

Low temperature fabrication and characterizations of β -CaSiO₃ ceramics

L.H. Long^{a,c}, L.D. Chen^{a,*}, J. Chang^b

^aState Key Laboratory of High Performance Ceramics and Superfine Microstructure, Shanghai Institute of Ceramics, Chinese Academy of Sciences, Shanghai 200050, China

^bBiomaterials and Tissue Engineering Research Center, Shanghai Institute of Ceramics, Chinese Academy of Sciences, Shanghai 200050, China

^cGraduate School of Chinese Academy of Science, Shanghai 200050, China

Received 4 February 2005; received in revised form 21 February 2005; accepted 20 March 2005

Available online 16 June 2005

Abstract

Dense β -CaSiO₃ (low temperature phase) ceramics were fabricated by spark plasma sintering (SPS) in the temperature range of 700–950 °C using chemically precipitated amorphous powder. The maximum density of the obtained β -CaSiO₃ ceramics reached about 2.60 g/cm³ with grain size of 100–200 nm. The maximum bending strength was 190 MPa. The Young's modulus and fracture toughness reached 62.8 GPa and 1.67 MPa m^{1/2}, respectively. The simulated body fluid (SBF) tests showed that HAP grains covered the surface of β -CaSiO₃ ceramics completely after 3 days immersion. The thickness of HAP layer was about 15 μ m for the 3 days specimen and it reached 25 μ m for the 7 days immersion. SPS-sintered β -CaSiO₃ ceramics obtained by SPS had both good mechanical properties and excellent bioactivity. © 2005 Elsevier Ltd and Techna Group S.r.l. All rights reserved.

Keywords: C. Mechanical properties; SPS; CaSiO₃; Ceramics; Bioactivity

1. Introduction

Since the discovery of Bioglass by Hench et al. in 1971 [1,2], many glasses, glass–ceramics and ceramics have been found to bond to living bone and most of these materials contain CaO–SiO₂ [3,4]. Recent studies showed that CaSiO₃ (wollastonite) has good bioactivity and thus exhibits prospective application as biomaterials [5,6]. Since CaSiO₃ ceramic is poorly sinterable, many reports so far have only concerned the preparation of CaSiO₃ powder, α -CaSiO₃ (high temperature phase) coating as well as its composite biomaterials [6–10], while few concerned the preparation and properties of β -CaSiO₃ ceramics.

In recent years, a new rapid sintering technique, spark plasma sintering (SPS), has been developed, which is featured with high heating rate and is suitable to densify ceramics at lower temperatures suppressing grain growth. Some biomaterials have also been fabricated by SPS [11,12], and it was reported that the SPS-sintered HAP ceramics

have better bioactivity than that obtained by a conventional hot-pressing method [13].

In this study, amorphous CaSiO₃ powder was prepared through a chemical precipitation method and β -CaSiO₃ ceramics were fabricated by SPS. Their mechanical properties and bioactivity were evaluated.

2. Experimental methods

CaSiO₃ powder for the present study was prepared through a chemical precipitation method. Ca(NO₃)₂·4H₂O and Na₂SiO₃·9H₂O were used as raw materials. They were separately dissolved in de-ionized water in the concentration of 0.3 mol/l. The pH value was adjusted to be 12 using NH₄OH. The obtained Na₂SiO₃ solution was dropped into Ca(NO₃)₂ solution with vigorous agitation. The reactant mixture was filtered, then washed three times with de-ionized water and finally washed by ethanol. The powder was dried at 80 °C, and then calcined at 700 °C for 2 h.

The obtained CaSiO₃ powder was put into a graphite die (30 mm in diameter) and sintered using Spark Plasma

* Corresponding author. Tel.: +86 21 52412520; fax: +86 21 52413122.
E-mail address: cld@mail.sic.ac.cn (L.D. Chen).

