



**CERAMICS** INTERNATIONAL

www.elsevier.com/locate/ceramint

Ceramics International 37 (2011) 249-255

### Morphology control of $\alpha$ -Al<sub>2</sub>O<sub>3</sub> platelets by molten salt synthesis

Zhu Li-hui \*, Huang Qing-wei

Shanghai Key Laboratory of Modern Metallurgy, Materials Processing, P.O. Box 15#, Shanghai University, 149 Yanchang Road, Shanghai 200072, People's Republic of China

Received 25 May 2010; received in revised form 30 June 2010; accepted 25 August 2010 Available online 29 September 2010

### Abstract

It is of great importance to control the morphology of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> plate-like powders since  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> platelets with different shapes are needed in various applications. This paper was focused on how to control the morphology of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> platelets by molten salt synthesis. Results show that the morphology of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> platelets is affected by the heating temperature, heating time, the molten salts species, the weight ratio of salt to powders, additives and the addition of nano-sized seeds. Especially, it is very effective to control the morphology of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> platelets by adjusting the addition of additives such as Na<sub>3</sub>PO<sub>4</sub>·12H<sub>2</sub>O and TiOSO<sub>4</sub>.  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> flakes with irregular shape are obtained by the addition of Na<sub>3</sub>PO<sub>4</sub>·12H<sub>2</sub>O and TiOSO<sub>4</sub> makes it possible to obtain thin  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> platelets with discal shape. A small amount of nano-sized seeds addition also has a strong effect on the size of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> platelets. However, if the seeds are added too much, the overlapping and abnormal crystal growth of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> platelets occur, and the size distribution becomes nonuniform. The effect mechanism of additives and seeds on the morphology of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> platelets was also discussed in this paper.

© 2010 Elsevier Ltd and Techna Group S.r.l. All rights reserved.

Keywords: B. Platelets; D. α-Al<sub>2</sub>O<sub>3</sub>; Molten salt synthesis; Additives; Seeds

#### 1. Introduction

Plate-like α-Al<sub>2</sub>O<sub>3</sub> powders are applied widely because they exhibit excellent properties, which derive from  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> and the special two-dimensional structure. For example, α-Al<sub>2</sub>O<sub>3</sub> platelets can be added into the ceramics as seeds to induce the abnormal grain growth, leading to the improvement of fracture toughness [1]. Textured Al<sub>2</sub>O<sub>3</sub> ceramics with anisotropic properties can be prepared by templated grain growth when suitable α-Al<sub>2</sub>O<sub>3</sub> platelets are chosen as templates [2]. Due to high aspect ratio and heat conductivity, α-Al<sub>2</sub>O<sub>3</sub> platelets can be added into the plastics to improve the thermal conductivity [3–5]. The commercial slurry for the primary polishing of hard disc is superior in good dispersion stability and desirable orientation when  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> platelets are added [4]. Since α-Al<sub>2</sub>O<sub>3</sub> platelets with different shapes are needed in various applications [4,5], it is of great importance to control the morphology.

Molten salt synthesis is one of the most important techniques to prepare plate-like  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> powders [5–10]. Compared with the conventional solid-state reaction, the preparation temperature and time can be greatly reduced because of high diffusivity of the components in the molten salt. Besides, the morphology of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> can be easily changed by many factors, for example, the precursors, the molten salts, heating temperature, heating time and so on. In this paper, α-Al<sub>2</sub>O<sub>3</sub> platelets were obtained by molten salt synthesis using Al<sub>2</sub>(SO<sub>4</sub>)<sub>3</sub> as raw materials, which was often chosen during molten salt synthesis in the previous investigations [5–9]. First, the effects of heating temperature, heating time, the molten salts species, and the weight ratio of salt to powders on the morphology of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> were studied. Considering that the addition of some additives not only decreases the transformation temperature of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> [11–14], but also changes the morphology of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> platelets obviously [5,8], it is necessary to investigate to control the morphology of α-Al<sub>2</sub>O<sub>3</sub> platelets by additives. As the most often-used additives, PO<sub>4</sub><sup>3-</sup> (introduced by trisodium phosphate) and Ti<sup>4+</sup> (introduced by titanyl sulfate) were chosen in this paper to study the influence of additives on the morphology of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> platelets synthesized by molten salt synthesis. The

