

## Short communication

Preparation and characterization of  $\text{Eu}^{3+}$ -doped fluorapatite nanoparticles by a hydrothermal methodYuxiu Sun<sup>a,\*</sup>, Hua Yang<sup>a</sup>, Dongliang Tao<sup>b</sup><sup>a</sup>Tianjin Key Laboratory of Structure and Performance for Functional Molecules, College of Chemistry, Tianjin Normal University, Xiqing district binshui west road 393, Tianjin 300387, PR China<sup>b</sup>College of Chemistry and Chemical Engineering, Fuyang Normal College, Anhui 236041, PR China

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## Abstract

In this study, luminescent  $\text{Eu}^{3+}$ -doped fluorapatite ( $\text{Eu}^{3+}$ :FA) was successfully synthesized by a hydrothermal process by using  $\text{Na}_2\text{EDTA}$ /citric acid (CA) as the chelating agent. X-ray diffraction (XRD), transmission electron microscopy (TEM) and photoluminescence spectra (PL) were utilized to characterize the obtained nanoparticles. The results showed that  $\text{Eu}^{3+}$  ions entered the FA lattice and occupied  $\text{Ca}^{2+}$  sites, which resulted in a decrease in the values of the lattice parameters. The as-prepared luminescent samples exhibit sphere-like aggregates that are constructed of small nanospheres and that are well dispersed with a good size distribution. Upon excitation by UV radiation, the  $\text{Eu}^{3+}$ :FA samples demonstrate the characteristic  $^5\text{D}_0$ – $^7\text{F}_{1-2}$  emission lines of  $\text{Eu}^{3+}$ . Finally, the influence of  $\text{Na}_2\text{EDTA}$ /CA on the morphology of  $\text{Eu}^{3+}$ :FA nanopowder was also discussed.

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**Keywords:** Fluorapatite; Rare earth ions; Photoluminescence; Hydrothermal method

## 1. Introduction

Recently, calcium apatites [ $\text{Ca}_{10}(\text{PO}_4)_6(\text{OH}, \text{F}, \text{Cl})_2$ ], the principle inorganic components of bone and teeth, have been attracting significant interest in the fields of biomedicine, dental implants, catalysis and environment engineering, due to their bioactive, biocompatible and osteoconductive properties [1–5]. Among the family of calcium apatites, fluorapatite (FA) is considered an alternative biomaterial due to its low solubility and good biocompatibility in comparison to hydroxyapatite (HA) [6]. Additionally, calcium apatites can also be used in other fields. For example, trivalent rare-earth-ion-doped calcium apatites are employed as biological fluorescent probes due to their good luminescent properties [7]. Moreover, some cationic substitutions have potential applications in the fields of luminescence, water purification and bioceramics [8–10].

The trivalent ions of the 14 stable elements, from La through Lu, have ion radii ranging from slightly greater to slightly smaller than that of the  $\text{Ca}^{2+}$  ion. Moreover, having greater coordination numbers (generally greater than six) and showing little stereochemical preference in their coordination chemistry,  $\text{Ln}^{3+}$  ions exhibit a preference toward oxygen donor ligands, similar to the  $\text{Ca}^{2+}$  ion. In addition, the additional charge on  $\text{Ln}^{3+}$  ions is not a deterrent to their replacing  $\text{Ca}^{2+}$  [11].

Among the rare earth elements from La to Lu,  $\text{Eu}^{3+}$  ions have a simple electronic energy level scheme and hypersensitive transitions, which have been utilized extensively in color televisions and high-efficiency fluorescent lamps. The  $\text{Eu}^{3+}$ -doped calcium apatite is also believed to be a good biological probe candidate because of its more stable luminescence over time in comparison with fluorescent organic molecules [12–16] and its low toxicity and good biocompatibility [17]. Therefore, calcium apatites with the general formula  $\text{Ca}_{10}(\text{PO}_4)_6(\text{X})_2$  ( $\text{X}=\text{OH}$  or  $\text{F}$ ) are good hosts for  $\text{Eu}^{3+}$  doping because  $\text{Eu}^{3+}$  has an ion radius similar to that of the  $\text{Ca}^{2+}$  in the calcium apatite lattice.

