

# Large-scale synthesis of WO<sub>3</sub> nanoplates by a microwave-hydrothermal method

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## Abstract

Tungsten oxide (WO<sub>3</sub>) nanoplates were synthesized by a 270 W microwave-hydrothermal reaction of Na<sub>2</sub>WO<sub>4</sub>·2H<sub>2</sub>O and citric acid (C<sub>6</sub>H<sub>8</sub>O<sub>7</sub>·H<sub>2</sub>O) in deionized water. X-ray diffraction (XRD), scanning electron microscopy (SEM), high resolution transmission electron microscopy (HRTEM), and selected area electron diffraction (SAED) were used to reveal the synthesis of WO<sub>3</sub> complete rectangular nanoplates in the solution of 0.2 g citric acid for 180 min, with O–W–O FTIR stretching modes at 819 and 741 cm<sup>−1</sup>, and two prominent O–W–O Raman stretching modes at 804 and 713 cm<sup>−1</sup>. The 2.71 eV indirect energy gap, and 430–460 nm blue emission wavelength range of WO<sub>3</sub> complete rectangular nanoplates were determined using UV–visible and photoluminescence (PL) spectrometers. The formation mechanism was also proposed according to the experimental results.

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## 1. Introduction

At present, a great deal of effort has been focused on the nanostructured materials with zero-dimensional quantum dots, one-dimensional nanofibers, nanotubes and nanorods, and two-dimensional nanoplates and nanodisks, due to their novel properties. The unique photocatalytic and electrochromic properties of WO<sub>3</sub> [1,2] are generally well known for multiple potential applications, such as H<sub>2</sub>S and NH<sub>3</sub> gas sensing [3,4], and electrochromic and photocatalytic materials [2]. For stoichiometric WO<sub>3</sub>, tungsten ions have +6 valence oxidation state with the empty 5d shells. It is a distorted ReO<sub>3</sub> type, composed of corner and/or edge shared WO<sub>6</sub> octahedrons, caused by the 2nd order Jahn–Teller effect [1,5]. Tungsten oxide is an *n*-type indirect band gap (*E<sub>g</sub>*) semiconductor with *E<sub>g</sub>* of 2.5–2.9 eV [1,2], influenced by morphologies, crystalline degree, structural order–disorder, and synthetic methods.

In the present research, WO<sub>3</sub> nanoplates were synthesized by a microwave-hydrothermal method, without the use of any additives. This process is rapid and benign to the environment, and may lead to large-scale synthesis.

## 2. Experiment

In the present research, 0.3332 g Na<sub>2</sub>WO<sub>4</sub>·2H<sub>2</sub>O and each of 0.1, 0.2, and 0.3 g citric acid (C<sub>6</sub>H<sub>8</sub>O<sub>7</sub>·H<sub>2</sub>O) were dissolved in 40 ml deionized water, to form S1, S2, and S3 solutions. The pH of the solutions was adjusted to 1 using 37% HCl. Each of the solutions was transferred in lab-made autoclaves, which were tightly closed, and processed at 270 W microwave radiation for 30, 60, 90, 120, and 180 min. The final products were then characterized by different methods: X-ray diffractometer (XRD, SIEMENS D500, Germany) operating at 20 kV, 15 mA, and using Cu-K<sub>α</sub> line, in combination with the database of the Joint Committee on Powder Diffraction Standards (JCPDS) [6]; scanning electron microscope (SEM, JEOL JSM-6335F, Japan) operating at 15 kV; transmission electron microscope (TEM, JEOL JEM-2010, Japan), high resolution transmission electron microscope (HRTEM) and selected area electron diffractometer

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(SAED) operating at 200 kV; Fourier transform infrared spectrometer (FTIR, Bruker Tensor 27, Germany) with KBr as a diluting agent and operated in the range of 4000–400  $\text{cm}^{-1}$ ; Raman spectrometer (T64000 HORIBA Jobin Yvon, U.S.A.) using a 50 mW and 514.5 nm wavelength Ar green laser; UV–visible spectrometer (Lambda 25 PerkinElmer, U.S.A) using a UV lamp with the resolution of 1 nm; and photoluminescence (PL) spectrometer (LS 50B PerkinElmer, U.S.A) using a 315 nm excitation wavelength at room temperature.

