

Preparation and characterization of CeO₂/TiO₂ nanoparticles by flame spray pyrolysis

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Abstract

Nanocrystalline TiO₂, CeO₂ and CeO₂-doped TiO₂ have been successfully prepared by one-step flame spray pyrolysis (FSP). Resulting powders were characterized with X-ray diffraction (XRD), N₂-physisorption, Transmission Electron Microscopy (TEM) and UV–Vis spectrophotometry. The TiO₂ and CeO₂-doped TiO₂ nanopowders were composed of single-crystalline spherical particles with as-prepared primary particle size of 10–13 nm for Ce doping concentrations of 5–50 at%, while square-shape particles with average size around 9 nm were only observed from flame-made CeO₂. The adsorption edge of resulting powder was shifted from 388 to 467 nm as the Ce content increased from 0 to 30 at% and there was an optimal Ce content in association with the maximum absorbance. This effect is due to the insertion of Ce^{3+/4+} in the TiO₂ matrix, which generated an n-type impurity band.

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1. Introduction

Titanium dioxide (TiO₂) has attracted strong attention due to its outstanding mechanical, thermal, electrical and photocatalytic properties. It has found appeal to many potential applications including a semiconductor in dye-sensitized solar cell [1], water and air treatment materials [2,3], photocatalytic splitting of water for green-energy hydrogen fuel [4,5] and catalysts and catalyst supports [6]. Its performance in these applications depends on its physical and chemical properties which are related to synthetic conditions. In order to improve the properties of TiO₂, many research groups attempted to modify TiO₂ visible wavelength by doping by elements [7,8].

Cerium oxide (CeO₂) is rare-earth oxide material used in the fields of photoluminescence, photosensitive material and anti-UV radiation. It has attracted large attention in catalysis due to

its ability to store/release oxygen as an oxygen reservoir via the redox shift between Ce⁴⁺ and Ce³⁺ under oxidizing and reducing conditions. Improvement of photocatalytic activity by addition of Ce atoms in TiO₂ has already been reported [9–11]. The CeO₂–TiO₂ nanopowders have not been used in the photocatalyst field but also in the other catalytic applications [12].

Many techniques have been applied for preparing of TiO₂, CeO₂ and CeO₂–TiO₂ powders like; chemical vapor deposition [13], precipitation [14], magnetron sputtering [15], hydrothermal synthesis [16], sol gel synthesis [17], and spray pyrolysis [18]. Flame synthesis is a commercial process to make nanoparticles in large quantity and low cost [19,20]. Flame spray pyrolysis (FSP) is a technique that is capable of producing a wide variety of product compositions, which can be used such sensors [21], catalysts and catalyst supports [22–24]. TiO₂ and CeO₂ have already been made by FSP with close control of particle size, morphology, and crystallinity [25,26]. Here we report a possibility of rapid, flame synthesis of TiO₂, CeO₂ and CeO₂–TiO₂ powders with characterization of their morphology, particle size and UV absorption with increasing Ce content.

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2. Experimental

2.1. Powder synthesis

Precursor solutions were prepared by dissolving appropriate amounts of titanium (IV) butoxide ($\text{Ti}(\text{OC}_4\text{H}_9)_4$, 97% Aldrich) and ammonium cerium (IV) nitrate ($(\text{NH}_4)_2\text{Ce}(\text{NO}_3)_6$, 99.99% Aldrich) in anhydrous ethanol ($\text{C}_2\text{H}_5\text{OH}$, T.J.). The metal concentration was kept constant at 0.3 M. The Ce contents were varied from 5 to 100 at%.

Nanocrystalline TiO_2 , CeO_2 and CeO_2 – TiO_2 powders were synthesized using flame spray pyrolysis (FSP) technique. In a typical run of the flame spray reactor, the liquid precursor mixture was fed into the center of a methane/oxygen flame by a syringe pump (at precursor feed rate of 5 ml/min in this experiment) and dispersed by oxygen (Thai Industrial Gas Co., Ltd.), to forming the fine spray. At the nozzle, the pressure drop at the capillary tip was kept constant at 1.5 bar by adjusting the orifice gap area. The spray flame was ignited by a smaller flame ring issuing from an annular gap. This premixed methane/oxygen (Thai Industrial Gas Co., Ltd.) supporting flame was kept constant with flow rate of 4.5 L/min and fuel/oxygen ratio of 1.5/3 L methane/L oxygen. A sintered metal plate ring (8 mm wide, starting at a radius of 8 mm) provided an additional oxygen sheath flow (5 L/min) surrounding the supporting flame. Particle products were collected on a glass microfibre filter (Whatman) with the aid of a vacuum pump.