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Thus far, many routes have been reported for the synthesis of rare-earth-doped apatite, including coprecipitation [18,19], the microemulsion-mediated hydrothermal process [20,21], the sol–gel combustion method [22], the thermal decomposition method [23], the high-temperature solid-state reaction [24] and microwave irradiation [25]. Nevertheless, some disadvantages exist for these processes, such as their time-consuming nature, the sophisticated effect parameters, chemical contamination, the microlevel particle size, the high reaction temperature and the high cost of the experimental equipment. Therefore, finding mild, facile and convenient synthesis procedures is quite advantageous. Moreover, there are few reports on the synthesis and characterization of  $\text{Eu}^{3+}$ -doped fluorapatite nanoparticles to our knowledge.

In this work, luminescent  $\text{Eu}^{3+}$ -doped fluorapatite  $\{\text{Ca}_{10}(\text{PO}_4)_6\text{F}_2, \text{Eu}^{3+}:\text{FA}\}$  was successfully synthesized via the chelating-reagent-assisted hydrothermal process. This research aims to prepare  $\text{Eu}^{3+}$ -doped fluorapatite nanoparticles using a more convenient process and to discuss the effects of  $\text{Eu}^{3+}$  incorporation into the structure of FA and  $\text{Na}_2\text{EDTA}/\text{CA}$  on the morphology of  $\text{Eu}^{3+}:\text{FA}$  nanoparticles.

## 2. Experimental

All the reagents were purchased without further purification.  $\text{Eu}(\text{NO}_3)_3$  was obtained by dissolving stoichiometric  $\text{Eu}_2\text{O}_3$  in dilute  $\text{HNO}_3$  solution under vigorous stirring. The superfluous  $\text{HNO}_3$  was driven off until the  $\text{Eu}(\text{NO}_3)_3$  crystal powder was obtained.

### 2.1. Synthesis of $\text{Eu}^{3+}:\text{FA}$ nanoparticles

In a typical experiment for the preparation of  $\text{Eu}^{3+}:\text{FA}$  nanoparticles,  $0.7 \times 10^{-3}$  mol of  $\text{Na}_2\text{EDTA}$  and  $2.5 \times 10^{-3}$  mol of CA were dissolved in distilled water. Under continuous stirring,  $2.5 \times 10^{-3}$  mol of  $\text{Ca}(\text{NO}_3)_2$  and  $2.5 \times 10^{-5}$  mol of  $\text{Eu}(\text{NO}_3)_3$  were added to the mixture, and a white emulsion-like mixture was obtained. Then,  $1.5 \times 10^{-3}$  mol of  $\text{Na}_3\text{PO}_4$  and  $0.5 \times 10^{-3}$  mol of NaF were added to the above mixture, and a colorless and transparent solution was obtained. The pH value was adjusted to 5.0–5.8 by using 1 mol/L NaOH. The mixture was then transferred into a Teflon-lined stainless autoclave and kept at  $160^\circ\text{C}$  for 12 h. The precipitates were separated by centrifugation and then dried in an oven at  $70^\circ\text{C}$  for 6 h.

### 2.2. Characterization

Powder X-ray diffraction (XRD) patterns were obtained on a German Bruker AXS D8 ADVANCE diffractometer. The XRD data were recorded by using Cu K radiation ( $\lambda=0.15406$  nm) at a scanning rate of  $0.02^\circ/\text{s}$  in the  $2\theta$  range from  $20^\circ$  to  $60^\circ$ . The obtained experimental patterns were compared with the standards compiled by the Joint

Committee on Powder Diffraction and Standards (JCDPS), which involved card # 15-0876 for FA.

The crystallite size of the FA nanopowders was determined by using the Scherrer equation [26]

$$D_{hkl} = \frac{k\lambda}{B \cos \theta}$$

where  $D$  is the crystallite size (nm),  $\lambda$  is the wavelength of the monochromatic X-ray beam (nm;  $\lambda=0.154$  nm for Cu K radiation),  $\theta$  is the diffraction angle ( $^\circ$ ),  $k$  is a constant varying with crystal habit and chosen to be 0.89 ( $k=0.89$ ), and  $B$  is the full width at half maximum for the diffraction peak under consideration (rad). For this purpose, three diffraction peaks, (002), (202) and (222), which have the advantages of being well separated and having high intensities, were chosen for the measurement. The half-widths were calculated by the SigmaPlot software.

Lattice parameters ( $c$  and  $a$ ) were calculated from peaks (002) and (211), respectively, using the standard HCP unit cell plane spacing relationship [27]

$$\frac{1}{d^2} = \frac{4}{3} \left( \frac{h^2 + hk + k^2}{a^2} \right) + \frac{l^2}{c^2}$$

where  $d$  is the distance between adjacent planes in the set of Miller indices ( $hkl$ ).

