

## TiO<sub>2</sub> nanofibers doped with rare earth elements and their photocatalytic activity

M. Shamshi Hassan<sup>a</sup>, Touseef Amna<sup>b</sup>, O-Bong Yang<sup>c</sup>, Hyun-Chel Kim<sup>d</sup>, Myung-Seob Khil<sup>a,\*</sup>

<sup>a</sup>Department of Organic Materials and Fiber Engineering, Chonbuk National University, Jeonju 561-756, Republic of Korea

<sup>b</sup>Center for Healthcare Technology Development, Chonbuk National University, Jeonju 561-756, Republic of Korea

<sup>c</sup>School of Semiconductor and Chemical Engineering, Chonbuk National University, Jeonju 561-756, Republic of Korea

<sup>d</sup>Korean Institute for Knit Industry, Iksan 570-330, Republic of Korea

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### Abstract

In the present study rare earth doped (Ln<sup>3+</sup>-TiO<sub>2</sub>, Ln = La, Ce and Nd) TiO<sub>2</sub> nanofibers were prepared by the sol–gel electrospinning method and characterized by XRD, SEM, EDX, TEM, and UV-DRS. The photocatalytic activity of the samples was evaluated by Rhodamine 6G (R6G) dye degradation under UV light irradiation. XRD analysis showed that all the synthesized pure and doped titania nanofibers contain pure anatase phase at 500 °C but at 700 °C it shows both anatase and rutile phase. XRD result also shows that Ln<sup>3+</sup>-doped titania probably inhibits the phase transformation. The diameter of nanofibers for all samples ranges from 200 to 700 nm. It was also observed that the presence of rare-earth oxides in the host TiO<sub>2</sub> could decrease the band gap and accelerate the separation of photogenerated electron–hole pairs, which eventually led to higher photocatalytic activity. To sum up, our study demonstrates that Ln<sup>3+</sup>-doped TiO<sub>2</sub> samples exhibit higher photocatalytic activity than pure TiO<sub>2</sub> whereas Nd<sup>3+</sup>-doped TiO<sub>2</sub> catalyst showed the highest photocatalytic activity among the rare earth doped samples.

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**Keywords:** Ln-doped titania; Electrospinning; Nanofibers; Dye degradation

### 1. Introduction

Among various semiconductors titania is widely used photocatalyst due to its strong oxidizing power, stable at different pH and favorable band gap energy. However titania also has some disadvantages like low surface area, fast recombination and wavelength maximum lies in UV region. Adsorption behavior and the separation efficiency of electron–hole pairs are two important factor on which photodegradation of TiO<sub>2</sub> strongly depends [1]. High specific surface area of catalysts generally increases the adsorption efficiency. Surface recombination can be avoided by transition metals doping [2], coupling with other semiconductors [3], noble metals deposition [4] and rare earth ions doping [5,6], whereas, volume recombination can be reduced by preparing titania in nanoscale. In recent years, many groups have examined the effect of metal doping on photocatalytic properties of TiO<sub>2</sub>. Doping with

lanthanides is a method to shift the maximum of absorption as well as enhance the photocatalytic activity. In addition to the methods of catalyst preparation, the photoactivity of the doped TiO<sub>2</sub> catalysts depends substantially on the nature of the dopant and its concentration [7].

One dimensional porous titania nanofibers consisting of well crystallized anatase nanocrystals is very advantageous for developing active photocatalytic materials for environmental purification and green energy [8]. Such materials are usually having large surface area and high porosity. A larger surface area provides more surface active sites for the adsorption of reactants molecules, which thus make the photocatalytic process more efficient [9]. Nanofibers have large surface area and regular morphology on nanometer scale. Due to their one dimensional morphology, the nanofibers sediment readily from their aqueous dispersions, compared with P25.

In order to investigate the light absorption ability of TiO<sub>2</sub> doped with different lanthanum metal and the effects of doping on the photocatalytic activity, the present work was designed to prepare Ln-doped TiO<sub>2</sub> nanofibers by electrospinning and characterized by XRD, TEM, and DRS patterns. Finally,

\* Corresponding author. Tel.: +82 63 270 4635; fax: +82 63 270 4249.

E-mail address: [mskhil@jbnu.ac.kr](mailto:mskhil@jbnu.ac.kr) (M.-S. Khil).

pristine and doped  $\text{TiO}_2$  were tested for photocatalytic degradation of R6G.