### 3. Results and discussion

#### 3.1. XRD

Comparing XRD spectra (Fig. 1) of the products synthesized from S2 solution for 30, 60, 90, 120, and 180 min with the JCPDS database [6], they corresponded with pure monoclinic  $\text{WO}_3$  (no. 72-1465) for 180 min long. When the processing time was shortened, mixed phases of monoclinic  $\text{WO}_3$  and orthorhombic  $\text{WO}_3 \cdot \text{H}_2\text{O}$  (no. 84-0886) were detected [6].

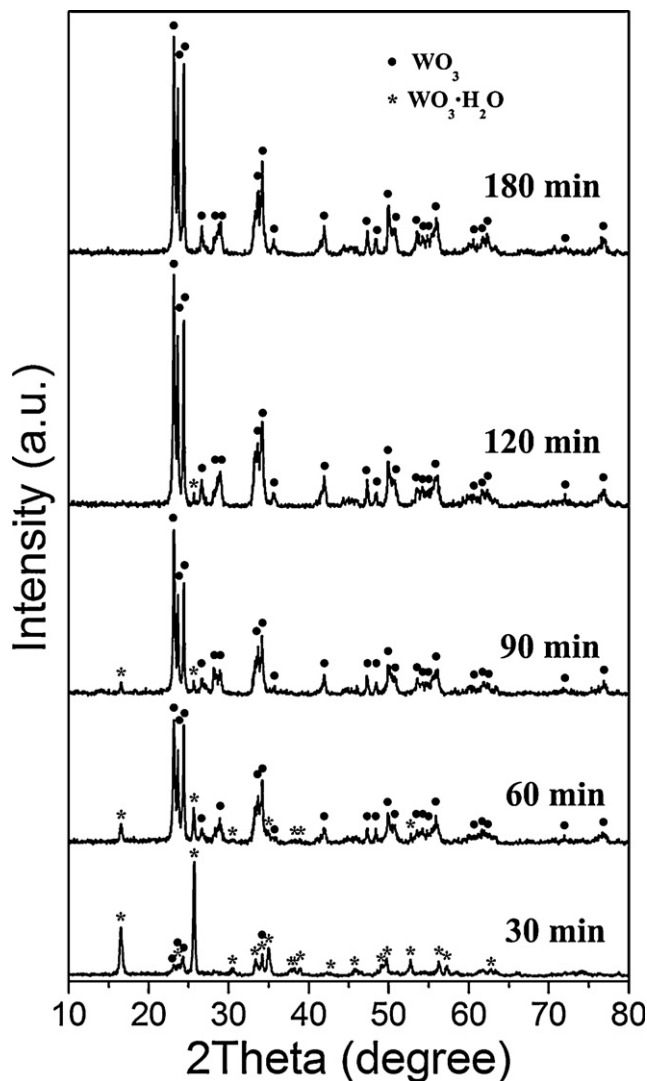


Fig. 1. XRD spectra of the products synthesized from S2 solution for different lengths of time.

#### 3.2. SEM

SEM images of the products synthesized from different solutions and lengths of time are shown in Fig. 2. For the S2 solution and 30 min long (Fig. 2a), the product was irregular nanoplates with different orientations. Upon increasing the processing time from 30 min to 60, 90, 120, and 180 min (Fig. 2b–e) in sequence, they gradually transformed into rectangular nanoplates, and were complete for 180 min long. Upon varying the contents of citric acid (S1, S2, and S3 solutions) at the fixed processing time of 60 min (Fig. 2b, f and g), the product became complete rectangular nanoplates for using 0.3 g citric acid in the S3 solution. The present research proved that  $\text{WO}_3$  morphologies were controlled by the lengths of time and contents of citric acid in the solutions.

