

Short communication

Low firing and microwave dielectric properties of BaTi_4O_9 with $\text{B}_2\text{O}_3\text{--ZnO--La}_2\text{O}_3$ glass addition

Enxiang Guan, Wei Chen, Lan Luo*

Shanghai Institute of Ceramics, Chinese Academy of Sciences, Shanghai 200050, PR China

Received 16 January 2006; received in revised form 4 February 2006; accepted 15 March 2006

Available online 25 July 2006

Abstract

A high- Q low firing ceramics material was fabricated from a composite of 80 wt% of BaTi_4O_9 powder and 20 wt% of $\text{B}_2\text{O}_3\text{--ZnO--La}_2\text{O}_3$ glass flux. The sample sintered at 900 °C for 3 h had the maximum bulk density. $\text{B}_2\text{O}_3\text{--ZnO--La}_2\text{O}_3$ glass was found to act as the sintering aid. The material sintered at 900 °C for 3 h consists of BaTi_4O_9 , LaBO_3 , an unidentified crystal phase and residual glass phase, and possesses excellent microwave dielectric properties: permittivity $k \approx 27$, quality factor $Q \times f \approx 20,000$ GHz, temperature coefficient of resonant frequency $\tau_f \approx 6.5$ ppm/°C.

© 2006 Elsevier Ltd and Techna Group S.r.l. All rights reserved.

Keywords: LTCCs; $\text{B}_2\text{O}_3\text{--ZnO--La}_2\text{O}_3$ glass; BaTi_4O_9

1. Introduction

Low-temperature co-fired ceramics (LTCCs) have been widely investigated due to their applications for the multilayer microwave components which can miniaturize the microwave devices. In the multilayer structures, the sintering temperature of the dielectric materials has to be reduced to or below 900 °C so as to co-fire with highly conductive embedded electrode such as Ag (the melting point of Ag is about 961 °C).

Generally speaking, adding low softening temperature glass is an effective, cheap approach to lower the densification temperature of ceramics [1–4]. Several researches including $\text{BaO--TiO}_2\text{--WO}_3$ with $\text{ZnO--B}_2\text{O}_3$ glass addition [5], $\text{BaTi}_4\text{O}_9/\text{Ba}_2\text{Ti}_9\text{O}_{20}$ with $\text{MgO--CaO--SiO}_2\text{--Al}_2\text{O}_3$ glass addition [6], $\text{Ba}_2\text{Ti}_9\text{O}_{20}$ with $3\text{ZnO--B}_2\text{O}_3$ glass addition [7] have been studied; however, the sintering temperature of the composites was generally higher than 900 °C which is still high for co-firing of Ag inner electrodes. $\text{BaO--Nd}_2\text{O}_3\text{--TiO}_2$ ceramic with $\text{La}_2\text{O}_3\text{--B}_2\text{O}_3\text{--TiO}_2$ glass addition has also been investigated by Jung et al. [8], although temperature of densification could be lowered below 900 °C, the fraction of glass additive was as high as 60 wt%, which certainly will influence the microwave

properties of $\text{BaNd}_2\text{Ti}_5\text{O}_{14}$ greatly. To weaken the destructive effect of the glass additives on the intrinsic microwave properties of the ceramics, researchers have to find good glass flux which could decrease the sintering temperature greatly with smaller amount addition but not influence the microwave properties of the host material too much.

$\text{B}_2\text{O}_3\text{--ZnO--La}_2\text{O}_3$ (BZL) glass is used as optical glass with high refractive index. It has not been found reported the literature as a sintering aid for LTCCs applications. In present study, we found the $\text{B}_2\text{O}_3\text{--ZnO--La}_2\text{O}_3$ glass was available as sintering aid in reducing the sintering temperature of the BaTi_4O_9 -based composites, and the sintering process and the microwave dielectric properties of the sintered composites were also investigated.

2. Experiments

$\text{B}_2\text{O}_3\text{--ZnO--La}_2\text{O}_3$ glass with the composition of B_2O_3 60 mol%, ZnO 30 mol%, La_2O_3 10 mol% was prepared using reagent grade H_3BO_3 , ZnO and La_2O_3 as the raw materials. The mixture was melted in a platinum crucible at 1100 °C for 1 h and the glass was quenched in water into fragments which were then pulverized and screened through a 80 mesh sieve to become the BZL glass powder. The BaTi_4O_9 was synthesized by conventional solid-state-reaction method using reagent grade BaCO_3 and TiO_2 as starting powders. BaCO_3 and TiO_2 ,

* Corresponding author. Tel.: +86 21 5241 6365; fax: +86 21 5241 3903.

