

# Depolarization temperature and piezoelectric properties of $(\text{Bi}_{1/2}\text{Na}_{1/2})\text{TiO}_3$ – $(\text{Bi}_{1/2}\text{Li}_{1/2})\text{TiO}_3$ – $(\text{Bi}_{1/2}\text{K}_{1/2})\text{TiO}_3$ lead-free piezoelectric ceramics

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## Abstract

The phase transition temperature and piezoelectric properties of  $x(\text{Bi}_{1/2}\text{Na}_{1/2})\text{TiO}_3$ – $y(\text{Bi}_{1/2}\text{Li}_{1/2})\text{TiO}_3$ – $z(\text{Bi}_{1/2}\text{K}_{1/2})\text{TiO}_3$  [ $x + y + z = 1$ ] (abbreviated as BNLKT100 $y$ –100 $z$ ) ceramics were investigated. BNLKT100 $y$ –100 $z$  ceramics were prepared by conventional ceramic fabrication. The depolarization temperature  $T_d$  was determined by the temperature dependence of the dielectric and piezoelectric properties. This study focuses on the effect of  $\text{Li}^{1+}$  and  $\text{K}^{1+}$  ions on  $T_d$  and the piezoelectric properties of BNT ceramics. BNLKT100 $y$ –100 $z$  ( $y = 0$ –0.08) has a morphotropic phase boundary (MPB) between rhombohedral and tetragonal phases at  $z = 0.18$ –0.20, and high piezoelectric properties were obtained at the MPB composition. The piezoelectric constant  $d_{33}$  increased with increasing  $y$ ; however,  $T_d$  decreased above  $y = 0.06$ . The  $d_{33}$  and  $T_d$  values of BNLKT4-20 and BNLKT8-20 were 176 pC/N and 171 °C, and 190 pC/N and 115 °C, respectively.

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**Keywords:** C. Piezoelectric properties; Morphotropic phase boundary; Depolarization temperature; Lead-free piezoelectric ceramics

## 1. Introduction

Lead-free piezoelectric ceramics for replacing  $\text{Pb}(\text{Zr,Ti})\text{O}_3$  (PZT)-based ceramics have recently become required from the viewpoint of environmental protection. Therefore, recently various lead-free piezoelectric solid solutions with a perovskite structure, such as  $\text{KNbO}_3$  [KN]-,  $(\text{Bi}_{1/2}\text{K}_{1/2})\text{TiO}_3$  [BKT]-,  $\text{BaTiO}_3$  [BT]-, and  $(\text{Bi}_{1/2}\text{Na}_{1/2})\text{TiO}_3$  [BNT]-based ceramics have been actively studied [1–12].

BNT has a perovskite structure with rhombohedral symmetry at room temperature. The phase transition temperature of the rhombohedral–tetragonal phase  $T_{R-T}$  and the Curie temperature of the tetragonal–cubic phase  $T_C$  are approximately 300 and 540 °C on heating, respectively, for a BNT single crystal [13]. Moreover, BNT-based ceramics shows relatively high piezoelectric properties; therefore, they are highly promising lead-free piezoelectric materials [8–12]. However, a significant problem of BNT ceramics is their low

depolarization temperature  $T_d$  of 185 °C for practical use as a piezoelectric actuator. In our previous paper, the  $T_d$  had been investigated accurately by measuring the temperature dependence of the piezoelectric properties of such ceramics [10]. In addition, how to determine the phase transition temperatures in detail for BNT–BKT–BT ternary systems had been demonstrated in our previous work [12] and the behavior of  $T_d$  and  $T_{R-T}$  for BNT-based solid solutions had been revealed [12,14,15].