2.2. Powder characterization

X-ray diffraction (XRD) patterns were recorded with a Siemens D5000 using nickel filtered $\text{Cu K}\alpha$ radiation. The crystallite size (d_{XRD}) was determined using the Scherrer equation and α -alumina as the external standard. The rutile content was calculated from diffraction peak by using the equation as shown below:

$$W_R = \left[1 + 0.8 \frac{I_A}{I_R} \right]^{-1} \quad (1)$$

where I_A and I_R represented integrated intensities of anatase and rutile diffraction peak, respectively. The BET surface area, average pore size diameters, and pore size distribution are determined by physisorption of nitrogen (N_2) using a BEL-SORP automated system. The morphology and particle size of powders were observed using a JEOL-JEM 200CX transmission electron microscope. The UV–Vis absorption spectra and band gap energy of nanopowders were obtained by using Lambda 650 UV/Vis spectrophotometer to study in electronic properties. The spectrophotometer was scanned around wavelength of 200–900 nm and obtained by UV WinLab software from Perkin-Elmer. The estimated band gap energy from the absorption spectra for catalysts were found by extrapolation the energy intercept of a plot of $(\alpha h\nu)^{1/2}$ versus $h\nu$ yield E_i for the allowed indirect transition ($n = 2$ from the follow equation)

$$\alpha = \frac{B(h\nu - E_g)^n}{h\nu} \quad (2)$$

where B is constant, E_g the optical band gap, α the absorption coefficient, $h\nu$ the photon energy, and n is an index. XPS experiments were carried out at National Synchrotron Research Center, using a VG Scientific $\alpha 110$ energy analyzer with $\text{Al K}\alpha$ (1486.6 eV). The excitation scan of 400 W, analyzer voltage of 2.6 kV and the pass energy of the analyzer was 50 eV. Peak assignments were calculated using Thermo Avantage software and all spectra are referenced to an adventitious C 1s binding energy of 285.0 eV.

3. Results and discussion

Fig. 1 shows the XRD patterns of the flame-made TiO_2 , CeO_2 and CeO_2 – TiO_2 powder. Flame-made TiO_2 powder exhibited the characteristic peaks of anatase titania with some contamination of rutile phase and flame-made CeO_2 shows a typical diffraction pattern of cubic fluorite CeO_2 . In the CeO_2 – TiO_2 mixed oxides, the anatase phase contents decreased with increasing of Ce content and transformed completely after Ce loading was 30 at%. Additional peaks of cubic CeO_2 were also observed at Ce contents of 5 at% and higher. The diffraction peaks due to cubic fluorite CeO_2 in CeO_2 – TiO_2 mixed oxides appear with the increasing concentration of CeO_2 for 5 at% of Ce.

TEM images of all powders are presented in Fig. 2. The flame-made TiO_2 powders consist of predominately spherical particles with the size around 10–20 nm with a low degree of aggregation. Adding 5–30 at% of Ce had no effect on the particle shape and aggregation state. The square-shape particles with average size around 8.2 nm were observed for pure flame-made CeO_2 .

Table 1 summarizes the physiochemical properties of all FSP-made powder. Rutile contents increased from 17 to 77% as the Ce contents increased from 0 to 10 at% and then only rutile phase was observed after the Ce loading was higher than 30 at%. The crystallite size of anatase phase decreased from 17 to 14 nm as increasing Ce content from 0 to 10 at%. In a similar way, the crystallite size of rutile phase decreased from 18 to

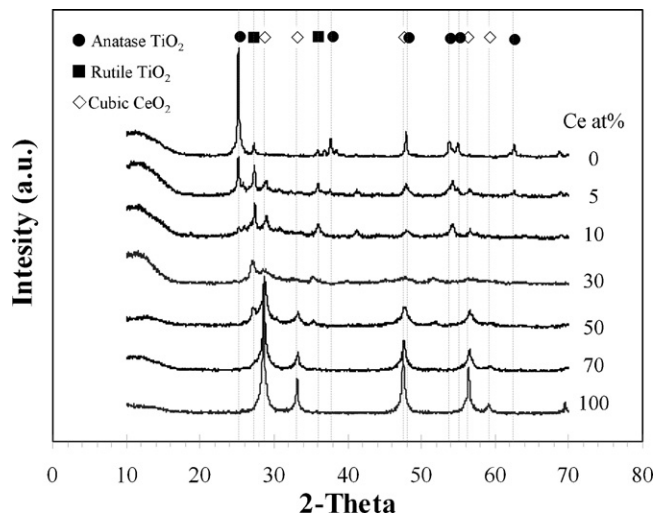


Fig. 1. XRD patterns of the FSP-derived TiO_2 , CeO_2 and CeO_2 – TiO_2 powders.

