

Combustion synthesis and spectroscopic characterisation of LaAlO_3 nanophosphors doped Er^{3+} ions[☆]

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Abstract

Lanthanum aluminate (LaAlO_3) nanophosphors doped with erbium were successfully synthesized by a combustion process. The powders were characterized X-ray diffraction (XRD), scanning electron microscopy (SEM), transmission electron microscopy (TEM) and fluorescence spectroscopy. Pure LaAlO_3 phase was obtained at 800 °C heated for 4 h, with an average crystal size, as determined by TEM, of 60 nm. Infrared emission spectra and decay times of main luminescence transitions were measured at room temperature. A strong emission is reported at 1550 nm from the ($^4I_{13/2} \rightarrow ^4I_{15/2}$) transition, which intensity depends on the Er concentration.

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1. Introduction

Lanthanum aluminate (LaAlO_3) with a perovskite-type structure has at room temperature a rhombohedral structure with R-3c symmetry and it undergoes phase transition into the Pm-3 m cubic phase at $T_c = 527$ °C [1]. Typically, LaAlO_3 has been prepared by conventional solid-state reaction of Al_2O_3 and La_2O_3 in the temperature range of 1500–1700 °C [2,3]. But this method suffers from many inherent shortcomings, such as the high-temperature heat treatment which has a detrimental effect on the grain size.

Recently LaAlO_3 has been successfully prepared by a novel two-step process [4] and microwave irradiation [5]. Moreover, various wet and soft chemical methods including the polymerized complex method using citric acid and ethylene glycol route have been reported [6]. Several low temperature (750–900 °C) chemical routes are used for preparing finer and

homogeneous powders of LaAlO_3 like polyvinyl alcohol (PVA) with metal nitrate synthesis [7], sol–gel process [8–10], EDTA gel route [11,12], co-precipitation method [13,14], pyrolysis using triethanolamine [15] and combustion synthesis with urea and hydrazine as fuel [16–19].

This paper presents the synthesis and characterization of $\text{LaAlO}_3:\text{Er}^{3+}$ phosphors prepared by combustion synthesis [20,21], which has the advantage of being simple, fast and economical in doping. The structural details and the optical properties of the synthesized phosphors have been investigated by X-ray diffraction (XRD), transmission electron microscopy (TEM), scanning electron microscopy (SEM), and fluorescence spectroscopy. The photoluminescence emission and excitation as well as the decay times were measured.

2. Experimental procedure

The starting materials were lanthanum nitrate hexahydrate [$\text{La}(\text{NO}_3)_3 \cdot 6\text{H}_2\text{O}$] (98%), aluminum nitrate nonahydrate [$\text{Al}(\text{NO}_3)_3 \cdot 9\text{H}_2\text{O}$] (99%), erbium(III) nitrate pentahydrate [$\text{Er}(\text{NO}_3)_3 \cdot 5\text{H}_2\text{O}$], and glycine [$\text{H}_2\text{NCH}_2\text{COOH}$] (99%).

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$\text{La}(\text{NO}_3)_3 \cdot 6\text{H}_2\text{O}$ and $\text{Al}(\text{NO}_3)_3 \cdot 9\text{H}_2\text{O}$ $\text{Er}(\text{NO}_3)_3 \cdot 5\text{H}_2\text{O}$ and $\text{H}_2\text{NCH}_2\text{COOH}$ were dissolved in distilled water. Er^{3+} ions doped lanthanum aluminate with general formula $(\text{La}_{1-x}\text{Er}_x)\text{AlO}_3$ were prepared with different concentrations of Er ($x=2\%$, 5% , 10%). During the process, the molar ratio of glycine to total metal cations concentration G/M was 2 and the cation ratio of La:Al was 1:1. Glycine was used as a fuel. The resulting solution was magnetically stirred at 85°C to get a clear and uniform solution. The solution slowly turned viscous on continued heating for about 1.5 h, becoming a highly viscous gel. Throughout the process, no signs of precipitation or turbidity were observed. The gel was put into a vacuum oven and kept at 110°C for 24 h, undergoing rapid dehydration and foaming followed by decomposition and generating a solid precursor. Finally this solid precursor was ground to fine powders and was calcined at different temperatures (600°C , 700°C and 800°C) for 4 h to obtain pure LaAlO_3 .