## 2. Experimental details

In a typical procedure [10], PVAc solution (18 wt%) was prepared by dissolving PVAc in DMF (2,2-dimethyl formamide) under magnetic stirring for 6 h. 5 g of titanium isopropoxide (TTIP) was taken in a separate bottle and add a few drops of acetic acid till the solution became transparent. Appropriate amount (1.0 wt%) of alcoholic solution of  $\text{Ln}(\text{NO}_3)_3$  was added into the TTIP solution. Then 6 g of PVAc solution was mixed slowly into TTIP under vigorous stirring. The obtained solution was put in a 10 ml syringe with a stainless steel needle. A copper pin connected to a high voltage generator was inserted in the solution as a positive terminal whereas a ground iron drum covered by polyethylene sheet served as counter electrode. The distance between the syringe needle tip and collector was fixed at 18 cm at 15 kV. The as synthesized mat of composite fiber was initially dried at 80 °C for 24 h under vacuum, and then calcined at 500 and 700 °C for 2 h in air atmosphere with a heating rate of 2 °C/min.

The crystallinity of pristine  $\text{TiO}_2$  and Ln-doped  $\text{TiO}_2$  catalysts were analyzed by X-ray powder diffraction recorded on a Rigaku/Max-3A X-ray diffractometer with  $\text{CuK}\alpha$  radiation ( $\lambda = 1.540 \text{ \AA}$ ). To examine the microstructure, the synthesized products were characterized by scanning electron microscopy (SEM, JEOL JSM6700, Japan) and transmission electron microscope (TEM, JEOL JEM-2010). The light absorbance of the samples was measured by using UV–vis diffused reflectance spectrum (UV-DRS, 525 Shimadzu).

The photocatalytic degradation was carried out by mixing 15 mg of the pure  $\text{TiO}_2$  or Ln-doped  $\text{TiO}_2$  photocatalysts into the 100 ml of (10 ppm) R6G aqueous solution under continuous stirring. The experiments were performed at room temperature and prior to irradiation, the slurry was aerated for 30 min to reach adsorption equilibrium followed by UV irradiation. Aliquots were withdrawn from the suspension at specific time intervals and centrifuged immediately at 12,000 rpm. The absorbance of the R6G solution was measured with a UV–vis spectrophotometer (Shimadzu UV-3101).

## 3. Results and discussion

Fig. 1 shows the XRD patterns of pure and Ln-doped titania samples at 500 °C. The crystalline phase of all pure and doped titania is showing the anatase phase [11] at 500 °C. Xu et al. [12] reported that rare earth metal salts can be changed into rare earth oxides ( $\text{La}_2\text{O}_3$ ) during the calcination process. Therefore, it can be inferred that the  $\text{Ln}^{3+}$  ions were dispersed on the surface of  $\text{TiO}_2$  in the form of small  $\text{Ln}_2\text{O}_3$  clusters. When the samples were calcined at 700 °C, the XRD spectra show the presence of rutile phase in addition to anatase phase (Fig. 2). It can be seen that the intensity of anatase phase decreased, while the intensity of rutile phase increased as the temperature was raised. The percentage of anatase ( $X_A$ ) in the sample can be estimated from the respective integrated XRD peak intensity

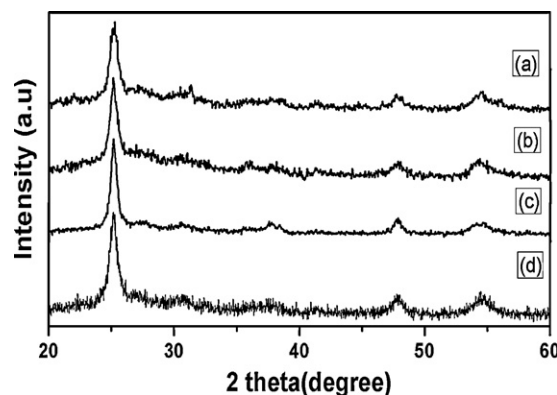


Fig. 1. XRD pattern of (a)  $\text{TiO}_2$ , (b)  $\text{La-TiO}_2$ , (c)  $\text{Ce-TiO}_2$ , and (d)  $\text{Nd-TiO}_2$  nanofibers at 500 °C.