<sup>\*</sup> Corresponding author. Tel.: +86 21 5633 1462; fax: +86 21 5633 3080. E-mail address: lhzhu@mail.shu.edu.cn (Z. Li-hui).

platelet morphology of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> powders was also significantly influenced by the addition of seeds [4,5,9], thus the effect of nano-sized  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> seeds on the morphology of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> platelets was also studied. At the same time, the effect mechanism of Na<sub>3</sub>PO<sub>4</sub>·12H<sub>2</sub>O, TiOSO<sub>4</sub> and seeds on the morphology of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> platelets was discussed.

### 2. Experimental procedure

α-Al<sub>2</sub>O<sub>3</sub> platelets were prepared according to the flow chart shown in Fig. 1. Al<sub>2</sub>(SO<sub>4</sub>)<sub>3</sub>·18H<sub>2</sub>O (purity: 99.5%, Meixing Chemical Company, Shanghai) and Na<sub>2</sub>CO<sub>3</sub> (purity: 99.5%, Hongguang Chemical Company, Shanghai) were used as raw materials. Either mixture sulfate salts  $(Na_2SO_4:K_2SO_4 = 1:1,$ molar ratio) or mixture chloride salts (NaCl:KCl = 1:1, molar ratio) were used as molten salts when considering that the mixture salts are more beneficial to the development of α-Al<sub>2</sub>O<sub>3</sub> platelets than pure salt. During the preparation of saturated solution A, Al<sub>2</sub>(SO<sub>4</sub>)<sub>3</sub>·18H<sub>2</sub>O and the salts were mixed according to the ratio of 1:2, 1:4 and 1:6 respectively, and then were dissolved in de-ionized water at 70 °C. The saturated solution B was obtained by dissolving Na<sub>2</sub>CO<sub>3</sub> in deionized water. Solution B was added slowly into solution A at 70 °C and kept stirring rapidly for 15 min. The obtained gel was dried at 120 °C for 24 h and then the product was heated at 1100-1300 °C for 0.5-8 h. The final powders were ultrasonic cleaned with de-ionized water repeatedly to remove the remained salt and then dried.

Either 0.17–0.68% Na<sub>3</sub>PO<sub>4</sub>·12H<sub>2</sub>O or 1–12% TiOSO<sub>4</sub> was added during the formation of sol–gel to control the morphology of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> powders. The addition amount of Na<sub>3</sub>PO<sub>4</sub>·12H<sub>2</sub>O and TiOSO<sub>4</sub> is presented by weight, respectively in terms of P<sub>2</sub>O<sub>5</sub> and TiOSO<sub>4</sub>, relative to the weight of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> particles. In order to study the effect of nano-sized  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> seeds on the morphology of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> platelets, 0.1–1 wt.% globular  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> powders with the average diameter of 60 nm were added during the formation of sol–gel. Mixture

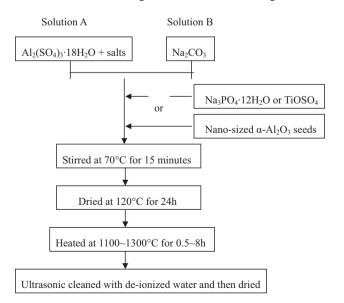


Fig. 1. Flow chart of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> platelets.

chloride salts which were composed of NaCl and KCl according to molar ratio of 1:1 were used as molten salts.

The phase assembly of the final powders was examined by X-ray diffraction analysis (RIGAKU, D/MAX-RB) with CuK $\alpha$  radiation ( $\lambda = 1.5418$  Å). The morphology of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> particles was observed by scanning electron microscope (SEM, S-570).

