

## Short communication

Pressureless sintering of machinable  $\text{Al}_2\text{O}_3/\text{LaPO}_4$  composites in  $\text{N}_2$  atmosphereGuoliang Gong<sup>a,b,\*</sup>, Baolin Zhang<sup>a</sup>, Hanrui Zhang<sup>a</sup>, Wenlan Li<sup>a</sup><sup>a</sup> Shanghai Institute of Ceramics, Chinese Academy of Sciences, Shanghai 200050, PR China<sup>b</sup> Graduate School of the Chinese Academy of Sciences, Beijing 100039, China

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## Abstract

$\text{Al}_2\text{O}_3/\text{LaPO}_4$  composites were fabricated by pressureless sintering in  $\text{N}_2$  atmosphere. The effect of sintering temperature and sintering time on densification was examined, as well as the dependence of mechanical properties and microstructure on  $\text{LaPO}_4$  content. Provided the ratio of La/P was close to 1:1, no reactions were observed after 2 h at 1650 °C in  $\text{N}_2$  atmosphere. The 30 wt.%  $\text{LaPO}_4/\text{Al}_2\text{O}_3$  composites could be machined using cemented carbide drills as those pressureless sintered in air.  $\text{LaAl}_{11}\text{O}_{18}$  was formed when the composite was sintered at 1700 °C for 1 h in  $\text{N}_2$  atmosphere, and the composites were not machinable.

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## 1. Introduction

Ceramic materials like alumina have been widely used as engineering materials. However, their difficulty in machining restricts their use due to high strength and high hardness. In recent years, attempts have been made to develop machinable ceramics [1–4]. According to the research of Morgan et al., two-phase composites consisting of rare-earth phosphates (e.g.,  $\text{CePO}_4$  and  $\text{LaPO}_4$ ) and refractory oxides (e.g.,  $\text{Al}_2\text{O}_3$ ,  $\text{ZrO}_2$  and mullite) can be cut and drilled using conventional tungsten carbide metal-working tools instead of expensive diamond tools.  $\text{Al}_2\text{O}_3/\text{LaPO}_4$  composites are stable in oxidizing environments [5,6]. When the stoichiometric  $\text{LaPO}_4$  is used, there are no reactions at temperatures up to 1750 °C in air.

Recently, machinable  $\text{Al}_2\text{O}_3/\text{LaPO}_4$  composites have been fabricated by different sintering method (pressureless sintering, hot pressing sintering) [2,7,8]. However, there are no reports on fabrication of  $\text{Al}_2\text{O}_3/\text{LaPO}_4$  composites by pressureless sintering in  $\text{N}_2$  atmosphere considering higher

sintering temperature than that of hot pressing may lead to carbothermic reduction of  $\text{LaPO}_4$  [9].

In the present work, the pressureless sintering of  $\text{Al}_2\text{O}_3/\text{LaPO}_4$  composites in  $\text{N}_2$  atmosphere was researched. The compatibility of the system  $\text{Al}_2\text{O}_3/\text{LaPO}_4$  composites in  $\text{N}_2$  atmosphere up to 1700 °C was examined. The effects of sintering temperature and sintering time on the densification were examined. The relationship of mechanical properties and microstructure with  $\text{LaPO}_4$  content was also revealed.

## 2. Experimental procedure

$\text{LaPO}_4$  powder was prepared by calcining rhabdophane-type  $\text{LaPO}_4 \cdot 0.5\text{H}_2\text{O}$  at 1300 °C for 6 h. The calcined powder was milled with silicon nitride balls in ethanol for 3 days and sieved through 200-mesh screen. X-ray diffraction (XRD) patterns of the calcined powder showed that only  $\text{LaPO}_4$  phase existed. The La/P atomic ratio of the calcined powder (ICP) was 0.99. Commercial alumina powder was used.