E-mail address: LanLuo@sunm.shcnc.ac.cn (L. Luo).

with molar ratio of 1:4, were mixed in a plastic bottle with distilled water and ZrO₂ balls for 5 h. The mixture was then dried and calcined at 1180 °C for 2 h to form BaTi₄O₉ phase.

Eighty weight percent of BaTi₄O₉ and 20 wt% BZL glass powders were mixed and milled with ZrO₂ balls and deionized water for 24 h. After drying, the mixture was granulated by adding 7 wt% polyvinyl alcohol (PVA) solution. Then the pellets 15 mm in diameter and 9 mm in thickness were fabricated by uniaxial pressing at 1000 kg/cm², followed by sintering between 880 and 940 °C for 3 h in air with a heating rate of 5 °C/min. The bulk densities of the sintered pellets were measured by the Archimedes method. The crystal phases present in samples were identified by X-ray diffraction analysis (XRD, D/max 2550 V Rigaku, Japan). Polished and HF-etched surface of sintered sample was observed by scanning electron microscopy (SEM) equipped with energy-dispersive spectroscopy (EDS: model INCA, Oxford Instruments, Bucks, UK). The microwave dielectric properties of sintered samples were measured among the frequency range of 5–6 GHz by network analyzer (HP8363A, Agilent, USA), the temperature coefficient of resonant frequency (τ_f) was measured within the range from –25 °C to 85 °C, and τ_f was defined by $(f(85) - f(-25)) / (f(25) \times 110)$, where $f(85)$, $f(-25)$ and $f(25)$ are the resonant frequencies at 85, –25 and 25 °C, respectively.

3. Results and discussion

3.1. Sintering process and microstructure

The bulk density of the pellets sintered between 880 and 940 °C is shown in Fig. 1. It was found that the sample sintered at 880 °C had a bulk density of only 3.0 g/cm³, whereas the one sintered at 900 °C for 3 h had already reached the maximum value of 4.3 g/cm³, and kept almost unchanged at 920 °C. It shows that the sample could be well-densified at and above 900 °C. It was believed that the B₂O₃–ZnO–La₂O₃ glass plays a crucial role in facilitating the densification of the composites. This result also indicated that the original B₂O₃–ZnO–La₂O₃

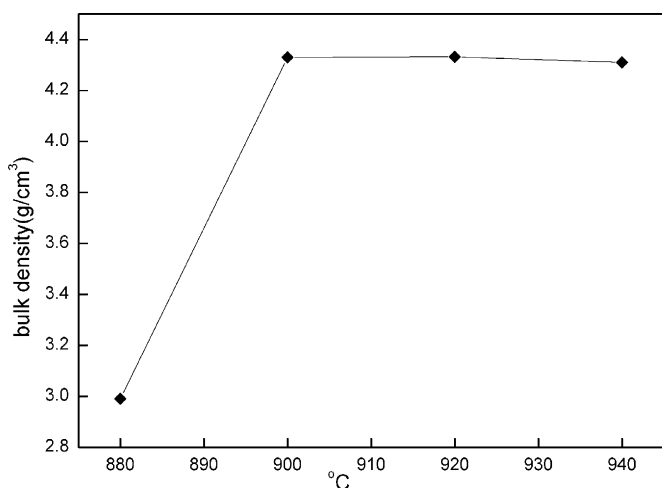


Fig. 1. Bulk density of the composites as a function of sintering temperature with dwell time for 3 h.