#### 3.3. SAED and HRTEM

SAED pattern (Fig. 3a) of the product synthesized from S2 solution for 180 min was indexed [7] to correspond with the (0 2 2), (0 0 2), (0  $\bar{2}$  2), (0 2 0), (0  $\bar{2}$  0), (0 2  $\bar{2}$ ), (0 0  $\bar{2}$ ) and (0  $\bar{2}$   $\bar{2}$ ) planes, specified as single crystalline  $\text{WO}_3$  [6]. For the present analysis, electron beam was in the [1 0 0] direction. The (0 0 2) and (0 2 0) crystallographic planes with the respective 0.384 nm and 0.376 nm spaces at two positions were also detected in HRTEM images (Fig. 3b and c). These two planes were at right angle with each other – in accordance with the interpreted SAED pattern.

#### 3.4. FTIR

FTIR spectra of the products synthesized from S2 solution for different lengths of time are showed in Fig. 4a. Obviously, the major vibrations associated with O–H stretching of residual water were detected at 3613–3134  $\text{cm}^{-1}$ , C=O stretching modes at 1628  $\text{cm}^{-1}$ , C–O stretching modes of carboxyl at 947  $\text{cm}^{-1}$ , O–W–O stretching modes at 819 and 741  $\text{cm}^{-1}$ , and W–O–W stretching modes at 659  $\text{cm}^{-1}$  [5,8,9]. Upon increasing the processing time from 30 min to 60, 90, 120, and 180 min, the O–H and C=O stretching modes were gradually decreased and no longer detected for 180 min long. Still, a little of C–O stretching mode was detected.

#### 3.5. Raman analysis

A definite existence of the products synthesized from S2 solution for different lengths of time was revealed by Raman spectra (Fig. 4b). Well-defined vibration modes centered at 804, 713, 610, 326, 275, 243, and 184  $\text{cm}^{-1}$  were detected. Comparing Raman spectra of the present analysis and the  $\text{WO}_3$  phase reported in the literatures [1,3,5,10–12], these products are monoclinic structure formed by O–W–O microcrystals connected with each other by W–O–W bonds, with the terminal W=O bonds at their surfaces. Two main peaks are typical O–W–O stretching modes of crystalline  $\text{WO}_3$  at 804  $\text{cm}^{-1}$  (symmetric) for the shorter bonds, and 713  $\text{cm}^{-1}$  (asymmetric) for the longer bonds. Weak peaks at 610  $\text{cm}^{-1}$  were assigned as the O–W–O stretching modes of  $\text{WO}_3$ .

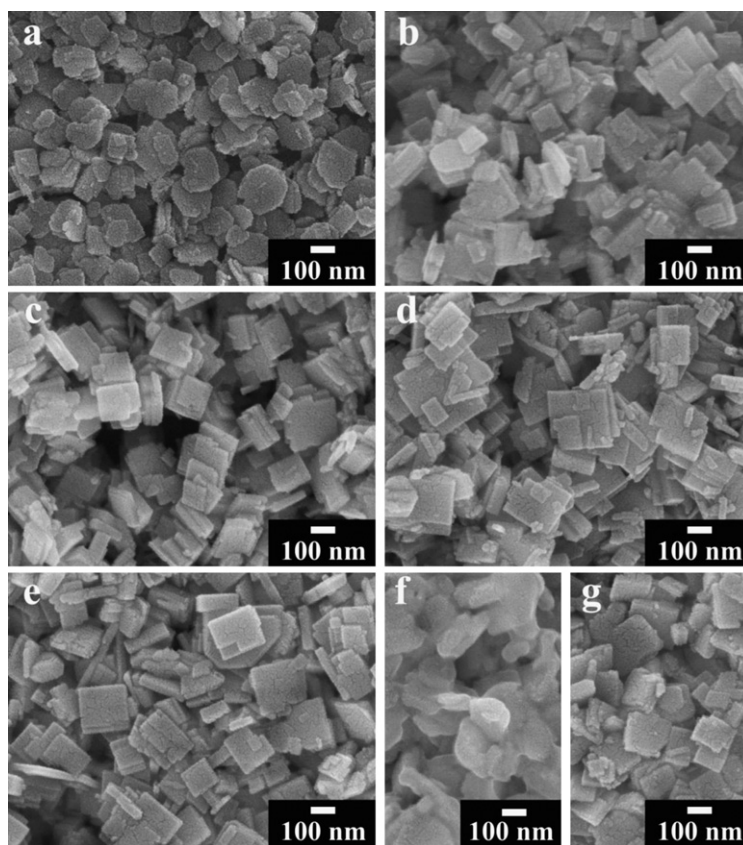


Fig. 2. SEM images of the products synthesized from (a–e) S2 solution for 30, 60, 90, 120, and 180 min, and (f and g) S1 and S3 solutions for 60 min, respectively.