3. Experimental

The X-ray powder diffraction (XRD) patterns of all the samples were recorded on a Philips X'Pert system (PW3020 vertical goniometer and PW3710 MPD control unit) with $\text{CuK}\alpha_{1,2}$ radiation ($\lambda_1=1.54059\text{ \AA}$ and $\lambda_2=1.54442\text{ \AA}$). In order to improve the signal to noise ratio, at least three runs (collected with 10 s/step and $0.05^\circ/\text{step}$) were measured.

The scanning electron images of samples were recorded with scanning electron microscope (SEM) JEOL JSM-5600LV, operated at 20 kV equipped with an Oxford Instruments ISIS series 300 EDS detector.

The morphology of products was characterized by transmission electronic microscopy (TEM) (Tecnai G2 ultra Twin). TEM images were taken at 300 kV with a JEOL JEM-3010 instrument, with an ultra-high resolution pole-piece (0.17 nm point resolution), equipped with a Gatan multi-scan CCD camera (Mod. 794) and an oxford EDS microanalysis detector. The powdered samples were dispersed in ethanol by sonication for approximately 5 min and deposited onto a holey carbon film grid.

Photoluminescence measurements were performed with a Fluorolog3-21 system (Horiba Jobin Yvon). A 450 W xenon arc lamp was used as a broadband excitation source and a double Czerny-Turner monochromator was used to select the excitation wavelength for photoluminescence excitation. All emission spectra were obtained using the same amount of powder, measured at room temperature and recorded under the same conditions.

The analysis of the emitted luminescence signal from the samples was performed by using a iHR320 single grating monochromator and a R5509 Hamamatsu photomultiplier tube detector. Time resolved characterization was recorded in multi channel single photon counting (MCSPC) mode by exciting the sample with a Nd-YAG laser NT 342/3/UVE/AW Ekspla, which can be tuned in the whole range $210\text{--}2300\text{ nm}$. The pulse duration is about 6 ns at 10 Hz repetition rate.

The particle size was estimated from the X-ray line broadening of the (110) diffraction peak using the Scherrer

formula [22].

$$D = 0.9\lambda / \beta \cos \theta, \quad (1)$$

where D is the particle size in nm, λ the radiation wavelength, θ the diffraction peak angle and β is the corrected line width at half-peak intensity. The correction for instrumental peak broadening was made using the Warren formula: $\beta = (b_{\text{obs}}^2 - b^2)^{1/2}$, where b_{obs} is the line width at half-peak intensity related to LaAlO_3 powder and b is the line width of the (110) diffraction peak of the LaAlO_3 .

4. Results and discussion

4.1. X-ray diffraction

The X-ray diffraction patterns of $\text{LaAlO}_3:\text{Er}^{3+}$ are shown in Fig. 1. According to XRD analysis (Fig. 1), the LaAlO_3 obtained at 800°C crystallizes in a pure rhombohedral perovskite structure (JCPDS no. 01-082-0478) with space group $R\bar{3}c$ (No.167), with unit cell dimensions $a=5.37\text{ \AA}$ and $c=13.10\text{ \AA}$.

So all diffraction peaks in these XRD patterns could be attributed to the rhombohedral perovskite crystal structure of LaAlO_3 which confirm that the doping concentration does not influence the crystalline phase formation. The particles size of LaAlO_3 powder calcined at 800°C for 4 h calculated from the Scherrer formula is about 60 nm .

4.2. TEM and SEM analysis

Fig. 2a presents TEM micrographs of $\text{LaAlO}_3:\text{Er}$ (5%) powders calcined at 800°C , showing that the powders are composed by monocrystalline nanoparticles and exhibit the formation of aggregates among them. The nanoparticles have a polyhedral morphology with particle size about 60 nm , which agrees very well with the results given by the Scherrer formula.

Fig. 2b illustrates the high resolution TEM images (HRTEM), pointing out that the samples are perfectly crystalline, as can be seen by the uniform distribution of lattice planes.

