H}_2\text{O}$  and  $\text{Mg}(\text{OH})_2$ . This observation was disputed by Abdelrazig and Sharp [16].

The mechanism of setting remains imperfectly explained. Some authors consider that it is a through-solution process including the dissolution of MAP and a part of MgO [8,17–19]. Ions in solution form hydrates that gather around a nucleation point [19]. Neiman and Sarma [8] observed the presence of an amorphous phase within the first hours, surrounding excess MgO and the filler. Colloidal hydrated particles form around the nuclei of struvite ( $\text{NH}_4\text{MgPO}_4 \cdot 6\text{H}_2\text{O}$ ). These particles then initiate a setting phenomenon through a gel-like interaction. This initial reaction takes place in an aqueous medium involving several molecules of  $\text{NH}_4\text{H}_2\text{PO}_4$  with an equivalent number of MgO molecules in solution, giving rise to a multimolecular

\* Corresponding author. Tel.: +33-4-72-43-82-96; fax: +33-4-78-94-98-07.

E-mail address: jean.pera@insa-lyon.fr (J. Péra)

framework of  $\text{NH}_4\text{MgPO}_4 \cdot 6\text{H}_2\text{O}$ , which is schematically represented as  $(\text{NH}_4\text{MgPO}_4 \cdot 6\text{H}_2\text{O})_n$ . With a low value of  $n$ , such as  $n = 3$ , it forms a ring of  $-\text{O}-\text{P}-\text{O}-\text{Mg}-\text{O}-\text{P}-$  chain, and the ring is surrounded by hydrogen-rich species, such as  $\text{H}_2\text{O}$  and  $\text{NH}_4^+ \cdot \text{H}_2\text{O}$  groups. These groups form hydrogen bonds with excess water,  $\text{NH}_4\text{H}_2\text{PO}_4$ , and other  $(\text{NH}_4\text{MgPO}_4 \cdot 6\text{H}_2\text{O})_n$  structures, resulting in colloidal-type particles. These particles then coagulate around excess  $\text{MgO}$  and filler  $\text{SiO}_2$  particles, thus initiating the setting reaction. At longer times, Neiman and Sarma [8] observed a crystalline reaction product, which was mostly struvite, in agreement with the observations of Abdelrazig et al. [17,19]. Sarkar [20] proposed another mechanism: an insoluble diffusion barrier coating is formed around  $\text{MgO}$  grains, consisting of polyphosphate units cross-linked with  $\text{Mg}^{2+}$  ions and coinciding with the onset of setting. With time, this gel slowly crystallizes into an interlocking microstructure of struvite, contributing significantly to the onset of hardening and stiffening of the cement paste.

In a recent study [21], Hall and Stevens have shown by X-ray diffraction that the major reaction product was struvite. Evidence was also found to suggest the presence of an amorphous or poorly crystalline phase with a lower degree of hydration in mortars prepared with water contents less than 8%.

The development of catalysis and physics of thin layers has focused several investigations on the dissolution mechanisms of  $\text{MgO}$  [22]. It has been found that the dissolution kinetics of  $\text{MgO}$  depends on a number of intrinsic and extrinsic factors such as the extent of dissolution, pH, surface structure, surface hydration, species in solution, and impurity and dopant concentrations.

Based on some of these findings, this paper aims to propose a new hypothetical mechanism for the setting of magnesium-phosphate cement.

## 2. Methods

### 2.1. Materials

Two commercially available varieties of dead-burned magnesia ( $\text{MgO}$ ) were used for our investigations. They were obtained by calcining magnesium hydroxy-carbonate  $[\text{Mg}_5(\text{OH})_2(\text{CO}_3)_4]$  at  $1,500^\circ\text{C}$  ( $m_1$ ) and  $1,100^\circ\text{C}$  ( $m_2$ ), respectively. The physical properties of these products are shown in Table 1. The theoretical specific surface area of each magnesia was computed from the particle size distribution, assuming that each grain was a perfect sphere.

