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# Preparation and luminescent properties of Eu-doped $Ln_2O_3$ (Ln = Gd, Lu) thin film by citrate sol–gel process

Qingfeng Liu\*, Qian Liu, Lan Luo

State Key Laboratory of High Performance Ceramics and Superfine Microsturctures, Shanghai Institute of Ceramics, Chinese Academy of Sciences, Shanghai, China

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#### Abstract

 $Ln_2O_3$ :Eu (Ln = Gd, Lu) luminescent thin films were deposited on sapphire and Si substrates using citrate sol-gel technique. The prepared films in different process conditions were characterized using X-ray diffraction (XRD), scanning electron microscopy (SEM) and photoluminescence (PL) measurement. The influence of viscosity, heating rate and annealing atmosphere on the morphology of  $Ln_2O_3$ :Eu films were discussed in detail. It is indicated that crack-free films could be prepared after sintering at temperatures ranging from  $500-1100\,^{\circ}$ C with heating rate of  $2.5\,^{\circ}$ C/min. The photoluminescence properties of the films were highly dependent on the annealing atmosphere, and the highest emission intensity was obtained for the sample by annealing in water vapor/oxygen atmosphere.

Keywords: A. Sol-Gel processes; Thin films; Ln<sub>2</sub>O<sub>3</sub>

## 1. Introduction

Europium-doped oxides are some of the most promising phosphors for cathode ray tubes (CRT) and flat panel displays (FPD's) applications [1]. Recently, there has been a growth of interest in using europium-doped sesquioxide  $Ln_2O_3$  ( $Ln=Gd,\ Lu$ ) as scintillators for X-ray medical imaging systems owing to its high-effective atomic number. However, the diffusion loss of such bulk scintillators limited the development of high-resolution X-ray detectors. It has been reported that the use of scintillators deposited as thin films would be a suitable solution to reduce the diffusion loss [2,3].

Although thin  $Ln_2O_3$  film has been widely prepared by a variety of deposition techniques, such as electron beam evaporation [4], metallo-organic chemical vapour deposition (MOCVD) [5], ion beam epitaxial technique [6], there were only a few reports on Eu-doped  $Ln_2O_3$  thin film and its luminescence properties, which was prepared by sol–gel method [7,8]. In those cases, the sol–gel precursors used were rare

\* Corresponding author. Tel.: +86-21-52412404; fax: +86-21-52413122.

E-mail address: qfliu@mail.sic.ac.cn (Q. Liu).

earth isopropoxide, which suffer from high cost, toxicity and difficulty in controlling the experimental processes.

In this paper Ln<sub>2</sub>O<sub>3</sub>:Eu luminescent thin films were prepared by an alternative citrate sol–gel method, which is based on the chelation of a metallic cation by critic acid, and further polymerization promoted by the addition of ethylene glycol and consequent polyesterification [9–11]. The suitable conditions for producing high-quality Ln<sub>2</sub>O<sub>3</sub>:Eu thin films and relative luminescent properties were investigated.

# 2. Experimental

High purity gadolinium/lutetium oxide ( $Ln_2O_3$ , Ln = Gd, Lu; 99.99%) and europium oxide ( $Eu_2O_3$ ; 99.99%) were used as raw materials. After dissolved in concentrated HNO<sub>3</sub> (A.R., analytical reagent), citric acid (CA) was added to the solution in a molar ratio CA/Ln = 4:1, then ethylene glycol (EG) was added in a molar ratio CA/EG = 1:10. The solution was stirred for 2 h at 80 °C ready for film coating. The viscosity was adjusted by addition of water in the range of 15–40 cP. Silicon and sapphire crystal were cleaned carefully and used as the substrates. The films were deposited by a dip-coating process, with a withdrawal speed of 6 mm/min, and then treated at 110 °C for 60 min to induce the

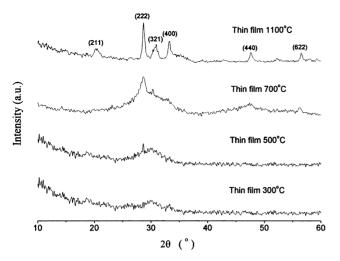


Fig. 1. XRD patterns of  $Gd_2O_3$ :Eu thin films with different annealing temperature.

polymerization. After pre-annealing, the films were calcined at  $500-1100\,^{\circ}\text{C}$  for 2 h, with the heating rate of 1, 2.5, 5 and  $10\,^{\circ}\text{C/min}$ . This deposition procedure was repeated three to five times in order to increase the films thickness.