#### 3. Results and discussion

## 3.1. Effect of processing parameters on the morphology of $\alpha$ -Al<sub>2</sub>O<sub>3</sub> platelets

X-ray diffraction analysis indicates single phase  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> is synthesized by molten salt synthesis when heated above 1100 °C. The increase of heating temperature and time helps to the development of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> and it is easy to obtain well-developed  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> platelets with big size. However, there is no obvious change in the shape and size of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> platelets when the heating temperature increases above 1200 °C or the time exceeds 4 h. Therefore, from the viewpoint of economy, the heating temperature should not be higher than 1200 °C and the heating time should not be longer than 4 h in order to obtain well-developed hexagonal  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> platelets. In subsequent experiments,  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> powders were synthesized at 1200 °C for 4 h.

In addition to the heating temperature and time, the morphology of α-Al<sub>2</sub>O<sub>3</sub> platelets is affected by the ratio of salt to powders. Fig. 2(a) shows the morphology of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> platelets heated at 1200 °C for 4 h in NaCl-KCl flux when the ratio of salt to powders is 2:1. Most of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> platelets are hexagonal, but there are some overlapped particles. The more the molten salts are used, the more the crystal growth space is provided and the higher diffusivity of the components in the molten salts is. Therefore, when the ratio of salt to powders increases to 4:1, the overlapping of powders improves, and the diameter of α-Al<sub>2</sub>O<sub>3</sub> platelets becomes bigger, see Fig. 2(b). But if the ratio of salt to powders is higher than 6:1, the diffusion distance of the components in the molten salts increases. On the contrary, it is not easy to obtain well-developed hexagonal α-Al<sub>2</sub>O<sub>3</sub> platelets, and the size is nonuniform, see Fig. 2(c). High ratio of salt to powders also means that much more water will be spent to remove the salts, so the ratio of salt to powders is fixed to be 4:1 in subsequent experiments.

When different fluxes are chosen, there is obvious difference in the morphology of  $\alpha\text{-}Al_2O_3$  platelets even though the heating temperature, heating time and the ratio of salt to powders are fixed. As shown in Fig. 2(d), the shape of  $\alpha\text{-}Al_2O_3$  particles becomes irregular and the size is nonuniform when  $Na_2SO_4-K_2SO_4$  is used.

# 3.2. Effect of additives on the morphology of $\alpha$ -Al<sub>2</sub>O<sub>3</sub> platelets

The morphology of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> platelets is changed easily by the additives rather than processing parameters. The addition of 0.17 wt.% Na<sub>3</sub>PO<sub>4</sub>·12H<sub>2</sub>O only leads to a little decrease in the diameter of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> platelets. When 0.34–0.68 wt.% Na<sub>3</sub>PO<sub>4</sub>·12H<sub>2</sub>O are added, the thickness of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> is reduced

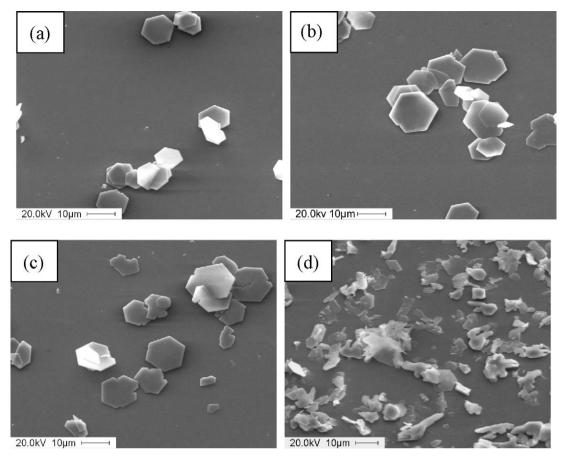


Fig. 2. The morphology of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> platelets heated at 1200 °C for 4 h using different molten salts and different ratio of salt to powders. (a) NaCl–KCl, ratio of salt to powders 2:1; (b) NaCl–KCl, ratio of salt to powders 4:1; (c) NaCl–KCl, ratio of salt to powders 6:1; and (d) Na<sub>2</sub>SO<sub>4</sub>–K<sub>2</sub>SO<sub>4</sub>, ratio of salt to powders 4:1.

greatly. At the same time, the agglomeration of  $\alpha\text{-}Al_2O_3$  powders is inhibited effectively and the particles are overlapped no longer. However, the shape of  $\alpha\text{-}Al_2O_3$  flakes becomes quite irregular and the size distribution becomes very broad. SEM micrograph of  $\alpha\text{-}Al_2O_3$  flakes added by 0.51 wt.% Na\_3PO\_4·12H\_2O in NaCl–KCl flux is shown in Fig. 3(a).  $\alpha\text{-}Al_2O_3$  flakes synthesized in Na\_2SO\_4–K\_2SO\_4 flux show similar morphology, see Fig. 3(b).