The mixtures of  $\text{Al}_2\text{O}_3$  and  $\text{LaPO}_4$  (10, 30, 50 and 80 wt.%) powders were ball-milled and sieved using a 200-mesh screen. Pure  $\text{Al}_2\text{O}_3$ ,  $\text{LaPO}_4$  and the mixtures of  $\text{Al}_2\text{O}_3$

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and  $\text{LaPO}_4$  were uniaxially dry-pressed at 50 MPa to bars (4 mm  $\times$  5 mm  $\times$  40 mm), then cold isostatic pressed at 200 MPa. The samples were placed in an alumina crucible, covered with alumina powders, and the alumina crucible was placed in a graphite die. These details are important to decrease carbothermic reduction of  $\text{LaPO}_4$ . The samples were sintered at 1400–1700 °C for 1–5 h in  $\text{N}_2$  atmosphere.

The bulk density and relative density were measured by the Archimedes method. The bending strength was measured by the three-point bending method with specimen dimensions of (3 mm  $\times$  4 mm  $\times$  36 mm), at a bending span of 30 mm, and cross-head speed of 0.5 mm/min. The machinability was tested using tungsten carbide drills at 450 rpm, with a drop of water placed at the drill tip at the beginning of each run. The crystalline phases of composites were determined by X-ray diffraction (XRD) with  $\text{Cu K}\alpha$ , operating at 40 kV and 100 mA. The fracture surface of composites was observed under scanning electron microscopy (SEM).

### 3. Results and discussion

#### 3.1. Chemical compatibility

The compatibility of the  $\text{Al}_2\text{O}_3/\text{LaPO}_4$  system was investigated by analyzing the possible existing phases using X-ray diffraction (XRD). Fig. 1 shows the corresponding XRD results. It can be seen that there were only  $\text{LaPO}_4$  and  $\text{Al}_2\text{O}_3$  phase when the sample was sintered in  $\text{N}_2$  at 1650 °C for 2 h. The result indicates that both phases of the system were compatible at 1650 °C. However,  $\text{LaAl}_{11}\text{O}_{18}$  was found in the outer layer when sintered up to 1700 °C. Morgan et al. reported that there were no reactions in  $\text{Al}_2\text{O}_3/\text{LaPO}_4$  system up to 1750 °C in air [5]. It shows  $\text{Al}_2\text{O}_3/\text{LaPO}_4$  system is more stable in air atmosphere. The reason is that the carbon of the graphite die diffused through the crucible and led to carbothermic reduction of the  $\text{LaPO}_4$ . This

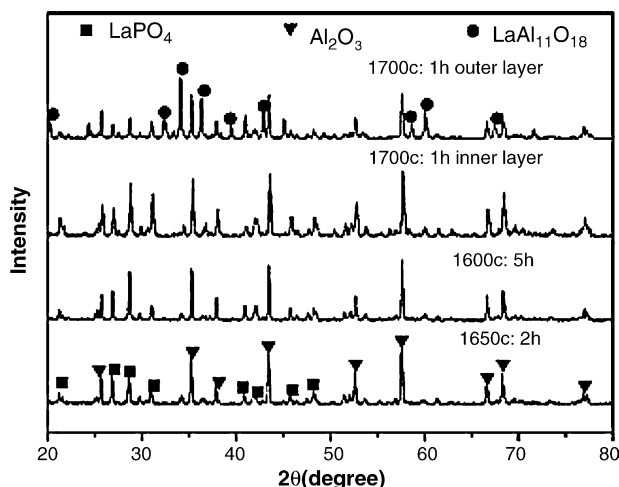


Fig. 1. XRD patterns of the  $\text{Al}_2\text{O}_3/\text{LaPO}_4$  composites.

resulted in a loss of phosphorus and  $\text{LaAl}_{11}\text{O}_{18}$  could be formed in the presence of excess La.