Table 1
Ion concentration of SBF in comparison with human blood plasma

	Concentration (mM)							
	Na ⁺	K ⁺	Ca ²⁺	Mg ²⁺	HCO ₃ ⁻	Cl ⁻	HPO ₄ ²⁻	SO ₄ ²⁻
SBF	142.0	5.0	2.5	1.5	4.2	148.5	1.0	0.5
Blood plasma	142.0	5.0	2.5	1.5	27.0	103.0	1.0	0.5

Table 2
The properties of CaSiO₃ ceramics

Sinter temperature (°C)	Density (g/cm ³)	Water absorption (%)	Fracture strength, σ_f (MPa)	Fracture toughness, K_{Ic} (MPa m ^{1/2})	Young's modulus (GPa)
700	2.1	19	140 ± 9	0.72 ± 0.03	40.1 ± 4.0
750	2.6	2.1	190 ± 5	1.67 ± 0.05	62.8 ± 5.6
800	2.5	3.4	171 ± 12	1.58 ± 0.03	57.5 ± 8.4
850	2.3	4.5	164 ± 5	1.57 ± 0.06	51.5 ± 5.0
900	2.2	3.1	109 ± 9	1.54 ± 0.04	51.0 ± 4.7
950	2.2	3.2	100 ± 11	1.52 ± 0.05	50.0 ± 5.2

Sintering (Dr Sinter 2040, Sumitomo Coal Mining Co.) in vacuum (less than 10^{-4} Pa) under a pressure of 40 MPa. The heating rate was 150–200 °C/min and holding time at the sintering temperature was 5 min.

The crystalline phases of the chemically precipitated powder and sintered specimen were analyzed by XRD (Rigaku D/max 2200PC). The density was determined according to Archimedes principle. Microstructures were observed by scanning electron microscopy (SEM, JSM-6700F). The bending strength and Young's modulus were measured with a universal tester (Instron-1195) using a three-point test method with a span length of 20 mm and a crosshead speed of 0.5 mm/min. The fracture toughness, K_{Ic} , was evaluated by the single-edge notch beam method (Instron-1195, notch width: 0.25 mm, the notch depth: 2 mm) with six samples each data. The distribution of elements was analyzed by electron probe microanalyzer (EPMA, EPMA-8705QH₂, Shimadzu, Japan).

The simulated body fluid (SBF) was prepared by dissolving reagent-grade CaCl₂, K₂HPO₄·3H₂O, NaCl, KCl, MgCl₂·6H₂O, NaHCO₃ and Na₂SO₄ in de-ionized water and the ion concentrations were adjusted to be similar to those in human blood plasma (Table 1). The SBF was buffered at pH 7.4 with trimethanol aminomethane–HCl. The sintered CaSiO₃ ceramic (2 mm × 3 mm × 20 mm) was soaked in 21 ml SBF solution and maintained at 36.5 °C. The soaking period was 3 days and 7 days, respectively and SBF was renewed every day.

3. Results and discussions

3.1. Characterization of β -CaSiO₃ ceramics

Fig. 1 shows the XRD patterns of CaSiO₃ ceramics sintered at various temperatures. The as-prepared powder was mainly in amorphous state with small amount of CaO–SiO₂– n H₂O (Fig. 1). Samples sintered by SPS at 700 °C to

900 °C are of β -CaSiO₃ (low temperature phase) ceramics. Sintering temperature and some properties of sintered CaSiO₃ ceramics were summarized in Table 2. The density of CaSiO₃ ceramic sintered at 700 °C was 2.1 g/cm³ and it reached 2.60 g/cm³ for the sample sintered at 750 °C. However, the density decreased gradually and maintained about 2.2 g/cm³ after 900 °C. The water absorption decreased significantly from 19% for the samples sintered at 700 °C to about 3% for those sintered above 750 °C. The bending strength of CaSiO₃ ceramics sintered at 700 °C was 140 MPa, and reached the maximum value of 190 MPa at 750 °C, which was much higher than that of cortical bone and about two times as that of the sintered CaSiO₃ reported previously [10,14]. The Young's modulus and fracture strength showed the same changing tendency, which reached maximum value of 62.8 GPa and 1.67 MPa m^{1/2}, respectively, at 750 °C.

