

# Low-temperature sintering of $0.85\text{CaWO}_4\text{--}0.15\text{LaNbO}_4$ ceramics

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## Abstract

Microwave dielectric properties of  $0.85\text{CaWO}_4\text{--}0.15\text{LaNbO}_4$  (CWLN) ceramics were investigated as a function of  $\text{H}_3\text{BO}_3$ ,  $\text{Li}_2\text{CO}_3$  content and sintering temperature. With the co-addition of 3.0 wt.%  $\text{H}_3\text{BO}_3\text{--}1.0$  wt.%  $\text{Li}_2\text{CO}_3$ , the sintering temperature could be effectively reduced from 1150 °C for pure CWLN ceramics to 900 °C without any degradation of dielectric properties. These results are due to the enhancement of the sinterability of CWLN by liquid phase sintering. For the specimens with  $\text{H}_3\text{BO}_3\text{--Li}_2\text{CO}_3$  sintered at 900 °C for 3 h, the dielectric constant ( $K$ ) did not changed remarkably. However, the quality factor ( $Qf$ ) and the temperature coefficient of resonant frequency ( $TCF$ ) increased up to  $y = 1.0$  of 3.0 wt.%  $\text{H}_3\text{BO}_3\text{--}y$  wt.%  $\text{Li}_2\text{CO}_3$ , and then decreased due to the formation of the secondary phases. Typically,  $K$  of 11.8,  $Qf$  of 45,200 GHz and  $TCF$  of  $-23.1$  ppm/°C were obtained for the specimens of CWLN with 3.0 wt.%  $\text{H}_3\text{BO}_3\text{--}1.0$  wt.%  $\text{Li}_2\text{CO}_3$  sintered at 900 °C for 3 h.  
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## 1. Introduction

The rapidly growing wireless industry needs a new high-performance material to build low loss and thermally stable integrated packages such as filters, duplexers, voltage-controlled oscillators and antennas [1,2]. Multilayer microwave components have been investigated to miniaturize resonant devices for volume efficiency. In multilayer structures, the ceramics with low sintering temperature are needed to co-fire with low-cost and low-melting point electrodes, such as silver with a melting point of 961 °C. However, most of the ceramic materials with good dielectric properties have a sintering temperature above 1150 °C. Therefore, the addition of sintering aids such as low-melting point glasses and/or oxides are required to reduce the sintering temperature, due to the most effective, simple and inexpensive method.

Several kinds of low-melting point sintering aids and a various synthetic glasses have been investigated to reduce the sintering temperature of microwave dielectric materials effectively [3–7]. Typically, Boron oxide and lithium oxide are the well-known liquid phase sintering promoters, and multi-component glasses are more effective than single component

glass to reduce the sintering temperature of ceramics with good microwave dielectric properties [8].

In our preliminary experiment,  $0.85\text{CaWO}_4\text{--}0.15\text{LaNbO}_4$  (CWLN) ceramics sintered at 1150 °C for 3 h showed a good microwave dielectric properties;  $K = 11.7$ ,  $Qf = 61,600$  GHz and  $TCF = -15.3$  ppm/°C. Although the microwave dielectric properties of the CWLN ceramics are good, it is not easy to apply this ceramics for the practical devices since the sintering temperature is relatively high.

In this work,  $\text{H}_3\text{BO}_3$  and  $\text{Li}_2\text{CO}_3$  were selected to search the new sintering additives suitable to CWLN ceramics. Dependence of the sinterability on the relative amount of  $\text{H}_3\text{BO}_3$  to  $\text{Li}_2\text{CO}_3$  was discussed. Physical and microwave dielectric properties of CWLN ceramics were investigated as a function of the sintering temperature and  $\text{H}_3\text{BO}_3$  and/or  $\text{Li}_2\text{CO}_3$  content.

