

Facile synthesis and luminescent properties of $\text{Y}_2\text{O}_3:\text{Eu}^{3+}$ nanophosphors via thermal decomposition of cocrystallized yttrium europium propionates

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Abstract

$\text{Y}_2\text{O}_3:\text{Eu}^{3+}$ nanophosphors were fabricated by direct thermal decomposition of monohydrate yttrium europium propionates, which can be facilely obtained via cocrystallization. The composition and structure of the cocrystallized product were confirmed by X-ray diffractionmetry (XRD), simultaneous differential scanning calorimetry (DSC) and thermogravimetric analysis (TGA) as well as inductively coupled plasma mass (ICP-MS) spectrometry. The effects of decomposition temperature on the resultant microstructures such as crystallinity, particle size and particle dispersibility were investigated by XRD and transmission electron microscopy (TEM). The dependence of their luminescent properties on decomposition temperature was also studied by fluorescence spectrophotometry. The results showed that the microstructures and luminescent intensities of $\text{Y}_2\text{O}_3:\text{Eu}^{3+}$ nanophosphors could be tuned by varying calcination temperatures of cocrystallized yttrium europium propionates from 600 to 1000 °C.

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1. Introduction

$\text{Y}_2\text{O}_3:\text{Eu}^{3+}$ is an efficient red-emission phosphor and has been widely used in fluorescent lamps (FL) and cathode ray tube (CRT) [1,2]. The studies during the last decade show that $\text{Y}_2\text{O}_3:\text{Eu}^{3+}$ nanophosphors has significant promise in field emission displays (FEDs) and plasma display panels (PDP) [3,4].

Many approaches have been developed for the fabrication of $\text{Y}_2\text{O}_3:\text{Eu}^{3+}$ nanostructures, such as precipitation [5–7], hydrothermal or solvothermal method [8,9], mechanochemical processing [10], spray pyrolysis [11], microemulsion [12], sol–gel [13] and combustion synthesis [14], etc. Among the various routes to the synthesis of $\text{Y}_2\text{O}_3:\text{Eu}^{3+}$

nanostructures based on thermal decomposition of precursors, two categories can be divided according to the precursor constitution, i.e. precipitates and gel. It is noticeable that the microstructures and physical properties of final products depend largely on the characteristics of the precursors obtained by controlled synthesis and the thermal decomposition conditions adopted. Therefore, choice and controlled synthesis of a suitable precursor is crucial to the fabrication of desired nanostructures with unique properties.

Numerous attempts have been made to synthesize $\text{Y}_2\text{O}_3:\text{Eu}^{3+}$ nanostructures via thermal decomposition of organometallic complex xerogel [13,15–17]. However, these efforts are limited to industrialized production due to complicated process, high cost and long synthetic duration. In our previous works, we have proposed a novel and simple method for the preparation of ultrafine and nano-sized ceria particles via thermal decomposition, wherein

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the soluble crystalline hydrate cerium carboxylates were used as the precursors rather than the traditional precipitates and gel [18,19]. So far, to the best of our knowledge, there have been a few of reports about the thermal decomposition behavior of the rare earth propionates including Ce, La and Lu [19–21]. In this paper, we will put forward a novel and facile approach for the synthesis of $\text{Y}_2\text{O}_3:\text{Eu}^{3+}$ nanophosphors via thermal decomposition of monohydrate yttrium europium propionates, which can be also easily obtained from cocrystallization. Their luminescent properties dependent on decomposition temperature have also been investigated. This method for synthesis of $\text{Y}_2\text{O}_3:\text{Eu}^{3+}$ nanophosphors is based on thermal decomposition of soluble salt grains, which extends the scope of precursors besides precipitates and gel. Furthermore, due to many advantages such as easy operation, feasible cost and simple process, this new synthetic route to $\text{Y}_2\text{O}_3:\text{Eu}^{3+}$ nanophosphors may find promising applications in the large scale production of $\text{Y}_2\text{O}_3:\text{Eu}^{3+}$ nanophosphors.

