

# The effect of sintering atmosphere on the microwave dielectric properties of $V_2O_5$ doped $BiNbO_4$ ceramics

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## Abstract

Sintering behavior and microwave dielectric properties of  $BiNbO_4$  doped with  $V_2O_5$  sintered under ambient and  $N_2$  atmosphere were investigated. The densification temperature and dielectric constant of  $BiNbO_4$  fired under ambient atmosphere decreased from 975 to 850 °C and 45.6 to 42.2, respectively, as the amount of  $V_2O_5$  increased from 0.1 to 3.2 mol%.  $BiNbO_4$  ceramics was very sensitive to low oxygen partial pressure atmosphere and can be severely reduced under  $H_2$  atmosphere. The apparent density of ceramics sintered under high purity  $N_2$  was smaller than that sintered under ambient atmosphere due to the production of vacancies defects. The formation of oxygen vacancies did not change the crystal structure or decay microwave properties. On the contrary, better microwave properties:  $\epsilon_r = 42.7$ ,  $Q_f = 28,500$  were gotten when  $BiNbO_4$  ceramics with 0.8 mol%  $V_2O_5$  additives sintered under  $N_2$  atmosphere.

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## 1. Introduction

Multilayer microwave devices have developed to reduce the size of band-pass filters and antenna duplexers in mobile radio communication systems. In multilayer structures, low sintering temperature dielectric materials were needed to co-fire with low melting point inner electrode such as copper or silver. Bismuth-based dielectric ceramics were well known as low-fire materials and had been studied for multilayer ceramic capacitors; the microwave dielectric properties of  $BiNbO_4$  ceramics were first reported by Kagata et al. [1]. Compared to other materials in microwave devices using noble metal electrode, base metal electrode (BME) copper can be applied to  $BiNbO_4$  systems due to the added CuO sinter additive in the past study [2]. Some research [3] showed that Ag could react with  $BiNbO_4$  ceramics, so Cu appeared to be the only useable BME for  $BiNbO_4$  system.

A major problem of BME was that the devices must be fired under low oxygen partial pressure atmosphere to protect Cu from oxidation. Under that atmosphere the  $BiNbO_4$  could be slightly reduced, forming oxygen vacancies and

other defects in the lattices. However the affects of sintering atmosphere on the sintering behavior and microwave dielectric properties of  $BiNbO_4$  ceramics doped with  $V_2O_5$  were not reported yet. It was interesting to clarify the relationship above. The aim of this paper was just to investigate the influence of sintering atmosphere on the sintering behavior and microwave dielectric properties of  $BiNbO_4$  ceramics.

## 2. Experiment procedures

Pure starting materials  $Bi_2O_3$ ,  $Nb_2O_5$  (>99.95%) were mixed according to the composition  $BiNbO_4$ . Mixtures were milled with  $ZrO_2$  balls for 24 h in deionized (DI) water then dried. After drying, the powders were ground and calcined at 800 °C for 5 h. The calcined materials were grinded and mixed with  $V_2O_5$  ( $x = 0.1, 0.2, 0.4, 0.8, 1.6, 3.2$  mol%, V1–V6) as the sinter aids and then sieved using a 200-mesh screen. PVA were added as binder before the powder pressed uniaxially into one pellet 12 mm in diameter 1 mm in thickness and another 15 mm in diameter and 6–8 mm in thickness. The smaller pellets were sintered at temperatures from 825 to 975 °C for 1 h under ambient atmosphere. Examinations of the shrinkage factors were acted out to determine the proper sintering temperature. Shrinkage factor was defined

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as:

$$\alpha = \frac{D_0 - D_T}{D_0}$$

where  $D_0$ ,  $D_T$  stand for the diameters of green and fired specimens, respectively.

Protect gas in our study were high purity  $N_2$  (>99.999%). Specimen's color sintered under  $N_2$  were found from light yellow to green with the increase of  $V_2O_5$  content, that very different from the samples sintered under ambient atmosphere, from light yellow to red. This implied the valence of element V had changed due to the low oxygen partial pressure atmosphere.