The alteration of mechanical properties of CaSiO₃ was mainly due to the change of the microstructure of the sintered

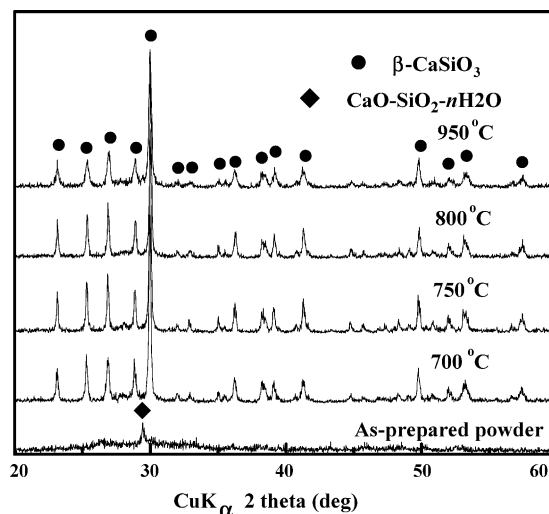
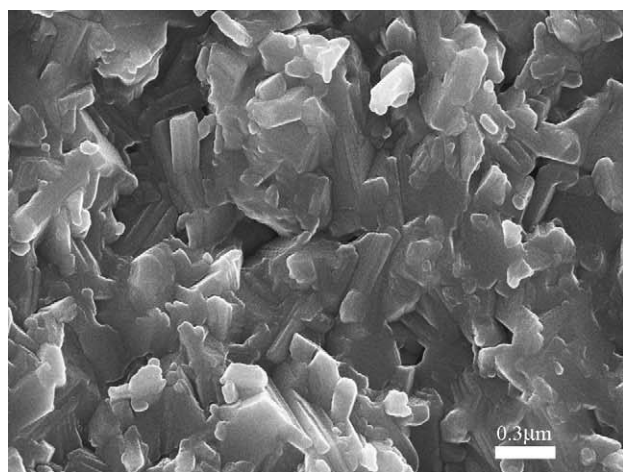
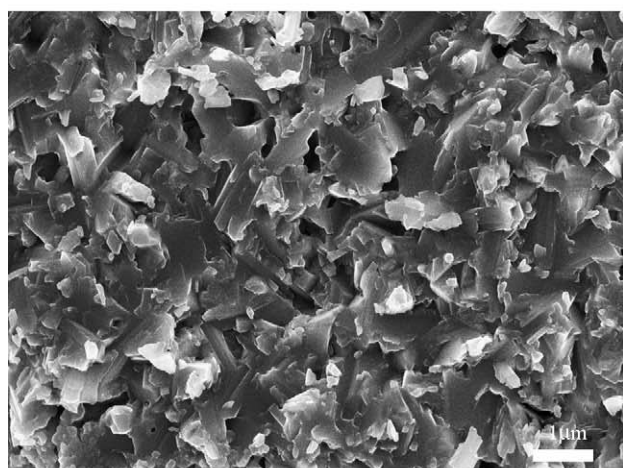


Fig. 1. XRD patterns of CaSiO₃ powder and ceramics sintered by SPS at various temperatures.

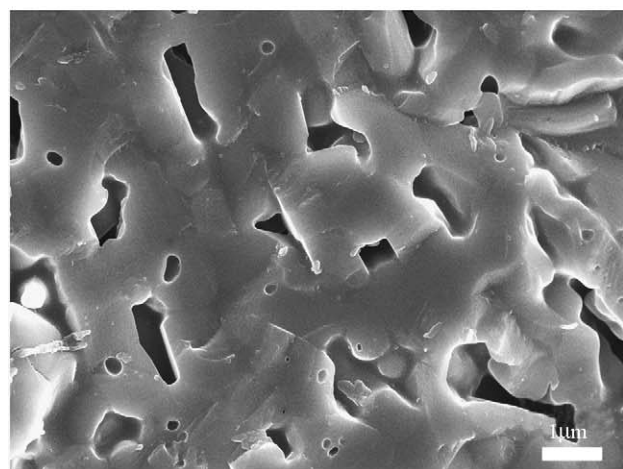
β -CaSiO₃ ceramics as shown in Fig. 2. The SEM micrographs of fractured surface of sintered CaSiO₃ ceramics indicated that the average grain size for the sample sintered at 750 °C was 100–200 nm (Fig. 2A). The fracture was mainly due to