Recently, an excellent piezoelectric constants  $d_{33}$  ( $d_{33}$  meter value) of 231 pC/N was obtained at the morphotropic phase boundary (MPB) composition of  $(\text{Bi}_{1/2}\text{Na}_{1/2})\text{TiO}_3$ – $(\text{Bi}_{1/2}\text{Li}_{1/2})\text{TiO}_3$ – $(\text{Bi}_{1/2}\text{K}_{1/2})\text{TiO}_3$  ternary systems [11], however, the relationship between  $T_d$  and  $d_{33}$  did not clarify. Therefore, the purpose of the present study is to clarify the relationship between  $T_d$  and the piezoelectric properties of  $(\text{Bi}_{1/2}\text{Na}_{1/2})\text{TiO}_3$ – $(\text{Bi}_{1/2}\text{Li}_{1/2})\text{TiO}_3$ – $(\text{Bi}_{1/2}\text{K}_{1/2})\text{TiO}_3$  ternary systems.

## 2. Experimental procedure

$x(\text{Bi}_{1/2}\text{Na}_{1/2})\text{TiO}_3$ – $y(\text{Bi}_{1/2}\text{Li}_{1/2})\text{TiO}_3$ – $z(\text{Bi}_{1/2}\text{K}_{1/2})\text{TiO}_3$  [ $x + y + z = 1$ ] (abbreviated as BNLKT100 $y$ –100 $z$ ) ceramics

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were prepared by conventional ceramic fabrication process. Raw materials with purities higher than 99.9% were used to prepare the ceramics. Sintered ceramics were then cut and polished to measure the electrical properties. The crystal structures and lattice constants of the sintered ceramics were determined by X-ray powder diffraction analysis using an X-ray diffractometer (Rigaku; RINT2000). The temperature dependences of the dielectric properties were measured using an automated dielectric measurement system with a multifrequency LCR meter (YHP 4275A and WK6440B). The piezoelectric properties were measured by a resonance–antiresonance method using an impedance analyzer (HP 4294A).

Rectangular shapes of  $2\text{ mm} \times 2\text{ mm} \times 5\text{ mm}$  were prepared for piezoelectric measurements. These samples were poled in stirred silicone oil by applying DC electrical fields of 4–6 kV/cm at RT. The electromechanical coupling factor of the longitudinal extensional mode  $k_{33}$  was calculated from the series and parallel resonance frequencies  $f_s$  and  $f_p$ . The piezoelectric constant  $d_{33}$  was calculated using the following formula of

$$d_{33} = k_{33} \sqrt{\epsilon_{33}^T \cdot s_{33}^E}, \quad (1)$$

where  $\epsilon_{33}^T$  is the free permittivity, which is measured after poling at 1 kHz, and  $s_{33}^E$  is the elastic constant.

### 3. Results and discussion

The densities of the ceramics were measured by the Archimedes method. The ratios of the measured densities to the theoretical densities of the sintered ceramics were all higher than 96%. The phases of BNLKT100y–100z were characterized using X-ray diffraction patterns. A single-phase perovskite structure was obtained for BNLKT0–100z. Although BNLKT100y–0 shows a single-phase perovskite structure below  $y = 0.24$ , some impurity phases were observed for BNLKT28–0. This means that the solid solubility limit of  $\text{Li}^+$  ( $y$ ) is 0.24 for BNLKT100y–0 because of the small ionic radius. The depolarization temperature  $T_d$  and the piezoelectric constant  $d_{33}$  of BNLKT0–0 (BNT) were 185 °C and 69.8 pC/N, respectively.  $d_{33}$  increased with increasing  $y$  for BNLKT100y–0, and the highest  $d_{33}$  was obtained for BNLKT16–0. On the other hand,  $T_d$  of  $y < 0.06$  was higher than that of BNT for BNLKT100y–0, and  $T_d$