The addition of  $TiOSO_4$  is helpful in the formation of hexagonal  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> platelets synthesized in NaCl-KCl flux.

Meanwhile,  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> particles tend to become small and thick, see Fig. 4. When the addition amount of TiOSO<sub>4</sub> increases from 1 wt.% to 6 wt.%, the average diameter of plate-like particles decreases from about 12.2  $\mu$ m to 8.6  $\mu$ m, and the average thickness increases quickly from 0.7  $\mu$ m to 2.1  $\mu$ m. There is a slight change in the diameter and thickness of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> platelets when more than 6 wt.% TiOSO<sub>4</sub> are added.

As indicated above,  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> flakes with irregular shape are obtained by the addition of Na<sub>3</sub>PO<sub>4</sub>·12H<sub>2</sub>O, while thick  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> particles with hexagonal shape are obtained by the

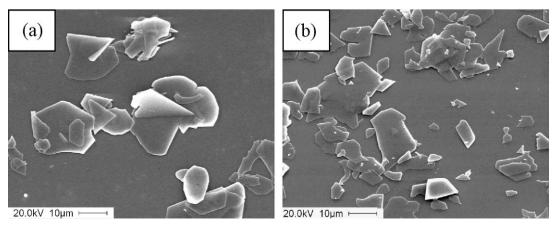


Fig. 3. The morphology of α-Al<sub>2</sub>O<sub>3</sub> platelets added by 0.51 wt.% Na<sub>3</sub>PO<sub>4</sub>·12H<sub>2</sub>O in different fluxes: (a) NaCl–KCl and (b) Na<sub>2</sub>SO<sub>4</sub>–K<sub>2</sub>SO<sub>4</sub>.

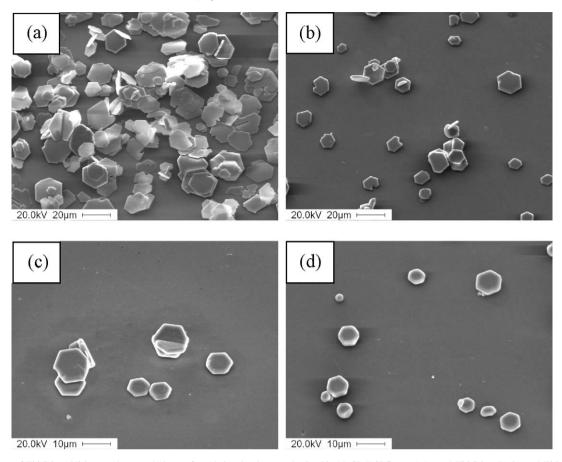


Fig. 4. The effect of TiOSO<sub>4</sub> addition on the morphology of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> platelets synthesized in NaCl–KCl flux: (a) 1 wt.% TiOSO<sub>4</sub>; (b) 3 wt.% TiOSO<sub>4</sub>; (c) 6 wt.% TiOSO<sub>4</sub>; (d) 11 wt.% TiOSO<sub>4</sub>.