The photoluminescence spectrum of the sample was recorded with a Hitachi F-4500 Fluorescence Spectrophotometer at room temperature. Transmission electron microscope (TEM) images were recorded on an FEI Tecnai G2 S-Twin with an acceleration voltage of 200 kV.

## 3. Results and discussion

### 3.1. Phase structure analysis

In the present study, the composition of the synthetic  $\text{Eu}^{3+}$ -doped fluorapatite depends on the initial  $(\text{Eu}+\text{Ca}):\text{P}:\text{F}$  mole ratio in the starting solution. The designated extent of substitution of  $\text{Ca}^{2+}$  by  $\text{Eu}^{3+}$  was determined by the  $x$  value in the general formula of  $\text{Eu}_x\text{Ca}_{10-x}(\text{PO}_4)_6\text{F}_2$ , where  $x=0.1$ . The atomic  $(\text{Eu}+\text{Ca}):\text{P}$  ratio was 1.67, which is the stoichiometric ratio of fluorapatite.

Fig. 1 shows the XRD patterns of  $\text{Eu}^{3+}:\text{FA}$  and the standard data for the hexagonal fluorapatite. Fig. 1(a) shows the XRD patterns of the  $\text{Eu}^{3+}:\text{FA}$  sample, wherein FA diffraction peaks were observed according to the standard card of FA (JCPDS #15-0876). No other metal oxide phases related to the doping component can be detected in the  $\text{Eu}^{3+}$ -doped calcium apatites. To clearly investigate the influence of the addition of  $\text{Eu}^{3+}$  in the FA crystallite, the lattice parameters of  $\text{Eu}^{3+}:\text{FA}$  powders are listed in Table 1.

The calculated lattice constants of the above samples are compared with those of the JCPDS no. 15-0876. It is shown that the calculated lattice constants of  $a=b=0.9372$  nm and  $c=0.6881$  nm are for  $\text{Eu}^{3+}:\text{FA}$ , and the standard

data of  $a=b=0.9384$  nm and  $c=0.6884$  nm are for standard FA. It is evident that  $\text{Eu}^{3+}$  ions ( $0.95$  Å) entered the FA lattice and occupied  $\text{Ca}^{2+}$  ( $0.99$  Å) sites, and the incorporation of  $\text{Eu}^{3+}$  ions into the fluorapatite resulted in the decrease of the lattice parameters.

The crystallite sizes of the prepared  $\text{Eu}^{3+}$ :FA sample calculated by using the XRD data are shown in Table 1. The crystallite size of the obtained powders is  $33.3$  nm.

### 3.2. TEM analysis

Fig. 2 shows the size and morphology of  $\text{Eu}^{3+}$ :FA powder samples obtained in the presence of  $\text{Na}_2\text{EDTA}/\text{CA}$

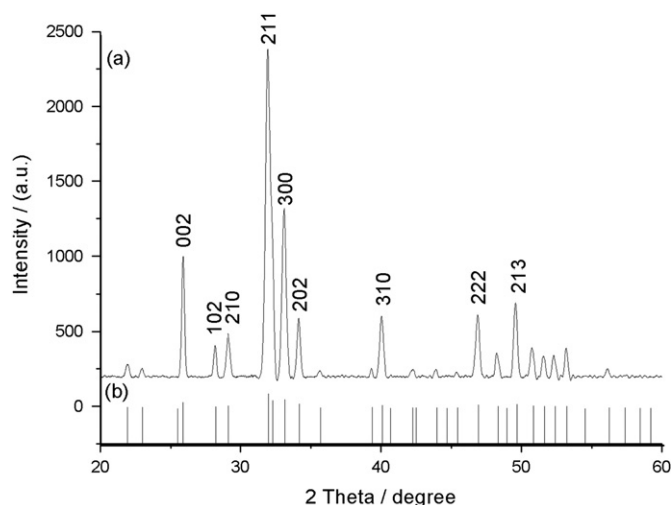


Fig. 1. XRD patterns of the synthetic  $\text{Eu}^{3+}$ -doped fluorapatite (a) and standard pure fluorapatite (b).

Table 1  
Lattice parameters and crystallite size of the obtained powders.

Sample	<i>a</i> -axis (nm)	<i>c</i> -axis (nm)	Crystallite size (nm)
$\text{Eu}^{3+}$ :FA	0.9372	0.6881	33.3
Standard FA	0.9384	0.6884	–

at  $160$  °C for  $12$  h. The morphology of the nanopowder indicates that it is composed of sphere-like agglomerates with narrow particle size distributions. In Fig. 2(a), the sizes of the agglomerates are in the range  $30$ – $46$  nm. As observed at higher magnifications (Fig. 2(b)), these agglomerates are formed by nanoparticles with sphere-like shapes ranging from  $5$  to  $9$  nm.

