

# Hydrothermal synthesis and visible-light photocatalytic activity of $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> composite hollow microspheres

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## Abstract

Novel  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> composite hollow spheres were successfully synthesized by a template-assisted precipitation reaction using urea as a precipitating agent and carbon spheres as templates in a mixed solvent of water and ethanol, and then calcined at 400 °C for 4 h. The as-prepared samples were characterized by X-ray diffraction, scanning electron microscopy, transmission electron microscopy, nitrogen adsorption–desorption isotherms, and vibrating sample magnetometer. The influence of calcination temperature and the molar ratio of titanium to iron (*R*) on the photocatalytic activity of the samples was investigated. The results indicated that the composite spheres show magnetic characteristics at room temperature and good photocatalytic activity under visible-light irradiation compare to the single-component  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> particles. This method can be further applied to synthesize nanocomposites of magnetic metal oxide and other metal oxide.

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## 1. Introduction

In recent years, micrometer- and nanometer-sized hollow structures have attracted a great deal of attention because of their low density, high surface area, good surface permeability and large light-harvesting efficiencies [1–4]. Moreover, these materials can be widely applied in photoelectric devices, catalysis, drug delivery, chromatography separation, and chemical reactors [5–9]. Various synthetic methods were explored to prepare hollow nanomaterials including Ionic liquids (ILs), self-assembly techniques, hydrothermal techniques, template-assisted techniques, and chemically induced self-transformation [10–12]. Up to now, template-assisted synthetic method has proved to be the most-applied and most effective route to fabricate inorganic hollow structures.

Titanium dioxide (TiO<sub>2</sub>), as one of the most important transition-metal functional oxides, has attracted extensive attention during the past decades for its superior physical and chemical properties and a wide variety of potential use in diverse fields such as solar energy conversion, environmental purification, and water treatment [13–16]. In particular, TiO<sub>2</sub> hollow nanomaterials and nanocomposites have received more and more attention owing to their high photocatalytic activity, chemical stability, low cost and nontoxicity [17,18]. However, because of its wide band-gap energy (3.02 eV), TiO<sub>2</sub> can only harvest the spectrum with wave lengths in the near-ultraviolet (UV) region shorter than 387 nm, which accounts for merely 4–5% of the solar spectrum. Moreover, TiO<sub>2</sub> follows a relatively high electron–hole recombination rate, which is detrimental to its photoactivity. To solve this issue, different approaches such as transition metal doping, inorganic dye-sensitizing, valuable metal deposition and coupling titania with other semiconductors have been devoted to enhancing the photocatalytic activity of TiO<sub>2</sub> in which the response of the semiconductor was extended toward the visible region [19–21]. Up to now, it is still a great

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challenge to effectively immobilize or separate the  $\text{TiO}_2$  particles in the photocatalytic system. Magnetic separation provides a very convenient approach for removing and recycling magnetic particles/composites by applying an appropriate magnetic field [22,23]. Compared to conventional nanopowder photocatalysts,  $\text{TiO}_2$  magnetic composites such as  $\text{Fe}_2\text{O}_3$ – $\text{TiO}_2$  or  $\text{Fe}_3\text{O}_4$ – $\text{TiO}_2$  can be regarded as a promising photocatalyst for the environmental purification at the industrial scale as they can be more readily separated from the slurry system by the magnetic separation after photocatalytic reaction and recycled. Very recently, Yu et al. [24] fabricate a hierarchical porous  $\gamma$ - $\text{Fe}_2\text{O}_3$ @ $\text{SiO}_2$ @ $\text{TiO}_2$  composite photocatalyst with superior photocatalytic properties by an effective three-step approach. Mou et al. [25] developed an asymmetric shrinkage approach for the fabrication of magnetic  $\gamma$ - $\text{Fe}_2\text{O}_3$ /TiO<sub>2</sub> Janus hollow bowls (JHBs) by constructing a precursor solution pair during the solvents evaporation process. Moreover the as-obtained products show an efficient visible-light photocatalytic activity and convenient magnetic separation for water purification.

Herein, novel  $\alpha$ - $\text{Fe}_2\text{O}_3$ /TiO<sub>2</sub> composite hollow spheres are successfully fabricated using carbon spheres prepared from saccharide solution as templates, and their visible-light photocatalytic activity and environment application are carefully investigated. In addition, we also investigated the effects of the molar ratio of iron to titanium ( $R$ ) on the microstructures and properties, especially on the photocatalytic property.