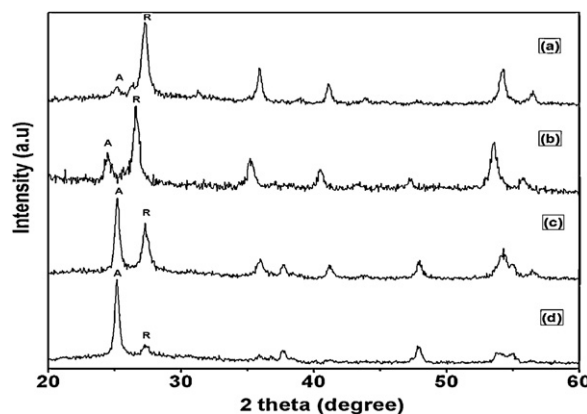


Fig. 2. XRD pattern of (a)  $\text{TiO}_2$ , (b)  $\text{La-TiO}_2$ , (c)  $\text{Ce-TiO}_2$ , and (d)  $\text{Nd-TiO}_2$  nanofibers at 700 °C.

using the following equation [13],  $X_A (\%) = 100 / (1 + 1.265 I_R / I_A)$ , where  $I_A$  represents the intensity of anatase peak and  $I_R$  is the intensity of rutile peak and  $X_A$  is the weight percentage of anatase in the sample. The percentage of rutile phase is more in bare titania than lanthanide ion doped titania (Table 1). From this result may say that  $\text{Ln}^{3+}$  doping probably inhibits the phase transformation. From the (1 0 1) peak of anatase  $\text{TiO}_2$  at 500 °C, the average size of crystallite ( $D$ ) was calculated (Table 1) using the Scherrer equation:  $D = K\lambda / \beta \cos\theta$  where  $\beta$  is the full width half maximum (FWHM) of the  $2\theta$  peak,  $K$  is the shape factor of the particles (it equals to 0.95),  $\theta$  and  $\lambda$  are the incident angle and the wavelength of the X-rays, respectively. However, no diffraction peaks of rare earth oxides appeared even when the samples were calcined at 700 °C, which might be because the content of rare earth oxides was below the detection limit of XRD analysis and large difference between the ionic radii of  $\text{Ti}^{4+}$  and lanthanide ions ( $\text{Ln}^{3+}$ ) [14].

Table 1  
Comparison of properties of titania and doped titania nanofibers.

Sample	$\text{TiO}_2$	$\text{TiO}_2\text{-La}$	$\text{TiO}_2\text{-Ce}$	$\text{TiO}_2\text{-Nd}$
B.Gap (eV)	3.48	3.17	3.14	3.1
Cryst. size (nm)	29.54	12.51	11.49	10.2
$X_A (\%)$	12.5	35	62.5	86.8

