

# Influence of calcination temperature on the piezoelectric properties of $\text{Ag}_2\text{O}$ doped $0.94(\text{K}_{0.5}\text{Na}_{0.5})\text{NbO}_3\text{--}0.06\text{LiNbO}_3$ ceramics

Moon-Soon Chae, Kyung-Su Lee, Jung-Hyuk Koh\*

*Department of Materials Engineering, Kwangju University, Seoul 139-701, South Korea*

Available online 16 October 2012

## Abstract

The effects of calcination temperature on the bulk density, piezoelectric, and ferroelectric properties were investigated for the  $\text{Ag}_2\text{O}$  doped  $0.94(\text{K}_{0.5}\text{Na}_{0.5})\text{NbO}_3\text{--}0.06\text{LiNbO}_3$  ceramics. The calcination temperatures were varied from 750 to 950 °C by 50 °C differences. An tetragonal XRD pattern, consistent with single-phase  $0.94(\text{K}_{0.5}\text{Na}_{0.5})\text{NbO}_3\text{--}0.06\text{LiNbO}_3$  was obtained after calcination at 850 °C for 2 h. And the experimental results showed that  $\text{Ag}_2\text{O}$  doped  $0.94(\text{K}_{0.5}\text{Na}_{0.5})\text{NbO}_3\text{--}0.06\text{LiNbO}_3$  ceramics calcined at 850 °C had a remnant polarization  $P_r=24.5\text{ }\mu\text{C}/\text{cm}^2$ , bulk density  $=4.32\text{ g}/\text{cm}^3$ , piezoelectric constant  $d_{33}=282\text{ pC}/\text{N}$  and electromechanical coefficient  $k_p=37.8\%$ .

© 2012 Elsevier Ltd and Techna Group S.r.l. All rights reserved.

**Keywords:**  $\text{Ag}_2\text{O}$ ; Lead-free; Calcination temperature

## 1. Introduction

Until now, piezoelectric ceramics could probably be represented by  $\text{Pb}(\text{Zr,Ti})\text{O}_3$  (PZT) based ceramics; such as  $\text{Pb}(\text{Ni,Nb})\text{O}_3\text{--}\text{Pb}(\text{Zr,Ti})\text{O}_3$  (PNN–PZT) [1],  $\text{Pb}(\text{Mg,Nb})\text{O}_3\text{--}\text{Pb}(\text{Zr,Ti})\text{O}_3$  (PMN–PZT) [2],  $\text{Pb}(\text{Ni}_{1/3}\text{Nb}_{2/3})\text{O}_3\text{--}\text{Pb}(\text{Zn}_{1/3}\text{Nb}_{2/3})\text{O}_3$  (PNN–PZN) [3], and  $\text{Pb}(\text{Mg}_{1/3}\text{Nb}_{2/3})\text{O}_3\text{--}\text{PbTiO}_3$  (PMN–PT) [4]. However, lead based piezoelectric materials pose serious problems of toxicity and volatilization. Therefore, there have been strong demands for the development of lead-free piezoelectric ceramics to reduce the environmental problems.

In the past few years, many papers have reported the use of bismuth–titanate ( $\text{Bi}_4\text{Ti}_3\text{O}_{12}$ ) [5] and potassium sodium niobate ( $\text{K}_{0.5}\text{Na}_{0.5})\text{NbO}_3$  (KNN) [6] families to replace lead-based piezoelectric ceramics. Among the candidates for lead-free piezoelectric ceramics, KNN ceramic has been recently considered as a promising material for lead-free piezoelectric systems due to its high piezoelectric coefficient of 416 pC/N and high Curie temperature of 253 °C [7]. Pure KNN ceramics are very difficult to densify by ordinary sintering process. It is generally believed that high sintering

temperature can improve the density and piezoelectric properties. However, in pure KNN ceramics fabricated by the conventional sintering methods, densification is known to be difficult because the phase stability is limited to 1140 °C according to the phase diagram for  $\text{KNbO}_3\text{--}\text{NaNbO}_3$  [8]. Therefore, in order to improve density and piezoelectric properties, various techniques of synthesis have been utilized, such as hot pressing and spark plasma sintering process, to obtain KNN ceramics with high densities [9,10]. Recently, normal sintering processes have been successfully applied to the fabrication of dense KNN-based ceramics [11,12]. Some dopants can be added to the KNN ceramics to improve the piezoelectric properties as well. Dopants, such as  $\text{CuO}$ ,  $\text{ZnO}$ ,  $\text{MnO}_2$ , and  $\text{LiTaO}_3$  can increase the density by lowering the sintering temperature of ceramics [13–15]. Guo et al. reported that  $(\text{K}_{0.5}\text{Na}_{0.5})\text{NbO}_3\text{--}\text{LiNbO}_3$  ceramics have a high piezoelectric constant ( $d_{33}=235\text{ pC}/\text{N}$ ) and electromechanical coefficient ( $k_p=44\%$ ) by employing the conventional ceramics process [16].