For the acidic component, reagent grade MAP was used. Fine siliceous sand (S) was also used for preparation of mortars. The mixture proportions of mortars were:  $\text{MgO}:\text{NH}_4\text{H}_2\text{PO}_4:\text{S} = 1:1:2$ . A liquid:solid ratio of 1:4 was employed. The mixture was poured into cylinders ( $\varnothing = 45$  mm; height = 25 mm) for the setting time measurements. The mortar started to set when the Vicat's needle encountered some difficulty in penetrating the mortar; this needle was

Table 1  
Physical properties of  $\text{MgO}$

Reference	BET specific surface area ( $\text{m}^2/\text{g}$ )	Theoretical specific surface area ( $\text{m}^2/\text{g}$ )	Particle size distribution ( $\mu\text{M}$ )		
			<300	<10	$D_{50}$
$m_1$	0.34	0.12	100%	7%	66
$m_2$	11.9	0.28	100%	23%	63

not equipped with the usual 700-g mass. The final set was measured as for normal Portland cement. This process allowed a distinction between the initiation of setting and the end of setting, which cannot be done applying the usual process due to a very short setting time.

### 2.2. Test methods

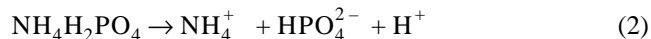
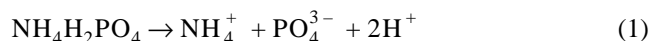
The interface tension between magnesia and an aqueous solution was assessed by measuring the wetting angle ( $\theta$ ) obtained when a drop of this solution reached magnesia. There was no wetting when  $\theta = 0^\circ$ , and wetting was perfect when  $\theta = 180^\circ$ . The angle was measured using a binocular lens (Olympus France, Runcis, France).

Ionic chromatography analyses were carried out using a DIONEX DX 1000 chromatograph (Dionex, Jouyen Josas, France), on solutions containing either magnesia and water or magnesia, MAP, and water. The identification of hydrates was done by means of X-ray diffraction and the  $\text{MgO}$  surfaces were observed using a scanning electron microscope (HITACHI S8000-high resolution (Hitachi Elextence, Verrieres le Buisson, France).

## 3. Results and discussion

### 3.1. Dissolution of MAP and $\text{MgO}$

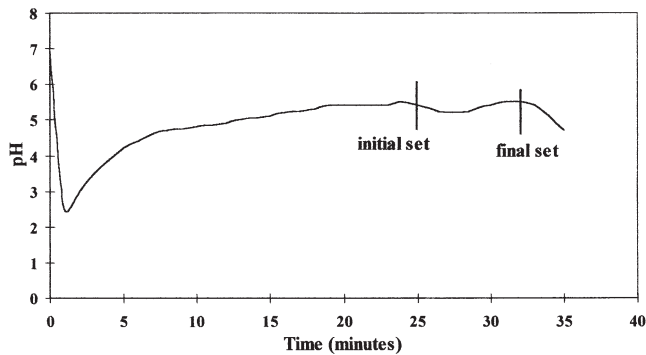
The MAP dissolves rapidly to achieve saturation according to Eqs. (1), (2), (3), and the fractions of  $\text{PO}_4^{3-}$ ,  $\text{HPO}_4^{2-}$ , and  $\text{H}_2\text{PO}_4^-$  ions are given by Eqs. (4) and (5):



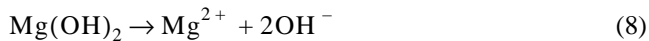
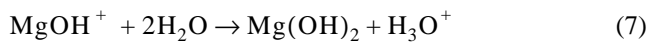
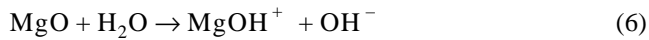
$$[\text{H}_2\text{PO}_4^-]/[\text{H}_2\text{PO}_4^-][\text{H}^+] = c^{\text{te}} \quad (4)$$

$$[\text{H}_2\text{PO}_4^{2-}]/[\text{PO}_4^{3-}][\text{H}^+] = c^{\text{te}} \quad (5)$$

Fig. 1 shows the variation of pH with time for mortar prepared with magnesia  $m_1$ . Within the first minute, the MAP dissolution leads to a pH drop. The proportions of ions  $\text{PO}_4^{3-}$ ,  $\text{H}_2\text{PO}_4^-$ , and  $\text{HPO}_4^{2-}$  are given by Eqs. (4) and (5). If MAP is introduced in excess, the remaining solid phase can only dissolve if either hydrates are formed, consuming phosphate and ammonium ions, or if the temperature increases, thus enhancing the solubility.