(A)



(B)



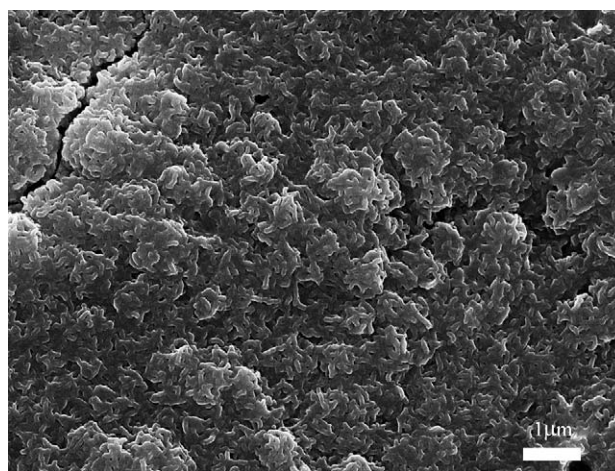
(C)

Fig. 2. SEM photographs of the fracture surfaces of CaSiO₃ ceramics sintered at different temperatures: (A) 750 °C; (B) 800 °C; (C) 900 °C.

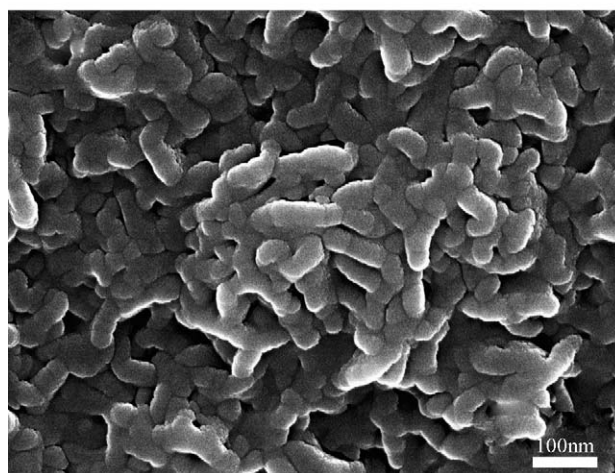
the intergrain cracking. The grain size increased with sintering temperature (Fig. 2B and C). Some transcrystalline fractures were observed for the sample sintered at 800 °C. For the sample sintered at 900 °C, the grains fully grew (Fig. 2C) and many large pores were enclosed within grains. In addition, the smooth fracture plane of the samples sintered at 950 °C showed that the samples sintered at 900 °C exhibited completely transcrystalline fracture. The rapid releasing of residual hydron at high temperature is attributed to the increase of pores and decrease of density after sintering at temperatures above 800 °C.

3.2. SBF immersion tests

Bioactivity was evaluated for β -CaSiO₃ ceramics sintered at 750 °C. Fig. 3 showed SEM micrographs of β -CaSiO₃ ceramics immersed in SBF for 3 days. The specimen surface was completely covered by rod-shaped, 50 nm width and 200 nm length (Fig. 3A and B) HAp grains. Also micro-cracks appeared on the surface (Fig. 3A). These indicated that HAp



(A)



(B)

Fig. 3. SEM photographs of the surface of CaSiO₃ ceramics immersed in SBF for 3 days. (B) is a magnification of (A).

deposited on β -CaSiO₃ ceramic immersed in SBF after 3 days. The thickness of HAp layer was about 15 μ m for the 3 days specimen and it reached 25 μ m for the 7 days immersion. The speed of HAp formation was faster than those previously reported [5,7], which might be attributed to the nano-structure of β -CaSiO₃ ceramic sintered by SPS at 750 °C.