Crystalline phases were analyzed by X-ray diffraction method. Microstructure was observed by a scanning electron microscope. Columns-like specimen were sintered under the optimized sintering temperature under various atmospheres for 2 h. Their densities were measured by the liquid Archimedes method using deionized water as the liquid. Measurements of the dielectric constant and the unloaded  $Q$  values on  $TE_{011}$  mode at 4–6 GHz were completed by Hakki and Coleman's dielectric resonator method after polishing the samples.

### 3. Results and discussion

Pure  $BiNbO_4$  had a crystal structure similar to  $SbTaO_4$  (orthorhombic) below 1020 °C, then transformed to high temperature phase (triclinic) gradually as temperature increased [4]. It was reported that adding proper dopant in this system could lower the phase transforming temperature.

In our experiments, the triclinic phase was not found due to the low sintering temperature, and according to the XRD results, the sample with the highest  $V_2O_5$  content sintered at 850 °C was typically  $\alpha$ - $BiNbO_4$  phase without any impurity phase. As shown in Fig. 1, the sintering atmosphere do not change the sample's phase structure, it revealed the amounts

Table 1

Shrinkage factor of samples with various  $V_2O_5$  content sintered at different temperatures under ambient atmosphere

Temperatures (°C)	Samples					
	V1	V2	V3	V4	V5	V6
825						0.124
850					0.132	0.128
875					0.134	0.127
900			0.127	0.134	0.131	0.127
925	0.123	0.131	0.129	0.132	0.13	
950	0.131	0.133	0.126	0.129		
975	0.133	0.133	0.127	0.124		
1000	0.133	0.134				

of vacancies in the lattice did not reach a level enough to changed the phase structure, but as will mention later did cause the apparent density decrease.

Shrinkage factor as a function of fire temperature of samples doped with various  $V_2O_5$  contents were listed in Table 1. Shrinkage factor increased first, reached a maximum approximately 0.13, then decreased slightly with the temperature increasing in most samples, thus proper sinter temperature can be drawn at the temperature get maximal shrinkage factor. Optimized sintering temperature decreased from 975 to 850 °C with  $V_2O_5$  content increased from 0.1 to 3.2 mol%.

The SEM revealed the changes in morphology in samples fired under optimized temperature with different content  $V_2O_5$  dopant. As shown in Fig. 2, poles were rarely found on the surface of samples even with very little  $V_2O_5$  dopants. It showed the added  $V_2O_5$  acted as efficient sintering aids. It was also interesting to find that the grain size was larger and more uniform in the samples with lower  $V_2O_5$  contents. Samples sintered under  $N_2$  showed the same micrograph.

From Table 2, we can draw the conclusion that ceramics doped with  $V_2O_5$  were all well densified, got a density higher than 95% theoretical densities (7.35 g/cm<sup>3</sup>) (estimated from the standard XRD patterns of  $BiNbO_4$ ). It was

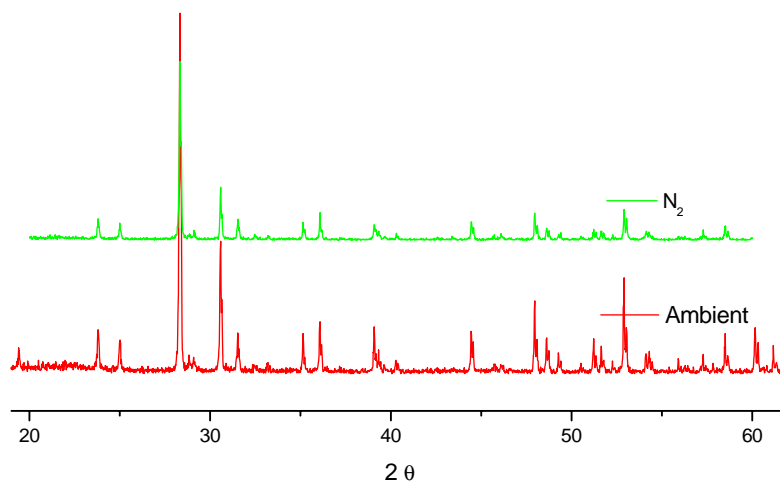


Fig. 1. The XRD patterns of V6 sintered at 850 °C under different atmospheres.