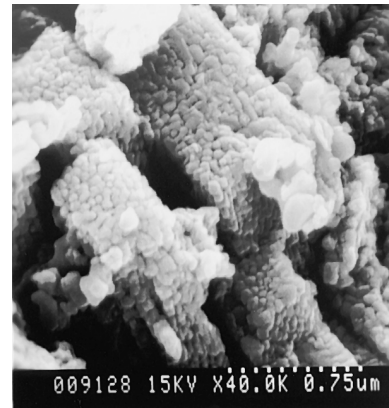
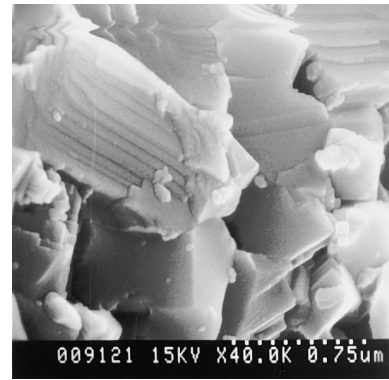
Fig. 1. Variation of pH with time. Mortar containing  $m_1$ .

Magnesia is slightly soluble in water and its dissociation occurs according to Eqs. (6), (7), and (8):



MgO surfaces must therefore successively adsorb one molecule of water, then two more, which allows the dissociation of  $\text{Mg}(\text{OH})_2$ , leading to one  $\text{Mg}^{2+}$  and two  $\text{OH}^-$  ions [20]. The presence of these  $\text{OH}^-$  ions explains the pH increase observed in Fig. 1. When the pH increases, MgO starts to dissolve.

The wetting angles and setting times of each magnesia ( $m_1$  or  $m_2$ ) are reported in Table 2. From Table 2, it appears that wetting of  $m_2$  was better than that of  $m_1$ . This is due to a difference in their surface structure, as shown in Fig. 2. Magnesia  $m_1$  exhibits crystalline structure well, while  $m_2$  presents a lot of surface defects. This is also confirmed by the ratio between the Brunauer, Emmett and Teller (BET) specific surface area and the theoretical value presented in Table 1: This ratio is 2.8 for  $m_1$  and 42.5 for  $m_2$ . This ratio represents the degree of deformation of a grain compared with a perfect sphere. The higher this ratio, the higher the degree of disorder in the solid. Magnesia  $m_2$  is therefore more disordered than  $m_1$ . This is due to the temperature of calcination at which  $m_2$  was prepared. Ahmed et al. [23] have shown that at temperatures greater than 1,100°C, a reconstruction of MgO surface occurred, limiting the amount of defects. Scamehorn et al. [24] have pointed out that the adsorption of water on MgO surface is highly dependent on the type of defects present on the surface: it is more difficult on a clean and smooth surface, which explains the best wetting of  $m_2$ . Ionic chromatography measurements (Fig. 3) show that the quantity of dissolved  $\text{Mg}^{2+}$  ions was more im-

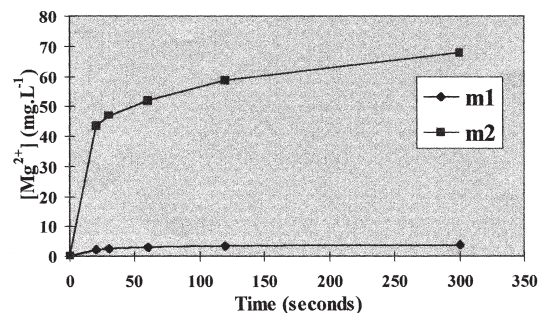
Fig. 2. Surface structure of  $m_1$  and  $m_2$ .

portant for  $m_2$  than for  $m_1$ . From these results, it seems that the higher the wetting, the higher the dissolution of magnesia, which leads to a shorter setting time: 1 min for  $m_2$  and 22 min for  $m_1$ .

