

Short communication

Sintering behavior of bulk $\text{SrBi}_2\text{Ta}_2\text{O}_9$ prepared by solid-state reaction

Baorang Li*, Longtu Li, Xiaohui Wang

State Key Laboratory of New Ceramics and Fine Processing, Department of Materials Science and Engineering, Tsinghua University, Beijing 100084, PR China

Received 6 March 2002; received in revised form 7 June 2002; accepted 10 July 2002

Abstract

The conventional solid-state reaction method has been used to prepare bulk $\text{SrBi}_2\text{Ta}_2\text{O}_9$ ceramics. The sintering behavior and decomposition of bulk $\text{SrBi}_2\text{Ta}_2\text{O}_9$ ceramics have been investigated. Higher sintering temperature ($> 1200^\circ\text{C}$) and longer sintering time (> 7 h) can result in decomposition of $\text{SrBi}_2\text{Ta}_2\text{O}_9$.

© 2002 Elsevier Science Ltd and Techna S.r.l. All rights reserved.

Keywords: Solid-state reaction; $\text{SrBi}_2\text{Ta}_2\text{O}_9$; Sintering temperature; Sintering time

Much attention has been paid to $\text{SrBi}_2\text{Ta}_2\text{O}_9$ thin films due to their potential applications in ferroelectric random access memories [1–4], while relatively little research on bulk $\text{SrBi}_2\text{Ta}_2\text{O}_9$ ceramics has been performed. However, studies on the characteristics of bulk $\text{SrBi}_2\text{Ta}_2\text{O}_9$ ceramics can enhance understanding of $\text{SrBi}_2\text{Ta}_2\text{O}_9$ thin films. In addition, for some processes such as RF sputtering and laser deposition, the preparation of dense $\text{SrBi}_2\text{Ta}_2\text{O}_9$ target ceramics is important. The bulk density, preferred orientation and stoichiometry of the target will affect the deposition rate, phase content and homogeneity of the thin film [5,6]. Thus in this paper the sintering behavior and decomposition of bulk $\text{SrBi}_2\text{Ta}_2\text{O}_9$ ceramics prepared by the conventional solid-state reaction have been investigated.

The starting materials were high purity tantalum pentaoxide ($> 99\%$), strontium carbonate ($> 99\%$) and bismuth oxide ($> 99\%$). The chemicals were mixed in equal molar ratio with deionized water and ball-milled for about four hours using zirconia balls. The obtained slurry was dried at 100°C and calcined at 1000°C for 2 h. In order to study the sintering behavior and decom-

position of bulk $\text{SrBi}_2\text{Ta}_2\text{O}_9$ ceramics in detail, the prepared $\text{SrBi}_2\text{Ta}_2\text{O}_9$ powders were uniaxially pressed into pellets and sintered at different temperatures for different times. Then the densities were measured (Archimedes principle).

Fig. 1 shows the shrinkage versus temperatures diagram of the $\text{SrBi}_2\text{Ta}_2\text{O}_9$ pellets. Up to 700°C the thermal expansion is nearly linear and the curve reaches a maximum at about 700°C . From this point the pellets shrink as the temperature increases, firstly at a relatively slow rate, until about 800°C , and then with a continuously increasing rate. A sudden step occurs at 1350°C .

In order to better understand the shrinkage of $\text{SrBi}_2\text{Ta}_2\text{O}_9$ during heating, the derivatives of Fig. 1 are shown in Fig. 2. Shrinkage starts at about 700°C . Beyond 700°C several valleys can be found and the rate is a step like function of the temperature.

The density of bulk $\text{SrBi}_2\text{Ta}_2\text{O}_9$ ceramic as a function of sintering temperature is shown in Fig. 3. The density tends to increase as the sintering temperature increases. At 1200°C the density reaches its maximum which is about 90% of theoretical density. SEM fractographs of the $\text{SrBi}_2\text{Ta}_2\text{O}_9$ pellets are presented in Fig. 4. The surface of the pellets sintered above 1100°C were examined by XRD. As shown in Fig. 5, decomposition starts at 1200°C and almost all the $\text{SrBi}_2\text{Ta}_2\text{O}_9$ disappears at 1300°C .

* Corresponding author.

E-mail address: libaorang99@mails.tsinghua.edu.cn (B. Li).