addition of TiOSO<sub>4</sub>. The combination addition of Na<sub>3</sub>PO<sub>4</sub>·12H<sub>2</sub>O and TiOSO<sub>4</sub> probably leads to the further change in the morphology of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> platelets. Fig. 5(a) and (b) shows α-Al<sub>2</sub>O<sub>3</sub> platelets synthesized in NaCl-KCl flux when 0.51 wt.%  $\text{Na}_3\text{PO}_4 \cdot 12\text{H}_2\text{O} + 3 \text{ wt.}\%$   $\text{TiOSO}_4$  and  $0.51 \text{ wt.}\% \text{ Na}_3\text{PO}_4\cdot 12\text{H}_2\text{O} + 12 \text{ wt.}\% \text{ TiOSO}_4$  are added, separately. The addition of Na<sub>3</sub>PO<sub>4</sub>·12H<sub>2</sub>O helps the formation of thin  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> platelets, but their shape is still irregular if only 3 wt.% TiOSO<sub>4</sub> is added, see Fig. 5(a). Thin  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> platelets with discal shape are obtained by the addition of 0.51 wt.%  $Na_3PO_4 \cdot 12H_2O + 12$  wt.%  $TiOSO_4$ , see Fig. 5(b). A similar phenomenon is also found in α-Al<sub>2</sub>O<sub>3</sub> platelets synthesized in Na<sub>2</sub>SO<sub>4</sub>-K<sub>2</sub>SO<sub>4</sub> flux, as shown in Fig. 5(c) and (d). Especially, when 0.51 wt.% Na<sub>3</sub>PO<sub>4</sub>·12H<sub>2</sub>O + 12 wt.% TiOSO<sub>4</sub> are added, not only thin  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> platelets with discal shape are obtained in Na<sub>2</sub>SO<sub>4</sub>-K<sub>2</sub>SO<sub>4</sub> flux, but also the overlapping and agglomeration of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> platelets are restrained, see Fig. 5(d). It is very effective to control the morphology of α-Al<sub>2</sub>O<sub>3</sub> platelets by adjusting the addition of Na<sub>3</sub>PO<sub>4</sub>·12H<sub>2</sub>O and TiOSO<sub>4</sub>.

# 3.3. Effect of nano-sized seeds on the morphology of $\alpha$ -Al<sub>2</sub>O<sub>3</sub> platelets

The size of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> platelets is affected significantly by the addition of a small amount of nano-sized seeds. Fig. 6 shows SEM micrographs of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> synthesized in NaCl–KCl flux

when added by 0.1 wt.%, 0.5 wt.% and 1 wt.% seeds. For convenience to compare, the micrograph of  $\alpha\text{-Al}_2O_3$  platelets without seed addition is also given. The average diameter of  $\alpha\text{-Al}_2O_3$  platelets decreases from about 12.2  $\mu m$  to 1.8  $\mu m$  quickly even though added by 0.1 wt.% seeds. With the increase of seed amount, the diameter and aspect ratio of  $\alpha\text{-Al}_2O_3$  platelets tends to decrease, and the size becomes nonuniform. When 0.5 wt.% seeds are added, there are some overlapped particles and a few  $\alpha\text{-Al}_2O_3$  crystals with abnormal growth. The addition of 1 wt.% seeds leads to more crystals with abnormal growth and agglomeration of powders, and the size distribution of  $\alpha\text{-Al}_2O_3$  platelets becomes very broad. It seems that the seeds are added too much to obtain a narrow size distribution of  $\alpha\text{-Al}_2O_3$  platelets.

### 4. Discussion

# 4.1. The effect mechanism of $Na_3PO_4$ ·12 $H_2O$ and $TiOSO_4$ on the morphology of $\alpha$ - $Al_2O_3$ platelets

The crystal structure of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> is composed of a hexagonal close-packed oxygen layer with Al<sup>3+</sup> occupying the interstitial sites.  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> belongs to trigonal system in which the lattice points occupy (0,0,0),(2/3,1/3,1/3) and (1/3,2/3,2/3) for hexagonal coordinate, so the  $\{0\ 0\ 0\ 1\}$  faces are hexagon.