Phase analysis of the films was performed by X-ray diffraction (XRD) using a Rigaku-Dmax 2550 V diffractometer. Surface morphology of the films was carried out using scanning electron microscopy (SEM). The emission spectra were taken on a Perkin-Elmer LS55 fluorescent spectrometer at room temperature.

## 3. Results and discussion

# 3.1. Phase identification by XRD

Fig. 1 shows the XRD patterns of  $Gd_2O_3$ :Eu thin films heat treated at 300, 500, 700 and  $1100\,^{\circ}$ C. It can be seen

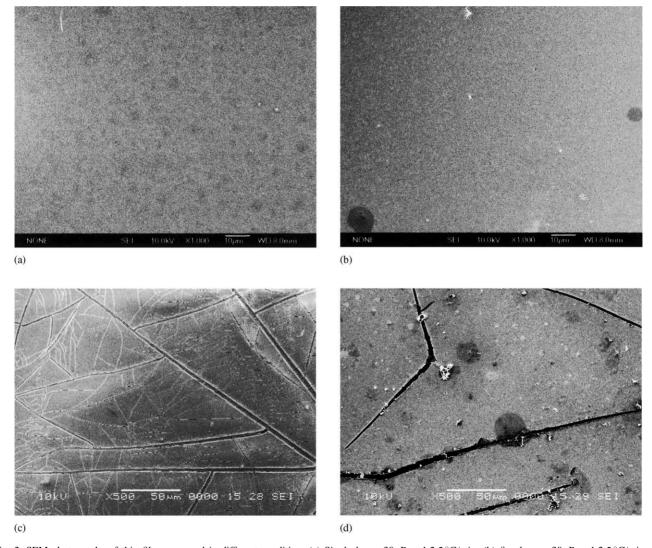


Fig. 2. SEM photographs of thin films prepared in different condition. (a) Single layer,  $30\,\text{cP}$  and  $2.5\,^\circ\text{C/min}$ ; (b) five layers,  $30\,\text{cP}$  and  $2.5\,^\circ\text{C/min}$ ; (c) single layer,  $40\,\text{cP}$  and  $2.5\,^\circ\text{C/min}$ ; (d) single layer,  $30\,\text{cP}$  and  $5\,^\circ\text{C/min}$ .

that thin films remain amorphous below  $300\,^{\circ}\text{C}$  and begin to crystallize at  $500\,^{\circ}\text{C}$ . With the increase of annealing temperature, the crystallinity of the films increases gradually. After annealed at  $1100\,^{\circ}\text{C}$ , thin film shows typical diffraction pattern of cubic phase and no second phase is observed.

# 3.2. Surface morphology

It is well known that the viscosity of the solution and heating rate are the most important parameters for preparation of crack-free films. Fig. 2 shows SEM photographs of thin films prepared at different viscosity and heating rate. It is clear that crack-free films can be obtained at low viscosity than 30 cP with heating rate of 2.5 °C/min (Fig. 2(a)), and no crack appears even when the thickness increased to 920 nm by five repeated coating process (Fig. 2(b)). This result is different with the result of Simoes, which indicated the critical thickness for crack-free films for PLZT films is about 350 nm [10]. However, either higher viscosity or higher heating rate will induce the stress due to volume shrinkage and result in the appearance of crack, as shown in Fig. 2(c) and (d). Obviously heating rate of 2.5 °C/min and viscosity low than 30 cP is the optimal condition for the high-quality film deposition.

#### 3.3. Luminescent properties of the films

Emission spectrum of Ln<sub>2</sub>O<sub>3</sub>:Eu thin films under 254 nm UV excited are shown in Fig. 3. The characteristic transitions  ${}^5D_0-{}^7F_J$  (J=1–4) of Eu<sup>3+</sup> ion are observed in both doped Lu<sub>2</sub>O<sub>3</sub> and Gd<sub>2</sub>O<sub>3</sub> cases. The dominate main emission peak correspond to the  ${}^5D_0-{}^7F_2$  transition at 611.0 and 611.6 nm, respectively. These luminescence spectra are in good agreement with previously reported luminescent measurements for polycrystalline powders [1].

Fig. 4 presents the emission spectra of the Ln<sub>2</sub>O<sub>3</sub> thin films annealed at various atmospheres at 1000 °C. It is shown that the annealing atmospheres affected the relative inten-

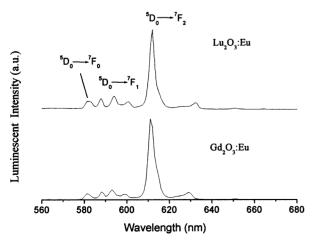
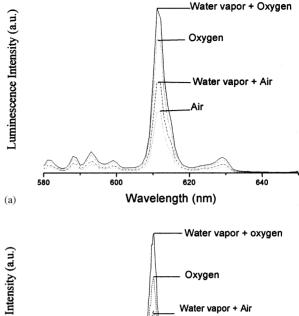


Fig. 3. Emission spectra of Ln<sub>2</sub>O<sub>3</sub>:Eu thin films under 254 nm UV excited.



