

Effects of Nd and Co co-doping on phase, microstructure and ferromagnetic properties of bismuth ferrite ceramics

Pornchanok Lawita^a, Pasinee Siriprapa^a, Anucha Watcharapasorn^{a,b},
Sukanda Jiansirisomboon^{a,b,*}

^aDepartment of Physics and Materials Science, Faculty of Science, Chiang Mai University, Chiang Mai 50200, Thailand

^bMaterials Science Research Center, Faculty of Science, Chiang Mai University, Chiang Mai 50200, Thailand

Available online 13 October 2012

Abstract

In this work, the samples of $\text{Bi}_{0.95}\text{Nd}_{0.05}\text{Co}_x\text{Fe}_{1-x}\text{O}_3$ (BNFCO) with $x=0, 0.03, 0.05$ and 0.07 mol fraction, were prepared by a solid state reaction. The effect of Nd and Co co-doping concentration on phase, densification, microstructure and ferromagnetic properties were examined. The BNFCO powders were prepared using a mixed-oxide method and calcined at 800°C for 2 h before being pressed and sintered at various temperatures in the range of $825\text{--}900^\circ\text{C}$ for 2 h. An increase in Co co-doping content increased density of the ceramics. Phase analysis by X-ray diffraction indicated the existence of rhombohedral phase for all BNFCO powders and ceramics. Microstructural investigation using the scanning electron microscope showed a reduction of grain size with increasing Co content. Magnetic hysteresis loops showed that remanent magnetization and coercive magnetic field of the Co-doped samples were improved. © 2012 Elsevier Ltd and Techna Group S.r.l. All rights reserved.

Keywords: A. Sintering; B. X-ray methods; C. Magnetic properties; D. Ferrites

1. Introduction

The multiferroic compound of BiFeO_3 or BFO has drawn a great amount of interest because of its rhombohedrally distorted perovskite-type structure that can be represented by a general formula ABO_3 , which belongs to the space group $R3c$ at room temperature. This allows a coexistence up to quite high temperature of ferroelectricity ($T_C \sim 850^\circ\text{C}$) and G-type antiferromagnetism ($T_N \sim 370^\circ\text{C}$) [1,2]. Therefore, it has attracted considerable attention because of their possible uses for future technology in information storage and sensors [3]. However, it has been known that BFO has a high leakage current and low resistance due to defects such as secondary phases and oxygen vacancies [4]. One way that has been investigated to improve properties of BFO is an A-site substitution, by displacement of the volatile Bi with rare earth elements such as La and Nd which used as substitute for Bi in BFO [5–10]. Mishra et al. reported that Nd-substituted BFO

nanoceramics showed high dielectric constant at high temperature and improvement in ferroelectric properties of BFO [9].

Recently, many research groups attempted to further improve magnetic characteristic of BFO, particularly enhancement of ferromagnetic properties of BFO ceramics by B-site substitution, by displacement of Fe in BFO with magnetic elements such as Co [11–13]. Apart from this, Zheng et al. reported that the substitution of Co in BFO ceramics had high remanent magnetization and improvement in magnetic properties of BFO [13].

The aim of this present work is to prepare and study the effect of Nd and Co co-doping on phase, densification, microstructure and ferromagnetic properties of BFO ceramics. The ceramic processing was carried out using a simple and cost effective solid-state mixed-oxide method. Phase, microstructure and ferromagnetic properties were investigated and discussed.

2. Material and methods

$\text{Bi}_{0.95}\text{Nd}_{0.05}\text{Co}_x\text{Fe}_{1-x}\text{O}_3$ or BNFCO ($x=0, 0.03, 0.05$, and 0.07 mol fraction) powders were prepared using the

*Corresponding author at: Department of Physics and Materials Science, Faculty of Science, Chiang Mai University, Chiang Mai 50200, Thailand. Tel.: +66 53 941921x631; fax: +66 53 943445.