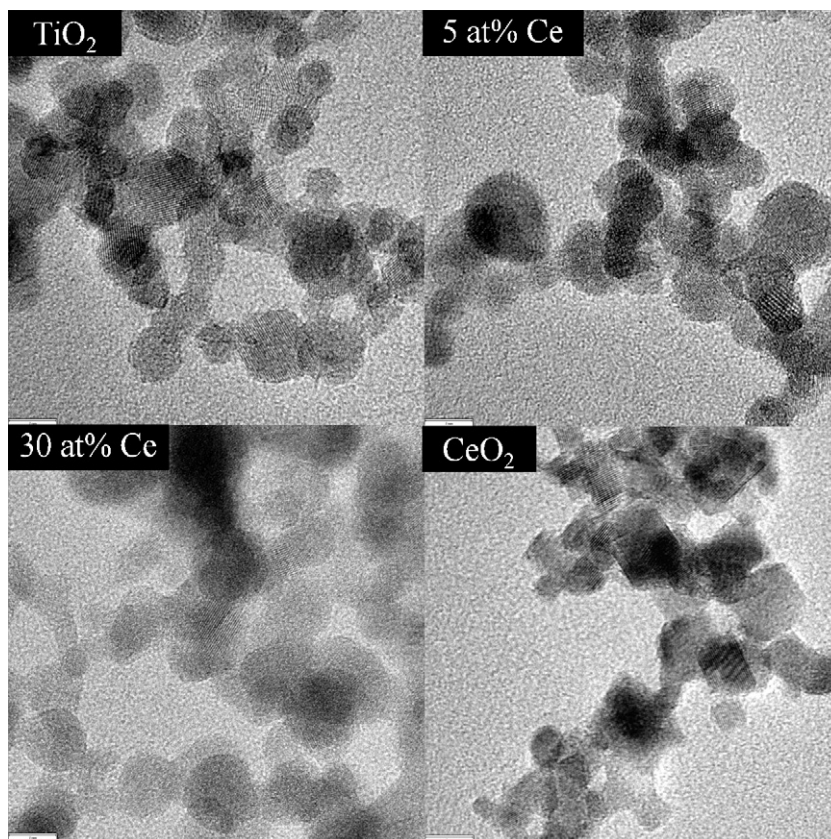


Fig. 2. TEM micrographs of the FSP-derived TiO_2 , CeO_2 and CeO_2 - TiO_2 powders.

7 nm as the increase in Ce content from 0 to 30 at%. The effect of decreasing of TiO_2 crystallite size and increasing of rutile contents after additions of second elements was also found in FSP-made Cu/TiO_2 and Fe/TiO_2 [27,28]. The authors proposed that doping by second element creates a higher number of defects, most likely oxygen vacancies, inside the TiO_2 crystal, thus accelerating the anatase to rutile transition and retard the crystallite growth. In our experiment, the specific surface area of powder decreased from 82.5 to 34.8 m^2/g with Ce doping from 0 to 30 at% and next for pure CeO_2 , BET surface area increased to 77.3 m^2/g .

Normally, the addition of Ce in TiO_2 retards the anatase–rutile phase transformation. It can be explained by the introducing of cerium ions into TiO_2 which produces some deformation of the lattice structure and additional deformation energy, followed by which retards the transition from anatase to

rutile phase [17,29–31]. However, in this work, addition of Ce atom accelerates the rutile phase transformation. This would be explained by the different preparation routes. In literatures, Ce modified TiO_2 particles were normally prepared via wet chemical routes, while the particle formation in FSP process occurred at high temperature at short residence time coupled with high quenching rate in gas phase [32]. Formation of nanoparticles by FSP was considered as follows: the sprayed droplets of precursor solution were evaporated and combusted as soon as they met the flame and released the metal atoms, then nucleation and growth of particles by coagulation and condensation occurred along the axial direction of the flame.