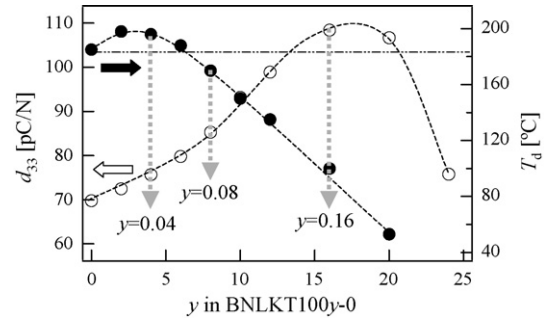


Fig. 1. Depolarization temperature  $T_d$  and piezoelectric constant  $d_{33}$  of BNLKT100y–0.

decreased with increasing  $y$  above 0.08. It is very important to increase  $d_{33}$  without reducing  $T_d$ , and the  $T_d$  and  $d_{33}$  values of BNLKT4–0 and BNLKT8–0 were 196 °C and 75.7 pC/N, and 170 °C and 85.3 pC/N, respectively. Therefore, in this study,  $T_d$  and the piezoelectric properties of BNLKT4–100z and BNLKT8–100z were investigated in detail.

The lattice constants  $a$  and  $c$ , and rhombohedrality  $90-\alpha$  and tetragonality  $c/a$  of BNLKT4–100z are shown in Fig. 2(a) and (b). Although BNLKT4–100z showed rhombohedral symmetry for  $z = 0-0.18$ , BNLKT4–20 showed tetragonal symmetry. This means the MPB between the rhombohedral and tetragonal phases is  $z = 0.18-0.20$  for BNLKT4–100z.

The temperature dependences of the coupling factor  $k_{33}$  and dielectric loss tangent  $\tan \delta$  after poling for BNLKT4–100z ( $z = 0.08, 0.12, 0.16, 0.20, 0.24$ , and  $0.28$ ) are shown in Fig. 3. It can be seen that  $T_d$  for the peak of  $\tan \delta$  is in good agreement with the values determined from  $k_{33}$  for BNLKT4–100z ( $z = 0.08, 0.12, 0.16, 0.20$ , and  $0.24$ ). However,  $T_d$  determined from  $\tan \delta$  was 10 °C higher than that determined from  $k_{33}$  for BNLKT4–28 because of the internal stress caused by the poling treatment.

Fig. 4 shows the variation in  $T_d$  as a function of  $z$  for BNLKT0–100z, BNLKT4–100z, and BNLKT8–100z. It is realized that the  $T_d$  of BNLKT4–100z is higher than that of BNLKT0–100y. On the other hand, the  $T_d$  of BNLKT8–100z is smaller than that of BNLKT0–100z. These results indicate that a small amount of  $\text{Li}^{1+}$  can increase  $T_d$ . Moreover, the variation in  $T_d$  as a function of  $z$  in BNLKT4–100z is similar to those in  $90-\alpha$  and  $c/a$ , as shown in Fig. 2(b). Therefore, it is considered that  $T_d$  is probably related to the magnitude of  $90-\alpha$  and  $c/a$ .

The piezoelectric properties were measured by the resonance–antiresonance method. The electromechanical cou-

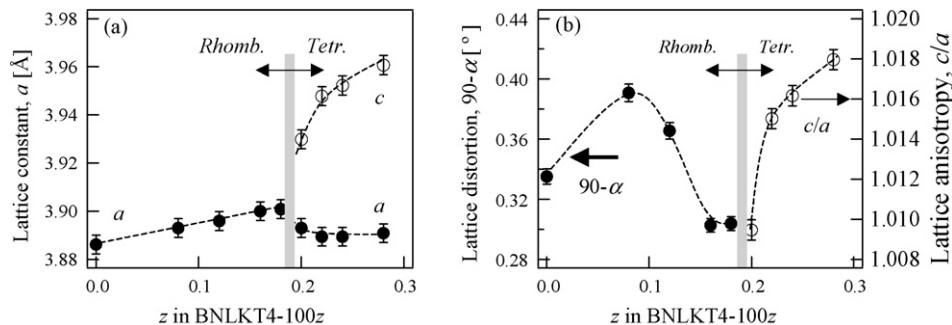


Fig. 2. Variations in (a) lattice constants  $a$  and  $c$ , and (b) rhombohedrality  $90-\alpha$  and tetragonality  $c/a$  as functions of  $z$  for BNLKT4–100z ceramics.