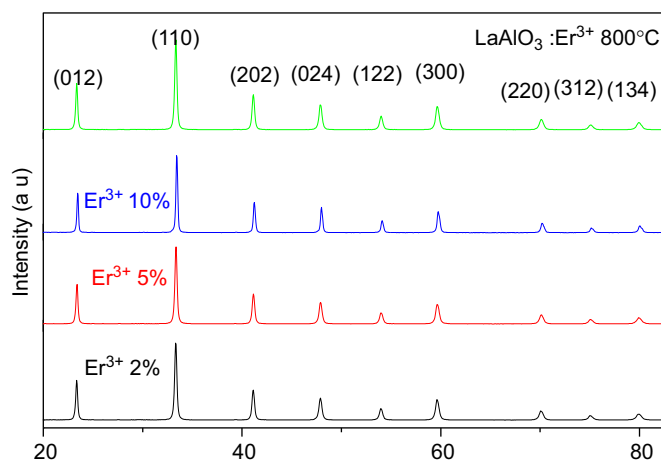


Fig. 1. X-ray diffraction patterns of $\text{LaAlO}_3:\text{Er}^{3+}$ observed and doping concentration variation.

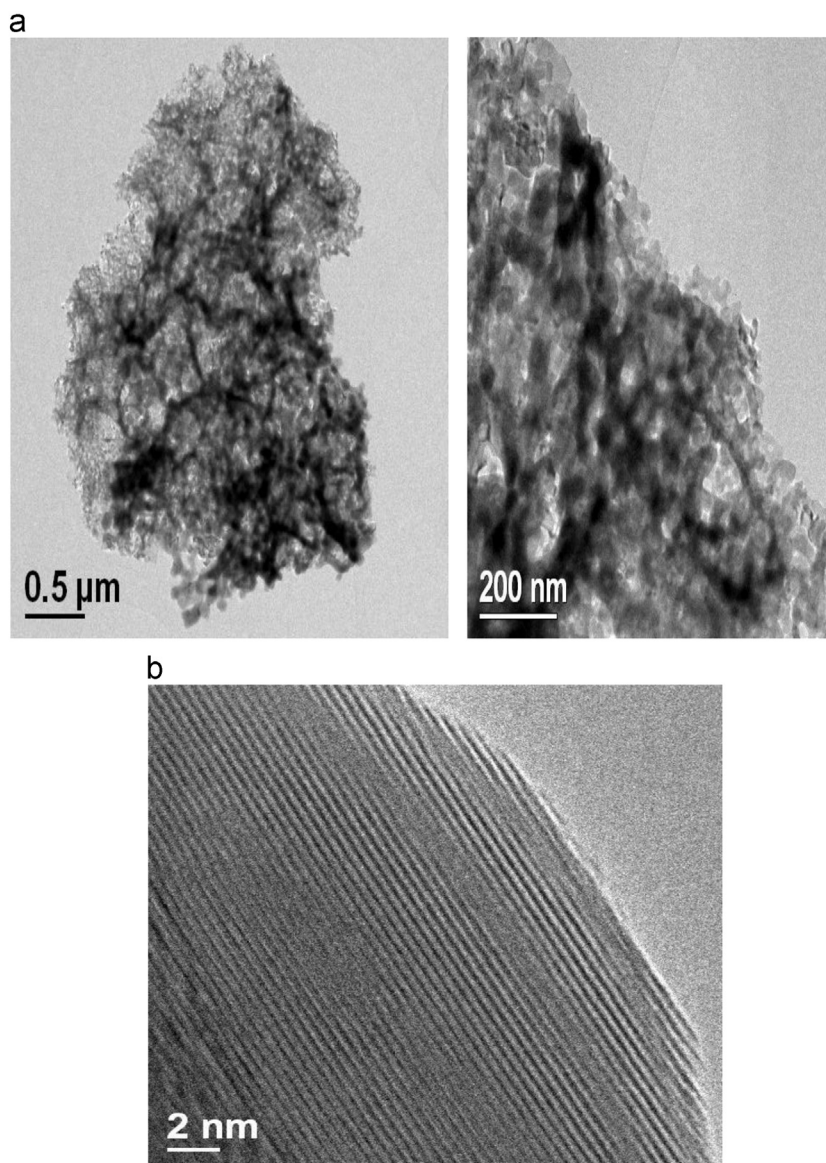


Fig. 2. (a) TEM micrograph and (b) HRTEM images and of LaAlO₃:Er (5%) obtained at 800 °C.

Fig. 3. shows SEM micrographs of LaAlO₃:Er (5%) annealed at 800 °C. The micrographs clearly indicate the porosity and agglomeration of polyhedral crystallites, which have no uniform shapes and sizes. This porosity can be assigned to the non-uniform distribution of temperature and mass flow in the combustion process. These porous powders are highly friable which facilitates easy grinding to obtain finer particles.