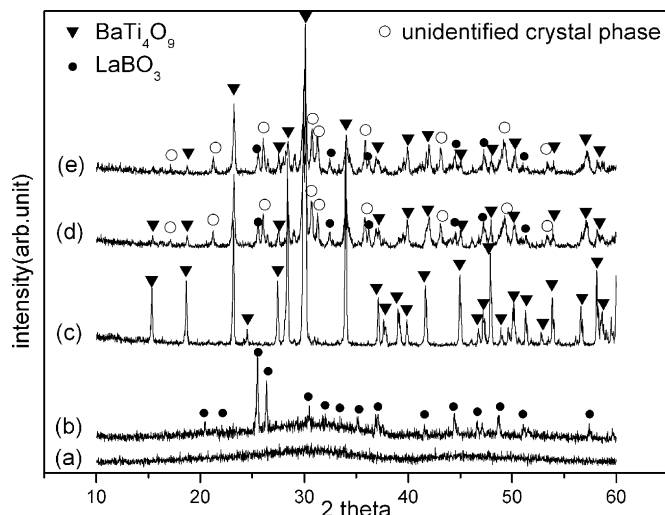


Fig. 2. X-ray diffraction patterns of (a) quenched BZL glass, (b) BZL glass heat treated at 900 °C, (c) calcined BaTi₄O₉ powder at 1180 °C, the composites sintered at (d) 900 °C and (e) 920 °C for 3 h.

glass creates a large liquid phase at about 900 °C but not at or below 880 °C.

From XRD patterns in Fig. 2(d and e), it was found that the composites sintered between 900 and 920 °C consisted of three crystalline phases: BaTi₄O₉ phase, LaBO₃ phase and an unidentified crystal phase. Fig. 2 also shows that the original glass was almost amorphous according to the dome outline of the XRD and LaBO₃ come from the crystallization of the B₂O₃–ZnO–La₂O₃ glass at about 900 °C. Because the raw materials were BaTi₄O₉ and the BZL glass, the unidentified crystal phase should be formed as the product of the interreaction between the glass flux and the BaTi₄O₉ phase. Due to the formation of the unknown phase, the sintering process obviously belonged to reactive sintering. Therefore, it was believed that dynamic mechanism of the chemical reaction was also one of the factors in facilitating the sintering process.

In the backscattered electron image (Fig. 3.) of the sample sintered at 900 °C for 3 h, three different areas were found in the picture: the grey area, the white area and the dark area. The white area and the grey area were confirmed by EDS as LaBO₃ and BaTi₄O₉, respectively, and the dark area consisted of the elements of Ba, Zn, Ti and O as shown in Table 1. From both Figs. 2 and 3, we know the BaTi₄O₉ phase was the majority phase of the composite. Combining with Fig. 2, we could deduce that the dark area corresponded to the unidentified phase in Fig. 2(d and e). Unfortunately, the unknown phase could not be identified by existing XRD data base. To identify the unknown phase, further investigations will be needed.

3.2. Microwave dielectric properties

Fig. 4 shows the permittivity (k) and quality factor ($Q \times f$) of the samples as a function of sintering temperature. As shown in Fig. 1, the bulk density of the sample increased rapidly from 880 to 900 °C, and remained unchanged from 900 to 920 °C. A little decrease in the bulk density of the sample appeared at 940 °C. Permittivity and quality factor of the samples exhibited

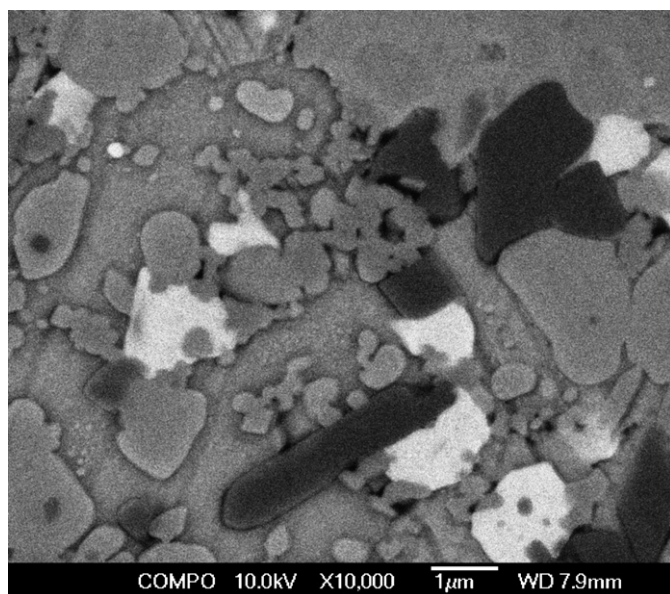


Fig. 3. Backscattered electron image of the sample sintered at 900 °C for 3 h.