## 2. Experimental

### 2.1. Synthesis of carbon spheres

All the reagents used in the experiments were in analytical grade (purchased from SCRC Chemical Co., China) and used without further purification. Carbon spheres were synthesized by the hydrothermal approach as reported previously [26]. In a typical synthesis, 6.0 g glucose was dissolved in 60 mL of distilled water under constant stirring. Then the aqueous solution was transferred to a 100 mL Teflon-lined stainless steel autoclave, maintained at 180 °C for 4 h. The black or puce precipitates were washed with distilled water and ethanol three times and dried at 60 °C for 8 h. In general, a longer hydrothermal reaction time results in the formation of larger carbonaceous polysaccharide spheres.

### 2.2. Synthesis of $\alpha$ - $\text{Fe}_2\text{O}_3$ /TiO<sub>2</sub> hollow structures

The  $\alpha$ - $\text{Fe}_2\text{O}_3$ /TiO<sub>2</sub> hollow structures were prepared as follows: 1.8 g  $\text{CO}(\text{NH}_2)_2$  and 0.3 g carbon spheres were dissolved in 54 mL ethanol/water mixed solvents with given volume ratio ( $V_e/V_w=48$  mL/6 mL). After vigorously magnetic stirring for 30 min by using a magnetic stirrer, a certain amount of  $\text{FeCl}_3$  and tetrabutyl titanate (TBOT) was added into the above solution under vigorously stirring for another 10 min. The molar ratio of titanium to iron ( $R$ ) varied from 2:1 to 2:3 (2:1, 2:1.5, 2:2, and 2:3). The mixture was then transferred into a 100 mL Teflon-lined stainless steel autoclave and sealed, and the autoclave was placed in a pre-heated oven

at 60 °C for 48 h and naturally cooled down to room temperature. The  $\alpha$ - $\text{Fe}_2\text{O}_3$ /TiO<sub>2</sub> precursor thus obtained was collected by centrifugation, washed with distilled water and dried under vacuum at 60 °C for 10 h. Finally, the prepared sample was heated in an air atmosphere to a certain temperature (400 °C) and held at this temperature for 4 h to remove the carbon spheres template.

### 2.3. Characterization

The X-ray diffraction patterns were recorded using a D8 advance (Bruker-AXS) diffractometer with Cu  $K\alpha$  radiation ( $\lambda=0.1546$  nm). The morphologies and structures of the samples were characterized by scanning electron microscopy (SEM, JEOL JXA-840A) and transmission electron microscopy (TEM) with a Japan JEM-100CX II transmission electron microscopy. The Brunauer–Emmett–Teller (BET) surface area ( $S_{\text{BET}}$ ) of the powders was analyzed by nitrogen adsorption in an ASAP2020 surface area and porosity analyzer (Micromeritics, USA). All the samples were degassed at 180 °C prior to nitrogen adsorption measurements. The BET surface area was determined by a multipoint BET method using the adsorption data in the relative pressure ( $P/P_0$ ) range of 0.05–0.25. Desorption isotherm was used to determine the pore size distribution via the Barret–Joyner–Halender (BJH) method with cylindrical pore size [27]. The nitrogen adsorption volume at the relative pressure ( $P/P_0$ ) of 0.994 was used to determine the pore volume and average pore sizes. Magnetic properties of the samples were measured by a model 4HF vibrating sample magnetometer (VSM, ADE Co. Ltd., USA).

### 2.4. Photocatalytic activity test

The evaluation of photocatalytic activity of the prepared samples for the photocatalytic decolorization of RhB aqueous solution was performed at ambient temperature. A 300 W halogen–tungsten through a UV-cutoff filter ( $>400$  nm,  $13.0 \pm 0.5$  mW cm<sup>2</sup> at 420 nm), which was positioned 10 cm away from the reactor, was used as a visible light source to trigger the photocatalytic reaction. Experiments were as follows: 0.05 g of the prepared  $\alpha$ - $\text{Fe}_2\text{O}_3$ /TiO<sub>2</sub> powder was dispersed in a 20 mL of RhB aqueous solution with a concentration of  $1 \times 10^{-5}$  M in a reaction cell. Before illumination, the suspensions were magnetically stirred in the dark for 60 min to ensure the establishment of an adsorption–desorption equilibrium between the photocatalyst powders and RhB. At given time intervals, 3 mL aliquots were sampled and centrifuged to remove photocatalyst powders. The filtrates were analyzed by recording the variations of the absorption-band maximum (553 nm) of RhB in the UV–vis spectrum. Total organic carbon of the solution was analyzed with the Apollo9000 TOC analyzer.

