

Short communication

Synthesis of ternary molybdenum carbosilicide

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Abstract

Dense compacts of $\text{Mo}_{\leq 5}\text{Si}_3\text{C}_{\leq 1}$ have been prepared by reactive hot pressing of a mixture of elemental powders of Mo, Si and C. Synthesis temperatures and soaking times have been optimized for obtaining pure $\text{Mo}_{\leq 5}\text{Si}_3\text{C}_{\leq 1}$. The hardness of pure sample is 14.7 GPa. © 2002 Elsevier Science Ltd and Techna S.r.l. All rights reserved.

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1. Introduction

Molybdenum carbosilicide, $\text{Mo}_{\leq 5}\text{Si}_3\text{C}_{\leq 1}$, a Nowotny phase, [1] is the only ternary phase in the Mo–Si–C system, as identified by Nowotny et al. [2] It belongs to a class of compounds of the general formula $\text{M}_{\leq 5}\text{Si}_3\text{C}_{\leq 1}$ where M = Zr, Nb, Mn or Mo, [1] has a melting point of about 2100 °C [3], and crystallizes in a hexagonal structure (D_{8h}) [1,4]. Its crystallographic formula may be written as $\text{Mo}_3^{\text{I}}\text{Mo}_2^{\text{II}}\text{Si}_3\text{C}_{\leq 1}$ where C atoms occupy partially the octahedral sites formed by Mo^{I} atoms. It is proposed [3] that $\text{Mo}_{\leq 5}\text{Si}_3\text{C}_{\leq 1}$ could not only be a reinforcement second phase but also a new matrix phase for composites based on Mo–Si–C systems. Shobu and coworkers have used $\text{Mo}_{\leq 5}\text{Si}_3\text{C}_{\leq 1}$ as infiltrants to produce composites with SiC. [5–7]

Preparation of $\text{Mo}_{\leq 5}\text{Si}_3\text{C}_{\leq 1}$ as a single-phase product is reported to be difficult owing to the small stability region in the phase diagram [8,9]. Suzuki and Niihara have prepared $\text{Mo}_{4.8}\text{Si}_3\text{C}_{0.6}$ as a single phase by reactive hot pressing of elemental powders at 30 MPa and 1500 °C for 1–2 h [3]. Parthe and Jeitschko have prepared a compound of this composition starting from Mo, Si and C powders by hot pressing and annealing at 1600 °C for 12 h in a vacuum furnace [1]. We attempted to synthesize single phase $\text{Mo}_{4.8}\text{Si}_3\text{C}_{0.6}$ in a shorter duration of time by reactive hot pressing of elemental powders at 26 MPa and temperatures ranging from 1600 to 1750 °C in argon atmosphere and succeeded in

obtaining a nearly single-phase product at 1700 °C in a time, as low as 8 min.

2. Experimental

Compounds of the nominal composition $\text{Mo}_{4.8}\text{Si}_3\text{C}_{0.6}$ have been prepared starting from stoichiometric amounts of elemental Mo powder (Aldrich, 99%), silicon powder (Aldrich, 99%) and carbon soot (home made), taken according to the equation



The powders were mixed together and hot pressed in a BN coated graphite die (12 mm i.d.) at 26 MPa for 8–30 min in an argon atmosphere at temperatures ranging from 1600 to 1750 °C. After the synthesis, the pellets were polished with 1 µm diamond paste and characterized by powder x-ray diffraction (Cu K_{α} , Philips Powder Diffractometer), density measurement (Archimedes principle), scanning electron microscopy (Jeol JSM-5600 LV Scanning Electron Microscope) and Vickers hardness measurement (Shimadzu HMV-2000 micro hardness tester).

3. Results and discussion

In all the preparations, $\text{Mo}_{\leq 5}\text{Si}_3\text{C}_{\leq 1}$ is the observed major phase confirming the X-ray diffraction pattern reported in the literature [1,4,10]. The small amounts of

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impurity phases are Mo_5Si_3 , Mo_2C and some unidentified phases. Table 1 lists the phases present in these preparations.