E-mail address: sukanda.jian@cmu.ac.th (S. Jiansirisomboon).

solid-state mixed-oxide method. The starting binary oxide powders were Bi_2O_3 (99.9%, Aldrich), Nd_2O_3 (99.9%, Aldrich), Co_3O_4 (99%, Aldrich) and Fe_2O_3 (99%, Riedel-de Haën). The powders were weighed according to their stoichiometric compositions with 5% Bi_2O_3 excess. These oxides were mixed in ethanol, ball milled for 24 h, and dried in an oven for 24 h. The dried mixtures were calcined at 800 °C for 2 h and then ground. The BNFCO powders were then pressed under uniaxial hydraulic at a pressure of 1 t using a few drops of 3 wt% polyvinyl alcohol as a binder. The pellets were sintered at various temperatures in the range of 825–900 °C for 2 h. The optimum sintering temperature for high density ceramics was determined and selected for further characterizations. Phases of the selected ceramics were characterized using an X-ray diffractometer (XRD, Philips model X-pert). Density was measured by Archimedes' method. Microstructure of the ceramics was investigated by scanning electron microscope (SEM, JEOL JSM-6335F). Magnetic hysteresis (M–H) loops were characterized using a vibrating sample magnetometer (VSM, Lake Shore 7400 Series).

3. Results and discussion

X-ray diffraction patterns of $\text{Bi}_{0.95}\text{Nd}_{0.05}\text{Co}_x\text{Fe}_{1-x}\text{O}_3$ (BNFCO) powders were obtained at a calcination temperature of 800 °C for 2 h dwell time as shown in Fig. 1. The patterns of BNFCO powders with different concentration of Co_3O_4 dopant were well matched with JCPDS No. 86-1518, which indicated an existence of polycrystalline rhombohedral distorted perovskite structure with a space group $R3c$. These patterns could be well indexed with the pattern of BFO, but there was also a small amount of a second phase presented. This phase was identified to be $\text{Bi}_{25}\text{FeO}_{40}$ which likely occurred due to an excessive Bi used for compensating volatilization during synthesis [14].

X-ray diffraction patterns of BNFCO ceramics sintered at 850 °C for 2 h are shown in Fig. 2. The patterns for all samples were indexed as rhombohedral distorted

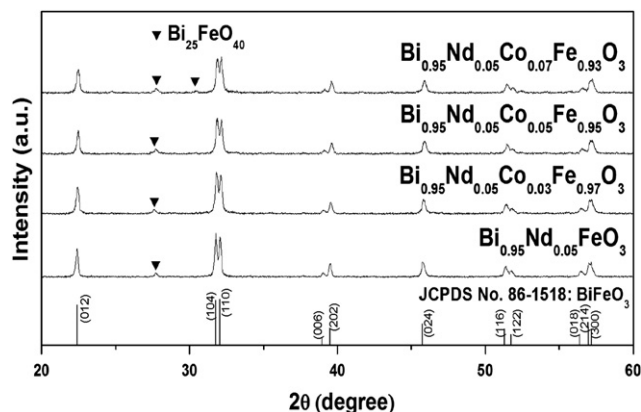


Fig. 1. X-ray diffraction patterns of $\text{Bi}_{0.95}\text{Nd}_{0.05}\text{Co}_x\text{Fe}_{1-x}\text{O}_3$ calcined powders.

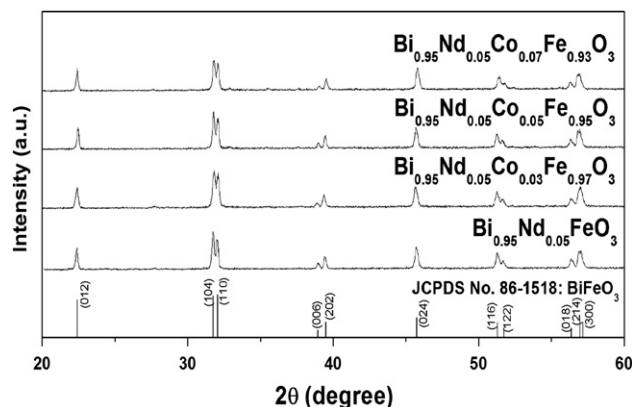


Fig. 2. X-ray diffraction patterns of $\text{Bi}_{0.95}\text{Nd}_{0.05}\text{Co}_x\text{Fe}_{1-x}\text{O}_3$ ceramics sintered at 850 °C for 2 h.