## 2. Experimental procedure

High-purity ( $\geq 99.9\%$ ) oxide powders of  $\text{CaCO}_3$ ,  $\text{La}_2\text{O}_3$ ,  $\text{WO}_3$ , and  $\text{Nb}_2\text{O}_5$  were used as starting powders. The powders were separately prepared according to the desired stoichiometric  $\text{CaWO}_4$  and  $\text{LaNbO}_4$  and ground with  $\text{ZrO}_2$  balls for 24 h in ethanol. Prepared powders of  $\text{CaWO}_4$  and  $\text{LaNbO}_4$  were dried and calcined at 700 °C, and 1100 °C for 3 h, respectively. The calcined powders were mixed according to the composition of  $0.85\text{CaWO}_4\text{--}0.15\text{LaNbO}_4$  (CWLN) and calcined at 1100 °C

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for 3 h. The calcined CWLN powders were re-milled for 24 h with the addition of  $\text{H}_3\text{BO}_3\text{--Li}_2\text{CO}_3$ , and pressed into 15 mm diameter disks under the pressure of  $1500 \text{ kg/cm}^2$ , isostatically. These disks were sintered from  $850^\circ\text{C}$  to  $950^\circ\text{C}$  for 3 h in the air.

Crystalline phases of the specimens were identified by the powder X-ray diffraction analyses (D/Max-3C, Rigaku, Japan). Microstructure of the sintered specimens was observed by a scanning electron microscope (JEOL, JSM 820, Japan). The sintered density of the specimens was measured by the Archimedes method. Dielectric constant ( $K$ ) and unloaded  $Q$  value at 9–11 GHz were measured by the post-resonant method [9]. The temperature coefficient of the resonant frequency ( $TCF$ ) was measured by the cavity method [10] in the temperature range from  $20^\circ\text{C}$  to  $80^\circ\text{C}$ .

### 3. Results and discussion

Powder X-ray diffraction patterns of  $0.85\text{CaWO}_4\text{--}0.15\text{LaNbO}_4$  (CWLN) with  $\text{H}_3\text{BO}_3\text{--Li}_2\text{CO}_3$  specimens sintered at  $900^\circ\text{C}$  for 3 h are shown in Fig. 1. For the co-addition of  $\text{H}_3\text{BO}_3$  and  $\text{Li}_2\text{CO}_3$ , a single phase with scheelite structure was obtained at a given composition ranges, however, the secondary phases such as  $\text{LaNbO}_4$ ,  $\text{LaNb}_5\text{O}_{14}$  and  $\text{La}_{14}\text{W}_8\text{O}_{45}$  were detected for the specimens with 3.0 wt.%  $\text{H}_3\text{BO}_3\text{--}1.5$  and/or 2.0 wt.%  $\text{Li}_2\text{CO}_3$ . Also, there were no remarkable changes in XRD patterns with  $\text{H}_3\text{BO}_3\text{--Li}_2\text{CO}_3$  content.

Fig. 2 shows the apparent densities of CWLN with  $\text{H}_3\text{BO}_3\text{--Li}_2\text{CO}_3$  as a function of sintering temperature, and  $\text{H}_3\text{BO}_3$  content, respectively. The apparent density of the pure CWLN specimens was remarkably increased with the sintering temperature, and shows a maximum value at  $1150^\circ\text{C}$ . As shown in Fig. 2(a), the apparent density of the specimens with 3.0 wt.%  $\text{H}_3\text{BO}_3\text{--}y$  wt.%  $\text{Li}_2\text{CO}_3$  was increased with increasing sintering temperature. These results are due to the increase of liquid phase ( $849^\circ\text{C}$ :  $\text{B}_2\text{O}_3\text{--Li}_2\text{O}$ ) by the co-addition of

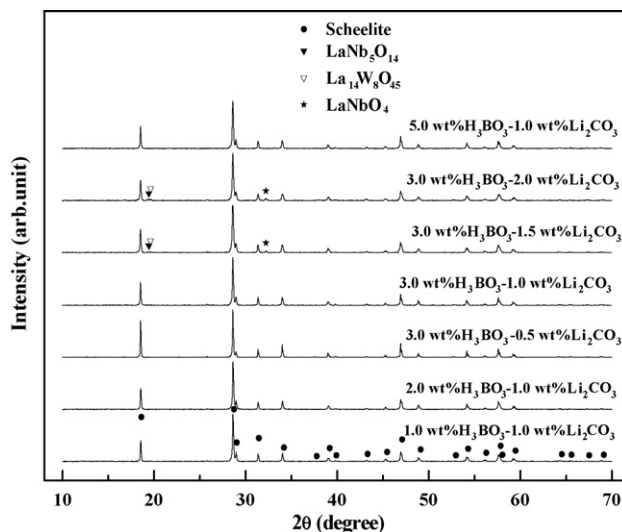


Fig. 1. X-ray diffraction patterns of  $0.85\text{CaWO}_4\text{--}0.15\text{LaNbO}_4$  with  $\text{H}_3\text{BO}_3\text{--Li}_2\text{CO}_3$  content sintered at  $900^\circ\text{C}$  for 3 h.