## 3. Results and discussion

Fig. 1 shows XRD patterns of the  $\alpha$ - $\text{Fe}_2\text{O}_3$ /TiO<sub>2</sub> samples prepared at  $R=2:1.5$  at 60 °C for 48 h and calcined at 400 °C

for 4 h. The major diffraction peaks are consistent with either those of the JCPDS file No. 21-1272 or those of the JCPDS file No. 33-0664, indicating that the products mainly consist of the tetragonal rutile  $\text{TiO}_2$  phase and the hexagonal  $\alpha\text{-Fe}_2\text{O}_3$  phase. In addition, some peaks (located at  $18.0^\circ$ ,  $25.5^\circ$ ,  $32.5^\circ$ ) that do not match either of these phases are from the solid solution of iron titanium oxide phase ( $\text{Fe}_2\text{TiO}_5$ ), which is probably due to the formation of a Fe-rich layer at the surface of  $\text{TiO}_2$ .

The size and morphology of the products were identified by SEM, TEM and HRTEM. Fig. 2a shows SEM image of the as-prepared  $\text{C}@ \alpha\text{-Fe}_2\text{O}_3/\text{TiO}_2$  composite nanostructures with diameters ranging from  $0.5 \mu\text{m}$  to  $1 \mu\text{m}$  obtained via the hydrothermal process before calcination. As shown Fig. 2b,

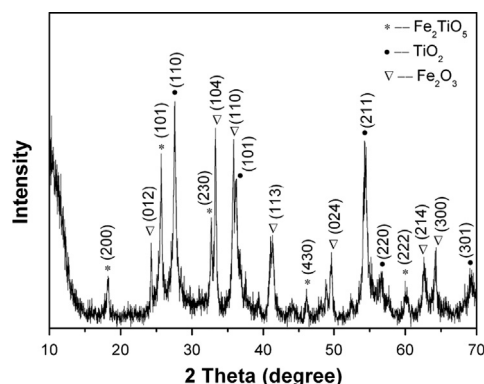


Fig. 1. XRD patterns of the products prepared at  $R=2:1.5$  and calcined at  $400^\circ\text{C}$  for 4 h.

it can be clearly seen that diameters of  $\alpha\text{-Fe}_2\text{O}_3/\text{TiO}_2$  hollow spheres decrease drastically and are about  $600 \text{ nm}$  in size after calcination. To get more information about the hollow structures, the samples were characterized by TEM. Fig. 2c is the typical TEM image of  $\alpha\text{-Fe}_2\text{O}_3/\text{TiO}_2$  hollow spheres, further confirming the hollow structure with a shell thickness of about  $60 \text{ nm}$ . It can be found that the shells of  $\alpha\text{-Fe}_2\text{O}_3/\text{TiO}_2$  hollow spheres are composed of randomly aggregated nanoparticles with sizes of about  $15 \text{ nm}$ . A HRTEM image reveals more details concerning the hybrid structure (Fig. 2d). The lattice fringe spacing of  $0.37 \text{ nm}$  and  $0.35 \text{ nm}$  are consistent with the interplanar distances of (012) plane of  $\alpha\text{-Fe}_2\text{O}_3$  and (110) plane of  $\text{TiO}_2$ .

To examine the surface area, as well as the pore size distribution, nitrogen adsorption/desorption analysis was conducted. As shown in Fig. 3a, the BET surface area of the as-prepared  $\alpha\text{-Fe}_2\text{O}_3/\text{TiO}_2$  hollow spheres is  $31.8 \text{ m}^2/\text{g}$ , which is higher than that of commercial iron oxide particles [28]. Fig. 3a shows the sample exhibits a type IV isotherm with a H3 hysteresis loop according to Brunauer–Deming–Deming–Teller (BDDT), indicating the presence of mesopores ( $2\text{--}50 \text{ nm}$ ). In addition, the observed hysteresis loops shifts to a high relative pressure  $P/P_0 \approx 1$ , suggesting the presence of large pores ( $> 50 \text{ nm}$ ). The corresponding pore size distribution curve is shown in Fig. 3b. It can be seen that the  $\alpha\text{-Fe}_2\text{O}_3/\text{TiO}_2$  hollow spheres contained small mesopores (ca.  $3 \text{ nm}$ ) and larger mesopores with a maximum pore diameter of ca.  $26 \text{ nm}$ . This result is consistent with the TEM observation (Fig. 2d). According to previous reports [27–29], the larger specific BET surface