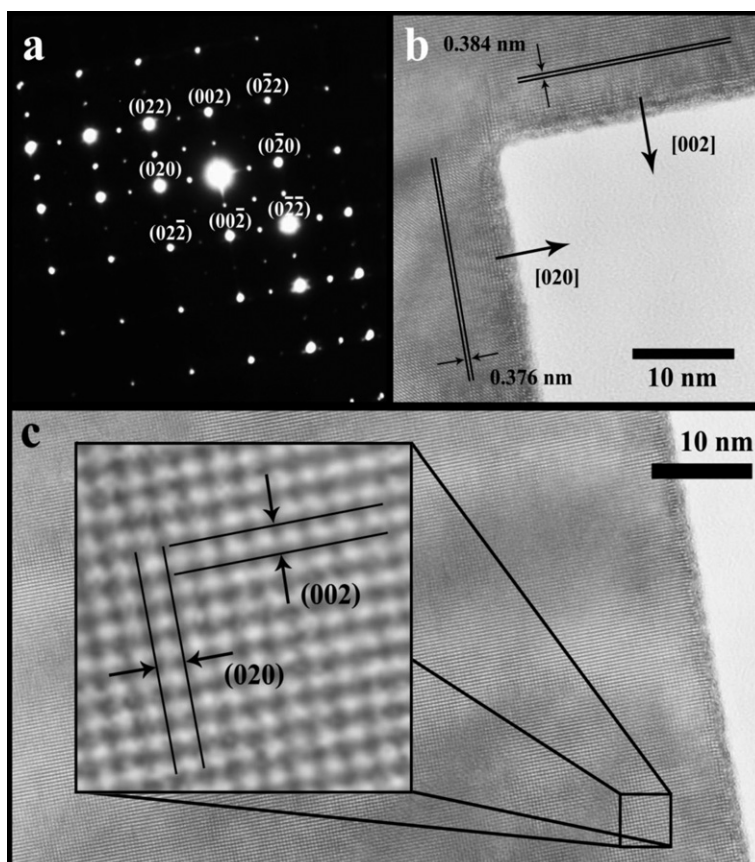


Fig. 3. (a) SAED pattern, and (b and c) HRTEM images of the product synthesized from S2 solution for 180 min.