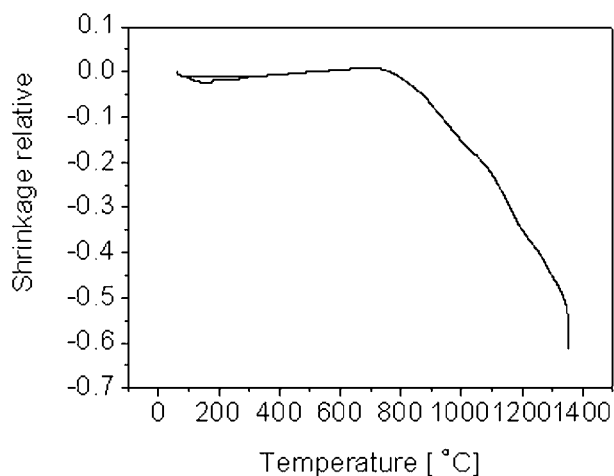


Fig. 1. The shrinkage diagram of the $\text{SrBi}_2\text{Ta}_2\text{O}_9$ pellets.

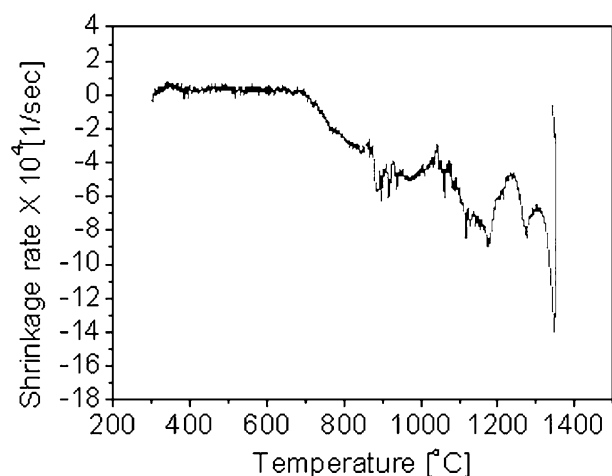


Fig. 2. The shrinkage rate diagram of the $\text{SrBi}_2\text{Ta}_2\text{O}_9$ pellets.

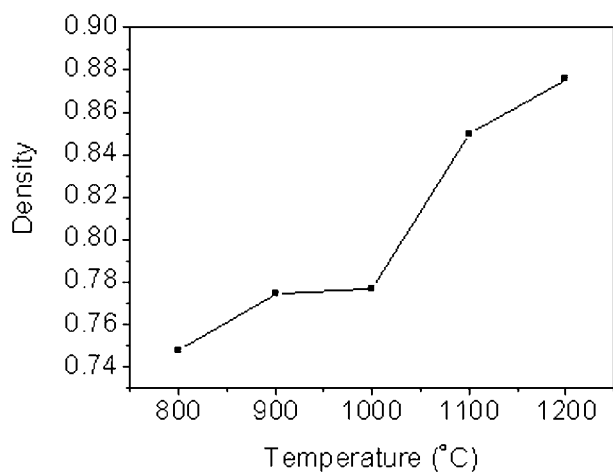
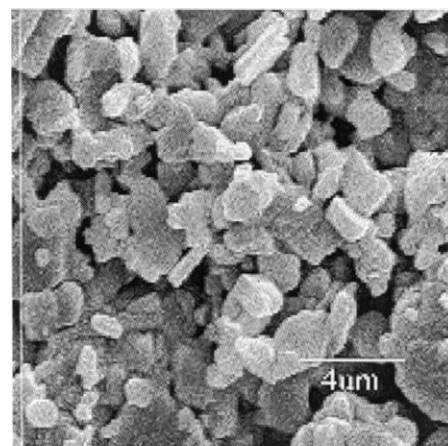
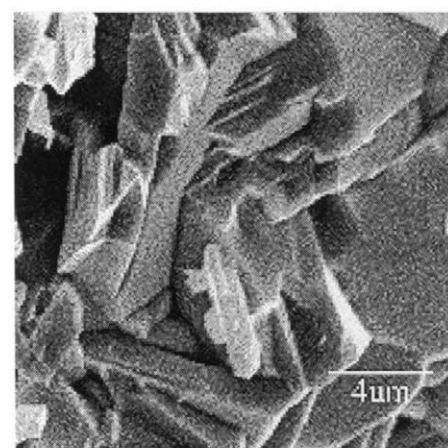


Fig. 3. Density of the $\text{SrBi}_2\text{Ta}_2\text{O}_9$ pellets as a function of sintering temperatures while the length of sintering time are 2 h.



800°C



1200°C

Fig. 4. SEM fractographs of the $\text{SrBi}_2\text{Ta}_2\text{O}_9$ pellets sintered at 800 and 1200 °C while the length of sintering time are 2 h.