The results presented in Fig. 4 point out that the dissolution of magnesia also depends upon the quantity of MAP present in the solution. The higher the MAP concentration, the higher the magnesia dissociation. Therefore, when the pH becomes acidic due to the introduction of MAP, magnesia dissolves, yielding two  $\text{OH}^-$  ions and brings back the pH to a neutral value. The lower the pH, the higher the dissolution. Therefore, the dissociation of magnesia is an acid-base

Table 2  
Properties of  $m_1$  and  $m_2$

Reference	Wetting angle (°)	Setting time (min)
$m_1$	30	22
$m_2$	140	1

Fig. 3. Dissolution of  $m_1$  and  $m_2$ .

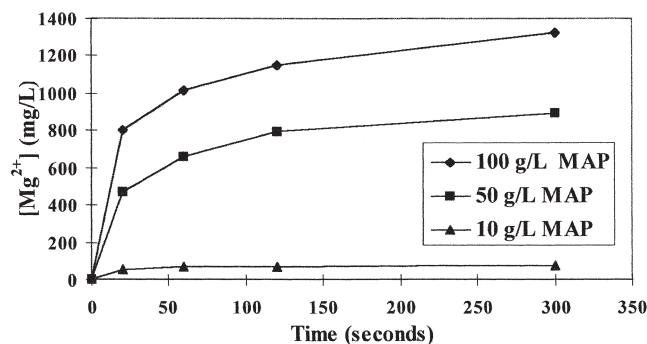


Fig. 4. Dissolution of magnesium  $m_1$  as a function of MAP concentration.

reaction; more specially, the magnesium surface is basic with structural defects and the solvent is acidic.

Fig. 5 shows the wetting angle of a drop of water and a drop of 100 g/L MAP solution on magnesium  $m_1$  surface. This angle is about the same in each case.

As presented in Table 3, although the wetting angle is the same, the quantity of dissolved  $Mg^{2+}$  ions is much more important in MAP solution than in water. Furthermore, despite a higher wetting ( $140^\circ$  instead of  $30^\circ$ ), the dissociation of magnesium are two distinct phenomena. To be dissolved magnesium needs an initial wetting but, as shown by Heidberg et al. [25], wetting can remain undissociative.

### 3.2. Crystals growth

Gonzalez-Munoz et al. [26] have shown that the production of struvite could be achieved with the physical presence of bacteria. From this study, it seems that struvite crystallization is achieved in two ways. First, the metabolic activity of the bacteria, with  $NH_4^+$  release and the consequent pH increase. Second, the cellular membrane fraction resulting from autolysis induces the nucleation process. Struvite crystallizes in both liquid and solid media.

To verify such an assumption, we carried out two series of setting tests: in the first one, the mixing water was deionized; in the second one, the mixing water was tap water having stagnated for several days to allow the development of bacteria. In each case, the setting time was the same and X-ray diffraction analyses show that struvite was the main hydrate to be formed. The presence of bacteria does not affect the production of struvite, and it seems that in magnesium-phosphate cements, the development of hydrates occurs without nucleation. To confirm this missing nucleation, the following solution was prepared: 50 g  $MgSO_4 \cdot 7H_2O$  and 24 g  $NH_4H_2PO_4$  were dissolved in 150 g water. The solution was then filtered and after 1 h, no germination was observed. The introduction of magnesium led to the occurrence of crystals within 20 min. With  $m_2$ , these crystals appeared within 1 min and were identified by X-ray diffraction as struvite. These results confirm the rapid setting of magnesium-phosphate cements, which seems due to a reaction involving magnesium surface.