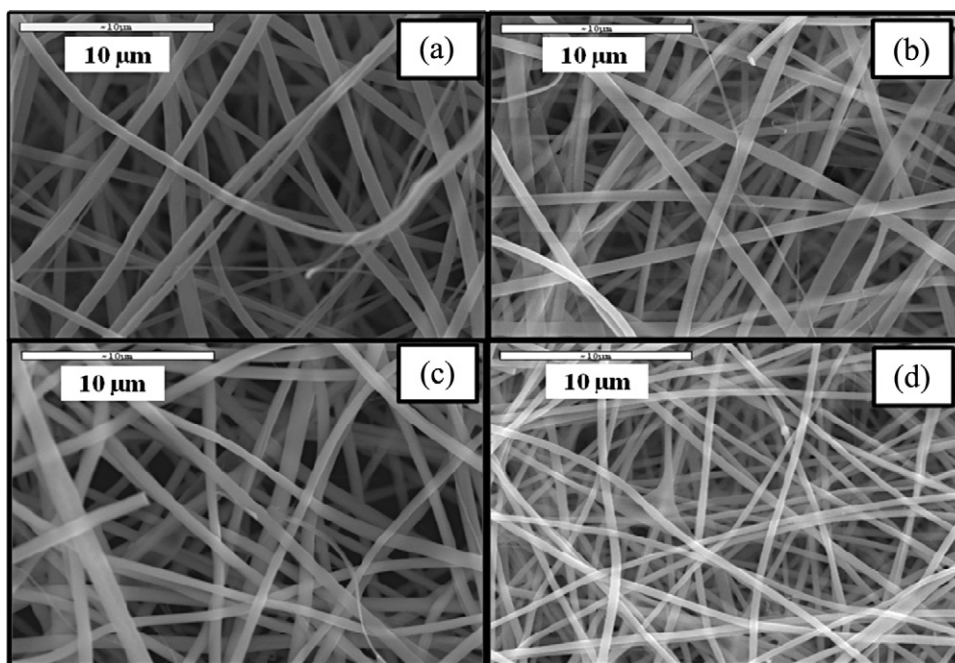


Fig. 3. SEM micrographs of (a)  $\text{TiO}_2$ , (b)  $\text{La-TiO}_2$ , (c)  $\text{Ce-TiO}_2$ , and (d)  $\text{Nd-TiO}_2$  nanofibers.

The SEM images of the synthesized  $\text{TiO}_2$  and rare earth doped  $\text{TiO}_2$  photocatalyst samples are shown in Fig. 3. Each individual nanofiber has smooth surface morphology and uniform diameter. Fig. 4 shows the energy-dispersive X-ray spectroscopy (EDX) analysis of the Ln-doped titania nanofibers. It has been observed that the sample contains Ti, Ln (Ln = Nd, Ce, La) and O; no other element impurity is detected, indicating the final product is free of impurity and composed of  $\text{Ln}_2\text{O}_3$  and  $\text{TiO}_2$ . The microstructures of the nanofibers were further investigated by TEM. Fig. 5 shows typical TEM images

of pristine and doped titania nanofibers at  $700^\circ\text{C}$ . Typical porous structures of the nanofibers were observed which is known to possess superior photocatalytic activity. The average nanofibers diameter of all the samples was found to be around 250 nm, 260 nm, 700 nm and 340 nm for  $\text{TiO}_2$ ,  $\text{La-TiO}_2$ ,  $\text{Ce-TiO}_2$  and  $\text{Nd-TiO}_2$  respectively. According to the crystal lattice parameters,  $\text{Ln}_2\text{O}_3$  and  $\text{TiO}_2$  cannot combine in a single crystal. Accordingly, a main  $\text{TiO}_2$  nanofiber doped with  $\text{Ln}_2\text{O}_3$  nanoparticles is the expected structure of the obtained product. To affirm this hypothesis, HR-TEM analysis of Nd-doped  $\text{TiO}_2$

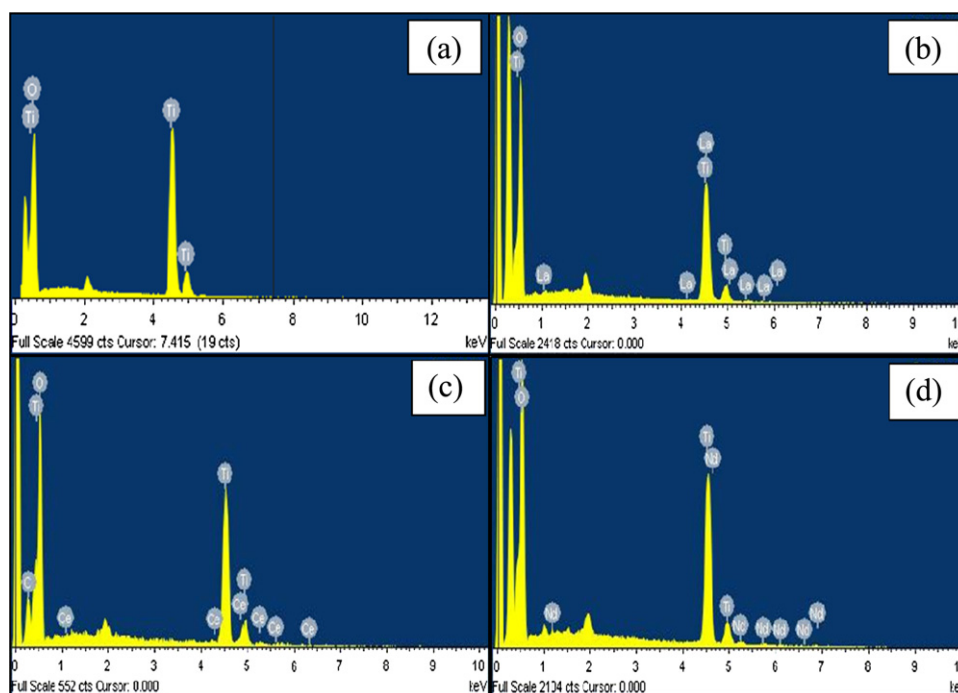


Fig. 4. EDX spectra of (a)  $\text{TiO}_2$ , (b)  $\text{La-TiO}_2$ , (c)  $\text{Ce-TiO}_2$ , and (d)  $\text{Nd-TiO}_2$  nanofibers.