2. Experimental procedures

2.1. Synthesis

Y_2O_3 (Purity: $\text{Y}_2\text{O}_3/\text{REO} \geq 99.999\%$) and Eu_2O_3 (Purity: $\text{Y}_2\text{O}_3/\text{REO} \geq 99.999\%$) were supplied by Ganzhou Qindong Rare Earths Group Co. Ltd. Propionic acid was of analytical grade and supplied by Sinopharm Chemical Reagent Co. Ltd. All the chemicals were used as received without further purification.

In a typical synthetic procedure, the mixture of Y_2O_3 and Eu_2O_3 with molar ratio of $\text{Y}/\text{Eu}=96.1/3.9$ were completely dissolved in excess amount of propionic acid solution under continuous agitating and heating. Then, by repeating the processes of evaporating, cooling and filtering, all the large white crystals of yttrium europium propionates were crystallized out of the propionic acid solution. The final products were collected after the cocrystallized yttrium europium propionates were dried at 80°C for 12 h and then calcined in a muffle furnace under air at the selected temperatures for 3 h.

2.2. Characterizations

Simultaneous DSC–TGA of the crystalline particles were conducted on a thermal analyzer (Model Q-600, TA Instruments) at a heating rate of $20^\circ\text{C min}^{-1}$ under static air from room temperature to 950°C . The microstructures of the particles were observed via a transmission electron microscope (Model 200CX, JEOL Co.). Phase identification was performed on an X-ray diffractometer (Model D8 Advance, Bruker Co.) operating at 40 kV/50 mA using $\text{Cu-K}\alpha$ radiation with a scanning speed of $0.15^\circ 2\theta$ per minute. Photoluminescence excitation (PLE) and photoluminescence (PL) properties of the samples were collected on a fluorescence spectrophotometer (Model F-4500,

Hitachi Co.) at room temperature with a 300 W Xe-lamp as the light source. PMT voltage of the measurements was 400 V, the excitation and emission slits were set at 2.5 nm. The contents of Eu in the as-calcined samples were determined by ICP-MS spectrometer (Model ELAN DRC-e, Perkin Elmer, Inc.).

3. Results and discussion

In order to identify the phase of the grains crystallized from propionic acid solution of Y_2O_3 and Eu_2O_3 , their XRD pattern was collected. As shown in the inset of Fig. 1, the characteristic peaks of the as-obtained grains are found to correspond well with those of monohydrate yttrium propionate (JCPDF 00-037-1559). Due to the small doping of Eu^{3+} , the characteristic peaks of monohydrate europium propionate were not observed. To further confirm the composition of the same sample, simultaneous DSC–TGA

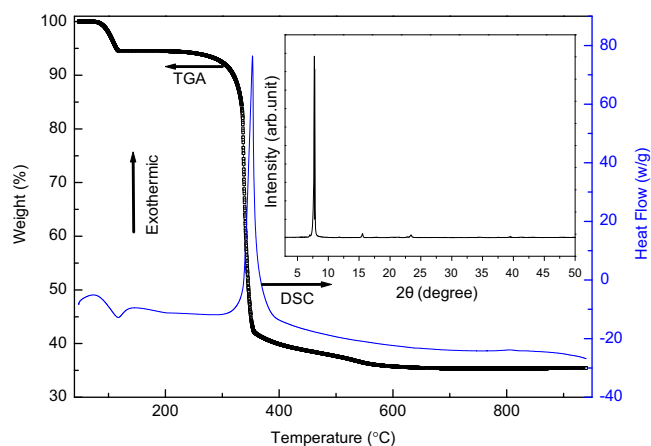


Fig. 1. DSC–TGA curves and XRD pattern (inset) of the crystalline particles obtained via dissolving mixture of yttria and europia (with molar ratio of $\text{Y}/\text{Eu}=96.1/3.9$) with excess propionic acid and subsequent cocrystallization.