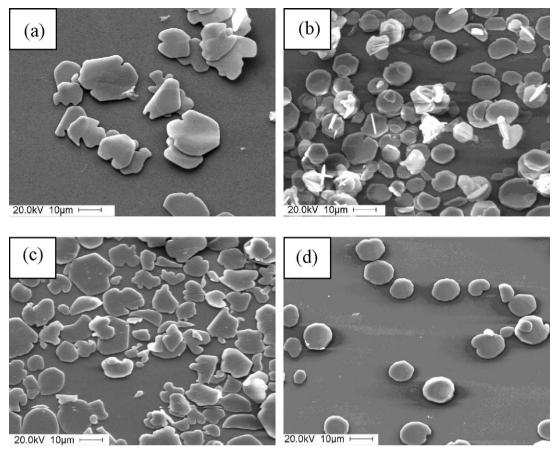


Fig. 5. The effect of combination addition of  $Na_3PO_4\cdot 12H_2O$  and  $TiOSO_4$  on the morphology of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> platelets in different fluxes: (a) NaCl-KCl, 0.51 wt.%  $Na_3PO_4\cdot 12H_2O + 3$  wt.%  $TiOSO_4$ ; (b) NaCl-KCl, 0.51 wt.%  $Na_3PO_4\cdot 12H_2O + 12$  wt.%  $TiOSO_4$ ; (c)  $Na_2SO_4-K_2SO_4$ , 0.51 wt.%  $Na_3PO_4\cdot 12H_2O + 3$  wt.%  $TiOSO_4$ ; (d)  $Na_2SO_4-K_2SO_4$ , 0.51 wt.%  $Na_3PO_4\cdot 12H_2O + 12$  wt.%  $TiOSO_4$ ;

The crystal development can be regarded as a "growth unit" course. It includes the formation of growth unit, the interfacial adsorption of growth unit, the movement of growth unit and the desorption of growth unit. According to the theoretical model of anionic coordination polyhedron growth units, the crystal growth and the final morphology are dependent on the crystallographic orientation and the combination manner of the growth units [15]. In molten salts, [Al–O<sub>6</sub>] octahedron is considered as the growth unit for  $\alpha$ -Al<sub>2</sub>O<sub>3</sub>. {10 \(\bar{1}\)0} faces often disappear, {0 0 0 1} faces appear predominantly and {11\(\bar{2}\)0} faces sometimes appear. Therefore,  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> tends to be hexagonal platelets.

During crystal growth, the morphology of particles may be changed when some ions in the molten salt are adsorbed on the crystal surfaces. When Na<sub>3</sub>PO<sub>4</sub>·12H<sub>2</sub>O is added, PO<sub>4</sub><sup>3-</sup> is inclined to adsorb on {0 0 0 1} faces where the apex angles of [Al–O<sub>6</sub>] octahedron is the least owing to the Coulomb force. Because PO<sub>4</sub><sup>3-</sup> has large ionic strength, the superimposition of growth units on {0 0 0 1} faces is effectively prohibited by PO<sub>4</sub><sup>3-</sup>, and the growth of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> in the thickness is restricted. The growth units tend to superimpose on the other two faces {1 0 \(\bar{1}\) 0} and {1 1 \(\bar{2}\) 0} when they combine with each other. As a result, thin and irregular platelets are finally obtained. Besides, the agglom-

eration and overlapping of platelets are effectively restrained owing to the electrostatic resistance and stereo-hindrance when enough  $PO_4^{3-}$  is adsorbed on the  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> crystal surfaces

When TiOSO<sub>4</sub> is added, Ti<sup>4+</sup> ions can diffuse into Al<sub>2</sub>O<sub>3</sub> to substitute Al<sup>3+</sup> ions at high temperatures [5]. In order to keep the electrostatic balance, three Ti<sup>4+</sup> ions diffuse into the crystal lattice to substitute four Al<sup>3+</sup> ions. As a result, the development of growth units and crystals may be affected by the lattice deformation due to the existence of extra Al<sup>3+</sup> vacancies. The growth velocity of  $\{1\,0\,\bar{1}\,0\}$  and  $\{1\,1\,\bar{2}\,0\}$  crystal faces is reduced and there is a minor difference in the growth velocity of various crystal faces. Thus the addition of TiOSO<sub>4</sub> is helpful in the formation of well-developed hexagonal  $\alpha\text{-Al}_2\text{O}_3$  particles with decreased size and increased thickness.

The morphology of  $\alpha$ -Al $_2$ O $_3$  platelets can be adjusted under the combined action of PO $_4$ <sup>3-</sup> and Ti<sup>4+</sup>. On the one hand, the superimposition of growth units on  $\{0\ 0\ 0\ 1\}$  faces is effectively inhibited by PO $_4$ <sup>3-</sup>. On the other hand, the difference in the growth velocity of various crystal faces is reduced by Ti<sup>4+</sup>. The combination addition of Na $_3$ PO $_4$ ·12H $_2$ O and TiOSO $_4$  makes it possible to obtain thin  $\alpha$ -Al $_2$ O $_3$  platelets with discal shape.