4. Conclusions

β -CaSiO₃ ceramics were densified by SPS at low temperature as 750 °C from amorphous CaSiO₃ powder synthesized through a chemical precipitation method. β -CaSiO₃ ceramics with grains of 100–200 nm were obtained at 750 °C. The obtained β -CaSiO₃ ceramics had good mechanical properties with bending strength of 190 MPa and fracture toughness of 1.67 MPa m^{1/2}. HAp deposited rapidly on the surface of β -CaSiO₃ ceramic in SBF. Therefore, β -CaSiO₃ ceramics sintered by SPS at 750 °C had good mechanical properties and bioactivity and might be a good candidate material for biomaterials.

Acknowledgements

This work is supported by National Science Foundation of China under grant nos. 50232020 and 50325208.

References

- [1] L.L. Hench, Ceramics: from concept to clinic, *J. Am. Ceram. Soc.* 74 (1991) 1487–1510.
- [2] L.L. Hench, R.J. Splinter, W.C. Allen, T.K. Greenlee, Bonding mechanisms at the interface of ceramic prosthetic materials, *J. Biomed. Mater. Res. Symp.* 2 (1971) 117–141.
- [3] W.P. Cao, L.L. Hench, Bioactive materials, *Ceram. Int.* 22 (1996) 493–507.
- [4] W. Min, Developing bioactive composite materials for tissue replacement, *Biomaterials* 24 (2003) 2133–2151.
- [5] P. Siriphannon, Y. Kameshima, A. Yasumori, K. Okada, S. Hayashi, Formation of hydroxyapatite on CaSiO₃ powders in simulated body fluid, *J. Eur. Ceram. Soc.* 22 (2002) 511–520.
- [6] P. Siriphannon, S. Hayashi, A. Yasumori, K. Okada, Preparation and sintering of CaSiO₃ from coprecipitated powder using NaOH as precipitant and its apatite formation in simulated body fluid solution, *J. Mater. Res.* 14 (1999) 529–536.
- [7] X.Y. Liu, X.B. Zheng, C.X. Ding, Apatite formed on the surface of plasma-sprayed wollastonite coating immersed in simulated body fluid, *Biomaterials* 22 (2001) 2007–2012.
- [8] T. Kokubo, S. Ito, M. Shigematsu, S. Sakka, T. Yamamuro, Fatigue and life time of bioactive glass–ceramic A-W containing apatite and Wollastonite, *J. Mater. Sci.* 22 (1987) 4067–4070.
- [9] P. Siriphannon, Y. Kameshima, A. Yasumori, K. Okada, S. Hayashi, Influence of preparation conditions on the microstructure and bioactivity of α -CaSiO₃ ceramics: formation of hydroxyapatite in simulated body fluid, *J. Biomed. Mater. Res.* 52 (2000) 30–39.
- [10] K.L. Lin, W.Y. Zhai, S.Y. Ni, J. Chang, Y. Zeng, W.J. Qian, Study of the mechanical property and in vitro biocompatibility of CaSiO₃ ceramics, *Ceram. Int.* 31 (2005) 323–326.
- [11] H.B. Guo, K. Aik Khorb, Y.C. Boeya, X. Miaoa, Laminated and functionally graded hydroxyapatite/yttria stabilized tetragonal zirconia composites fabricated by spark plasma sintering, *Biomaterials* 24 (2003) 667–675.
- [12] W. Li, L. Gao, Fabrication of HAp-ZrO₂ (3Y) nano-composite by SPS, *Biomaterials* 24 (2003) 937–994.
- [13] A. Nakahira, M. Tamai, H. Aritani, S. Nakamura, K. Yamashita, Biocompatibility of Dense Hydroxyapatite Prepared Using an SPS Process, Wiley Periodicals Inc., 2002 pp. 550–557.
- [14] T. Endo, S. Sugiura, M. Sakamaki, H. Takizawa, M. Shimada, Sintering and mechanical properties of β -wollastonite, *J. Mater. Sci.* 29 (1994) 1501–1506.