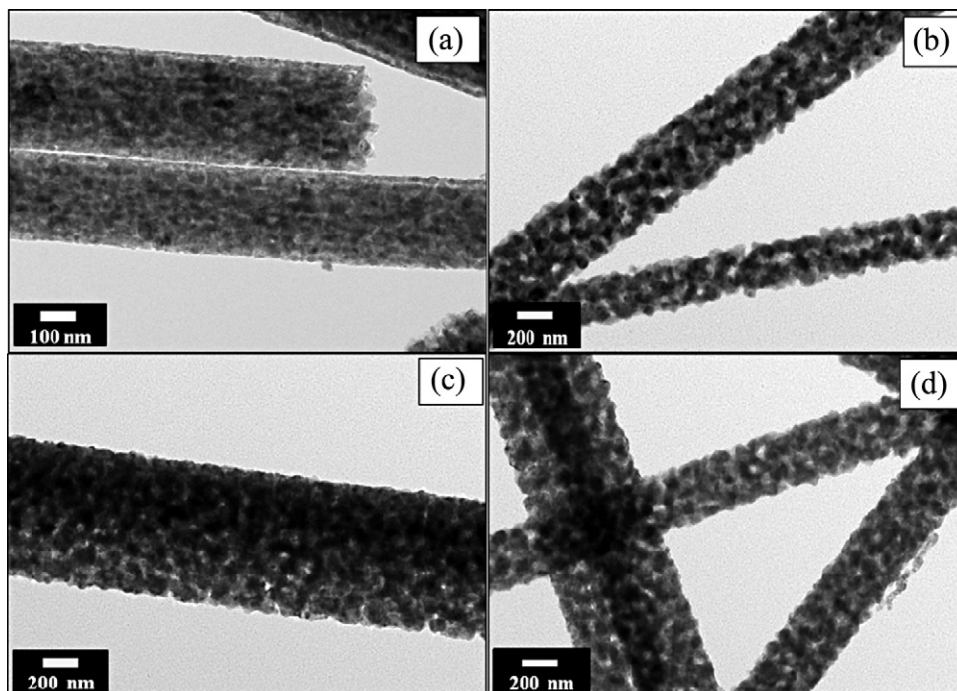


Fig. 5. TEM micrographs of (a)  $\text{TiO}_2$ , (b)  $\text{La-TiO}_2$ , (c)  $\text{Ce-TiO}_2$ , and (d)  $\text{Nd-TiO}_2$  nanofibers.

has been conducted (Fig. 6). As shown in the HR-TEM image (Fig. 6), the black nanoparticles do have different crystal lattice parameters than the matrix as the crystal plane are closer, so one can claim that these nanoparticles represent  $\text{Nd}_2\text{O}_3$ . Moreover, the selected area electron diffraction (SAED) pattern confirms the good crystallinity of the sample (inset Fig. 6).

The DRS spectra of pure  $\text{TiO}_2$  and  $\text{Ln}^{3+}$ -doped  $\text{TiO}_2$  samples calcined at  $700^\circ\text{C}$  are shown in Fig. 7. In comparison with those of pure  $\text{TiO}_2$ , the optical absorption edges of the  $\text{Ln}^{3+}$ - $\text{TiO}_2$  samples are shifted toward longer wavelength in the order of  $\text{Nd}^{3+}\text{-TiO}_2 > \text{Ce}^{3+}\text{-TiO}_2 > \text{La}^{3+}\text{-TiO}_2 > \text{TiO}_2$  photocatalyst. This red shift in the optical absorption band is due to the presence of the rare-earth ions, and it can be attributed to a charge-transfer transition between the lanthanide f electrons and the conduction or valence band of the host  $\text{TiO}_2$  [15].