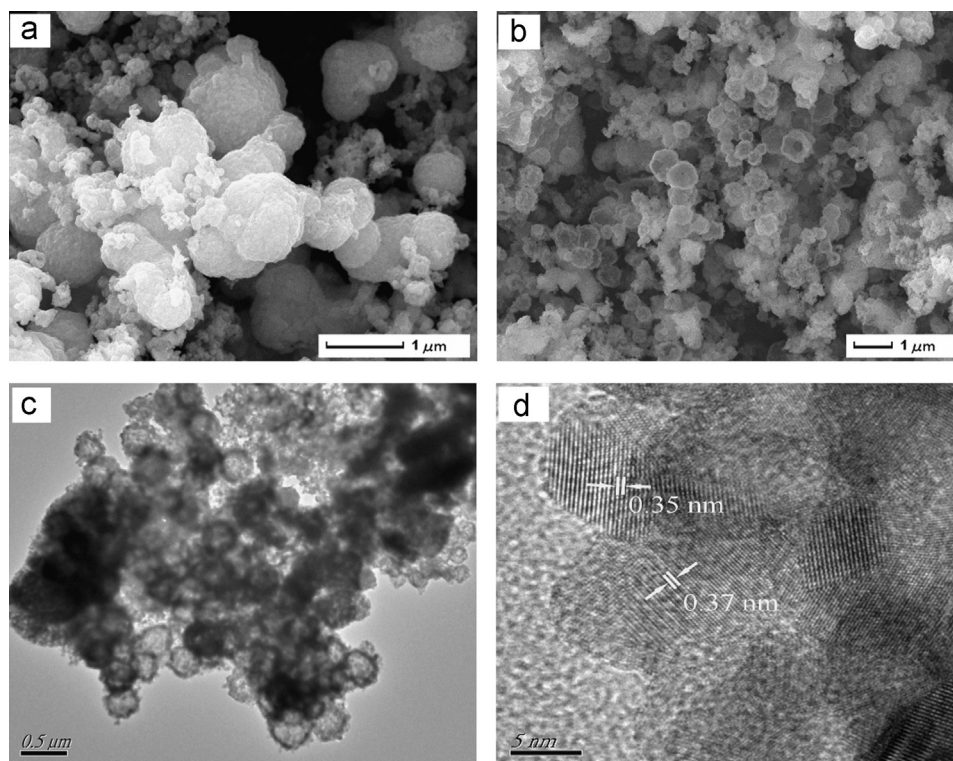


Fig. 2. SEM images of the  $\alpha\text{-Fe}_2\text{O}_3/\text{TiO}_2$  hollow spheres prepared at  $R=2:1.5$  before calcination (a) and after calcination (b) at  $400^\circ\text{C}$  for 4 h; (c, d) TEM and HR-TEM images of the  $\alpha\text{-Fe}_2\text{O}_3/\text{TiO}_2$  hollow structure after calcination.



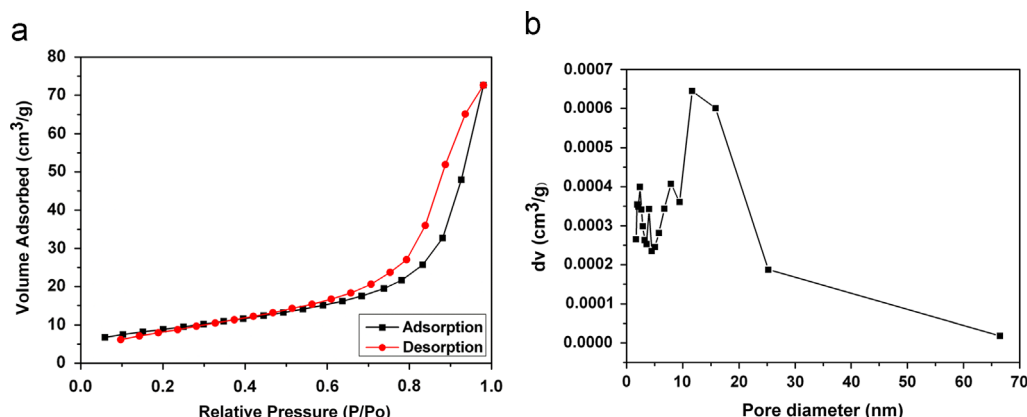


Fig. 3. Nitrogen adsorption–desorption isotherms (a) and pore-size distribution curve (b) for the as-obtained  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> hollow spheres.

area and mesoporous structure would be beneficial to adsorb the organic pollutants onto the surface of the photocatalyst.

