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Short communication

Preparation of alumina/carbon nanotubes composites by chemical precipitation

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Abstract

Continuous alumina coating on multi-walled carbon nanotubes (MWCNTs) was successfully prepared by a new method of chemical precipitation using aluminum nitrate and ammonia as starting materials. Structure and morphology of the alumina/multi-walled carbon nanotubes (Al₂O₃/MWCNTs) composites were characterized by transmission electron microscopy (TEM), high resolution transmission electron microscopy (HRTEM), energy-dispersive X-ray spectroscopy (EDX), X-ray diffraction (XRD), infrared spectra (IR), thermo gravimetric analysis (TG), differential thermal analysis (DTA) and N_2 adsorption–desorption. The results show that polyvinyl alcohol (PVA) modification on the surface of MWCNTs contributes to form continuous alumina coating, γ -Al₂O₃ layers with thickness of 1–3 nm cover the surface of MWCNTs and the original structure of MWCNTs is retained during the coating process.

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1. Introduction

Since the discovery of carbon nanotubes (CNTs), they have been receiving much attention over the past decade because of their properties such as small dimension, big area, high strength, unique conductivity and applications in reinforce of composites, field emission devices, chemical sensors and nanoelectrical devices [1–5]. CNTs can substitute carbon fibers as reinforcements in composites because of their tremendous properties. However, before CNTs are used as reinforcing additives in composites, there are two major obstacles such as the dispersibility of CNTs in matrix and undesirable interface reaction between CNTs and matrix at high temperature [6,7]. Coating technology on the surface of CNTs was the most efficient method to overcome these problems and improve the mechanical properties at the same time, such as alumina coating, titania coating, and etc. [8].

Presently, various methods have been reported to load CNTs with functional nanoparticles, such as decorating MWCNTs with iron oxide nanoparticles using a chemistry precipitation

method [9], Pt nanoparticles using supercritical carbon dioxide deposition [10], Cu and Ni using electroless plating method [11], ceria nanoparticles by supercritical hydrothermal synthesis method [12], anatase TiO₂ nanoparticles by sol–gel and rutile TiO₂ nanorods by hydrothermal methods [13].

However, there are few reports about the synthesis of alumina/multi-walled carbon nanotubes (Al₂O₃/MWCNTs) composites [14–16]. Li et al. immersed MWCNTs into aluminum nitrate solution, then calcinated under N₂ atmosphere at 500 °C to synthesize Al₂O₃/MWCNTs composites with sponge-like morphology, and their adsorbility of fluorinion in water was studied [14]. MWCNTs decorated with alumina nanoparticles were prepared by an impregnation method using aluminum isopropoxide (AlIP) as inorganic sources and decomposed by hydrolysis on the surface of carbon nanotubes [15]. Lee et al. filled and coated MWCNTs with alumina by chemical vapor deposition [16].

In this paper, in order to provide a favourable dispersibility and prevent harmful interface reaction for CNTs in matrix, we propose a simple chemical precipitation to synthesize continuous alumina coating on the surface of MWCNTs modified by PVA. The chemical precipitation process, the effects of PVA and the characterization of composites were discussed.

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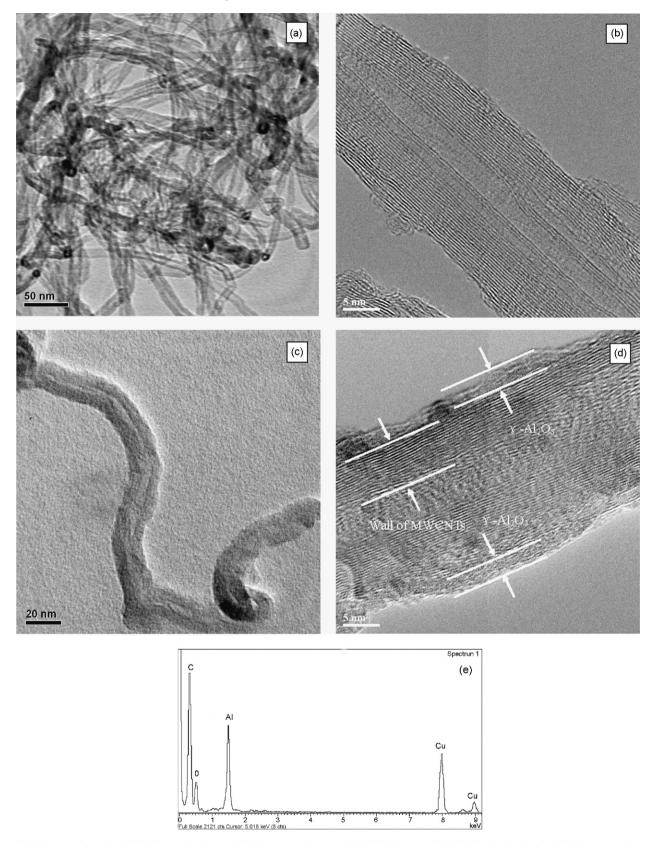