$\text{H}_3\text{BO}_3\text{--Li}_2\text{CO}_3$ . On the other hand, the apparent density of the specimens was decreased with  $\text{Li}_2\text{CO}_3$  content due to the increase of excess liquid phase. For CWLN with 3.0 wt.%  $\text{H}_3\text{BO}_3\text{--}0.5$  wt.%  $\text{Li}_2\text{CO}_3$ , the apparent density of the specimens shows a maximum value at  $900^\circ\text{C}$ , and the same value of density was obtained for the pure CWLN specimens sintered at  $1150^\circ\text{C}$ . Hence, the sintering temperature of CWLN could be reduced from  $1150^\circ\text{C}$  to  $900^\circ\text{C}$  by the co-addition of  $\text{H}_3\text{BO}_3\text{--Li}_2\text{CO}_3$ . Generally, the effectiveness of sintering aids depends on the several factors, such as sintering temperature, viscosity, solubility and glass wettability [11]. Viscosity, solubility and glass wettability of  $\text{H}_3\text{BO}_3\text{--Li}_2\text{CO}_3$  in this work might be probably changed with sintering temperature and  $\text{H}_3\text{BO}_3\text{--Li}_2\text{CO}_3$  content. For the specimens with  $x$  wt.%  $\text{H}_3\text{BO}_3\text{--}1.0$  wt.%  $\text{Li}_2\text{CO}_3$  sintered at  $900^\circ\text{C}$  (Fig. 2(b)), the density was

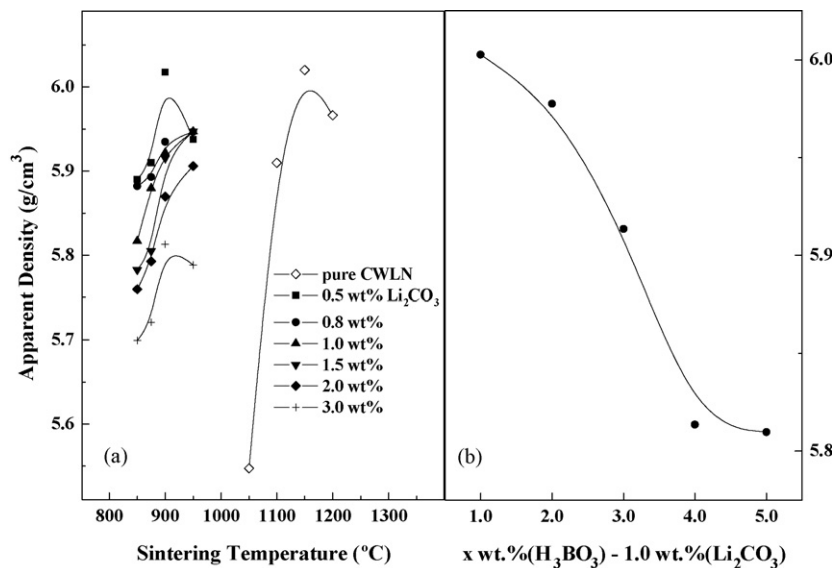


Fig. 2. Apparent density of  $0.85\text{CaWO}_4\text{--}0.15\text{LaNbO}_4$  specimens with (a) 3.0 wt.%  $\text{H}_3\text{BO}_3\text{--}y$  wt.%  $\text{Li}_2\text{CO}_3$  sintered at various temperature for 3 h (b)  $x$  wt.%  $\text{H}_3\text{BO}_3\text{--}1.0$  wt.%  $\text{Li}_2\text{CO}_3$  sintered at  $900^\circ\text{C}$  for 3 h.