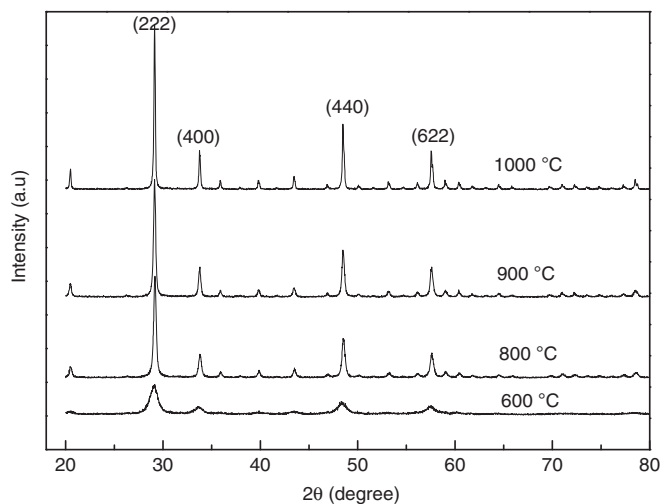


Fig. 2. XRD patterns of the samples obtained via decomposition of $\text{Y}_{0.961}\text{Eu}_{0.039}(\text{C}_2\text{H}_5\text{COO})_3 \cdot \text{H}_2\text{O}$ at different temperatures for 3 h.

was conducted. As revealed in Fig. 1, the decomposition of hydrate crystalline salts on heating consists of three stages in weight loss. The first weight loss (5.48 wt%) before 120 °C

results from the removal of one crystalline water, which is well consistent with the theoretical value of $Y_{0.961}Eu_{0.039}(C_2H_5COO)_3 \cdot H_2O$ (5.48 wt%). The second stage appears