Fig. 1. TEM images of pristine MWCNTs (a), alumina/MWCNTs composites (c) and HRTEM images of MWCNTs (b), alumina/MWCNTs composites (d), and EDX spectrum of composites (e).

2. Experimental

MWCNTs with purity of 95% were provided by Shenzhen Nanotech Port Co., Ltd., which were prepared by the catalytic decomposition of CH₄ using La₂NiO₄ as catalyst precursor. To remove the catalyst, MWCNTs were purified via being dispersed in a 1 M HNO₃ solution for more than 6 h, followed by filtering and washing with distilled water several times. PVA with polymerization degree of 1750 is commercially pure. Al(NO₃)₃·9H₂O, NH₃·H₂O, HNO₃, H₂SO₄ are analytic pure. All the reagents were used without further purification. A typical preparation procedure of Al₂O₃/MWCNTs composites was as follows: To modify the surface of MWCNTs, 1.0 g of purified MWCNTs was dispersed in a 100 ml PVA aqueous solution (2.5 wt%) for 6 h, then the suspension was filtered. washed with distilled water several times to remove the residual PVA, and dried at 60 °C in vacuum for 12 h. Subsequently, 0.5 g PVA-modified MWCNTs was dispersed into 100 ml Al(NO₃)₃ (10 wt%) solution by ultrasonic treatment. Then the ammonia solution with a concentration of 2.5 wt% was added dropwise into the suspension with vigorously agitation until the pH value reached 9.5. After agitated for 1 h, the insoluble black products were filtered, washed with distilled water and dried at 60 °C in vacuum for 12 h. Finally, the precipitated products were heated in N₂ atmosphere from room temperature too 500 °C at the heat rate of 3 °C/min and kept for 2 h, and about 0.6 g Al₂O₃/ MWCNTs composites were obtained.

The morphology and chemical component of the products were examined by means of transmission electron microscopy (TEM), high resolution transmission electron microscopy (HRTEM) and energy-dispersive X-ray spectroscopy (EDX) on a JEOL JEM-2010 equipped with an energy-dispersive X-ray spectrometer, employing an accelerating voltage of 200 kV. X-ray diffraction (XRD) was recorded on a BRUKER-AXS X-ray powder diffractometer with Cu Kα radiation ($\lambda = 0.154178$ nm) in the range of $10^{\circ} < 2\theta < 80^{\circ}$ at a scanning speed of 1°/min. The infrared absorption spectra (IR) was measured on an EQUINOX55 Fourier transform infrared spectroscope using KBr pellets from 400 to 4000 cm⁻¹ at a resolution of 2 cm⁻¹. The zeta potential was measured with a Zetaplus analyzer (Zetasizer 2000). Each sample was ultrasonicated for 1 h prior to analysis. The ionic strength was maintained at 10⁻³ M using NaCl. TG and DTA were performed on a TG 2050 thermo gravimetric analysis (TG) from room temperature to 800 $^{\circ}$ C and a DTA 1600 differential thermal analysis (DTA) from room temperature to 1000 °C at a heating rate of 10 °C/min in air. The N₂ absorption–desorption isotherms at liquid nitrogen temperature (78 K) were measured by ASAP2010 static volumetric absorption analyzer. BET surface area was calculated based on the adsorption data in the relative pressure range of $p/p_0 = 0.05-0.25$.