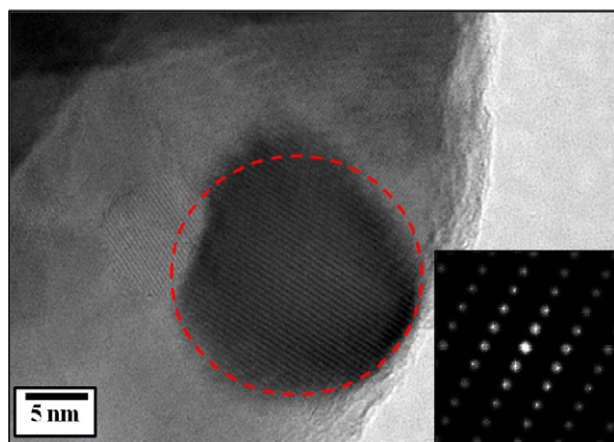


Fig. 6. High resolution TEM (HR-TEM) of  $\text{Nd}$ -doped  $\text{TiO}_2$ , the inset shows its SAED pattern.

Compared with the other  $\text{Ln}^{3+}\text{-TiO}_2$  samples, the  $\text{Nd}^{3+}\text{-TiO}_2$  photocatalyst had the highest absorption intensity and largest red shift, which indicates that the doping of  $\text{Nd}^{3+}$  should have the best photoresponse in the UV region, as well as the highest photocatalytic activity.

To investigate the photocatalytic activities of the  $\text{Ln}$ -doped titania samples, photocatalytic degradation experiments were carried out using R6G in an aqueous suspension under UV light irradiation. Fig. 8 shows the results of the photocatalytic degradation of R6G in presence of different rare earth doped  $\text{TiO}_2$  catalysts with the same dopant content (1.0 wt%) for 60 min of irradiation. The different rare earth loaded photocatalysts exhibited higher photocatalytic degradation activities than pure  $\text{TiO}_2$ , indicating the promoting effect of the rare earth modification. It has been reported in the literature that the photocatalytic activities of catalysts depend on many factors, such as crystallinity, surface properties, absorption

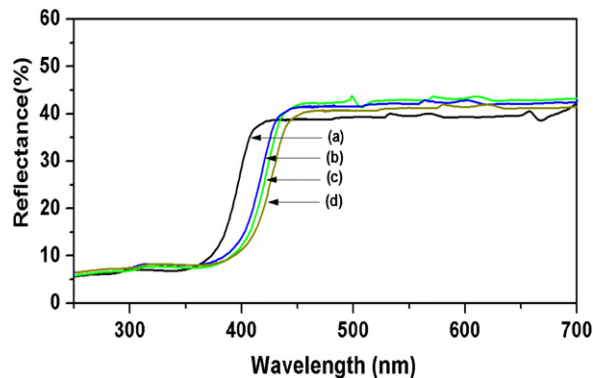


Fig. 7. UV-vis DRS spectra of (a)  $\text{TiO}_2$ , (b)  $\text{La-TiO}_2$ , (c)  $\text{Ce-TiO}_2$ , and (d)  $\text{Nd-TiO}_2$  nanofibers.

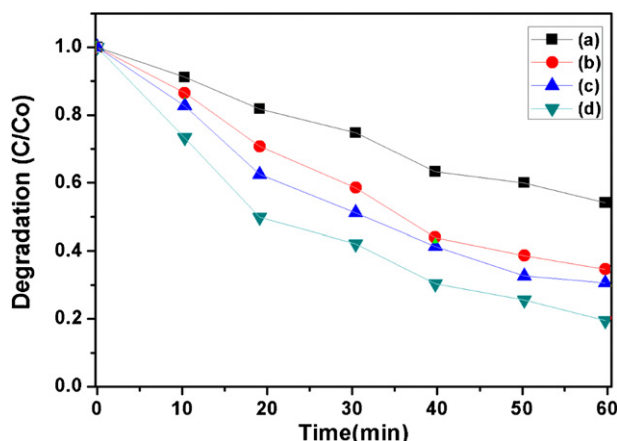


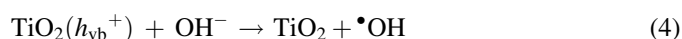
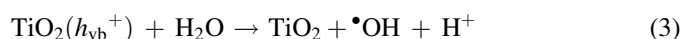
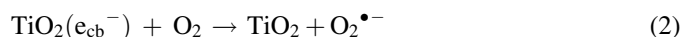
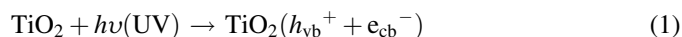
Fig. 8. Photocatalytic degradation of R6G over the different rare earth-doped TiO<sub>2</sub> catalysts (a) TiO<sub>2</sub>, (b) La-TiO<sub>2</sub>, (c) Ce-TiO<sub>2</sub>, and (d) Nd-TiO<sub>2</sub> nanofibers with the same doped contents (1.0 wt%).