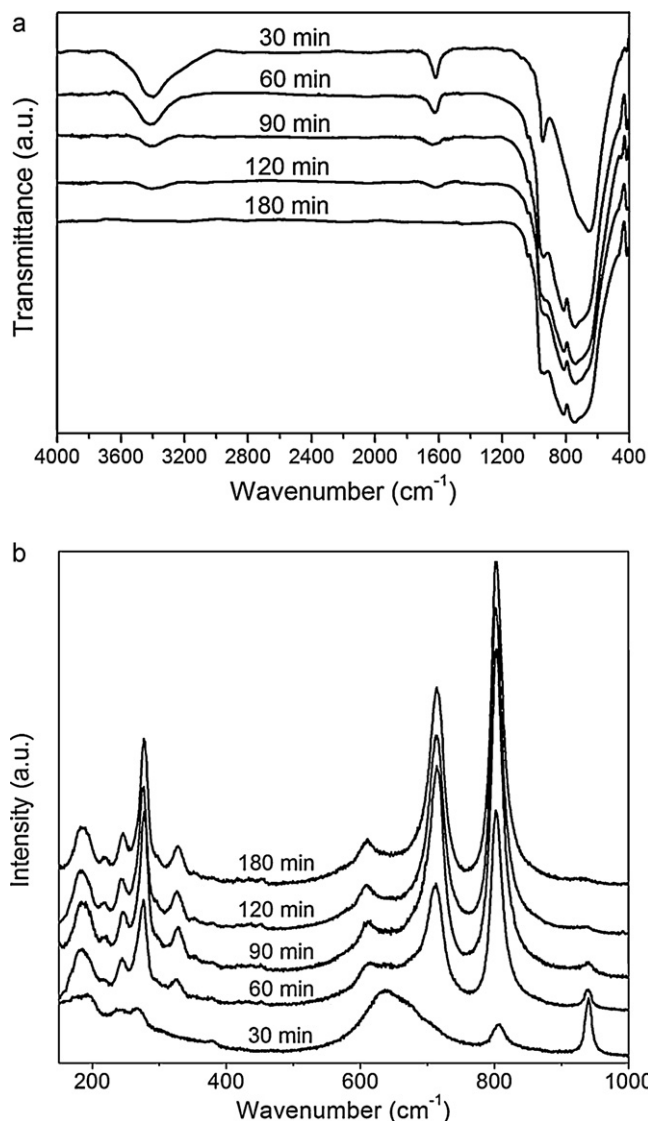
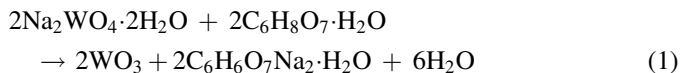


Fig. 4. (a) FTIR and (b) Raman spectra of the products synthesized from S2 solution for different lengths of time.

[5]. Those at 326, 275, and 243 cm<sup>-1</sup> are assigned as W–O–W bending modes of the bridging oxide. The peak at 184 cm<sup>-1</sup> is attributed to the lattice vibration of crystalline WO<sub>3</sub>. Additional mode at 941 cm<sup>-1</sup> belonging to W=O stretching [5] of the product processed for 30 min was detected. It was lessened for longer processing times, and no longer detected for 180 min long.

### 3.6. Proposed mechanism

During microwave-hydrothermal processing, Na<sub>2</sub>WO<sub>4</sub>·2H<sub>2</sub>O reacted with citric acid (C<sub>6</sub>H<sub>8</sub>O<sub>7</sub>·H<sub>2</sub>O) to form WO<sub>3</sub> molecules, which nucleated, and grew as WO<sub>3</sub> nanoplates.



Initially, the nanoplates were irregular. They became rectangular and more complete, when the processing time was lengthened (Fig. 5). It should be noted that the nanoplates

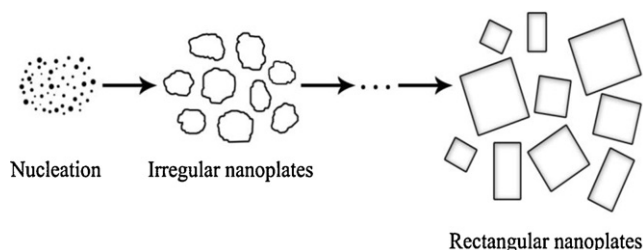


Fig. 5. Schematic diagram for the formation of rectangular nanoplates.

formed in different sizes, controlled by the initiation (born) of nuclei, aging time, growth rate, and surrounding concentration.

### 3.7. PL emission

The optical property (Fig. 6a) of WO<sub>3</sub> nanoplates synthesized from S2 and S3 solutions for different lengths of time was studied using 315 nm excitation wavelength. In this research, the emission peaks presented the bands over the 350–600 nm range with a strong blue emission peaks centered at the 430–460 nm wavelength range – in accordance with the reports of Wang et al. [2], and Lee et al. [13]. In the present research,