3. Results and discussion

Fig. 1a shows the TEM image of pristine MWCNTs. The pristine MWCNTs are curved and twisted with each other. Their length is several micrometers and the average external

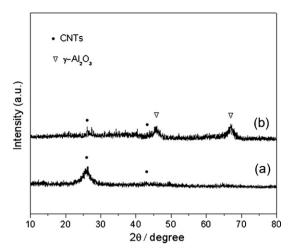


Fig. 2. XRD patterns of (a) pristine MWCNTs and (b) alumina/MWCNTs composites.

diameter is about 10-30 nm. HRTEM image of the MWCNTs indicates that their carbon atoms arrange orderly and their tubular structure is very clear (Fig. 1b). Fig. 1c represents the rough and compact surface of the Al₂O₃/MWCNTs composites, compared with the MWCNTs. It seems that the surface of nanotubes has been coated with some products, which are testified as y-Al₂O₃ in the XRD analysis (Fig. 2b). Fig. 1d shows a representative HRTEM image of the Al₂O₃/MWCNTs composites, illustrates clearly the periodic multiwalls of the MWCNTs and the γ -Al₂O₃ coating layers, and the thicknesses of the γ-Al₂O₃ layers are about 1-3 nm. Fig. 1e displays the EDX spectrum taken from the Al₂O₃/MWCNTs composites in Fig. 1d. Only the C-, O-, Al-related peaks are present in the EDX spectrum, revealing that the covered products is indeed Al₂O₃; the Cu-related peaks in the spectrum come from the Cu grids.

Fig. 2 displays the XRD patterns of pristine MWCNTs and $Al_2O_3/MWCNTs$ composites. The two peaks at 26.02° and 42.68° correspond to $(0\ 0\ 2)$ and $(1\ 0\ 1)$ reflections of MWCNTs, respectively (Fig. 2a) [17]. From Fig. 2b, the two peaks of MWCNTs are much lower than that of pure MWCNTs, and the two peaks at 45.86° and 67.03° are

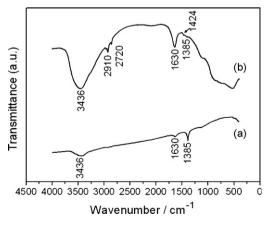


Fig. 3. IR patterns of (a) pristine MWCNTs and (b) alumina/MWCNTs composites.

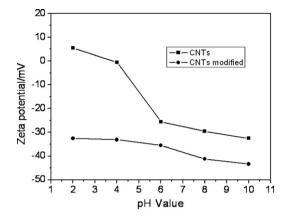


Fig. 4. Zeta potentials of pristine MWCNTs and MWCNTs modified by PVA.

assigned to (4 0 0) and (4 4 0) reflections of γ -Al₂O₃ (JCPDS Card 10-0425). Therefore, the composites should be considered as nanosized MWCNTs coated by γ -Al₂O₃.

IR spectra are helpful to further understand the formation of Al₂O₃/MWCNTs composites. The IR spectra of pristine MWCNTs are shown in Fig. 3a. The band at 1385 cm⁻¹ is assigned to the C–C vibration, the band at 3436 and 1630 cm⁻¹ are assigned to the O–H stretch and bend vibration of adsorbed water. In Fig. 3b, compared with pure MWCNTs (Fig. 3a), the very broad and smooth band from 400 to 1000 cm⁻¹ demonstrate the feature of nano-alumina [18]. The band at 1424, 2720 and 2910 cm⁻¹ correspond to the C–H stretch and bend vibration, originated from the surface of MWCNTs modified by PVA [19,20].

Fig. 4 shows zeta potential of pristine MWCNTs and MWCNTs modified by PVA. When the pH value of solution is above 4, the MWCNTs are negatively charged. The modification of MWCNTs by PVA further decreases the zeta potential value of all pH values studied. It means that the modification of MWCNTs by PVA can generate more functional groups on the surface graphite layers of MWCNTs.