4.3. Photoluminescence studies

Optical properties of Er³⁺ doped LaAlO₃ was investigated first by Deren and all [23].

The excitation spectrum recorded at 1550 nm of 2% Er³⁺ doped LaAlO₃ is shown in Fig. 4.

The excitation spectrum consists of two intense peaks at 380 nm and 520 nm, which are assigned to (⁴I_{15/2} → ⁴G_{11/2}) and (²H_{11/2} → ⁴I_{15/2}), followed by weak shoulders at 365 nm,

406 nm, 448 nm, 486 nm, 540 nm and 648 nm. This peaks are attributed to (⁴I_{15/2} → ⁴G_{9/2}),

(⁴I_{15/2} → ⁴H_{9/2}), (⁴I_{15/2} → ⁴F_{5/2}), (⁴I_{15/2} → ⁴F_{7/2}), (⁴S_{3/2} → ⁴I_{15/2}) and (⁴F_{9/2} → ⁴I_{15/2}) respectively.

The emission spectra under 378 nm excitation of Er³⁺ doped LaAlO₃ with different concentrations (2%, 5% and 10%) are shown in Fig. 5. The transition (⁴I_{13/2} → ⁴I_{15/2}) located at 1550 nm is the most intense.

The photoluminescence emission intensity increases with increasing Er³⁺ concentration (Table 1) from 2% to 5% and then decreases for higher concentrations, suggesting the occurrence of concentration quenching. The reason for concentration quenching is that the possibility of interaction between Er³⁺ ions also increases with increasing Er³⁺ concentration, allowing energy transfer and migration throughout the material until a defect can cause non radiative recombination. Therefore an excessive doping of rare earth ions

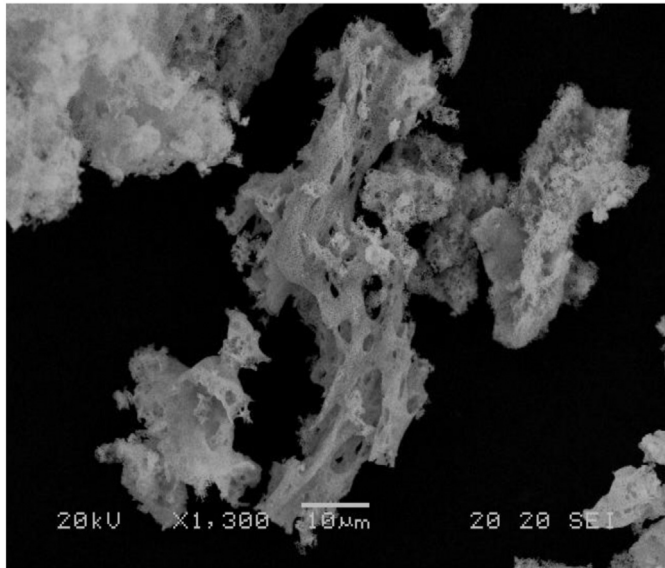


Fig. 3. SEM images of LaAlO₃:Er (5%) obtained at 800 °C.

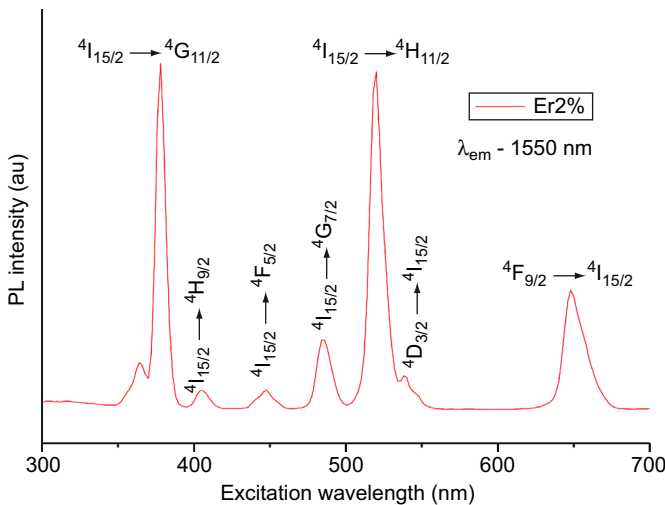


Fig. 4. Excitation spectrum of LaAlO₃:Er (2%).

should be avoided because it is detrimental for the phosphor efficiency.