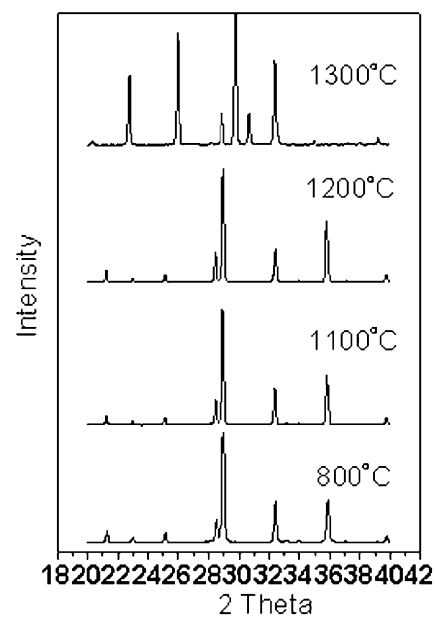


Fig. 5. XRD patterns of the $\text{SrBi}_2\text{Ta}_2\text{O}_9$ pellets sintered at 1100–1300 °C while the length of sintering time are 2 h.

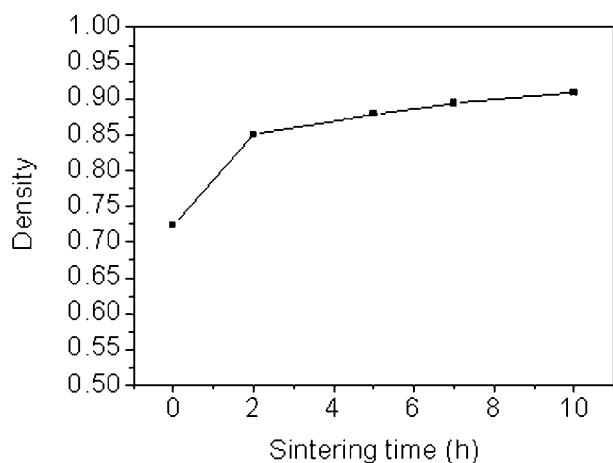


Fig. 6. Density of pellets as a function of sintering time while the sintering temperature is 1100 °C.

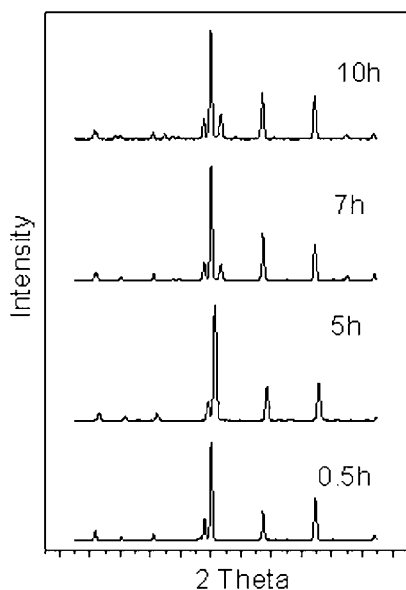


Fig. 7. XRD patterns of the $\text{SrBi}_2\text{Ta}_2\text{O}_9$ pellets sintered at 1100 °C for different length of sintering time.

In order to know the effect of the length of sintering time on the decomposition of bulk $\text{SrBi}_2\text{Ta}_2\text{O}_9$ ceramics, the pellets were sintered at 1100 °C for different times. Density versus sintering time is graphed in Fig. 6. It shows density increases from 70 to 85% of theoretical density within 2 h, then remains approximately constant and reaches 90% after 10 h. The surfaces of the pellets are also examined by XRD which is shown in Fig. 7. The results show that sintering for longer than 7 h results in decomposition of $\text{SrBi}_2\text{Ta}_2\text{O}_9$.

In conclusion, bulk $\text{SrBi}_2\text{Ta}_2\text{O}_9$ ceramic can be prepared by solid-state reaction and its density is about 90% of theoretical density at 1200 °C. Higher sintering temperature (> 1200 °C) and longer sintering time (> 7 h) can result in decomposition of $\text{SrBi}_2\text{Ta}_2\text{O}_9$.

Acknowledgements

This work was financially supported by the National High Technology Research and Development Program of China (863Program No. 2001AA320502).

References

- [1] Ho Nyung Lee, Alina Visinoiu, Stephen Senz, J. Appl. Phys. 11 (2000) 6658–6664.
- [2] Seshu B. Desu, Dillip P. Vijay, X. Zhang, Bao Ping He, Appl. Phys. Lett. 69 (12) (1996) 1719–1721.
- [3] Y. Ding, Appl. Phys. Lett. 76 (2000) 1103–1105.
- [4] Huei-Mei Tsai, Pang Lin, Tseung-Yuen Tseng, Appl. Phys. Lett. 72 (14) (1998) 1787–1789.
- [5] M. Descamps, D. Remiens, L. Chabal, B. Jaber, B. Thierry, Appl. Phys. Lett. 66 (6) (1995) 685–687.
- [6] D. Bhattacharya, R.K. Singh, P.H. Holloway, J. Appl. Phys. 70 (10) (1991) 5433–5439.