similar trend of change to that of bulk density with increasing sintering temperature. At 880 °C, due to the high porosity, the composite had low permittivity and quality factor. While at 900 °C, due to high densification, the sample exhibited great enhancement of permittivity and quality factor. The decrease in the permittivity and quality factor of the sample sintered at 940 °C is probably due to a little over-sintering of the sample. Generally speaking, the $Q \times f$ value of glass is much lower than pure microwave dielectric ceramic. In the present study

Table 1

The compositional data in Fig. 3 by EDS analysis

Three different areas	The compositional elements	Crystal phase
The grey area	Ba, Ti, O	BaTi ₄ O ₉
The white area	La, B, O	LaBO ₃
The dark area	Ba, Zn, Ti and O	Unknown phase

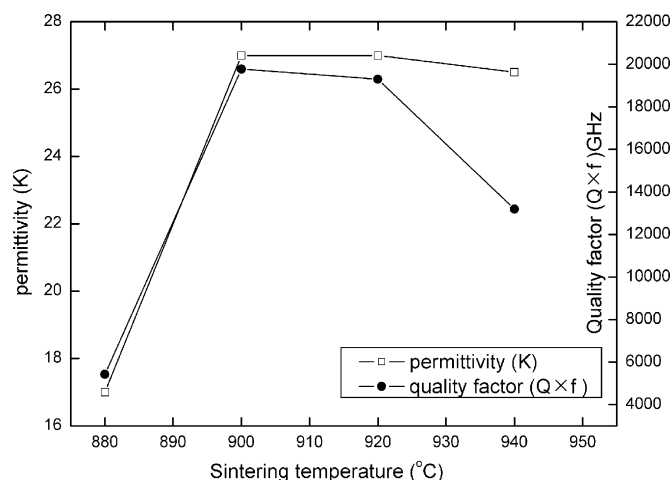


Fig. 4. Permittivity k and quality factor of the composites as a function of sintering temperature.

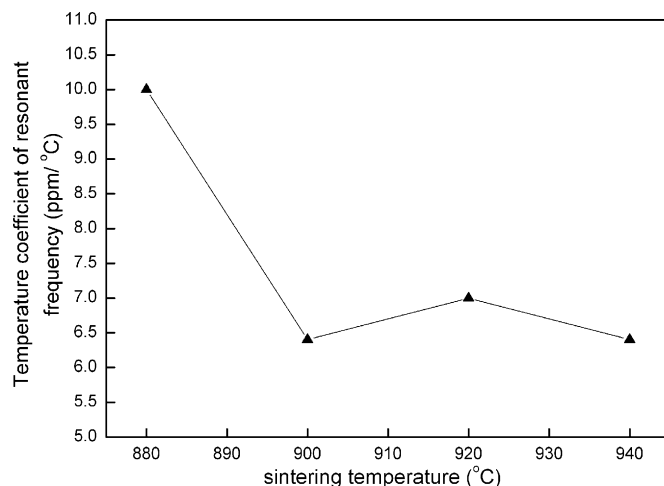


Fig. 5. The temperature coefficient of resonant frequency τ_f of the composites as a function of sintering temperature.

probably because of the smaller amount of glass addition, the sample sintered at 900 °C for 3 h exhibited a permittivity k of 27 and quality factor $Q \times f$ value of near 20,000 GHz. This is an excellent quality factor value compared with most LTCCs materials.

The temperature coefficient of resonant frequency τ_f was measured from the samples sintered at different temperatures for 3 h, and the result is shown in Fig. 5. A very good τ_f value of 6.5 ppm/°C was obtained for the sample sintered at 900 °C. The ceramics obtained in this work had a very complicated phase composition. Besides the main phase BaTi₄O₉, the samples contained the LaBO₃ phase precipitated from the BZL glass flux, the unidentified crystal phase and the residual glass phase. Although the τ_f value of LaBO₃ and BaTi₄O₉ have been reported to be 670 [9] and 16 ppm/°C [10], respectively, the τ_f value of the other two phases are not clear at present from the existing literature. Because the temperature coefficient of resonant frequency τ_f is determined by the collective effects of all phases [11], it could be only speculated that the τ_f value of the Ba–Zn–Ti–O crystal phase is negative which would offset the positive τ_f of LaBO₃ and BaTi₄O₉ to obtain the final smaller τ_f 6.5 ppm/°C of the composite sintered at 900 °C. To better understand the behavior of the τ_f of the materials, further investigation on the nature of individual crystal phase and the residual glass will be necessary.