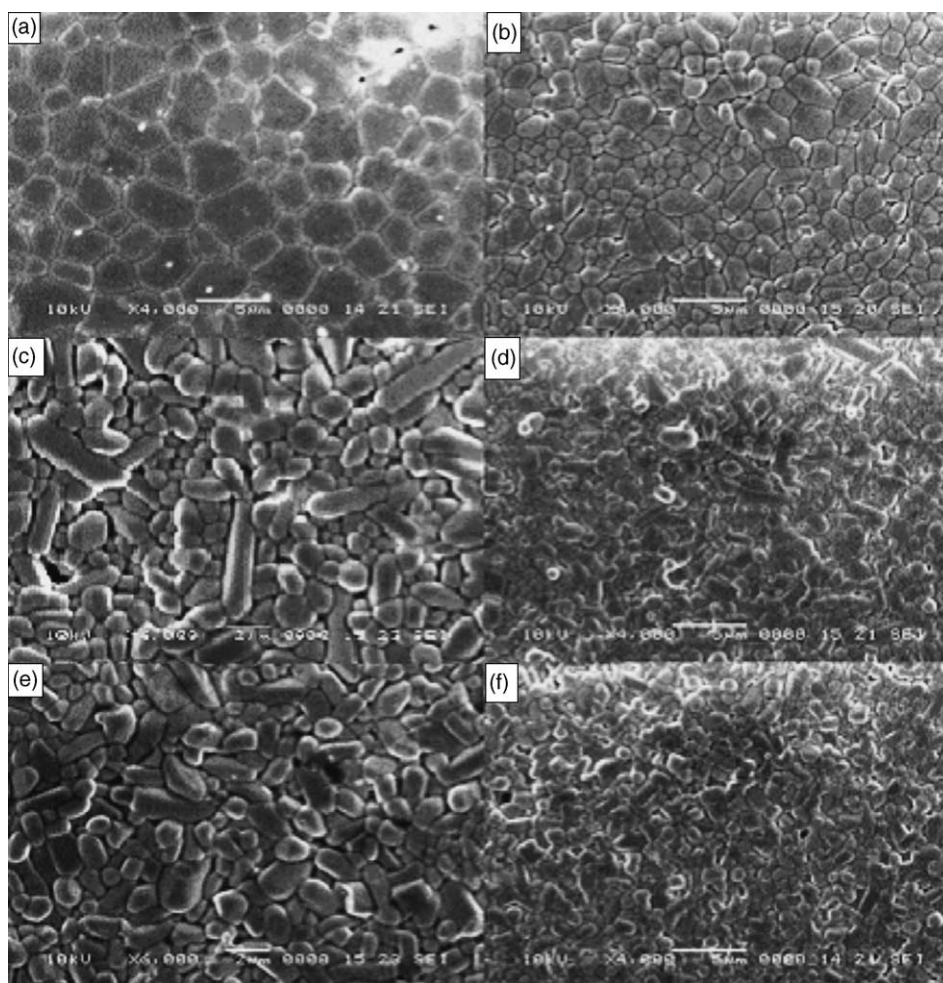
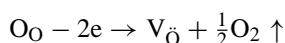


Fig. 2. SEM micrographs of V1 sintered at 950 °C under ambient (a) and N<sub>2</sub> (b); V4 sintered at 900 °C under ambient (c) and N<sub>2</sub> (d); V6 sintered at 850 °C under ambient (e) and N<sub>2</sub> (f).

reported that pure BiNbO<sub>4</sub> could not be densified under 1000 °C, so, a little amount of V<sub>2</sub>O<sub>5</sub> could be enough to densify BiNbO<sub>4</sub>. The density of samples sintered at optimized sintering temperature under ambient atmosphere first increased, come to a maximum 7.149 at  $x = 0.2$  mol% then decreased slowly as increasing V<sub>2</sub>O<sub>5</sub> content. The same trend was found in the samples sintered under N<sub>2</sub> but to our astonishment, all the densities of the samples sintered under N<sub>2</sub> were smaller than that sintered under ambient atmosphere at the same temperature. Excluding experimen-

tal error, this can only been explained by the formation of oxygen vacancies in the ceramics when sintered under low oxygen partial pressure atmosphere.