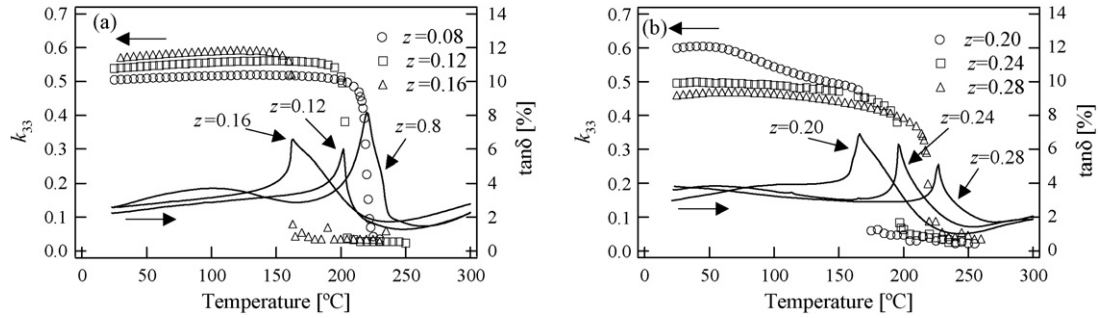


Fig. 3. Temperature dependences of coupling factor  $k_{33}$  and dielectric loss tangent after poling for BNLKT4-100 $z$ ; (a)  $z = 0.08, 0.12$ , and  $0.16$  and (b)  $z = 0.20, 0.24$ , and  $0.28$ .

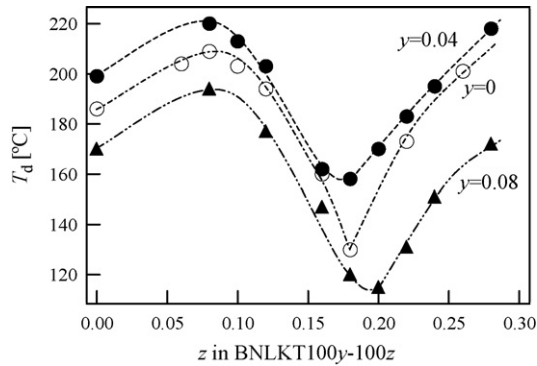


Fig. 4. Depolarization temperatures  $T_d$  of BNLKT0-100 $z$ , BNLKT4-100 $z$ , and BNLKT8-100 $z$  ceramics.

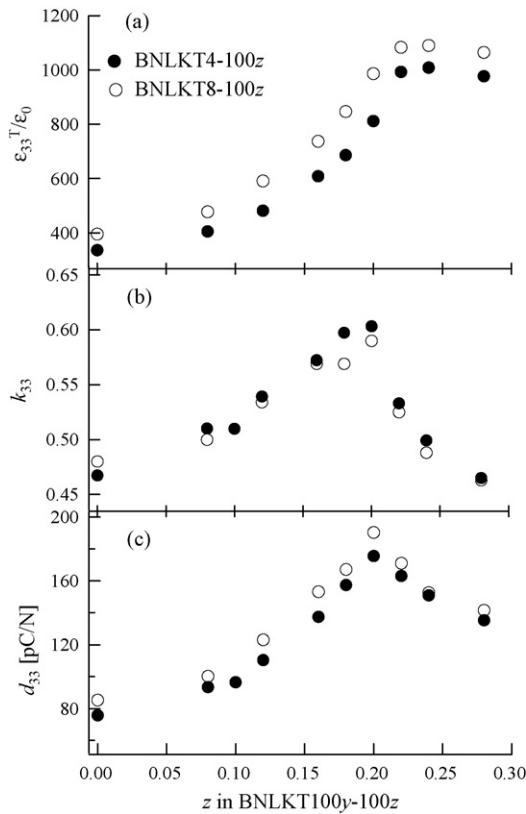


Fig. 5. (a) Free permittivities  $\epsilon_{33}^T/\epsilon_0$ , (b) electromechanical coupling factors  $k_{33}$ , and (c) piezoelectric constants  $d_{33}$  of BNLKT4-100 $z$  and BNLKT8-100 $z$  ceramics.