It is well known that the formation of  $\text{LaAl}_{11}\text{O}_{18}$  occurs in two stages [10–12]. The formation of  $\text{LaAlO}_3$  occurs at lower temperatures (<800 °C) and  $\text{LaAlO}_3$  is converted to  $\text{LaAl}_{11}\text{O}_{18}$  at higher temperatures (>800 °C). The reaction speed of the conversion of  $\text{LaAlO}_3$  to  $\text{LaAl}_{11}\text{O}_{18}$  is slow and more than 1800 °C is required for complete conversion for 24 h. In our experiments,  $\text{LaAlO}_3$  was not detected on the 30 wt.%  $\text{LaPO}_4/\text{Al}_2\text{O}_3$  composites sintered at 1700 °C in  $\text{N}_2$  atmosphere. The reason is that  $\text{LaAlO}_3$  formed at 1700 °C (>800 °C) instead of lower temperatures (<800 °C).  $\text{LaAlO}_3$  phase is less stable than  $\text{LaAl}_{11}\text{O}_{18}$  at 1700 °C. The formation of  $\text{LaAlO}_3$  and conversion of  $\text{LaAlO}_3$  to  $\text{LaAl}_{11}\text{O}_{18}$  occurred at the same time.

#### 3.2. Densification

The relationship of bulk density with sintering temperature and sintering time of the 30 wt.%  $\text{LaPO}_4/\text{Al}_2\text{O}_3$  composites (the ratio is chosen to ensure machinability) was summarized in Fig. 2. The densities of the composites increase with the increase of sintering temperature and sintering time. The bulk density of 30 wt.%  $\text{LaPO}_4/\text{Al}_2\text{O}_3$  composite sintered at 1650 °C for 2 h is close to theoretical density. There are reactions at higher sintering temperature, according to XRD results (Fig. 1). Therefore, the sintering temperature was determined as 1650 °C.

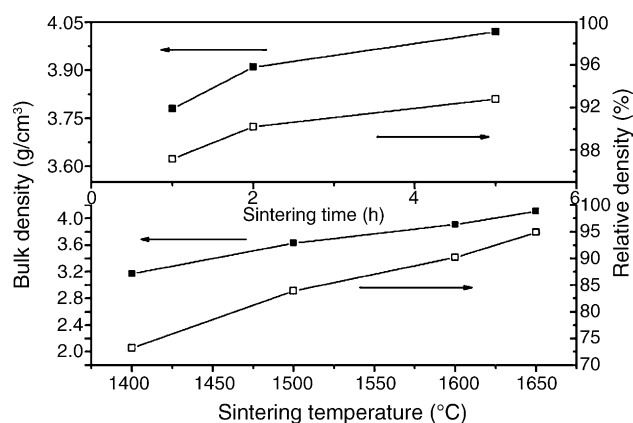


Fig. 2. Effects of sintering time and sintering temperature on the bulk densities and relative densities of the  $\text{Al}_2\text{O}_3/\text{LaPO}_4$  composites.

Table 1

Densities of the  $\text{Al}_2\text{O}_3/\text{LaPO}_4$  composites by pressureless sintering at 1650 °C

Content of $\text{LaPO}_4$ (wt.%)	Bulk density ( $\text{g}/\text{cm}^3$ )	Relative density (%)	Machinability
0	3.89	97.5	No
10	3.96	96.8	No
30	4.11	94.9	Yes
50	4.18	91.7	Yes
80	4.47	91.2	Yes
100	4.96	96.8	Yes