4. Conclusion

A high- Q low firing ceramics material was fabricated from a composite of 80 wt% of BaTi₄O₉ and 20 wt% of B₂O₃–ZnO–La₂O₃ glass flux. The B₂O₃–ZnO–La₂O₃ glass could be promising sintering aid in LTCCs.

The composite could be well-densified at 900 °C for 3 h; the B₂O₃–ZnO–La₂O₃ glass played crucial role in low-temperature sintering. The ceramics obtained in this work had a very complicated phase composition. Besides the main phase BaTi₄O₉, three other phases existed. They are the LaBO₃ phase which precipitated from the BZL glass flux, the unidentified

crystal phase which was formed as the product of interreaction between glass flux and BaTi_4O_9 , and the residual glass phase.

The composite sintered at 900 °C for 3 h obtained the maximum bulk density and possessed excellent microwave dielectric properties: permittivity k of 27, quality factor $Q \times f$ of near 20,000 GHz and temperature coefficient of resonant frequency τ_f of 6.5 ppm/°C.

References

- [1] T. Takada, S.F. Wang, S. Yoshikawa, S.J. Jang, R.E. Newnham, Effects of glass additions on $(\text{Zr}, \text{Sn})\text{TiO}_4$ for microwave applications, *J. Am. Ceram. Soc.* 77 (9) (1994) 2485.
- [2] C.M. Cheng, C.F. Yang, S.H. Lo, Sintering and dielectric properties of $\text{Ba}_2\text{Ti}_9\text{O}_{20}$ microwave ceramics by glass additions, *Jpn. J. Appl. Phys.* 36 (1997) L1604–L1607.
- [3] C.C. Lee, P. Lin, Effects of glass additions on microwave properties of $\text{BaO-La}_2\text{O}_3-4.7\text{TiO}_2$, *Jpn. J. Appl. Phys.* 37 (1998) 6048–6054.
- [4] I.M. Zajc, Drofenik, semiconducting BaTiO_3 ceramics prepared by low temperature liquid phase sintering, *J. Mater. Res.* 13 (3) (1998) 660–664.
- [5] T. Takada, S.F. Wang, S. Yoshikawa, S.-J. Jang, R.E. Newnham, Effect of glass additions on $\text{BaO-TiO}_2\text{-WO}_3$ microwave ceramics, *J. Am. Ceram. Soc.* 77 (7) (1994) 1909–1916.
- [6] Chien-min Cheng, Sintering $\text{BaTi}_4\text{O}_9/\text{Ba}_2\text{Ti}_9\text{O}_{20}$ -based ceramics by glass additions, *J. Eur. Ceram. Soc.* 20 (2000) 1061–1067.
- [7] Y.-C. Leea, W.-H. Leeb, F.-S. Shieuc, Microwave dielectric properties and microstructures of $\text{Ba}_2\text{Ti}_9\text{O}_{20}$ -based ceramics with $3\text{ZnO-B}_2\text{O}_3$ addition, *Eur. Ceram. Soc. Vol 25* (2005) 3459–3468.
- [8] B.-H. Jung, S.-J. Hwang, H.-S. Kim, Glass-ceramic for low temperature co-fired dielectric ceramic materials based on $\text{La}_2\text{O}_3\text{-B}_2\text{O}_3\text{-TiO}_2$ glass with BNT ceramics, *J. Eur. Ceram. Soc.* 25 (2005) 3187–3193.
- [9] O. Dernovsek, A. Naeini, G. Preu, W. Wersing, M. Eberstein, W.A. Schiller, LTCC glass-ceramic composites for microwave application, *J. Eur. Ceram. Soc.* 21 (2001) 1693–1697.
- [10] J.H. Choy, Y.S. Han, Microwave characteristics of BaO-TiO_2 ceramics prepared via a citrate route, *J. Am. Ceram. Soc.* 78 (5) (1995) 1169–1172.
- [11] A.E. PALADINO, Temperature-compensated $\text{MgTi}_2\text{O}_5\text{-TiO}_2$ dielectrics, *J. Am. Ceram. Soc.* 54 (1971) 168.