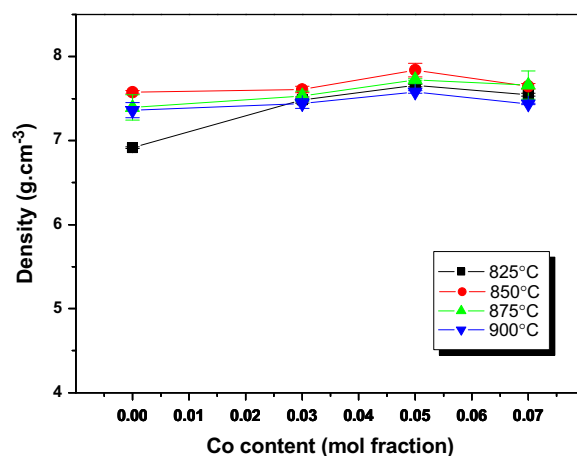


Fig. 3. Relationship between density and Co doping content of $\text{Bi}_{0.95}\text{Nd}_{0.05}\text{Co}_x\text{Fe}_{1-x}\text{O}_3$ ceramics.

perovskite structure similar to those found in the powders but without the presence of a second phase.

Fig. 3 illustrates a relationship between density of BNFCO ceramics and content of Co dopant at different sintering temperatures. Densities of the BNFCO ceramics were found to increase with increasing Co content. At 850 °C, it could be seen that the density values were rather similar for all samples. Increasing sintering temperature > 850 °C tended to reduce densities of all ceramics. From the density results, it was quite clear that the optimum sintering temperature for BNFCO ceramics was found to be 850 °C. However, the density values obtained at each temperature were rather close in a range of 7.6–7.8 g cm^{−3}.

SEM micrographs of sintered surfaces of BNFCO ceramics are shown in Fig. 4. Grain sizes of all samples were found to decrease with increasing Co content and homogeneous microstructure was also developed. Increasing Co content, the grains of BNFCO ceramics had more angular shape. It is believed to occur due to the surface energy of Co (2695 ergs/cm²) being higher than that of Fe (2525 ergs/cm²) [15], thus it is reasonable that Co doped

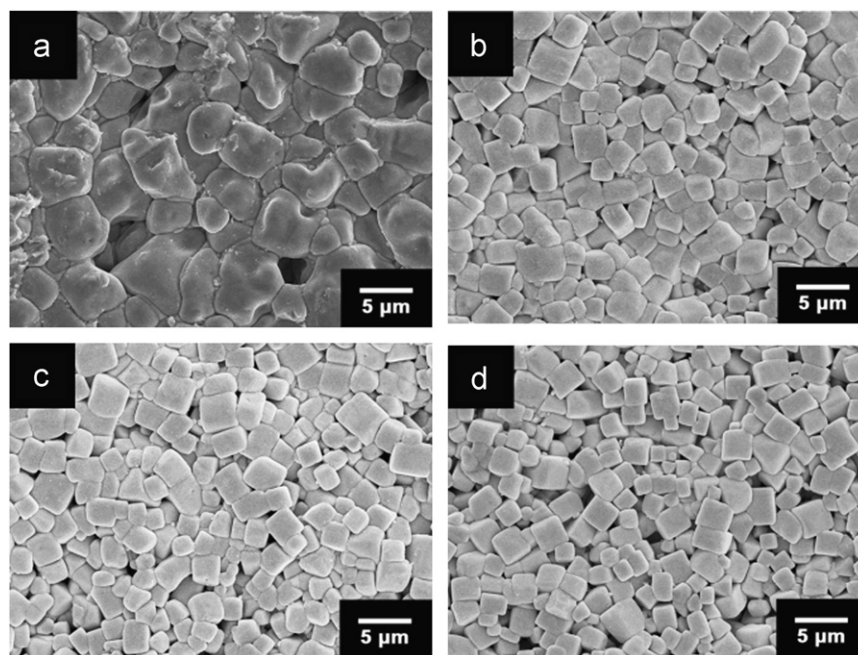


Fig. 4. SEM micrographs of $\text{Bi}_{0.95}\text{Nd}_{0.05}\text{Co}_x\text{Fe}_{1-x}\text{O}_3$ ceramics; (a)–(d) indicates $x=0, 0.03, 0.05$ and 0.07 mol fraction, respectively.