This range is consistent with the crystallite size of  $\text{Eu}^{3+}$ :FA powders calculated by using XRD data within the error range.

### 3.3. Photoluminescence properties

The emission spectrum of the  $\text{Eu}^{3+}$ :FA sample was recorded at an excitation wavelength of  $250$  nm, as shown in Fig. 3. The spectral features observed in the  $582$ – $603$  nm and  $603$ – $640$  nm ranges were ascribed to the  $\text{Eu}^{3+}$  ion  $^5\text{D}_0 \rightarrow ^7\text{F}_1$  and  $^5\text{D}_0 \rightarrow ^7\text{F}_2$  transitions, respectively. The

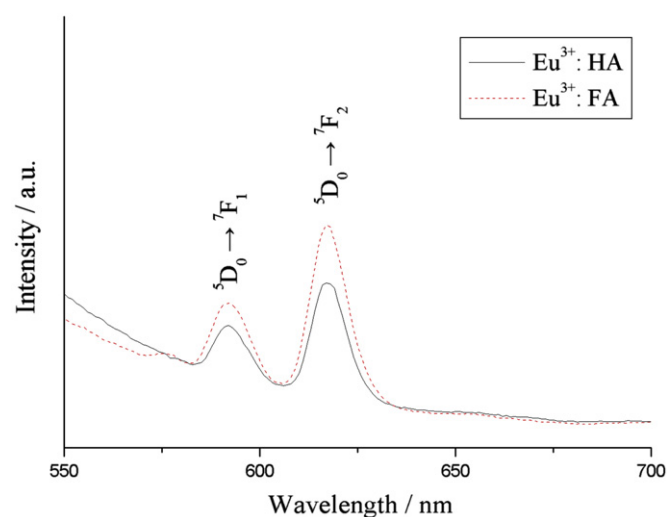


Fig. 3. Luminescence spectra of the  $\text{Eu}^{3+}$ :FA sample (dashed line) and the  $\text{Eu}^{3+}$ :HA sample (full line) ( $1\%$   $\text{Eu}^{3+}$  doping concentration,  $\lambda_{\text{ex}}=250$  nm).

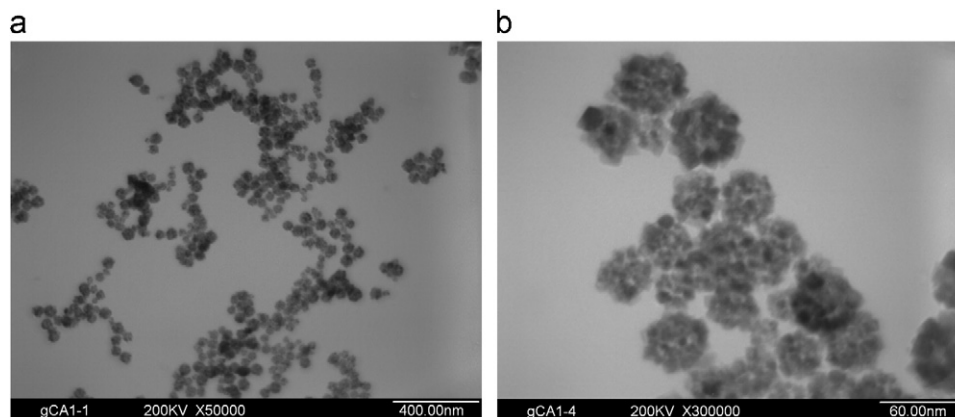


Fig. 2. (a) and (b) TEM images of  $\text{Eu}^{3+}$ :FA nanoparticles taken at different magnifications.