UV–visible spectra of pure TiO_2 , CeO_2 , and CeO_2 - TiO_2 nanoparticles are shown in Fig. 3. The absorption edge, λ° for all catalysts ranged between 467 and 388 nm. The adsorption at 388 nm is characteristic of TiO_2 and the CeO_2 - TiO_2

Table 1
The physiochemical properties of FSP-made powders.

Catalysts	Crystal phase	Anatase:rutile ratio	BET surface area (m^2/g)	Crystallite size		
				Anatase (nm)	Rutile (nm)	CeO_2 (nm)
TiO_2	Anatase–Rutile	17	82.5	17	18	n/a
5 at% CeO_2 - TiO_2	Anatase–Rutile	53	76.6	15	16	n/a
10 at% CeO_2 - TiO_2	Anatase–Rutile	77	74.5	14	9	n/a
30 at% CeO_2 - TiO_2	Rutile– CeO_2	100	34.8	n/a	7	5.5
50 at% CeO_2 - TiO_2	Rutile– CeO_2	100	54.3	n/a	7	10
70 at% CeO_2 - TiO_2	Rutile– CeO_2	100	72.8	n/a	n/a	12
CeO_2	CeO_2	–	77.3	n/a	n/a	7

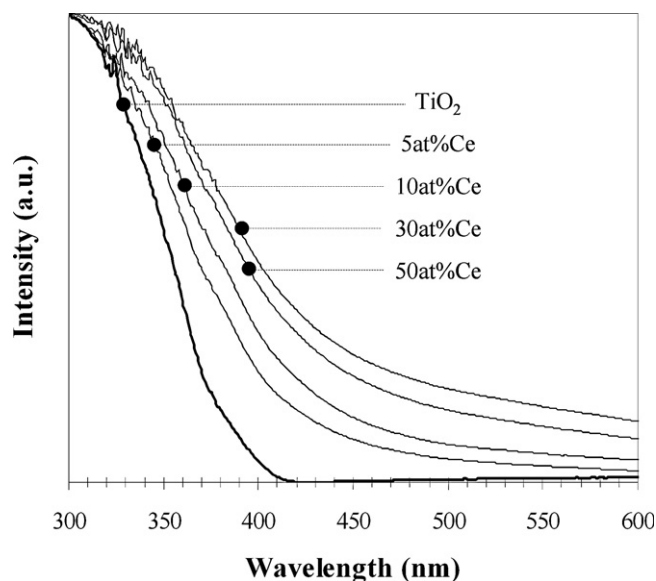


Fig. 3. UV–visible spectra of pure FSP-derived TiO_2 , CeO_2 , and CeO_2 – TiO_2 powders.

nanopowder shifted the adsorption band toward to visible range (300–800 nm). The adsorption edges of pure TiO_2 , 3 at% CeO_2 – TiO_2 , 5 at% CeO_2 – TiO_2 , 10 at% CeO_2 – TiO_2 , and 30 at% CeO_2 – TiO_2 were extended to about 388, 418, 420, 435, and 467 nm, respectively. The absorbance variation increases with increase of cerium content up to 30 at%. For higher Ce contents, the absorbance decrease. Red shifts of flame-made CeO_2 – TiO_2 would be attributed to the charge-transfer between the impurity band and the conduction band of TiO_2 [17]. This would occurred by the replacement of Ti^{4+} ions by $\text{Ce}^{4+/3+}$ sites to form an interfacial phases, which could be deemed and formed impurity band of interfacial with n-type identity. Moreover, the absorbance of the rare earth metal-doped sample was found to increase with increasing rare earth content. This suggest that the solid solution between CeO_2 and TiO_2 can be formed by using FSP technique and the maximum Ce loading amount by this method is 30 at%.

4. Conclusion

Nanocrystalline TiO_2 , CeO_2 and CeO_2 -doped TiO_2 have been successfully prepared by one-step flame spray pyrolysis (FSP). Resulting TiO_2 and CeO_2 -doped TiO_2 nanopowders were composed of single-crystalline spherical particles with as-prepared primary particle size of 10–13 nm for Ce doping concentrations of 5–50 at%. While the square-shape particles with average size around 8.2 nm were only observed in flame-made CeO_2 . The adsorption band of obtained powder was shifted to visible range (300–800 nm) as the Ce doping contents increased to 30 at%.

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