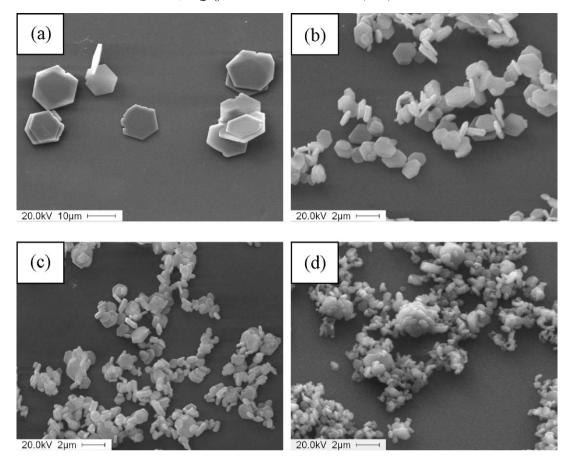


Fig. 6. The effect of nano-sized seeds on the morphology of α-Al<sub>2</sub>O<sub>3</sub> platelets: (a) 0 wt.% seeds; (b) 0.1 wt.% seeds; (c) 0.5 wt.% seeds; (d) 1 wt.% seeds.

## 4.2. The effect mechanism of nano-sized seeds on the morphology of $\alpha$ -Al<sub>2</sub>O<sub>3</sub> platelets

The formation of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> platelets in molten salt should be a nucleation-growth process. When solution B is added slowly into solution A, the following reaction will happen.

$$3Na_2CO_3 + Al_2(SO_4)_3 + 3H_2O = 3Na_2SO_4 + 2Al(OH)_3 + 3CO_2 \uparrow$$

During heating, the thermal decomposition of Al(OH)<sub>3</sub> results in the formation of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub>, which can act as the nuclei of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> platelets. When nano-sized  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> seeds are added, extra nucleation sites can be provided by the seeds since they have small size and the same crystal structure as  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> nuclei. In the later stage, α-Al<sub>2</sub>O<sub>3</sub> particles grow large from the nuclei provided by either the nano-sized seeds or α-Al<sub>2</sub>O<sub>3</sub> formed by the decomposition of Al(OH)<sub>3</sub>. Therefore, fine  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> platelets are obtained accompanied by the increase in the nuclei of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub>. However,  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> platelets directly developed from nano-sized α-Al<sub>2</sub>O<sub>3</sub> seeds have small aspect ratio because it is limited by globular shape of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> seeds. As a result, the diameter and aspect ratio of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> platelets tends to decrease when more nano-sized α-Al<sub>2</sub>O<sub>3</sub> seeds are added. If too much seeds are added, some seeds which close up to each other are probably integrated by α-Al<sub>2</sub>O<sub>3</sub> crystals precipitated during molten salt synthesis. The phenomena of overlapping and abnormal crystal growth maybe take place. The size distribution of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> platelets becomes very broad.

### 5. Conclusions

α-Al<sub>2</sub>O<sub>3</sub> platelets are synthesized by molten salt synthesis using Al<sub>2</sub>(SO<sub>4</sub>)<sub>3</sub> as raw materials, and their morphology can be changed by the heating temperature, heating time, the molten salts species, the weight ratio of salt to powders, additives and the addition of nano-sized seeds. Especially, it is very effective to control the morphology of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> platelets by adjusting the addition of additives such as Na<sub>3</sub>PO<sub>4</sub>·12H<sub>2</sub>O and TiOSO<sub>4</sub>. α-Al<sub>2</sub>O<sub>3</sub> flakes with irregular shape are obtained by the addition of Na<sub>3</sub>PO<sub>4</sub>·12H<sub>2</sub>O, while thick α-Al<sub>2</sub>O<sub>3</sub> particles with hexagonal shape are obtained by the addition of TiOSO<sub>4</sub>. The combination addition of Na<sub>3</sub>PO<sub>4</sub>·12H<sub>2</sub>O and TiOSO<sub>4</sub> makes it possible to obtain thin α-Al<sub>2</sub>O<sub>3</sub> platelets with discal shape. A small amount of nano-sized seeds addition also has a strong effect on the size of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> platelets. However, if the seeds are added too much, the overlapping and abnormal crystal growth of α-Al<sub>2</sub>O<sub>3</sub> platelets occur, and the size distribution becomes nonuniform.