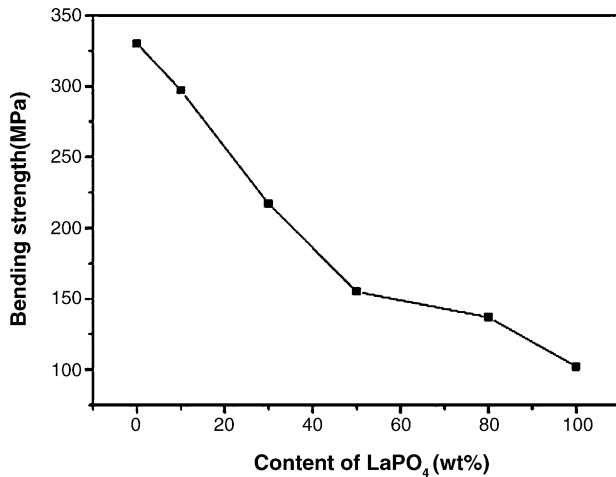


Fig. 3. Effect of LaPO<sub>4</sub> content on the bending strength Fig. 5. Hole drilled in Al<sub>2</sub>O<sub>3</sub>/LaPO<sub>4</sub> composite of Al<sub>2</sub>O<sub>3</sub>/LaPO<sub>4</sub> composites using tungsten carbide drills Second electron image Backscattered electron image (a) 10 wt.% LaPO<sub>4</sub> Second electron image Backscattered electron image (b) 50 wt.% LaPO<sub>4</sub>.

Table 1 shows the densities of the  $x\text{LaPO}_4/(1-x)\text{Al}_2\text{O}_3$  ( $x = 0, 10, 30, 50, 80$  and  $100\%$ ) composites sintered at  $1650^\circ\text{C}$  for 2 h. The densities of the composites increased with the increase of LaPO<sub>4</sub> content due to the higher density of LaPO<sub>4</sub>, while the relative densities of composites

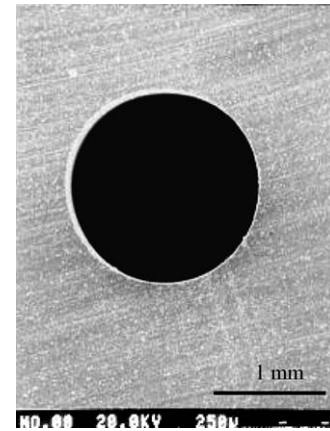


Fig. 5. Hole drilled in Al<sub>2</sub>O<sub>3</sub>/LaPO<sub>4</sub>.

decreased. The introduction of LaPO<sub>4</sub> particles in the matrix retarded the sintering.

### 3.3. Mechanical properties

Variations in bending strength with LaPO<sub>4</sub> content for Al<sub>2</sub>O<sub>3</sub>/LaPO<sub>4</sub> composites are shown in Fig. 3. As compared with those of Al<sub>2</sub>O<sub>3</sub> ceramics (330 MPa), the bending strength of composites remarkably reduced due to lower