It was known that metal oxide ceramics could be slightly reduced when sintered under reducing atmosphere. This procedure can be described as:



The formation of oxygen vacancies responds for many related properties changes in specimen sintered under low oxygen partial pressure atmosphere, such as conductance, density, and so on. In the case of density, the theoretical value of perfect BiNbO<sub>4</sub> can be calculated by:

$$\rho = [\text{Bi}_\text{Bi}]M_\text{Bi} + [\text{Nb}_\text{Nb}]M_\text{Nb} + [\text{O}_\text{O}]M_\text{O}$$

where  $M$  stands for the atomic weight. The sum is determined by both metal and oxygen contents, so the formation of large amounts of oxygen vacancies at the oxygen sites in the lattice can reduce the apparent density detectably.

Dielectric constants of BiNbO<sub>4</sub> doped with V<sub>2</sub>O<sub>5</sub> under microwave region were shown in Fig. 3. It decreased sharply

Table 2

Densities of samples with various V<sub>2</sub>O<sub>5</sub> contents sintered at optimized sintering temperature under ambient and N<sub>2</sub> atmosphere (ST: sintering temperature)

	Samples					
	V1	V2	V3	V4	V5	V6
ST (°C)	975	950	925	900	875	850
Air	7.093	7.149	7.143	7.128	7.104	7.047
N <sub>2</sub>	7.089	7.131	7.139	7.121	7.099	7.032

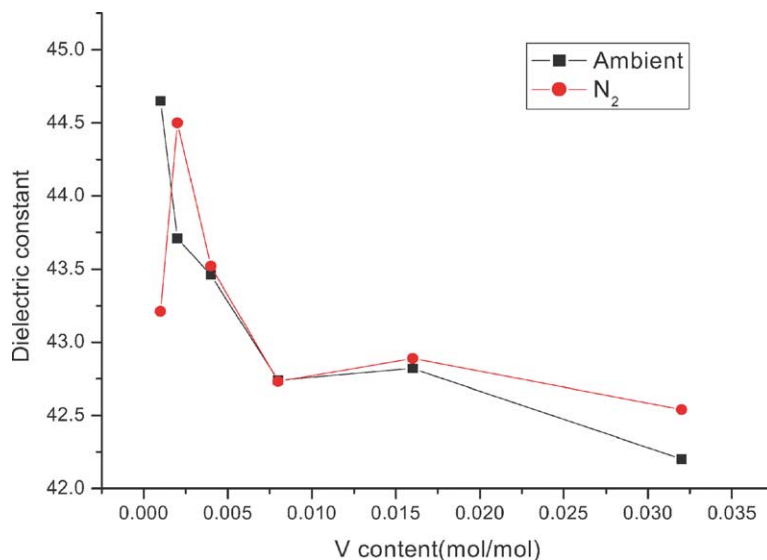


Fig. 3. Dielectric constant of BiNbO<sub>4</sub> sintered at optimized sintering temperature under different atmospheres as a function of V<sub>2</sub>O<sub>5</sub> contents.

from 44.7 at  $x = 0.1$  mol% to 42.7 at  $x = 0.8$  mol%, and then saturated at 42 with more V<sub>2</sub>O<sub>5</sub> additives when samples sintered under ambient atmosphere. The same trend was found in the samples sintered under N<sub>2</sub> atmosphere. Their permittivity appeared to be a little higher than that of the samples fired under ambient atmosphere, especially V<sub>2</sub>O<sub>5</sub> additives reached a higher level of contents.

The Qf values were strongly dependent on the amounts of V<sub>2</sub>O<sub>5</sub> additives and had the same trend as the bulk density of ceramics. As shown in Fig. 4, the Qf values increased from 4530 to 22,100 as the amount of V<sub>2</sub>O<sub>5</sub> additive increased from 0.1 to 0.8 mol% when the samples sintered under ambient atmosphere, then followed by a decrease to 1600 as V<sub>2</sub>O<sub>5</sub> content reached 3.2 mol%.