Fig. 4 shows the UV–visible absorption spectra of the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> hollow spheres. Pure TiO<sub>2</sub> only absorbed ultraviolet radiation of less than 400 nm (not shown here). The Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> composite hollow spheres samples exhibited not only stronger absorption in the ultraviolet region of less than 400 nm but also adequate and strong absorption in the visible light region of 400–700 nm. It indicates that Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> composite hollow spheres samples can also be used under visible light. In the present condition, Fe<sup>3+</sup> was doped with the TiO<sub>2</sub> lattice, resulted in the reduction of the band gap. Additionally, the band gap of Fe<sub>2</sub>O<sub>3</sub> was only 2.2 eV. The presence of this narrow band gap semiconductor, as well as the reduced TiO<sub>2</sub> band gap, greatly ameliorated the light absorption properties of the Fe<sub>2</sub>O<sub>3</sub>–TiO<sub>2</sub> composites, and enabled the absorption of visible light.

The photocatalytic activity of the samples was evaluated by monitoring the decoloration of RhB in aqueous solution. Prior to detailed evaluation of photocatalytic activities of the prepared samples, the concentration of RhB in the aqueous solution decreased slightly before illumination due to the adsorption in the dark. Most of the concentration decrease occurred during irradiation, and then the decrease in RhB concentration during irradiation was attributed to chemical reaction rather than adsorption. Additionally, it was also noticed that the degradation of RhB did not occur in the absence of catalysts under visible light under the experimental conditions. The low but not negligible activity can be attributed to a self-photosensitized process [29,29]. However, the efficiency is very low because of the fairly slow interfacial electron-transfer rate. For convenience, we assumed that the concentration of RhB after desorption–adsorption equilibrium was the initial concentration,  $C_0$ . Fig. 5a shows the temporal evolution of the absorption spectra of RhB under visible light irradiation in the presence of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> hollow spheres. The absorption peak at 553 nm drops gradually with increasing irradiation time and almost disappears after 180 min, indicating that the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> hollow spheres exhibits excellent photocatalytic activity during the reaction. Fig. 4b shows the effects of visible light irradiation time (t) on the normalized

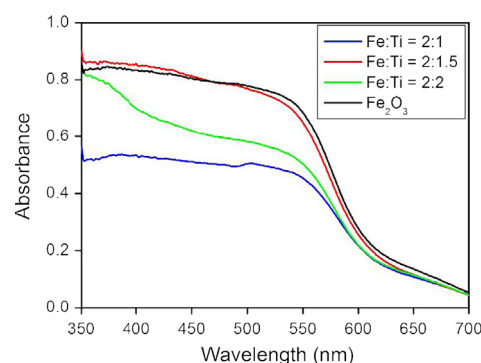


Fig. 4. UV–vis absorption spectrum of Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> composite hollow spheres.

RhB concentration ( $C_t/C_0$ ) for the RhB aqueous solutions containing different catalysts. It can be seen that when RhB solution was irradiated in the absence of photocatalyst, a decoloration rate of less than 10% after 180 min was detected, which is considered to come from the photolysis of organic dye RhB. However, when P25, pure Fe<sub>2</sub>O<sub>3</sub>,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> hollow spheres photocatalysts were employed, the decoloration rate of organic dye RhB after 180 min was about 50%, 75% and 98% respectively. To determine the mineralization degree of organic dye RhB, TOC analyses were performed on dispersions at different visible-light irradiation times. It can be clearly seen from TOC data (Fig. 5c) that TOC diminishes gradually as the exposure time increases and ca. 42% TOC removal is obtained at 360 min. It is then conclusive that the decoloration of RhB in the presence of photocatalyst is majorly attributed to the photocatalytic degradation of RhB along with slight photolysis of dye under visible light irradiation. Moreover, the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> hollow spheres exhibit better visible-light photocatalytic activity than that of the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> or pure TiO<sub>2</sub> sample. The enhanced visible-light photocatalytic activity can be attributed to the increase in specific surface area, the separation of photo-induced electron–hole pairs and hollow microarchitecture [30]. The magnetic properties of photocatalysts are favorable in the water treatment because it can be conveniently separated by magnetic separation technology [31]. Magnetic hysteresis measurement for  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> hollow spheres is carried out in applied magnetic field at