Recent studies [22] have shown that the dissociation of

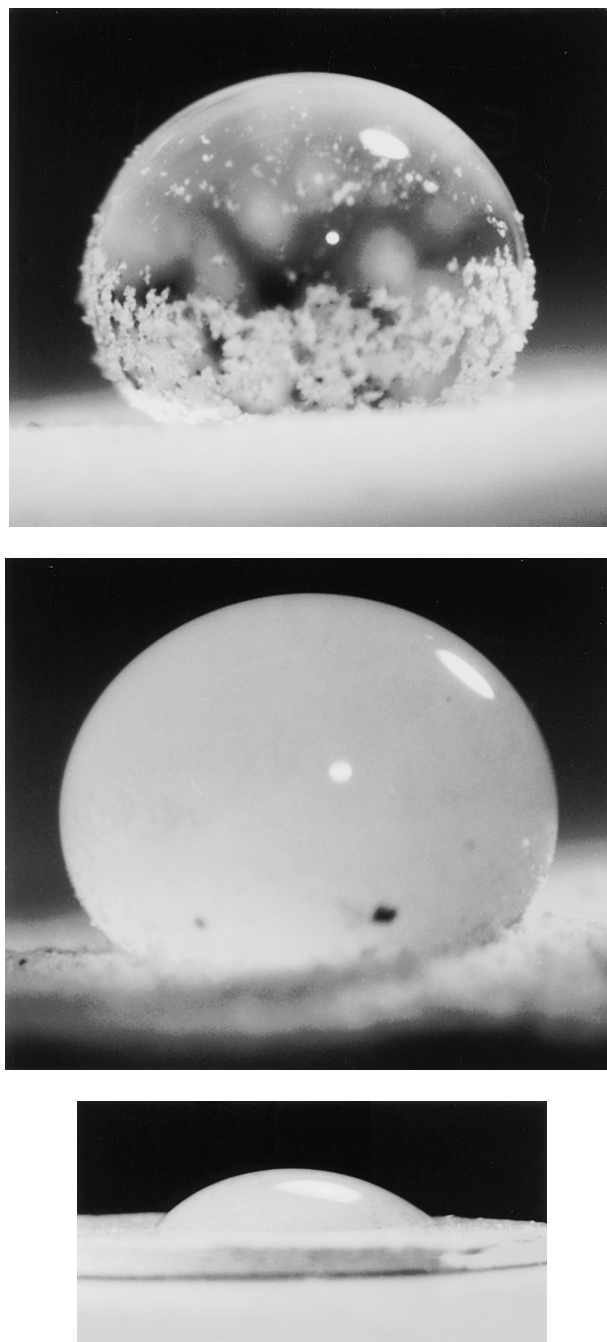


Fig. 5. Wetting angles of  $m_1$  and  $m_2$  with different solutions: (top)  $m_1$  and water; (middle)  $m_1$  and MAP solution; (bottom)  $m_2$  and water.

Table 3  
Wetting angle and dissolved  $Mg^{2+}$  ions

Magnesia	Solvent	Wetting angle ( $^\circ$ )	$Mg^{2+}$ ions within 1 min
$m_1$	Water	30	2.9 mg/L
$m_1$	100 g/L MAP solution	30	1,014.6 mg/L
$m_2$	Water	140	51.7 mg/L