### Acknowledgements

The work was supported by the Science Innovation Foundation of Shanghai Municipal Commission of Education

under the grant 09YZ26. The authors were grateful for the help provided by Ms. Zhang Qian-Ying and Ms. Dai Yue-Qin during the experiments.

### References

- Y. Yoshizawa, M. Toriyama, S. Kanzaki, Preparation of high fracture toughness alumina sintered bodies from bayer aluminum hydroxide, Journal of the Ceramic Society of Japan 106 (12) (1998) 1172–1177.
- [2] M.M. Seabaugh, I.H. Kerscht, G.L. Messing, Texture development by templated grain growth in liquid-phase-sintered α-alumina, Journal of the American Ceramic Society 80 (5) (1997) 1181–1188.
- [3] R.F. Hill, R. Danzer, Synthesis of aluminum oxide platelets, Journal of the American Ceramic Society 84 (3) (2001) 514–520.
- [4] T. Fukuda, R. Shido, Flaky-like alpha-alumina particles and method for producing the same, EN Patent 1 148 028 A2, 4 December 2001.
- [5] K. Nitta, T.M. Shau, J. Sugahara, Flaky aluminum oxide and pearlescent pigment and production thereof, EN Patent 0 763 573 A2, 5 September 1997.
- [6] S. Hashimoto, A. Yamaguchi, Synthesis of α-Al<sub>2</sub>O<sub>3</sub> platelets using sodium sulfate flux, Journal of Materials Research 14 (12) (1999) 4667–4672.
- [7] S. Hashimoto, A. Yamaguchi, Formation of porous aggregations composed of Al<sub>2</sub>O<sub>3</sub> platelets using potassium sulfate flux, Journal of the European Ceramic Society 19 (1999) 335–339.

- [8] S. Hashimoto, A. Yamaguchi, Effect of impurities on morphology of α-alumina platelets formed by using sodium sulfate flux, Advances in Science and Technology Part B 29 (2000) 711–718.
- [9] X.H. Jin, L. Gao, Size control of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> platelets synthesized in molten Na<sub>2</sub>SO<sub>4</sub> flux, Journal of the American Ceramic Society 87 (4) (2004) 533–540
- [10] L.-H. Zhu, Q.-W. Huang, W. Liu, Synthesis of plate-like α-Al<sub>2</sub>O<sub>3</sub> single-crystal particles in NaCl–KCl flux using Al(OH)<sub>3</sub> powders as starting materials, Ceramics International 34 (7) (2008) 1729–1733.
- [11] Y.Q. Wu, Y.F. Zhang, G. Pezzotti, Influence of AIF<sub>3</sub> and ZnF<sub>2</sub> on the phase transformation of gamma to alpha alumina, Materials Letters 52 (2002) 366–369.
- [12] Z.Y. Song, Y.C. Wu, Preparation of alumina ultrafine powders and its modification by titania doping, Journal of Chinese Ceramic Society 32 (8) (2004) 920–925.
- [13] L. Jiang, Y.S. Wu, Y.B. Pan, W.B. Lin, J.K. Guo, Influence of fluorides on phase transition of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> formation, Ceramics International 33 (6) (2007) 919–923.
- [14] H.J. Kim, T.G. Kim, J.J. Kim, S.S. Park, S.S. Hong, G.D. Lee, Influences of precursor and additive on the morphology of nanocrystalline α-alumina, Journal of Physics and Chemistry of Solids 69 (2008) 1521–1524.
- [15] W. Li, E. Shi, Z. Yin, Theoretical model of anionic coordination polyhedron growth units and the growth habits of crystals, Science in China 31 (6) (2001) 487–495.