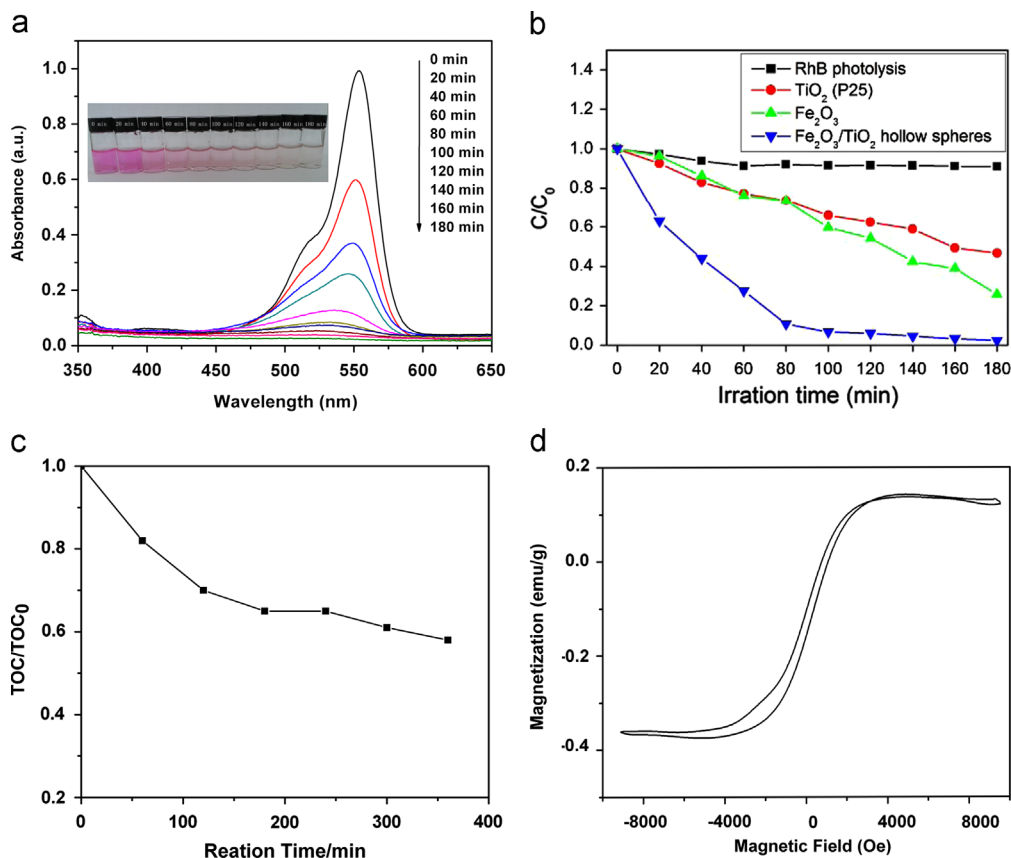


Fig. 5. Absorption spectra of the RhB solution containing  $\alpha-Fe_2O_3/TiO_2$  hollow spheres after different irradiation times with visible light (a); normalized RhB concentration versus visible light irradiation time for the RhB aqueous solutions with the different catalysts (b); Changes in solution TOC during RhB degradation (c). Initial solution:  $10 \mu mol L^{-1}$  RhB +  $25 \mu mol L^{-1}$   $\alpha-Fe_2O_3/TiO_2$  hollow spheres; 200 W metal halide lamp, pH 7.0; and Magnetic hysteresis loops of  $\alpha-Fe_2O_3/TiO_2$  hollow spheres (d).

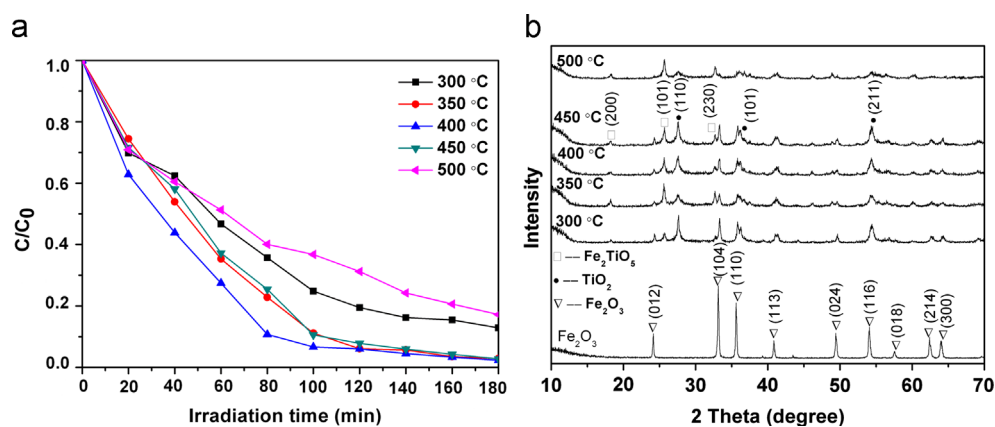


Fig. 6. Comparison of the photocatalytic activity of the samples after calcination at different temperature (300–500 °C) (a); XRD patterns of the products prepared after calcination at different temperature (300–500 °C) (b).