BNFO may lead to an increase in surface energy. From this result, the system would decrease the surface energy by reducing grain size. The result also showed that Co substitution resulted in an equilibrium shape of BNFCO crystalline while it was growing. Therefore, the grains were found to have angular shape rather than curved surfaces. This was because the angular surface had surface energy less than curved surfaces.

Fig. 5 shows plots of magnetic hysteresis (M–H) loops of BNFCO samples. The remanent magnetization and coercive magnetic field of the samples were improved due to Co doping. Under a maximum magnetic field of 7675 Oe, H_c values of $\text{BNFCO}_{x=0.03}$, $\text{BNFCO}_{x=0.05}$ and $\text{BNFCO}_{x=0.07}$ ceramics were 366 Oe, 390 Oe and 377 Oe, and M_r values were 0.21 emu/g, 0.53 emu/g and 0.92 emu/g, respectively. It was found that the remanent magnetization increased with increasing Co content. This is believed to occur due to the relative magnetic permeability (μ_r) of Co being 250, which shows the ferromagnetic behavior ($\mu \gg 1$) [16]. Co substitution may lead to an increase in ferromagnetic behavior of undoped sample. Therefore, the relative magnetic permeability of BNFCO should be increased with increasing Co content. Therefore, it is reasonable that Co doped BNFO may also lead to an increase in the remanent magnetization. Although the saturation magnetization of Fe ($M_s=1707$ Gauss) is more than that of Co ($M_s=1400$ Gauss) [16], but Co substituted into the lattice of BNFO ceramics leads to an increase in hard magnet behavior. This result was caused by a decrease in random switching of magnetic moment which was caused by Co ions. Moreover, a reduction of coercive magnetic field was observed with increasing Co-doping content. This was due to the addition of Co leading to a decrease in the distance between magnetic particles. As a result, the coupling of magnetic moment was increased.

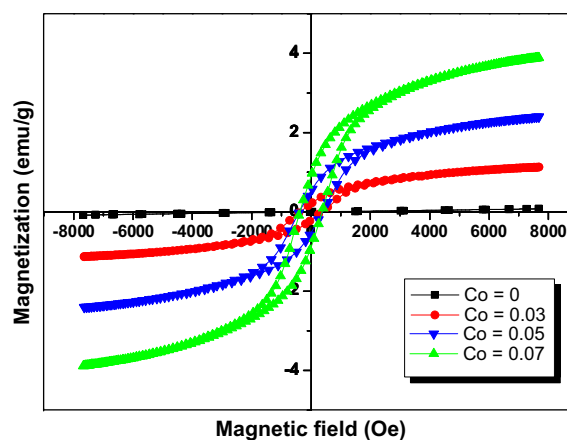


Fig. 5. Magnetic hysteresis loops of $\text{Bi}_{0.95}\text{Nd}_{0.05}\text{Co}_x\text{Fe}_{1-x}\text{O}_3$ ceramics.

Therefore, the external magnetic field to reverse the direction of magnetic moment was more easily.

4. Conclusions

In this study, $\text{Bi}_{0.95}\text{Nd}_{0.05}\text{Co}_x\text{Fe}_{1-x}\text{O}_3$ (when $x=0, 0.03, 0.05$ and 0.07 mol fraction) ceramics were successfully prepared by the solid state mixed-oxide method. BNFCO ceramics were identified by X-ray diffraction method as a rhombohedral structure. Increasing Co-doping content was found to decrease grain size and increase density of the ceramics. Addition of Co in BNFCO ceramics resulted in more angular shape. The remanent magnetization was found to increase while the coercive magnetic field decreased with increasing of Co-doping content.

Acknowledgments

This work is supported by the Thailand Research Fund (TRF), the National Research University Project under Thailand's Office of the Higher Education Commission (OHEC), the Faculty of Science and the Graduate School, Chiang Mai University.