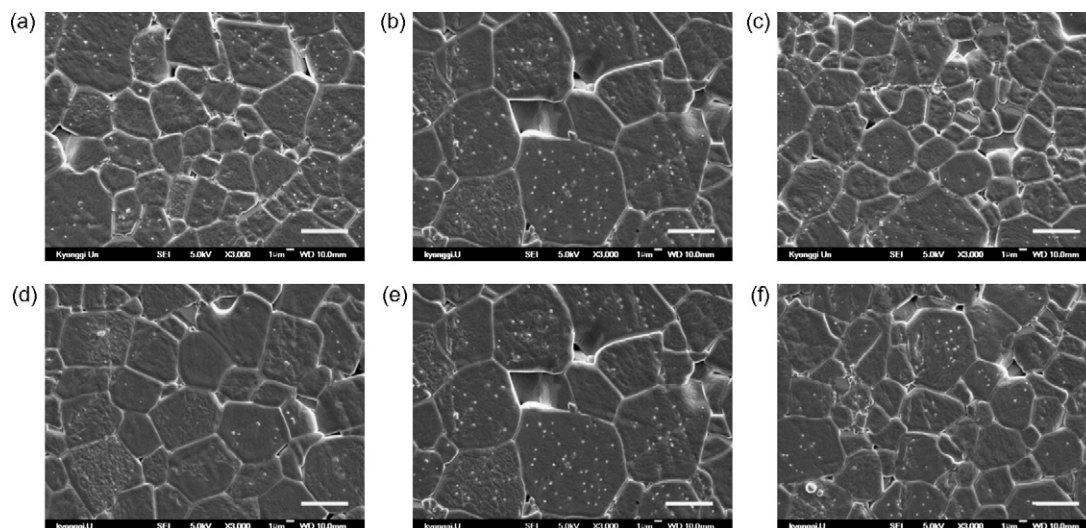


Fig. 3. SEM photographs of  $0.85\text{CaWO}_4\text{--}0.15\text{LaNbO}_4$  with  $x$  wt.%  $\text{H}_3\text{BO}_3\text{--}y$  wt.%  $\text{Li}_2\text{CO}_3$  specimens at  $900^\circ\text{C}$  for 3 h; (a)  $x = 1.0$  wt.%,  $y = 1.0$  wt.%; (b)  $x = 3.0$  wt.%,  $y = 1.0$  wt.%; (c)  $x = 5.0$  wt.%,  $y = 1.0$  wt.%; (d)  $x = 3.0$  wt.%,  $y = 0.5$  wt.%; (e)  $x = 3.0$  wt.%,  $y = 1.0$  wt.%; (f)  $x = 3.0$  wt.%,  $y = 1.5$  wt.% (bar =  $5\ \mu\text{m}$ ).

decreased with  $\text{H}_3\text{BO}_3$  content, which was similar tendency to the behavior of density of the specimens with  $\text{Li}_2\text{CO}_3$  content. Therefore, the apparent density of CWLN specimens sintered at low temperature was depended on the amount of liquid phase.

SEM photographs of CWLN with  $\text{H}_3\text{BO}_3\text{--Li}_2\text{CO}_3$  ceramics sintered at  $900^\circ\text{C}$  for 3 h are shown in Fig. 3. For the specimens with  $x$  wt.%  $\text{H}_3\text{BO}_3\text{--}1.0$  wt.%  $\text{Li}_2\text{CO}_3$ , the grain size of the specimens was increased up to  $x = 3.0$ , and then decreased for the further addition of  $\text{H}_3\text{BO}_3$ . For the specimens with  $3.0$  wt.%  $\text{H}_3\text{BO}_3\text{--}y$  wt.%  $\text{Li}_2\text{CO}_3$ , the grain size of the specimens was increased up to  $y = 1.0$ , and then decreased for the further addition of  $\text{Li}_2\text{CO}_3$ . For the specimens with  $3.0$  wt.%  $\text{H}_3\text{BO}_3\text{--}1.5$  and  $2.0$  wt.%  $\text{Li}_2\text{CO}_3$ , the secondary phases could not be observed in the SEM micrograph, even though the secondary phases were detected in the XRD patterns of Fig. 1.