Fig. 6 reports the decay time of ($^4I_{13/2} \rightarrow ^4I_{15/2}$) at 1550 nm. These curves are not decreasing in a single exponential way, so they were fitted by a double exponential function. For this reason the values of lifetimes were obtained by an average calculation [24]:

$$\tau_{\text{avg}} \cong \frac{A_1 \tau_1^2 + A_2 \tau_2^2}{A_1 \tau_1 + A_2 \tau_2}$$

For the three concentration of Er³⁺ ions (2%,5%, 10%) we can see a decreasing of lifetime with increasing concentration: from 6.90 ms for the sample containing 2% of Er³⁺ ions to 5.87 ms for the sample containing 5% and 3.86 ms for the 10% concentration sample, in agreement with the hypothesis of concentration quenching [21].

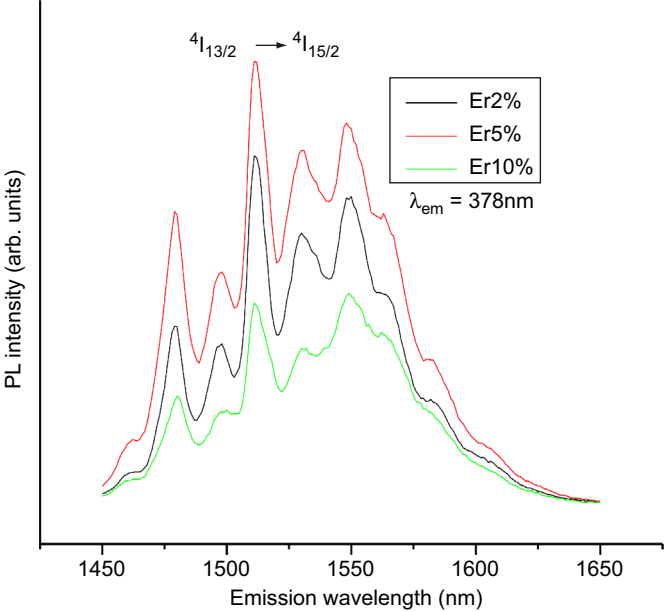


Fig. 5. Emission spectra of LaAlO₃:Er (2%,5%,10%) obtained at 800 °C.

Table 1
Dependence of the relative emission intensity of Er³⁺ on its doping concentration for Er-doped LaAlO₃.

Er concentration(%)	Normalized intensities of $^4I_{13/2} \rightarrow ^4I_{15/2}$
2%	0.78
5%	1
10%	0.45

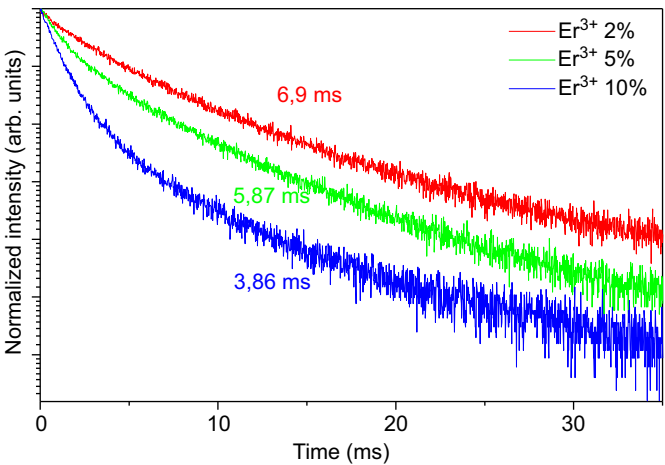


Fig. 6. Decay time of LaAlO₃:Er (2%, 5% and 10%).

5. Conclusion

A pure LaAlO₃ with a perovskite structure was obtained at 800 °C using a combustion method. The TEM image shows that we obtain a nanopowder with a particle size of about 60 nm. A strong luminescence at 1550 nm in the infrared (IR)

region due to the ($^4I_{13/2} \rightarrow ^4I_{15/2}$) transition has been observed in $\text{LaAlO}_3:\text{Er}^{3+}$ phosphors. This intensity increases up to 5% of Er^{3+} ions in the sample, and then it decreases because of concentration quenching.

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