properties, and so on [16]. In this case, the enhanced photocatalytic activity of Ln<sup>3+</sup>-TiO<sub>2</sub> can be explained that the presence of rare earth ions shifts the absorption edge to a longer wavelength in the order Nd<sup>3+</sup>-TiO<sub>2</sub> > Ce<sup>3+</sup>-TiO<sub>2</sub> > La<sup>3+</sup>-TiO<sub>2</sub> > TiO<sub>2</sub> (Fig. 7), which is in consistent with the order of the photocatalytic performances. A larger red shift indicates that a sample can absorb more photons. Therefore, the red shift in the absorption band is favorable for photocatalytic reaction. Our findings are in good agreement with the results of Fan et al. [17] and Zhang et al. [18]. Moreover, for Ln<sup>3+</sup>-doped TiO<sub>2</sub> samples, the presence of lanthanide ions could increase the interfacial charge transfer and inhibit the recombination of electron-hole pairs. As for the rare earth modified TiO<sub>2</sub> photocatalysts, lanthanide ions could act as effective electron scavengers to trap the conduction band electrons of TiO<sub>2</sub> [19]. In the presence of lanthanide loaded TiO<sub>2</sub> catalysts, the rare earth oxides present on the surface of the host TiO<sub>2</sub> could act as electron traps, which could enhance electron-hole separation and increase the quantum efficiency. Therefore, the photogenerated electrons could easily transfer from the interior to the surface of the catalyst, which would promote photocatalytic reaction.

On the basis of the above analysis, it is suggested that the enhanced photocatalytic activity of rare earth doped TiO<sub>2</sub> could be a synergetic effect of many factors, including red shifts to longer wavelengths, and enhanced rates of interfacial charge transfer. In the present study, it is worth mentioning that Nd<sup>3+</sup>-doped TiO<sub>2</sub> exhibited the highest dye degradation (~80%) among all photocatalysts. This may be attributed to the high percentage of anatase phase in the sample (Table 1) which is considered to have better photocatalytic activity than the rutile phase [20] and largest red-shift in the optical absorption band were exhibited by the Nd<sup>3+</sup>-loaded TiO<sub>2</sub> (Fig. 7).

Previously, the detailed mechanisms involved in semiconductor photocatalysis have been studied and reviewed [21–24]. Usually it is considered that the process is initiated by the band-gap excitation of TiO<sub>2</sub> under UV illumination ( $\lambda < 385$  nm). Excited by energetic photons, electrons (e<sup>-</sup>) are transferred from the valence band to the conduction band of TiO<sub>2</sub>, thus

generating a hole (h<sup>+</sup>) in the valence band (Eq. (1)). The electrons and holes diffuse to the surface of TiO<sub>2</sub> and react with the adsorbed O<sub>2</sub>, water and surface hydroxide group, respectively, leading to the production of O<sub>2</sub><sup>-</sup> (Eq. (2)) and •OH (Eqs. (3) and (4)). The resulting •OH radical, being a very strong oxidizing agent (standard redox potential +2.8 V) can oxidize dyes to the mineral end-products. According to this, the relevant reactions at the semiconductor surface causing the degradation of dyes can be expressed as follows:



where  $h\nu$  is photon energy required to excite the semiconductor electron from the valence band (VB) region to conduction band (CB) region.

#### 4. Conclusions

Pure and Ln-doped TiO<sub>2</sub> nanofibers were successfully synthesized by sol-gel electrospinning method and were assayed for photocatalytic activity. Ln<sup>3+</sup>-doped TiO<sub>2</sub> samples were found to exhibit higher photocatalytic activities than pure TiO<sub>2</sub> because of the synergetic effect of red shifts to longer wavelengths and specific characteristics of lanthanide ions. Nd<sup>3+</sup>-doped TiO<sub>2</sub> showed the highest activity among all the rare earth doped samples studied.

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