Figs. 4 and 5 show the microwave dielectric properties of CWLN with  $x$  wt.%  $\text{H}_3\text{BO}_3\text{--}1.0$  wt.%  $\text{Li}_2\text{CO}_3$  and  $3.0$  wt.%  $\text{H}_3\text{BO}_3\text{--}y$  wt.%  $\text{Li}_2\text{CO}_3$  ceramics sintered at  $900^\circ\text{C}$  for 3 h, respectively. The dielectric constant ( $K$ ) of the specimens was

not changed remarkably with  $\text{H}_3\text{BO}_3$ ,  $\text{Li}_2\text{CO}_3$  content. It has been reported [12] that  $Qf$  value is affected by secondary phase, density, impurities and grain size. For the specimens with  $\text{H}_3\text{BO}_3\text{--Li}_2\text{CO}_3$  sintered at  $900^\circ\text{C}$ , the effect of density on  $Qf$  value could be neglected because the relative density was higher than 96%. With an increase of  $\text{H}_3\text{BO}_3$ ,  $\text{Li}_2\text{CO}_3$  content,  $Qf$  value of the specimens was increased up to  $3.0$  wt.%  $\text{H}_3\text{BO}_3\text{--}1.0$  wt.%  $\text{Li}_2\text{CO}_3$ , and then decreased due to the decrease of grain size, as shown in Fig. 3.

Fig. 6 shows the temperature coefficient of resonant frequency ( $TCF$ ) of CWLN with  $3.0$  wt.%  $\text{H}_3\text{BO}_3\text{--}y$  wt.%  $\text{Li}_2\text{CO}_3$  ceramics sintered at  $900^\circ\text{C}$  for 3 h.  $TCF$  was largely depended on the temperature coefficient of dielectric constant ( $TCK$ ), which in turn depended on the dielectric constant if there were no secondary phases without the change of crystal structure. For the specimens with  $3.0$  wt.%  $\text{H}_3\text{BO}_3\text{--}y$  wt.%  $\text{Li}_2\text{CO}_3$ ,  $TCF$  of the specimens was increased up to  $y = 1.0$ , and then decreased due to the formation of the secondary phases such as  $\text{LaNb}_5\text{O}_{14}$ ,  $\text{La}_{14}\text{W}_8\text{O}_{45}$  and  $\text{LaNbO}_4$ , as confirmed in

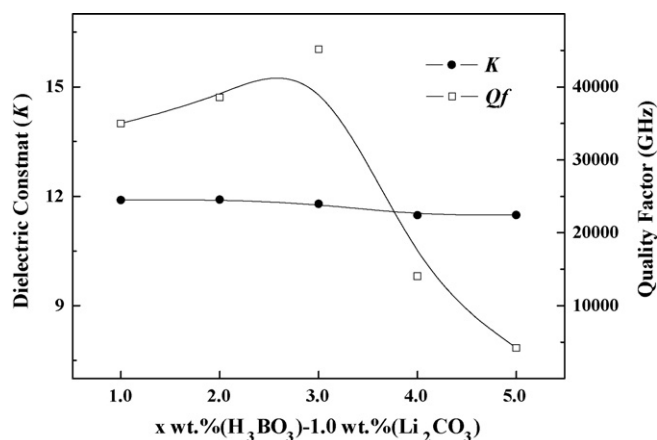


Fig. 4. Microwave dielectric properties of  $0.85\text{CaWO}_4\text{--}0.15\text{LaNbO}_4$  with  $x$  wt.%  $\text{H}_3\text{BO}_3\text{--}1.0$  wt.%  $\text{Li}_2\text{CO}_3$  specimens sintered at  $900^\circ\text{C}$  for 3 h.