For the samples prepared at 1600 and 1650 °C, the amount of the impurity phase, Mo_5Si_3 , is comparatively large. At 1700 °C, Mo_5Si_3 is absent. This indicates that during the reaction Mo and Si react to form Mo_5Si_3 first, which then reacts with Si and C to form $\text{Mo}_{4.8}\text{Si}_3\text{C}_{0.6}$. However, when the preparation temperature is raised to 1750 °C Mo_5Si_3 forms again, probably due to the decomposition of $\text{Mo}_{4.8}\text{Si}_3\text{C}_{0.6}$. It seems that a temperature of 1700 °C is ideal for short-duration synthesis. Fig. 1 shows the XRD patterns of samples prepared at various temperatures with a fixed soaking time of 15 min.

To find the effect of soaking time, we carried out synthesis experiments at 1700 °C by using different soaking times (Fig. 2). Nearly single phase $\text{Mo}_{\leq 5}\text{Si}_3\text{C}_{\leq 1}$ is formed at this temperature with a soaking time of just 8 min (Fig. 2a). For this sample only one impurity-line at $2\theta = 39.4^\circ$, which corresponds to 100% line of Mo_2C ,

Table 1

Impurity phases, density and hardness values of $\text{Mo}_{\leq 5}\text{Si}_3\text{C}_{\leq 1}$ prepared under different conditions

Soaking temperature, °C	Soaking time, min	A ^a	B ^a	C ^a	Density, g/cm ³	Hv, GPa
1600	15	16	—	—	7.63	14.8
1650	15	15	24	—	7.12	12.3
1700	8	—	4	—	7.74	14.7
1700	15	—	6	5	7.79	13.8
1700	30	—	9	—	7.64	16.4
1750	15	12	—	—	7.59	18.0

^a A = $100 \cdot I_{60}(\text{Mo}_5\text{Si}_3)/I_{100}(\text{Mo}_{4.8}\text{Si}_3\text{C}_{0.6})$, B = $100 \cdot I_{100}(\text{Mo}_2\text{C})/I_{100}(\text{Mo}_{4.8}\text{Si}_3\text{C}_{0.6})$, C = $100 \cdot I_{100}(\text{unidentified phase})/I_{100}(\text{Mo}_{4.8}\text{Si}_3\text{C}_{0.6})$.

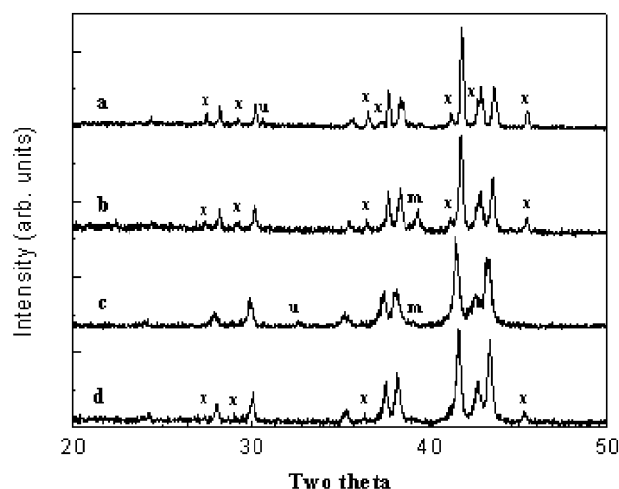


Fig. 1. X-ray powder patterns of $\text{Mo}_{\leq 5}\text{Si}_3\text{C}_{\leq 1}$ samples prepared at (a) 1600, (b) 1650, (c) 1700 and (d) 1750 °C, with a fixed soaking time of 15 min. (x = Mo_5Si_3 , m = Mo_2C , u = unidentified phase.)

is present. Therefore, under these synthesis conditions it is possible to obtain nearly single phase $\text{Mo}_{\leq 5}\text{Si}_3\text{C}_{\leq 1}$. Fig. 3 shows the SEM image of this sample. At higher soaking times, the amount of Mo_2C increases. Suzuki and Niihara [3] have observed the formation of Mo_2C in their preparations when the Si-content in the starting mixture was slightly less than that required for single-phase formation. Therefore, formation of Mo_2C in our preparations may be due to the volatilization of a small amount of Si from the reaction mixture during high-temperature treatment.