room temperature, with the field sweeping from  $-18$  to  $18$  kOe. As shown Fig. 4c, the magnetization measurement result of the  $\alpha-Fe_2O_3/TiO_2$  hollow spheres is indicative of the presence of ferromagnetic components. The saturation magnetization (Ms) is  $8.4$  emu/g. In addition, from Fig. 4c it can be seen that no saturation of the magnetization as a function of the field is observed up to the maximum applied magnetic field.

To further study the photocatalytic activity of the  $\alpha-Fe_2O_3/TiO_2$  hollow spheres, the time dependence of the relative concentration of RhB for the samples calcined at different temperature (300–500 °C), is depicted in Fig. 6a. It was easily be found that the photocatalytic activity of the sample  $\alpha-Fe_2O_3/TiO_2$  hollow spheres increased gradually with the increasing calcination temperature up to  $400$  °C. With the

further increase of the temperature to 450 °C, the photocatalytic activity decreased slightly. When the calcinations temperature reached 500 °C, the sample has the lowest photocatalytic activity. Fig. 5b shows XRD patterns of the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> samples calcined at different temperature for 4 h. The XRD results indicate that the sample calcined at 300 °C consists of both  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> and rutile TiO<sub>2</sub> along with a small amount of iron titanium oxide phase (Fe<sub>2</sub>TiO<sub>5</sub>). With the increase of calcination temperature (T), the intensities of diffraction peaks of Fe<sub>2</sub>TiO<sub>5</sub> increased. However, When T reached 500 °C, the relative peak intensities of TiO<sub>2</sub> phase decreases obviously, indicated low photocatalytic activity. The improved photocatalytic activity of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> samples calcined at 350–450 °C might be attributed to their good crystallinity, fewer defects and low recombination rate of the

electron–hole pair due to the synergetic effect among the components of TiO<sub>2</sub>, Fe<sub>2</sub>TiO<sub>5</sub>, Fe<sub>2</sub>O<sub>3</sub> in the nanostructure.

In addition, it is found that the molar ratio of iron to titanium ( $R=n(\text{Fe})/n(\text{Ti})$ ) plays a vital role on the photocatalytic activity of the final products. The XRD patterns of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> obtained at different  $R$  calcined at 400 °C are shown in Fig. 7a. At  $R=2:1$ , the sample consists of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> and rutile phase. When  $R$  increased to 2:1.5 and 2:2, the peaks located at 2 theta values of 18.09°, 25.55° and 32.51° suggested the appearance of a Fe<sub>2</sub>TiO<sub>5</sub> phase (JCPDS No. 41-1432), which was mixed with  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> and rutile phase. However, further increase  $R$  to 2:3, the sample is composed of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> and rutile phase, and no Fe<sub>2</sub>TiO<sub>5</sub> phase was observed. Fig. 7b–d show TEM images of as-prepared  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> obtained at various  $R$ . Fig. 6c–d clearly reveal the structure of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/