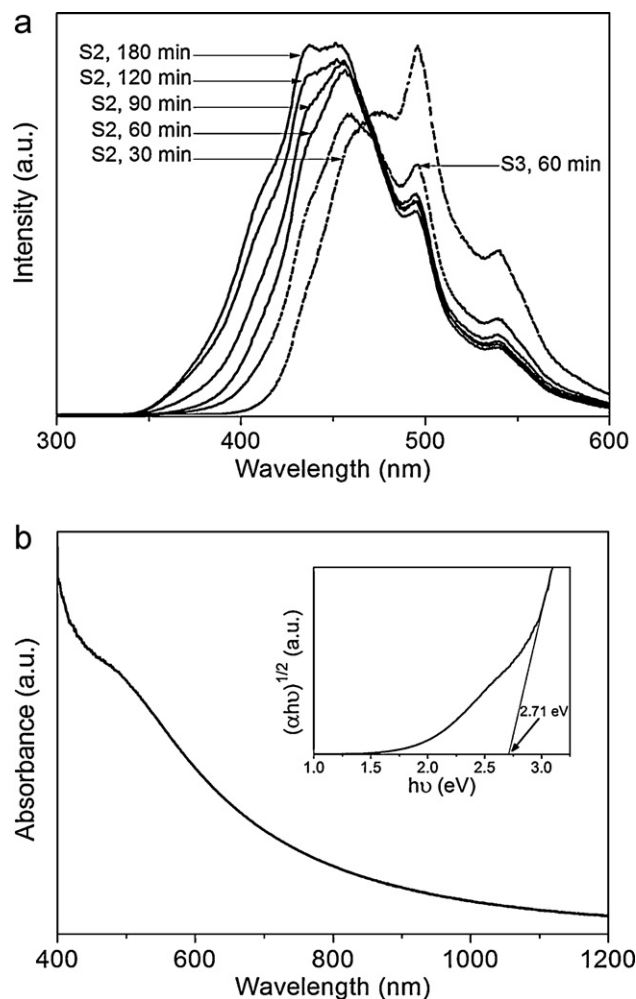


Fig. 6. (a) PL spectra of the products synthesized from S2 and S3 solutions for different lengths of time. (b) UV–visible absorption of the product synthesized from S2 solution for 180 min.



this PL emission was caused by electron–hole radiative recombination. Their shoulders were caused by localized states in the band gap due to impurities [13]. Some difference in the emission wavelengths was detected, caused by the phase, purity, morphologies, and synthesized conditions.

### 3.8. UV–visible absorption

UV–visible absorption (Fig. 6b) of the product synthesized from S2 solution for 180 min. By using Wood and Tauc equation [14] below.

$$\alpha h\nu = (h\nu - E_g)^n \quad (2)$$

where  $\alpha$  is the absorbance,  $h$  is the Planck constant,  $\nu$  is the photon frequency,  $E_g$  is the energy gap, and  $n$  is the pure numbers associated with the different types of charged transitions. For  $n = 1/2$ , 2,  $3/2$  and 3, the transitions are the direct allowed, indirect allowed, direct forbidden, and indirect forbidden, respectively. The absorption was controlled by two photon energy ( $h\nu$ ) ranges relative to energy gap ( $E_g$ ). For  $h\nu > E_g$ , absorption is linearly increased with the increasing of photon energy – caused by the transition of electrons from the topmost occupied state of valence band to the bottommost unoccupied state of conduction band. For  $h\nu < E_g$ , the absorption curve became different from linearity, caused by the charged transition relating to defects. In the present research, the indirect  $E_g$  was determined to be 2.71 eV (457.5 nm) – in accordance with the report of Kuzmin et al. [1], Wang et al. [2], and Su et al. [15]. It should be noted that this indirect  $E_g$  (457.5 nm) was in accordance with the 430–460 nm PL blue emission range as well.

### 4. Conclusions

WO<sub>3</sub> nanoplates were successfully synthesized by microwave-hydrothermal reaction of Na<sub>2</sub>WO<sub>4</sub>·2H<sub>2</sub>O and citric acid (C<sub>6</sub>H<sub>8</sub>O<sub>7</sub>·H<sub>2</sub>O) in deionized water for different lengths of time. By increasing the processing time of the 0.2 g citric acid solution to be 180 min, the product became pure monoclinic WO<sub>3</sub> with complete rectangular nanoplates. The indirect energy gap was determined to be 2.71 eV, and photoemission to be 430–460 nm wavelength range.

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