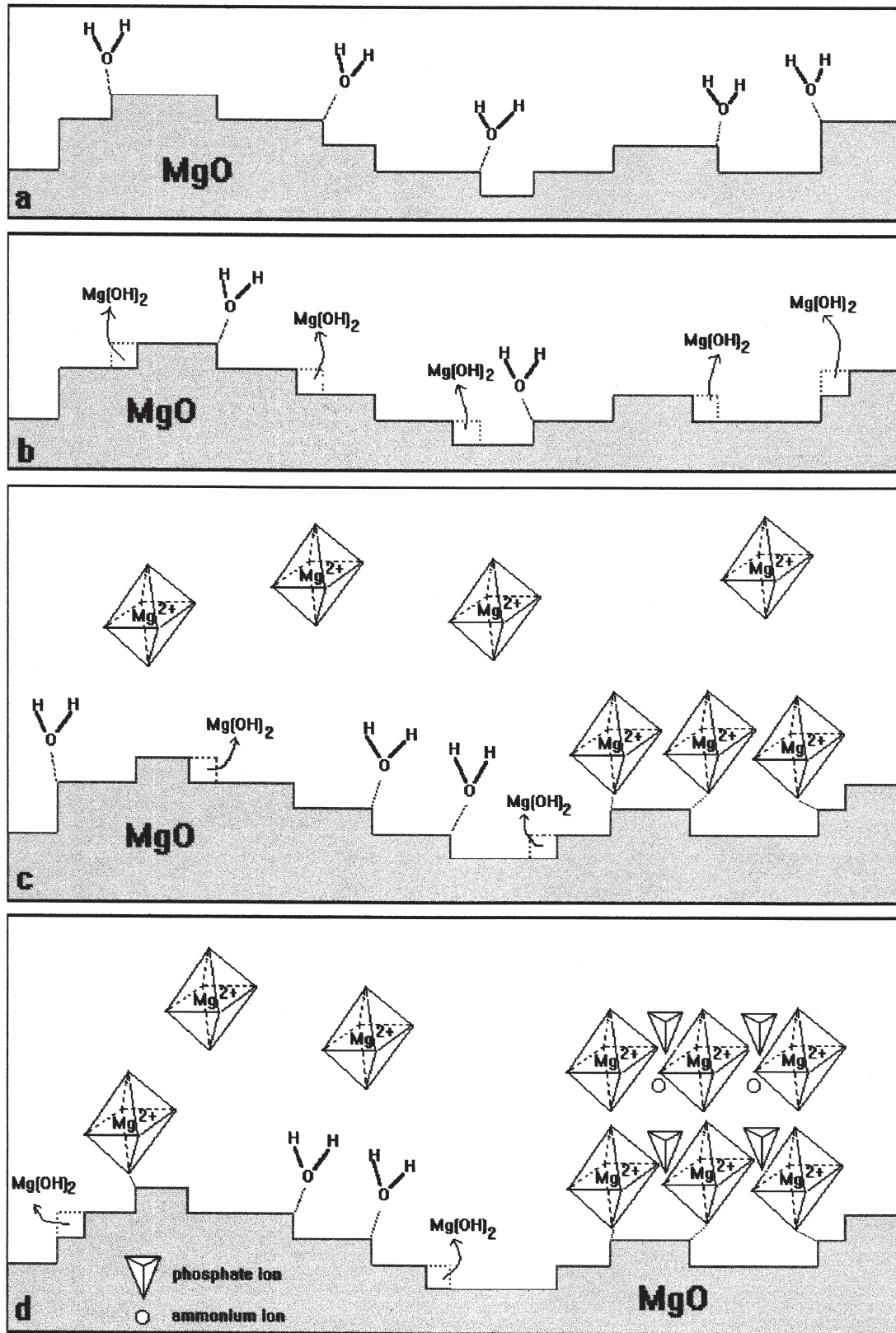


Fig. 6. Dissolution of MgO and development of crystals.

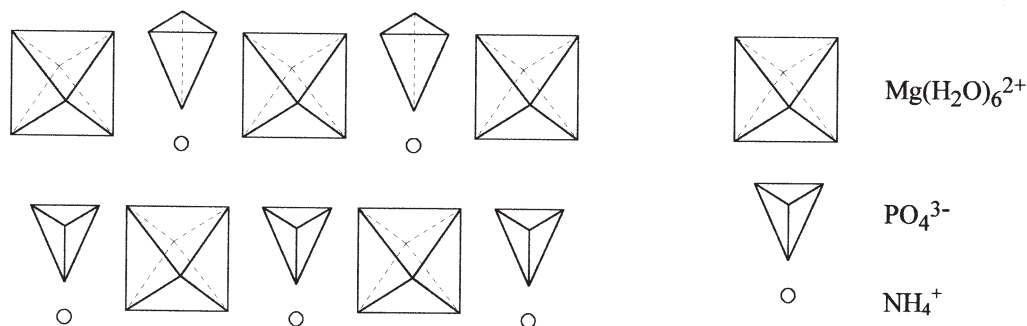


Fig. 7. Structure of struvite.

magnesia needed the successive adsorption of one molecule of water, then two more, which allowed an  $\text{Mg}(\text{OH})_2$  compound to leave the structure and immediately decompose into one  $\text{Mg}^{2+}$  ion and two  $\text{OH}^-$  ions (Figs. 6a and b). Once in solution,  $\text{Mg}^{2+}$  ions form an  $\text{Mg}(\text{H}_2\text{O})_n^{2+}$  complex, and generally  $n = 6$  [27]. During the wetting of magnesia surface, it is possible that an  $\text{Mg}(\text{H}_2\text{O})_6^{2+}$  octahedron substitutes to one molecule of water. The two supplementary molecules of water needed for the achievement of  $\text{MgO}$  dissociation therefore cannot approach to the surface. The  $\text{Mg}(\text{H}_2\text{O})_6^{2+}$  remains hooked to the surface. Several adsorbed complexes can thus present sites for crystallization, as shown in Fig. 6b.