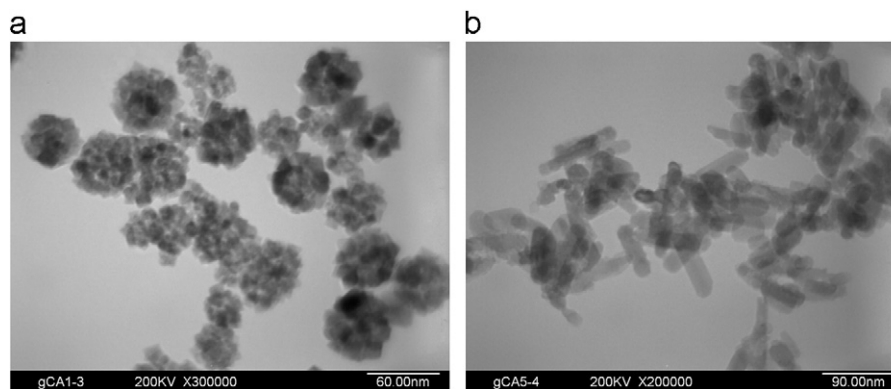


Fig. 4. TEM images of  $\text{Eu}^{3+}$ :FA nanoparticles obtained in the presence of  $\text{Na}_2\text{EDTA}/\text{CA}$  (a) and in the absence of  $\text{Na}_2\text{EDTA}/\text{CA}$  (b).

more efficient emission was observed for the hypersensitive  $^5\text{D}_0 \rightarrow ^7\text{F}_2$  transition with a maximum intensity at 617 nm.

We also found that the PL intensity of  $\text{Eu}^{3+}$ :FA increased slightly, in contrast to  $\text{Eu}^{3+}$ :HA sample, which was described in our recent report [19]. The comparison is based on the same  $\text{Eu}^{3+}$  doping concentration (1%), the same reaction time and temperature, and the fact that the chelating agent of CA/ $\text{Na}_2\text{EDTA}$  has no effect on the PL intensity in our present work. In addition, no significant modification of emission spectra was observed, expect for PL intensity. We believe that the increase of the PL intensity may be attributed to the replacement of  $\text{OH}^-$  by  $\text{F}^-$  in the structure of apatite crystals [28].

### 3.4. Effect of $\text{Na}_2\text{EDTA}/\text{CA}$

In our research, sphere-like  $\text{Eu}^{3+}$ :FA nanoagglomerates have been synthesized by a hydrothermal process by using CA and  $\text{Na}_2\text{EDTA}$  as chelating reagents. To investigate the effect of the chelating reagent on the  $\text{Eu}^{3+}$ :FA nanocrystals,  $\text{Eu}^{3+}$ :FA samples were also prepared in the absence of CA and  $\text{Na}_2\text{EDTA}$ .

Fig. 4(a) shows a typical TEM image of the sample obtained in the presence of  $\text{Na}_2\text{EDTA}/\text{CA}$ , which has a sphere-like morphology and is 30–46 nm in diameter. Fig. 4(b) shows the sample obtained in the absence of  $\text{Na}_2\text{EDTA}/\text{CA}$ . It is evident that rod-like  $\text{Eu}^{3+}$ :FA nanoparticles 48–69 nm in length and 16–25 nm in width were produced when  $\text{Na}_2\text{EDTA}/\text{CA}$  was not present, while sphere-like aggregates were obtained by using  $\text{Na}_2\text{EDTA}/\text{CA}$  as a chelating reagent. It is believed that citric species could promote the aggregation of apatite nanoparticles in a side-by-side manner due to the electrostatic attraction between the  $-\text{COO}^-$  groups and the calcium ions [29]. In our case,  $\text{Na}_2\text{EDTA}/\text{CA}$  may be absorbed on the surface of  $\text{Eu}^{3+}$ :FA crystals and prevent crystal growth along the  $c$ -axis. In addition, in the CA-enriched solution, to reduce the surface energy, sphere-like  $\text{Eu}^{3+}$ :FA nanocrystals with a higher energy would assemble together rapidly to form the sphere-like aggregates.

## 4. Conclusions

In summary, luminescent  $\text{Eu}^{3+}$ -doped fluorapatite ( $\text{Eu}^{3+}$ :FA) was successfully synthesized by a hydrothermal process by utilizing  $\text{Na}_2\text{EDTA}$  and CA as the growth control agents. The incorporation of  $\text{Eu}^{3+}$  ions into the fluorapatite resulted in a decrease in the values of the lattice parameters. The as-prepared luminescent samples exhibit sphere-like aggregates that are constructed of small nanospheres. Upon excitation by UV radiation, the  $\text{Eu}^{3+}$ :FA samples demonstrate the characteristic  $^5\text{D}_0 \rightarrow ^7\text{F}_{1-2}$  emission lines of the  $\text{Eu}^{3+}$  ion. The influence of  $\text{Na}_2\text{EDTA}/\text{CA}$  on the morphology of  $\text{Eu}^{3+}$ :FA crystals may be attributed to the fact that in the CA-enriched solution, to reduce the surface energy, sphere-like  $\text{Eu}^{3+}$ :FA nanocrystals with higher energy would assemble together rapidly to form the sphere-like aggregates.

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