Table 1 also includes densities of the compacts obtained. We find that the highest density is obtained when the preparation is carried out at 1700 °C for 15 min. The Vickers hardness value for the pure sample, as indicated in Table 1, is 14.7 GPa. These values are higher than those reported by Suzuki and Niihara [3] (11.0–12.6 GPa for monolithic $\text{Mo}_{\leq 5}\text{Si}_3\text{C}_{\leq 1}$). When the

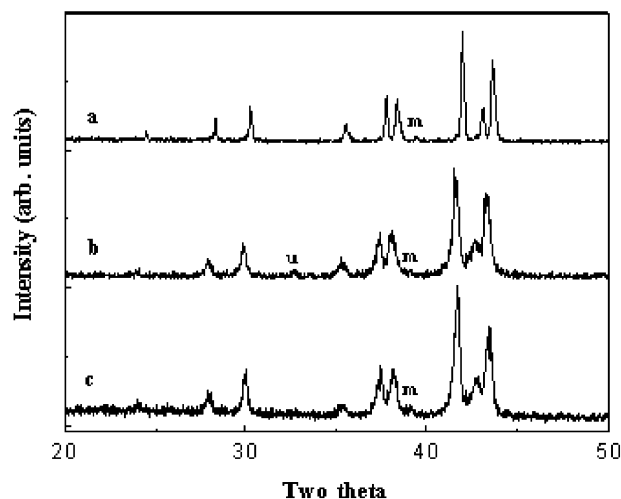


Fig. 2. X-ray powder patterns of $\text{Mo}_{\leq 5}\text{Si}_3\text{C}_{\leq 1}$ samples prepared at 1700 °C with soaking times of (a) 8 min, (b) 15 min and (c) 30 min. (m = Mo_2C , u = unidentified phase.)

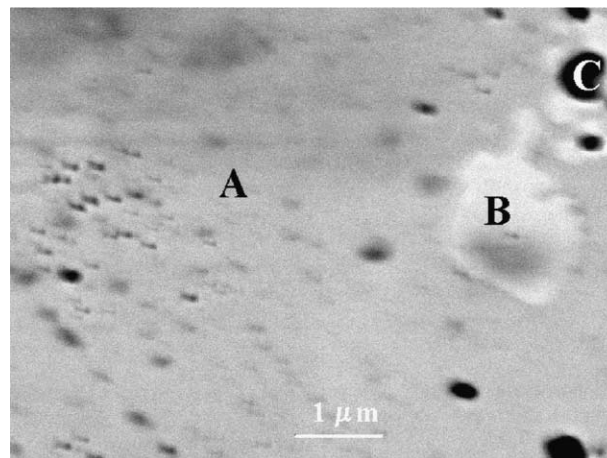


Fig. 3. Scanning electron micrograph of a sample prepared at 1700 °C with soaking of 8 min. (A = $\text{Mo}_{\leq 5}\text{Si}_3\text{C}_{\leq 1}$, B = Mo_2C , C = pore.)

impurity phases are present in considerable amounts, the hardness value is found to be different; the sample prepared at 1750 °C has a hardness of 18 GPa.

4. Conclusion

It is possible to prepare almost single-phase dense compacts of $\text{Mo}_{\leq 5}\text{Si}_3\text{C}_{\leq 1}$ by reactive hot pressing of elemental powders at 26 MPa and 1700 °C for 8–15 min. The materials so synthesized have Vickers hardness of 13–15 GPa.

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