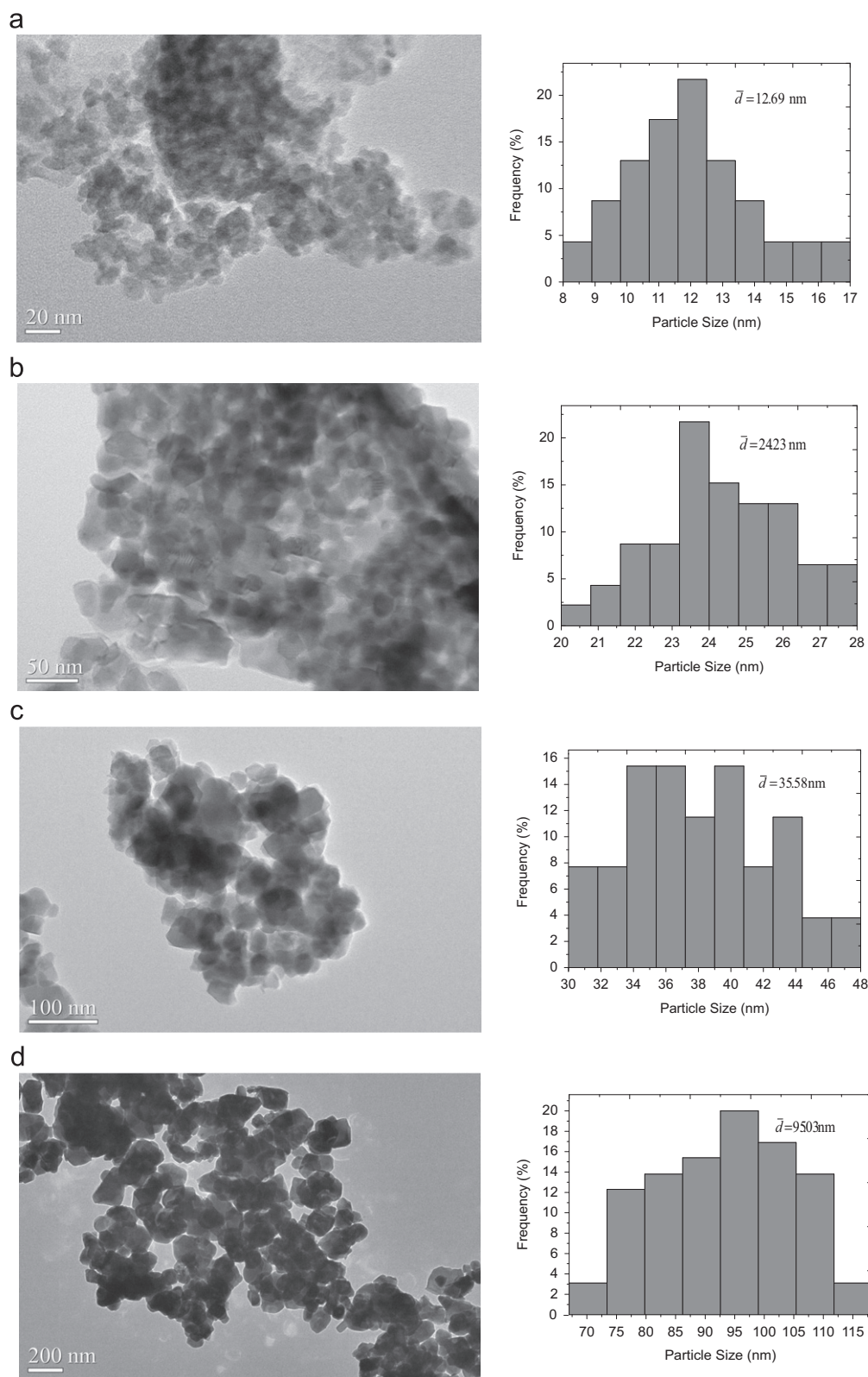


Fig. 3. TEM micrographs (left) and size distribution histograms (right) of the samples obtained via decomposition of $Y_{0.961}Eu_{0.039}(C_2H_5COO)_3 \cdot H_2O$ at different temperatures for 3 h: (a) 600 °C, (b) 800 °C, (c) 900 °C and (d) 1000 °C.

from 241.22 to 401.37 °C. A large weight loss in the TGA curve and a sharp strong exothermic peak at 353.01 °C in the DSC curve are observed simultaneously. This drastic exothermal process is ascribed to the decomposition of propionate group. Following this process is the third stage with gradual weight loss, which is possibly attributed to the further release of organic residues or carbon in the samples at higher temperature. The total weight loss is 64.64 wt%, which is in good agreement with theoretical value of 64.87 wt%. Moreover, the ICP-MS result confirms that the Eu content in the as-calcined sample agrees well with the designed Eu content. Based on the above data of XRD, DSC–TGA and ICP-MS, it can be concluded that the as-obtained crystalline grains are $Y_{0.961}Eu_{0.039}(C_2H_5COO)_3 \cdot H_2O$ and that the thermal decomposition processes of the yttrium europium propionates can be expressed as follows:

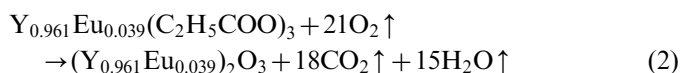


Fig. 2 shows the XRD patterns of the samples obtained via calcination of $Y_{0.961}Eu_{0.039}(C_2H_5COO)_3 \cdot H_2O$ at different temperatures for 3 h respectively. All the characteristic diffraction peaks of the four samples are consistent with cubic Y_2O_3 (JCPDS card no. 25-1011). No additional phase was detected in the XRD patterns. As shown in Fig. 2, with variation of the sintering temperature from 600 to 1000 °C, the intensities of the characteristic diffraction peaks become stronger, indicating a higher crystallinity of $Y_2O_3:Eu^{3+}$.

To investigate the microstructure evolution of the as-calcined particles with sintering temperature, a transmission electron microscope was employed for observation. TEM micrographs of the samples obtained via calcination of $Y_{0.961}Eu_{0.039}(C_2H_5COO)_3 \cdot H_2O$ at different temperatures for 3 h are presented in Fig. 3. As can be seen, with the increase of calcination temperature, the primary spherical-shaped particles became larger and more dispersive, the average particle sizes of the samples from the correspondent particle size histograms grow from 12.69 nm at 600 °C, 24.23 nm at 800 °C, 35.58 nm at 900 °C to 95.03 nm at 1000 °C, which are approximately close to the average crystallite sizes calculated according to Scherrer's formula. Notably, it is at 1000 °C that the nanoparticle agglomeration improves remarkably although the particles become larger quickly. When the precursor was calcined above 1100 °C for 3 h, the resultant particles would grow to over 100 nm in size.