The structure of struvite presented in Fig. 7 consists of

$\text{PO}_4^{3-}$  tetrahedra,  $\text{Mg}(\text{H}_2\text{O})_6^{2+}$  octahedra, and  $\text{NH}_4^+$  groups, held together by hydrogen bonds [28].

The  $\text{MgO}$  surface, partially covered by  $\text{Mg}(\text{H}_2\text{O})_6^{2+}$  complexes, ions  $\text{PO}_4^{3-}$ , and  $\text{NH}_4^+$  present in solution, can come close together and, owing to hydrogen bonds, generate a struvite network (Fig. 6d).

The crystals grow around magnesia grains and connect, coating filler grains and developing a cementitious matrix (Fig. 8). The reaction stops when the magnesia grains are entirely covered by hydrates and can no longer dissolve. The reaction can also stop when one of the two reactants (MAP or water) is missing.

#### 4. Conclusions

When a magnesia-phosphate cement is mixed with water, an exothermic reaction occurs, leading to a hardened product within a few minutes. The dissolution of magnesia is an acid-base reaction between basic surfaces of  $\text{MgO}$  and the acid MAP solution generated by the instantaneous dissolution of MAP in water. The setting time depends on the structure of  $\text{MgO}$  surface. The more disordered the surface, the more basic it is, and therefore the kinetics of wetting is high. The dissociation depends on both this wetting and the MAP concentration of the solution.

Struvite crystals do not appear after a nucleation process induced by a high degree of saturation. Crystallization starts when  $\text{Mg}(\text{H}_2\text{O})_6^{2+}$  complexes substitute to molecules of water during the wetting of  $\text{MgO}$  surfaces. Opposite to molecules of water that lead to the dissociation of magnesia,  $\text{Mg}(\text{H}_2\text{O})_6^{2+}$  complexes remain attached to the surface and progressively cover it.  $\text{PO}_4^{3-}$  and  $\text{NH}_4^+$  ions and  $\text{Mg}(\text{H}_2\text{O})_6^{2+}$  complexes can therefore develop a struvite network, from the surface, owing to hydrogen bonds.

#### References

- [1] D.M. Roy, New strong cement materials: Chemically bonded ceramics, *Science* 235 (1987) 651–658.
- [2] D.D. Double, Chemically bonded ceramics: Taking the heat out of making ceramics, *Journal of Materials Education* 12 (1990) 353–381.
- [3] S.S. Seehra, S. Gupta, S.S. Kumar, Rapid setting magnesium phos-



Fig. 8. Development of struvite crystals.