It was known that the dielectric loss of ceramics in the microwave region had an intrinsic and an extrinsic origin. Anharmonic phonon decay processes in the pure crystal lattice caused the intrinsic losses while crystal defects and microstructures caused the extrinsic losses. The XRD patterns showed us the BiNbO<sub>4</sub> ceramic had a single phase with various V<sub>2</sub>O<sub>5</sub> additives, it implied the extrinsic loss dominated the dielectric loss of the ceramics. From Fig. 2 minimum poles can be found on the surface of specimen doped with 0.1 mol% V<sub>2</sub>O<sub>5</sub>, this answered for the lower Qf of the sample. More V<sub>2</sub>O<sub>5</sub> additive more compact samples can be gotten, thus resulted in higher Qf value. The ceramics got a better figure of merit  $Qf = 22,100$  at  $x = 0.8$  mol%. However, on increasing the amount of V<sub>2</sub>O<sub>5</sub> from 0.8 to

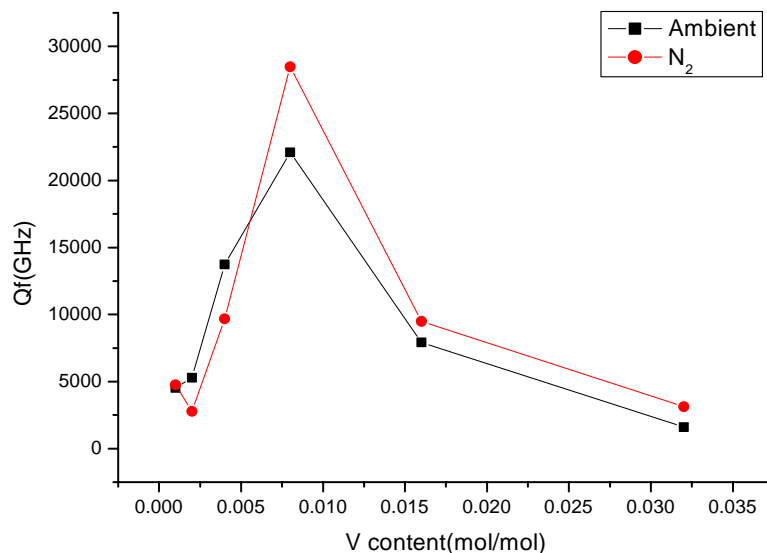


Fig. 4. Qf of BiNbO<sub>4</sub> sintered at optimized sintering temperature under different atmospheres as a function of V<sub>2</sub>O<sub>5</sub> contents.

3.2 mol%, Qf values were degraded abruptly. The reason might be the grain boundary phase or second phase in the ceramics caused by too many additives, and thus degraded the dielectric properties.

The Qf values of samples sintered under N<sub>2</sub> showed the same trend. The maximum was about 28,500 and was gotten at  $x = 0.8$  mol%. It was interesting to find that the Qf values were much higher when the samples sintered under N<sub>2</sub> than that sintered under ambient atmosphere. It was opposite to the trend of density. Since the XRD and SEM results showed us little difference between them, this can only be explained by the decrease of densities, i.e. the formation of oxygen vacancies. So it seems that proper content of oxygen vacancies can enhance the microwave dielectric properties of the ceramics.

#### 4. Conclusion

V<sub>2</sub>O<sub>5</sub> was an employable sintering aid for BiNbO<sub>4</sub> ceramic. BiNbO<sub>4</sub> can be densified below 1000 °C under ambient atmosphere with very little V<sub>2</sub>O<sub>5</sub>, and the ceramic with 0.8 mol% V<sub>2</sub>O<sub>5</sub> got better microwave dielectric properties:  $\epsilon_r = 42.7$ , Qf = 22,100. But V<sub>2</sub>O<sub>5</sub> doped BiNbO<sub>4</sub> ceramics were very sensitive to low oxygen partial pressure sintering atmosphere. And the ceramics were severely reduced under H<sub>2</sub> atmosphere, metal bismuth was found on the surface of the samples. A lot of oxygen vacancies formed in the lattices when BiNbO<sub>4</sub> sintered under high purity N<sub>2</sub> atmosphere, and thus changed the apparent density detectably.

But the formations of oxygen vacancies in the samples sintered under N<sub>2</sub> atmosphere did not reach a level enough to decay the microwave dielectric properties. On the contrary, proper content oxygen vacancies enhanced the Qf values of the BiNbO<sub>4</sub> ceramics doped with moderate content of V<sub>2</sub>O<sub>5</sub>, the BiNbO<sub>4</sub> with 0.8 mol% V<sub>2</sub>O<sub>5</sub> sintered under N<sub>2</sub> atmosphere showed excellent microwave dielectric properties:  $\epsilon_r = 42.7$ , Qf = 28,500.

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