Photoluminescence excitation (PLE) and photoluminescence (PL) of $(Y_{0.961}Eu_{0.039})_2O_3$ calcined at different temperatures for 3 h are shown in Fig. 4(a). The broad yet strong band with a maximum at 254 nm observed from the PLE spectrum is attributed to the transition by charge transfer, that is, electronic transition from the 2p orbital of O^{2-} to the 4f orbital of Eu^{3+} activators [22]. Moreover, all

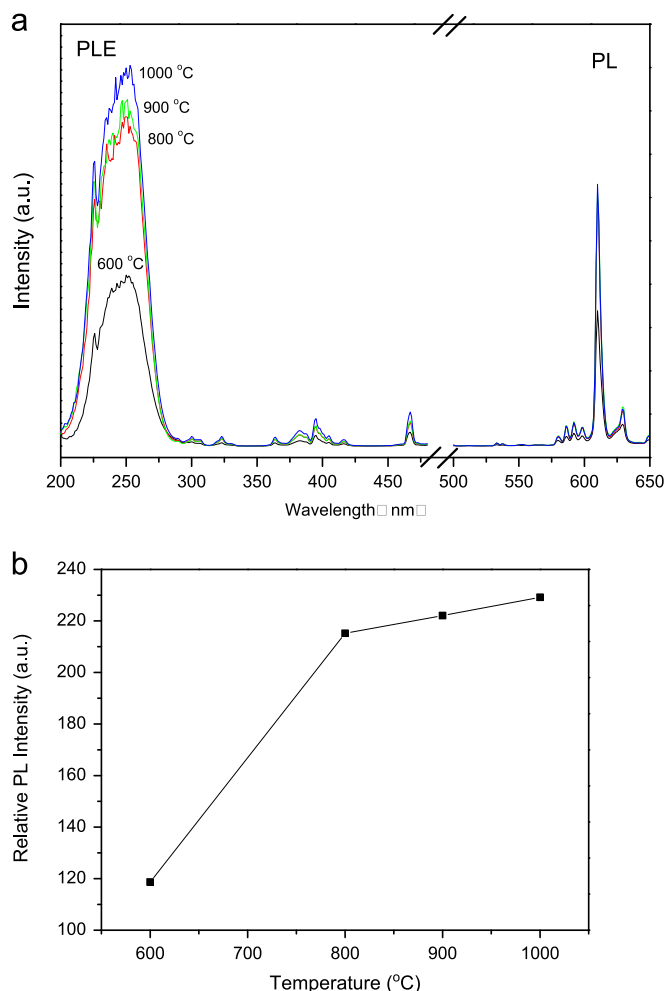


Fig. 4. (a) Room temperature PLE and PL spectra of the $(Y_{0.961}Eu_{0.039})_2O_3$ samples obtained via decomposition of $Y_{0.961}Eu_{0.039}(C_2H_5COO)_3 \cdot H_2O$ at different temperatures for 3 h; (b) The relative PL intensities of 610 nm emission as a function of calcination temperatures. The PLE spectra were obtained by monitoring the 610 nm emission while the PL spectra were obtained under UV excitation at 254 nm.

the samples exhibit emission at 610 nm corresponding to red emission due to $^5D_0 \rightarrow ^7F_2$ transition under the excitation of 254 nm [12]. It can be seen from Fig. 4(b) that decomposition temperature of the precursor has significant affects on luminescence of the resultant particles, especially below 800 °C. A 200 °C increase in the temperature from 600 to 800 °C yields an 81% increase in the intensity of the 610 nm emission. This may be mainly due to a sharp increase of $Y_2O_3:Eu^{3+}$ in crystallinity from 600 to 800 °C as can be seen in Fig. 2 [23].

4. Conclusions

We have developed a novel and simple route to $Y_2O_3:Eu^{3+}$ nanophosphors via the thermal decomposition of soluble salt crystal grains, which can be easily obtained via dissolving the designated molar ration of yttria and europia with excess propionic acid and subsequent cocrystallization. DSC–TGA and XRD analysis confirm that the

as-obtained crystalline grains are Eu-doped monohydrate yttrium propionates. The results reveal that the calcination temperature has significant effects on size, agglomeration and luminescence of the resultant particles. It can be concluded that the luminescent intensities of $\text{Y}_2\text{O}_3\text{:Eu}^{3+}$ nanophosphors may be adjusted by varying calcination temperature of cocrystallized yttrium europium propionates from 600 to 1000 °C. In prospect, this new synthetic route to $\text{Y}_2\text{O}_3\text{:Eu}^{3+}$ nanophosphors may find promising applications in the large scale production of $\text{Y}_2\text{O}_3\text{:Eu}^{3+}$ nanophosphors owing to such advantages as easy operation, feasible cost and simple process.

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