- phate cement for quick repair of concrete pavements—Characterisation and durability aspects, *Cem Concr Res* 23 (2) (1993) 254–266.
- [4] S. Popovics, N. Rajendran, M. Penko, Rapid hardening cements for repair of concrete, *ACI Materials J* 84 (1) (1987) 64–73.
  - [5] B. El Jazairi, Rapid repair of concrete pavings, *Concrete* 16 (2) (1982) 12–15.
  - [6] B. El Jazairi, The properties of hardened MPC mortar and concrete relevant to the requirements of rapid repair of concrete pavements, *Concrete* 21 (1987) 25–31.
  - [7] A.S. Wagh, J.C. Cunnane, D. Singh, D.T. Reed, S. Armstrong, W. Subhan, N. Chawla, Chemically phosphate ceramics for radioactive and mixed-waste solidification and stabilization, In: *Proceedings of the Conference: Technology and Programs for Radioactive Waste Management and Environmental Restoration, WM'93, Tucson, Vol. 2, 1993*, pp. 1613–1617.
  - [8] R. Neiman, A.C. Sarma, Setting and thermal reactions of phosphate investments, *J Dental Res* 59 (9) (1980) 1478–1485.
  - [9] P.A. Gilham-Dayton, The phosphate bonding of refractory materials, *Trans Br Ceram Soc* 62 (1963) 895–904.
  - [10] T.E. Moore, G.M. Watts, Investment materials, US Patent, 2, 1949, pp. 479–504.
  - [11] F.C. Allan, K. Asgar, Reaction of cobalt-chromium casting alloys with investments, *J Dental Res* 59 (9) (1980) 1478–1485.
  - [12] T. Awada, On setting mechanism and hygroscopic expansion of phosphate-bonded investment mixed with colloidal silica solution, *J Japan Dental Res Soc* 32 (1974) 6.
  - [13] C.P. Mabie, Petrographic study of the refractory performance of high-fusing dental alloy investments, *J Dental Res* 52 (1973) 96.
  - [14] M.C. Connaway-Wagner, W.G. Klemperer, J.F. Young, A comparative study of magnesia-orthophosphate and magnesia-tripolyphosphate cement, in: (S. Mindess, Ed.), *Ceramic Transactions, Advances in Cementitious Materials, Vol. 16*, The American Ceramic Society, Inc., Waterville, Ohio, 1991, pp. 679–688.
  - [15] T. Sugama, L.E. Kukacka, Magnesium monophosphate cements derived from diammonium phosphate solutions, *Cem Concr Res* 13 (1983) 499–506.
  - [16] B.E.I. Abdelrazig, J.H. Sharp, A discussion of the papers on magnesia-phosphate cements by T. Sugama and L.E. Kukacka, *Cem Concr Res* 15 (1985) 921–922.
  - [17] B.E.I. Abdelrazig, J.H. Sharp, B. El. Jazairi, The chemical composition of mortars made from magnesia-phosphate cement, *Cem Concr Res* 18 (1988) 415–425.
  - [18] T. Finch, J.H. Sharp, Chemical reactions between magnesia and aluminum orthophosphate to form magnesia-phosphate cements, *J Mat Sci* 24 (1989) 4379–4386.
  - [19] B.E.I. Abdelrazig, J.H. Sharp, B. El. Jazairi, The microstructure and mechanical properties of mortars made of magnesia-phosphate cements, *Cem Concr Res* 19 (1989) 247–258.
  - [20] A.K. Sarkar, Investigation of reaction/bonding mechanisms in regular and retarded magnesium-ammonium phosphate cement systems, *Ceramic Transactions, Cement Technology* 40 (1994) 281–288.
  - [21] D.A. Hall, R. Stevens, Effect of water content on the structure and mechanical properties of magnesia-phosphate cement mortar, *J Am Ceram Soc* 81 (6) (1998) 1550–1556.
  - [22] S.A. Holt, C.F. Jones, G.S. Watson, A. Crossley, C. Johnston, C.J. Sofield, S. Myhra, Surface modification of MgO substrates from aqueous exposure: An atomic force microscopy study, *Thin Solid Films* 292 (1997) 96–102.
  - [23] F. Ahmed, K. Sakai, H. Ota, R. Aoki, N. Ikemiya, S. Hara, Surface morphology for annealed and etched MgO (100), *Journal of Low Temperature Physics* 105 (5/6) (1996) 1343–1348.
  - [24] C.A. Scamehorn, N.M. Harrison, M.I. McCarthy, Water chemistry on surface defect sites: Chemidissociation versus physio-sorption on MgO (001), *Journal of Chemistry and Physics* 101 (2) (1994) 1547–1554.
  - [25] J. Heidberg, B. Redlich, D. Wetter, Adsorption of water vapor on the MgO (100) single crystal surface, *Ber Bunsenges, Phys Chem* 99 (II) (1995) 1333–1337.
  - [26] M.T. Gonzalez-Munoz, N. Ben Omar, M. Martinez-Canamero, M. Rodriguez-Gallego, A.L. Galindo, J.M. Arias, Struvite and calcite crystallization induced by cellular membranes of *Myxococcus xanthus*, *J Crystal Growth* 163 (1996) 434–439.
  - [27] E. Longo, A. Varela, C.V. Santelli, O.J. Whitemore, Model of interactions between magnesia and water, in: (W.D. Kingery, Ed.), *Advances in Ceramics, Vol. 10*, John Wiley & Sons, New York, NY, 1983, pp. 592–600.
  - [28] A. Whitaker, J.W. Jeffery, The crystal structure of struvite,  $\text{MgNH}_4\text{PO}_4 \cdot 6\text{H}_2\text{O}$ , *Acta Crystallographica* 6 (1979) 339–354.