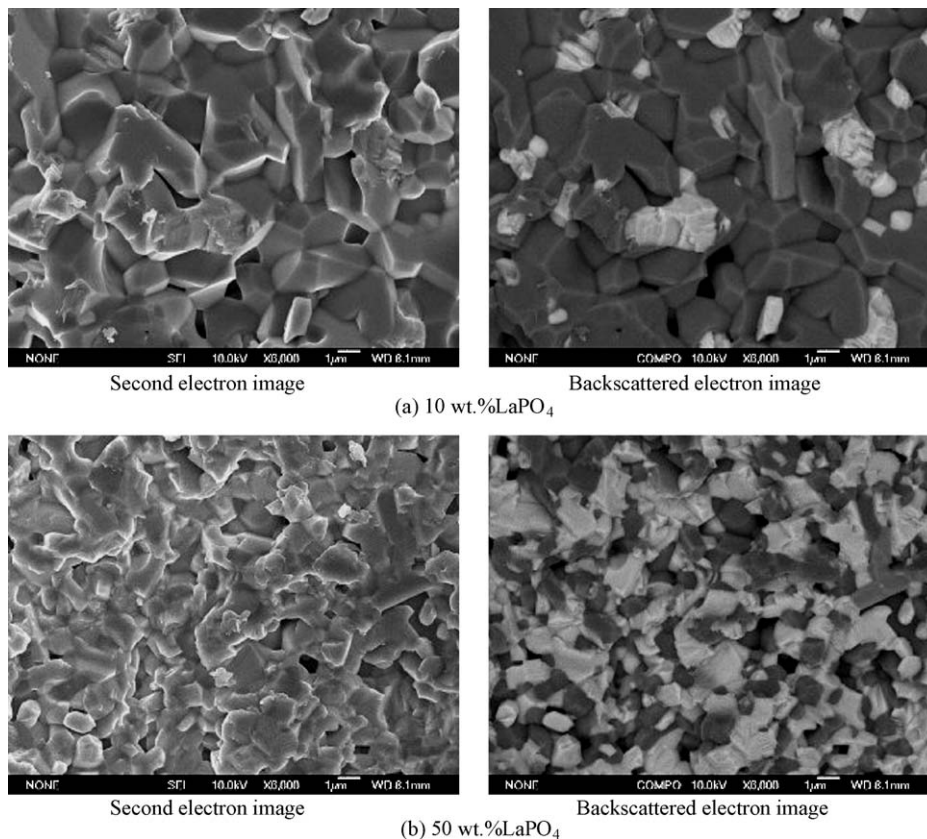


Fig. 4. SEM of Al<sub>2</sub>O<sub>3</sub>/LaPO<sub>4</sub> fracture surfaces at  $1650^\circ\text{C}$ .

bending strength of  $\text{LaPO}_4$  ceramics (102 MPa) and lower densification. With increase of  $\text{LaPO}_4$  content, the bending strength of composites decreased.

### 3.4. Microstructure

Fig. 4 shows the SEM micrographics of the fracture surface of 10 and 50 wt.%  $\text{LaPO}_4/\text{Al}_2\text{O}_3$  composites.  $\text{LaPO}_4$  grains distributed at the  $\text{Al}_2\text{O}_3$  grain boundaries and impeded the grain growth of  $\text{Al}_2\text{O}_3$ . The fracture mode of  $\text{Al}_2\text{O}_3$  grain in composites was mainly intergranular due to the weak bonding of  $\text{Al}_2\text{O}_3$  and  $\text{LaPO}_4$  phases.

### 3.5. Machinability

The monolithic  $\text{Al}_2\text{O}_3$  ceramics were not machinable, while the pure  $\text{LaPO}_4$  ceramics were easily machinable using tungsten carbide drills. The  $x\text{LaPO}_4/(1-x)\text{Al}_2\text{O}_3$  ( $x > 30\%$ ) composites could be machinable. Fig. 5 shows a hole drilled by a tungsten carbide drill on the 30 wt.%  $\text{LaPO}_4/\text{Al}_2\text{O}_3$  composites. The hole was cleanly drilled, with no evidence of cracking or chipping. The reason that composites were machinable is that layered  $\text{LaPO}_4$  phase and the weak interfaces between the two phases are helpful to the machinability, which was discussed by Davis et al. [2]. The 30 wt.%  $\text{LaPO}_4/\text{Al}_2\text{O}_3$  composites sintered at  $1700^\circ\text{C}$  were not machinable. The formation of  $\text{LaAl}_{11}\text{O}_{18}$  is detrimental to weak interface of  $\text{LaPO}_4$  and  $\text{Al}_2\text{O}_3$  phases.

## 4. Conclusion

Machinable  $\text{Al}_2\text{O}_3/\text{LaPO}_4$  composites ceramics could be fabricated by pressureless sintering in  $\text{N}_2$  atmosphere in a carbon furnace.  $\text{LaPO}_4$  is compatible with  $\text{Al}_2\text{O}_3$  in a carbon furnace at  $1650^\circ\text{C}$ .  $\text{LaAl}_{11}\text{O}_{18}$  formed due to the carbon

reduction reaction at  $1700^\circ\text{C}$ . The sinterability and bending strength of  $\text{Al}_2\text{O}_3/\text{LaPO}_4$  composites decreased with the increase  $\text{LaPO}_4$  content. The 30 wt.%  $\text{LaPO}_4/\text{Al}_2\text{O}_3$  composites sintered at  $1650^\circ\text{C}$  could be machined using cemented carbide drill. The formation of  $\text{LaAl}_{11}\text{O}_{18}$  is detrimental to machinability.

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