pling factor  $k_{33}$ , free permittivity  $\epsilon_{33}^T$ , and piezoelectric constant  $d_{33}$  of BNLKT4-100 $z$  and BNLKT8-100 $z$  are shown in Fig. 5. The  $\epsilon_{33}^T$  of BNLKT100 $y$ -100 $z$  increased markedly near the MPB composition, and the  $\epsilon_{33}^T$  values of the tetragonal side were higher than those of the rhombohedral side. Moreover, the  $\epsilon_{33}^T$  values of BNLKT8-100 $z$  were all higher than those of BNLKT4-100 $z$  because the  $T_d$  of BNLKT8-100 $z$  is lower than that of BNLKT4-100 $z$ , and the  $\epsilon_{33}^T/\epsilon_0$  values of BNLKT4-100 $z$  and BNLKT8-100 $z$  at  $z = 0.22$ – $0.28$  were approximately equal at 1000 and 1100, respectively. On the other hand, the  $k_{33}$  values of BNLKT4-100 $z$  were almost the same as those of BNLKT8-100 $z$ , as shown in Fig. 5(b), and the highest  $k_{33}$  values were 0.603 for BNLKT4-20 and 0.590 for BNLKT8-20.

Fig. 5(c) shows  $d_{33}$  as a function of  $z$  for BNLKT4-100 $z$  and BNLKT8-100 $z$ .  $d_{33}$  was calculated using formula (1). For the BNLKT100 $y$ -100 $z$  system, the  $d_{33}$  values of BNLKT8-100 $z$  were higher than those of BNLKT4-100 $z$ , because the  $\epsilon_{33}^T/\epsilon_0$  values of BNLKT8-100 $z$  were higher than those of BNLKT4-100 $z$ . The highest  $d_{33}$  values of BNLKT4-100 $z$  and BNLKT8-100 $z$  were 176 and 190 pC/N, respectively. Moreover, the highest  $d_{33}$  of BNLKT0-20 was 168 pC/N. These results indicate that the substitution of Li ions in the A-site of BNT is very effective for improving  $d_{33}$ .

From the viewpoint of the temperature property,  $T_d$  is actual working temperature; therefore, its  $T_d$  value is very important. The  $d_{33}$  values of MPB compositions were rather high; however, the  $T_d$  values of the MPB composition were very low. In this study, for BNLKT0-20, BNLKT4-20, and BNLKT8-20, showed 130, 171 and 115 °C, respectively. Although, high  $d_{33}$  ( $d_{33}$  meter value) of 231 pC/N was obtained for BNLKT10-20 [11], it is considered that the  $T_d$  of this composition is probably lower than 100 °C. Thus, considering its high  $d_{33}$  and high  $T_d$ , the MPB composition of BNLKT4-100 $x$  is thought to be the optimal composition for that of BNLKT100 $y$ -100 $z$  systems.

#### 4. Conclusions

In this study, the depolarization temperature  $T_d$  and piezoelectric properties of BNLKT100 $y$ -100 $z$  were investigated. This study revealed that a small amount of Li ions can increase  $d_{33}$  without reducing  $T_d$ . The  $d_{33}$  values of BNLKT8-100 $z$  were higher than those of BNLKT4-100 $z$ ; however, the  $T_d$  of BNLKT8-100 $z$  was lower than that of BNLKT4-100 $z$ . Considering its high  $d_{33}$  and high  $T_d$ , the MPB composition of

BNLKT4-100z is thought to be the optimal composition for that of BNLKT100y–100z systems.

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