Based on above analysis, a simulation scheme is proposed to illustrate the whole process for the preparation of Al₂O₃/MWCNTs composite (Fig. 5). After MWCNTs modified by PVA, the hydrophobic groups (–OH) of PVA incorporate with MWCNTs by Van Der Waals force, their hydrophilic groups extend into solution, and induce MWCNTs become more negative. Al³⁺ ions incorporate with hydrophilic group (–OH) by electrostatic force [21]. After aqueous ammonia was added dropwise into solution, Al³⁺ ions change to alumina hydrate

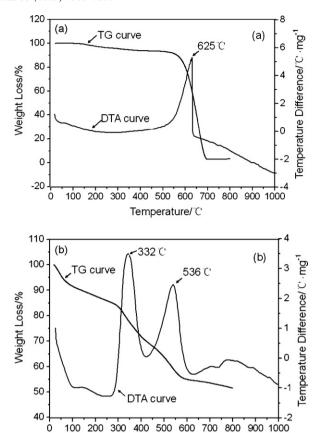


Fig. 6. TG and DTA curves of (a) pristine MWCNTs and (b) the precipitated products

Temperature/ C

which deposits on the surface of MWCNTs and transforms to γ -Al₂O₃ after calcinated at 500 °C for 2 h.

The heat treatment process of pristine MWCNTs and the precipitated products was analyzed by TG and DTA. According to the DTA curve of Fig. 6a, the exothermal peak around 625 °C can be assigned to the ignition of MWCNTs and there is a corresponding 98% weight loss in the TG curve (Fig. 6a). The ignition residues are chiefly ferric oxide and silicate, which are introduced in the growth process. On the other hand, there are visible differences in the composites (Fig. 6b). First, the total weight loss is about 46%, less than that of the MWCNTs. Second, there are two exothermic peaks at 332 and 536 °C, the exothermic peaks at 332 °C correspond to the ignition of PVA [22], and the exothermic peaks at 536 °C correspond to the

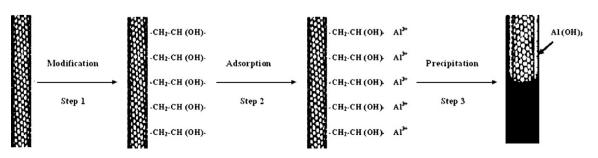


Fig. 5. Schematic representation of synthesis of alumina/MWCNTs composites.

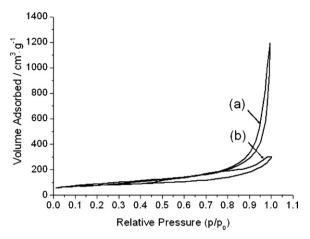


Fig. 7. Adsorption–desorption isotherms of (a) pristine MWCNTs and (b) alumina/MWCNTs composites.

Table 1 Nitrogen adsorption-desorption analysis data of MWCNTs and composites

Samples	Carbon nanotubes	Composites
BET surface area (m ² g ⁻¹)	314.03	168.39
Pore volume (cm ³ g ⁻¹)	1.85	0.42
Pore diameter (nm)	9.1	5.5

ignition of MWCNTs, which is much lower than that of pure MWCNTs. Finally, the broad endothermic scale from 220 to 270 °C is observed, which contribute to the decomposition of Al(OH)₃.

The nitrogen adsorption and desorption isotherms of pristine MWCNTs and Al₂O₃/MWCNTs composites are presented in Fig. 7. The isotherm of MWCNTs is type IV (BDDT classification) and has type H3 hysteresis loops, associated with aggregates of platelike particles giving rise to slit-like pores [23]. The isotherm of the composites is type IV and has two types of hysteresis loops. At low relative pressure $0.45 < p/p_0 < 0.85$, the hysteresis loop is type H2, which can be observed in the pores with narrow necks and wider bodies (ink-bottle pores) [23], perhaps because the Al₂O₃ block the pores of MWCNTs, when they deposit on the surface of MWCNTs. At high relative pressure between $0.85 < p/p_0 < 1$, the shape of the hysteresis loops is type H3. BET surface area, pore volume and pore diameter of MWCNTs and composites are listed in Table 1. After MWCNTs coated with Al₂O₃, the BET surface area, pore volume and pore diameter of MWCNTs significantly reduced from 314.03 to 168.39 cm² g⁻¹, 1.85 to $0.42 \text{ cm}^3 \text{ g}^{-1}$, 9.1 to 5.5 nm, respectively. This directly proved that Al₂O₃ cover the surface of the MWCNTs and block their pores.

4. Conclusions

The Al₂O₃/MWCNTs composites with continuous alumina coatings on the surface of MWCNTs were fabricated by chemical precipitation. During the process, PVA-modified MWCNTs and contributed to form the continuous alumina

coatings. The alumina coating was γ -Al₂O₃, and their thickness is about 1–3 nm.

Acknowledgments

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