References

- [1] I. Sosnowska, T. Peterlin-Neumaier, E. Steichele, Spiral magnetic ordering in bismuth ferrite, *Journal of Physics C: Solid State Physics* 15 (1982) 4835–4846.
- [2] J.R. Teague, R. Gerson, W.J. James, Dielectric hysteresis in single crystal BiFeO_3 , *Solid State Communications* 8 (1970) 1073–1074.
- [3] E. Ascher, H. Rieder, H. Schmid, H. Stössel, Some properties of ferromagnetoelectric nickel-iodine boracite, *Journal of Applied Physics* 37 (1966) 1404–1405.
- [4] V.R. Palkar, J. John, R. Pinto, Observation of saturated polarization and dielectric anomaly in magnetoelectric BiFeO_3 thin films, *Applied Physics Letters* 80 (2002) 1628–1630.
- [5] B.F. Yu, M.Y. Li, J. Wang, L. Pei, D.Y. Guo, X.Z. Zhao, Enhanced electrical properties in multiferroic BiFeO_3 ceramics co-doped by La^{3+} and V^{3+} , *Journal of Physics D: Applied Physics* 41 (2008) 185401–185405.
- [6] T. Kawae, H. Tsuda, H. Naganuma, S. Yamada, M. Kumeda, S. Okamura, A. Morimoto, Composition dependence in BiFeO_3 film capacitor with suppressed leakage current by Nd and Mn cosubstitution and their ferroelectric properties, *Japanese Journal of Applied Physics* 47 (2008) 7586–7589.
- [7] D.K. Pradhan, R.N.P. Choudhary, C. Rinaldi, R.S. Katiyar, Effect of Mn substitution on electrical and magnetic properties of $\text{Bi}_{0.9}\text{La}_{0.1}\text{FeO}_3$, *Journal of Applied Physics* 106 (2009) 024102–024111.
- [8] S. Karimi, I.M. Reaney, Y. Han, J. Pokorny, I. Sterianou, Crystal chemistry and domain structure of rare-earth doped BiFeO_3 ceramics, *Journal of Materials Science* 44 (2009) 5102–5112.
- [9] R.K. Mishra, D.K. Pradhan, R.N.P. Choudhary, A. Banerjee, Dipolar and magnetic ordering in Nd-modified BiFeO_3 nanoceramics, *Journal of Magnetism and Magnetic Materials* 320 (2008) 2602–2607.
- [10] A. Gautam, K. Singh, K. Sen, R.K. Kotnala, M. Singh, Crystal structure and magnetic property of Nd doped BiFeO_3 nanocrystallites, *Materials Letters* 65 (2011) 591–594.
- [11] Y.K. Jun, S.H. Hong, Dielectric and magnetic properties in Co- and Nb-substituted BiFeO_3 ceramics, *Solid State Communications* 144 (2007) 329–333.
- [12] W.S. Kim, Y.K. Jun, K.H. Kim, S.H. Hong, Enhanced magnetization in Co and Ta-substituted BiFeO_3 ceramics, *Journal of Magnetism and Magnetic Materials* 321 (2009) 3262–3265.
- [13] X. Zheng, Q. Xu, Z. Wen, X. Lang, D. Wu, T. Qiu, M.X. Xu, The magnetic properties of La doped and codoped BiFeO_3 , *Journal of Alloys and Compounds* 499 (2010) 108–112.
- [14] S.K. Singh, N. Menou, H. Funakubo, K. Maruyama, H. Ishiwara, (111)-textured Mn-substituted BiFeO_3 thin films on $\text{SrRuO}_3/\text{Pt}/\text{Ti}/\text{SiO}_2/\text{Si}$ structures, *Applied Physics Letters* 90 (2007) 242914.
- [15] T.A. Roth, The surface and grain boundary energies of iron, cobalt and nickel, *Materials Science and Engineering* 18 (1975) 183–192.
- [16] C. Kittel, *Introduction to Solid State Physics*, Seventh ed., John Wiley & Sons Ltd, New York, 1995.