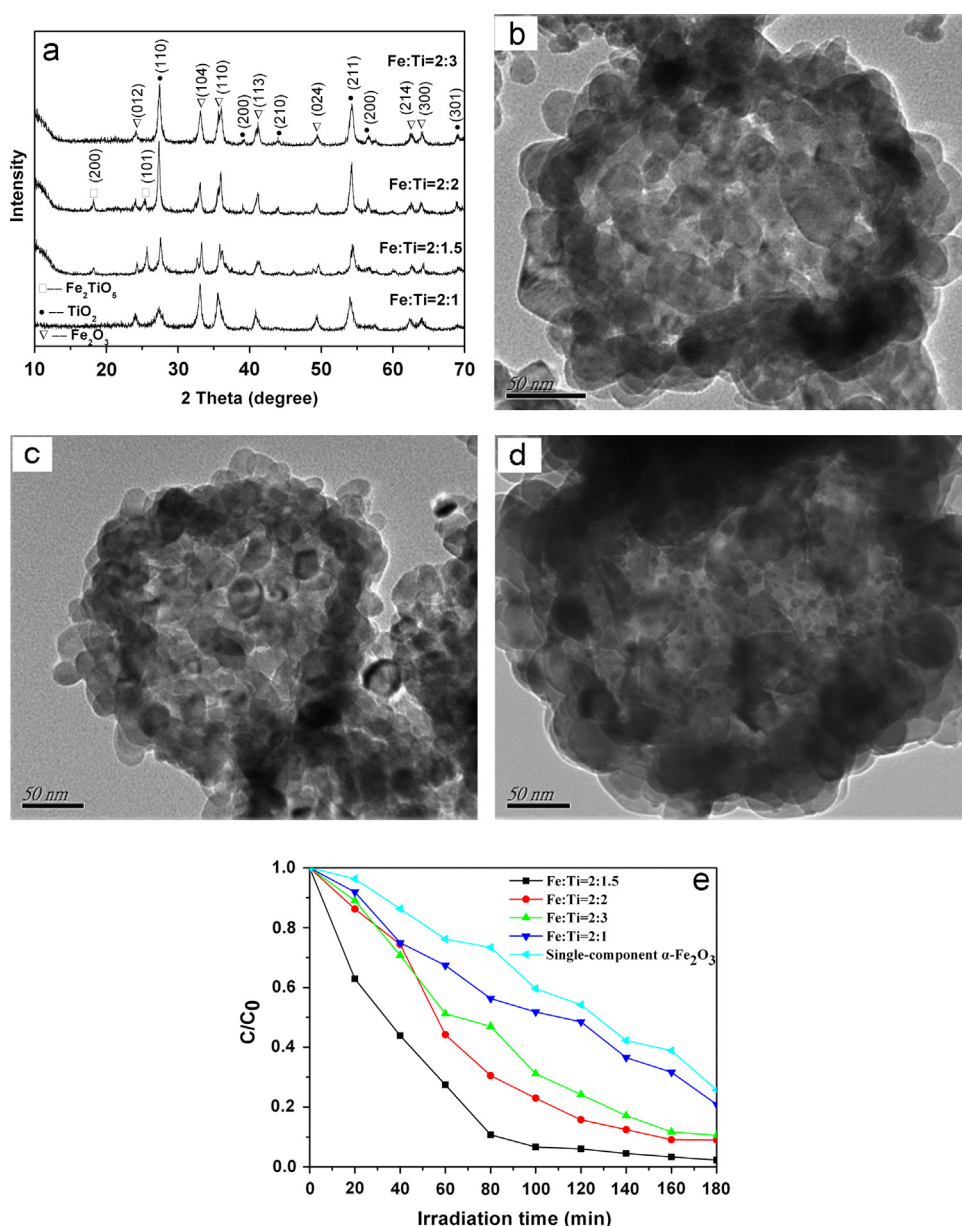


Fig. 7. XRD patterns (a) of the products prepared with varying  $R$  calcined at 400 °C for 4 h; TEM images of the products prepared with varying  $R$ : (b) 2:1, (c) 2:2, (d) 2:3, and comparison of the photocatalytic activity of the samples with varying  $R$  (e).

TiO<sub>2</sub> composite hollow spheres was composed of randomly aggregated nano-particles with sizes of about 18 nm. Moreover, the doped TiO<sub>2</sub> could hinder slightly the increase of the composite nanoparticles size.

The photocatalytic activity of the samples with varying *R* was evaluated by monitoring the degradation of RhB in aqueous solution. *R* exhibited obvious influence on the photocatalytic activity of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> composite hollow spheres (Fig. 7e). Compared with pure  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> composite hollow spheres with different mass ratios all exhibit better photocatalytic activity. In particular, the photocatalytic activity of the samples obtained at *R*=2:1.5 reaches a maximum value. The high photocatalytic activity might be partially due to its more intense absorption in the visible light range (Fig. 4), more red shift in the band gap transition, and larger surface compared with other Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> composites, which resulted in more photogenerated electrons and holes participating in the photocatalytic reactions under visible light and more pollutants adsorbed on the surface of the photo catalyst.

#### 4. Conclusions

In summary,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> composite hollow spheres were successfully synthesized on a large scale by a controlled template-assisted hydrothermal precipitation reaction. The results indicated that the composite spheres show a high photocatalytic activity on the degradation of an RhB solution. Calcination temperature and the molar ratio of titanium to iron (*R*) should play important roles on photocatalytic activity of the samples. Moreover, the prepared samples can be more readily separated and reused from the slurry system by the magnetic separation after photocatalytic reaction compared to conventional photocatalysts. In addition, this work also provided a simple and general method for fabricating nanocomposites of magnetic metal oxide and other metal oxide, which may find wider applications involve drug delivery, microcontainers, microreactors, sensors, and so forth.

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