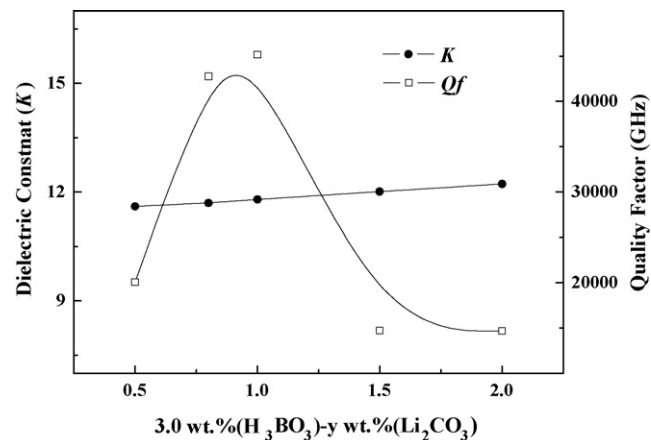


Fig. 5. Microwave dielectric properties of  $0.85\text{CaWO}_4\text{--}0.15\text{LaNbO}_4$  with  $3.0$  wt.%  $\text{H}_3\text{BO}_3\text{--}y$  wt.%  $\text{Li}_2\text{CO}_3$  specimens sintered at  $900^\circ\text{C}$  for 3 h.

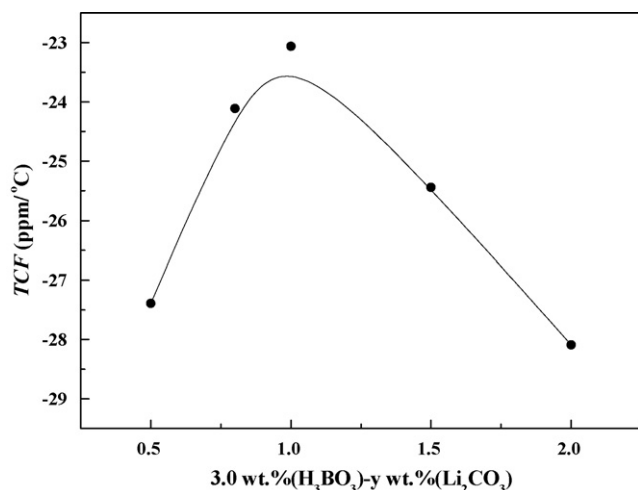


Fig. 6. TCF of  $0.85\text{CaWO}_4\text{--}0.15\text{LaNbO}_4$  with 3.0 wt.%  $\text{H}_3\text{BO}_3\text{--}y$  wt.%  $\text{Li}_2\text{CO}_3$  specimens sintered at  $900^\circ\text{C}$  for 3 h.

Fig. 1. Good microwave dielectric properties of  $K = 11.8$ ,  $Qf = 45,200$  GHz and  $TCF = -23.1$  ppm/ $^\circ\text{C}$  were obtained for the specimens with 3.0 wt.%  $\text{H}_3\text{BO}_3\text{--}1.0$  wt.%  $\text{Li}_2\text{CO}_3$  sintered at  $900^\circ\text{C}$ , which is  $250^\circ\text{C}$  lower than that of pure CWLN. These results are caused by the effect of  $\text{H}_3\text{BO}_3\text{--Li}_2\text{CO}_3$  on the sinterability.

#### 4. Conclusions

Effects of  $\text{H}_3\text{BO}_3\text{--Li}_2\text{CO}_3$  on the sinterability and microwave dielectric properties of  $0.85\text{CaWO}_4\text{--}0.15\text{LaNbO}_4$  (CWLN) ceramics were investigated. For the specimens sintered at  $900^\circ\text{C}$  for 3 h, the dielectric constant ( $K$ ) did not changed remarkably by the co-addition of  $\text{H}_3\text{BO}_3\text{--Li}_2\text{CO}_3$ . For the specimens with 3.0 wt.%  $\text{H}_3\text{BO}_3\text{--}1.0$  wt.%  $\text{Li}_2\text{CO}_3$ , the quality factor ( $Qf$ ) showed the maximum value due to the largest grain size, and the sintering temperature of CWLN ceramics was effectively reduced from  $1150^\circ\text{C}$  to  $900^\circ\text{C}$  without degradation of dielectric properties. These results are due to the enhancement of the sinterability of CWLN by liquid phase sintering. The temperature coefficient of resonant frequency (TCF) of the specimens increased up to  $y = 1.0$  of

3.0 wt.%  $\text{H}_3\text{BO}_3\text{--}y$  wt.%  $\text{Li}_2\text{CO}_3$ , and then decreased due to the secondary phases such as  $\text{LaNbO}_4$ ,  $\text{LaNb}_5\text{O}_{14}$  and  $\text{La}_{14}\text{W}_8\text{O}_{45}$ .

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