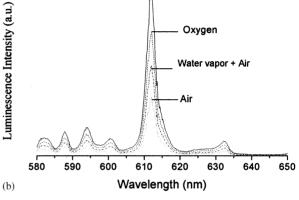


Fig. 4. Luminescent spectra of  $Ln_2O_3$ :Eu thin film annealed at various atmosphere. (a)  $Gd_2O_3$ ; (b)  $Lu_2O_3$ .

sities strongly. Oxygen atmosphere caused an increase of a factor two relative to air, and the introduction of water vapor could also improve the luminescent intensity. The highest emission intensity was obtained by annealing at water vapor/oxygen atmosphere. The atmosphere might help the decomposition of carbon in the film, which could reduce the luminescent efficiency [12].

### 4. Conclusion

The luminescent  $Ln_2O_3$ :Eu thin films were prepared by citrate sol–gel method. Heating rate of  $2.5\,^{\circ}$ C/min and viscosity lower than  $30\,^{\circ}$ CP were the optimal condition for the high-quality film deposition. The luminescent intensity increased with the increase of annealing temperature, and the annealing atmospheres affected the relative intensities strongly. The highest emission intensity was obtained for the film by annealing at water vapor/oxygen atmosphere at  $1000\,^{\circ}$ C.

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# References

- L. Sun, J. Yao, C. Liu, et al., Rare earth activated nanosized oxide phosphors: synthesis and optical properties, J. Lumin. 87–89 (2000) 447–450.
- [2] O. Pons Y Moll, A. Huignard, E. Antic-Fidancev, et al., Eu<sup>3+</sup>- and TM<sup>3+</sup>-doped yttrium oxide thin films for optical applications, J. Lumin. 87–89 (2000) 1115–1117.
- [3] G.A. Hirata, J. Mckittrick, M. Avalos-Borja, J.M. Siqueiros, D. Devlin, Physical properties of Y<sub>2</sub>O<sub>3</sub>:Eu luminescent films grown by MOCVD and laser ablation, Appl. Surf. Sci. 113–114 (1997) 509–514.
- [4] N.K. Sahoo, S. Thakur, M. Senthilkumar, D. Bhattacharyya, N.C. Das, Reactive electron beam evaporation of gadolinium oxide optical thin films for ultraviolet and deep ultraviolet laser wavelengths, Thin Solid Films 440 (2003) 155–168.
- [5] G. Bonnet, M. Lachkar, J.C. Colson, J.P. Larpin, Characterization of thin solid films of rare earth oxides formed by the metallo-organic chemical vapour deposition technique, for high temperature corrosion applications, Thin Solid Films 261 (1995) 31–36.

- [6] J.-P. Zhou, C.-L. Chai, S.-Y. Yang, Z.-K. Liu, Anomalous temperature dependence of photoluminescence from stoichiometric GD<sub>2</sub>O<sub>3-X</sub> film, J. Cryst. Growth 260 (1-2) (2004) 136–142.
- [7] A. Garcia-Murillo, C. Le Luyer, C. Garapon, C. Dujardin, et al., Optical properties of europium-doped Gd<sub>2</sub>O<sub>3</sub> waveguiding thin films prepared by the sol–gel method, Opt. Mater. 19 (2002) 161– 168.
- [8] A. Garcia-Murillo, C. Le Luyer, C. Dujardin, C. Pedrini, J. Mugnier, Elaboration and characterization of Gd<sub>2</sub>O<sub>3</sub> waveguiding thin films prepared by the sol–gel process, Opt. Mater. 16 (2001) 39–46.
- [9] M.P. Pechini, US Patent, 3 330 697, July 1967.
- [10] A.Z. Simoes, M.A. Zaghete, M. Cilense, J.A. Varela, et al., Preparation of 9/65/35 PLZT thin films deposited by a dip-coating process, J. Eur. Ceram. Soc. 21 (2001) 1151–1157.
- [11] Y.L. Chai, D.T. Ray, G.J. Chen, Y.H. Chang, Synthesis of  $La_{0.8}Sr_{0.2}Co_{0.5}Ni_{0.5}O_{3-\delta}$  thin films for high sensitivity CO sensing material using the Pechini process, J. Alloys Compd. 333 (2002) 147–153.
- [12] E. Zych, On the reasons for low luminescence efficiency in combustion-made  $Lu_2O_3$ :Tb, Opt. Mater. 16 (2001) 445–452.