In this study, we propose a compatible niobate component to improve the piezoelectric properties of  $(\text{K,Na})\text{NbO}_3$  based piezoelectric systems.  $\text{Ag}(\text{Ta,Nb})\text{O}_3$  ceramics, which has ferroelectric–antiferroelectric phase transition near 182 K, has perovskite structure at room temperature with lattice parameter  $a=b=3.92\text{ }\text{\AA}$  and  $c=3.93\text{ }\text{\AA}$  [17]. Both  $(\text{K}_{0.5}\text{Na}_{0.5})$

\*Corresponding author. Tel.: +82 010 9556 3608; fax: +82 2 554 1643.  
E-mail address: [jhkoh@kw.ac.kr](mailto:jhkoh@kw.ac.kr) (J.-H. Koh).

$\text{NbO}_3\text{--LiNbO}_3$  and  $\text{Ag}(\text{Ta,Nb})\text{O}_3$  have perovskite structures with similar lattice parameters. Moreover, Ag dopant plays the role of a sintering aid for low temperature co-fired ceramics.  $\text{Ag}_2\text{O}$  was employed as an additional dopant to improve the piezoelectric properties. From our previous study, we found that 7 mol%  $\text{Ag}_2\text{O}$  doped  $0.94(\text{K}_{0.5}\text{Na}_{0.5})\text{NbO}_3\text{--}0.06\text{LiNbO}_3$  lead-free piezoelectric ceramics have a high piezoelectric coefficient of 274 pC/N [18]. Therefore, we focused on the preparation of 7 mol%  $\text{Ag}_2\text{O}$  doped  $0.94(\text{K}_{0.5}\text{Na}_{0.5})\text{NbO}_3\text{--}0.06\text{LiNbO}_3$  lead-free piezoelectric ceramics at different calcination temperatures, and systemically investigate their dielectric and piezoelectric properties.

## 2. Experiment

The 7 mol%  $\text{Ag}_2\text{O}$  doped  $0.94(\text{K}_{0.5}\text{Na}_{0.5})\text{NbO}_3\text{--}0.06\text{LiNbO}_3$  ceramics were prepared by the conventional sintering process. Analytical-grade metal oxides and carbonate powders,  $\text{NaCO}_3$ ,  $\text{K}_2\text{CO}_3$ ,  $\text{Li}_2\text{CO}_3$ ,  $\text{Nb}_2\text{O}_5$ , and  $\text{Ag}_2\text{O}$  were employed in this process. They were milled for 24 h with zirconia balls and ethyl alcohol and then dried for 24 h. They were calcined at 850 °C for 5 h. After the calcinations,  $\text{Ag}_2\text{O}$  powders were added to  $0.94(\text{K}_{0.5}\text{Na}_{0.5})\text{NbO}_3\text{--}0.06\text{LiNbO}_3$  powders. They were ball milled again, dried for 24 h, and calcined for 2 h. The dried mixture was sieved through a 100-mesh sieve. These powders were mixed with a binder of 3 wt% PVA, and pressed into disks of 12 mm diameter and 1 mm thickness under 98 MPa. These disks were sintered in air at 1080 °C for 2 h. The sintered samples were polished to 0.7 mm and then thermal etched at 1000 °C for 20 min. The bulk density was calculated from mass and dimensions of ceramic pellets. The crystalline structures were investigated

by X-ray diffraction analysis employing Cu K $\alpha$  radiation (Rigaku, D/MAX-2500 V/PC). The microstructure was observed using SEM (Hitachi, S-4300). The frequency

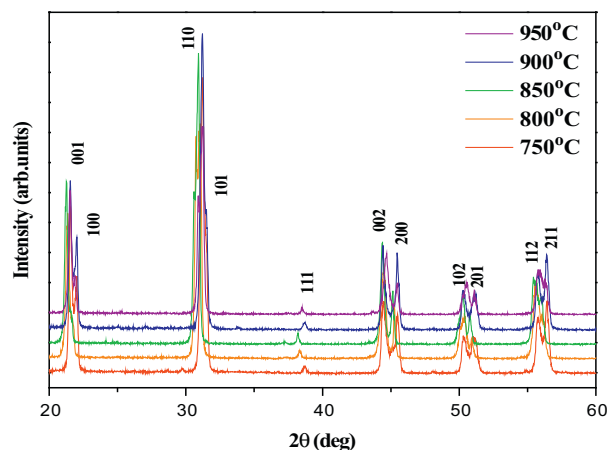


Fig. 1. XRD patterns of 7 mol%  $\text{Ag}_2\text{O}$  doped  $0.94(\text{K}_{0.5}\text{Na}_{0.5})\text{NbO}_3\text{--}0.06\text{LiNbO}_3$  ceramics calcined at different temperatures.

Table 1

Lattice parameters  $c$  and  $a$  of 7 mol%  $\text{Ag}_2\text{O}$  doped  $0.94(\text{K}_{0.5}\text{Na}_{0.5})\text{NbO}_3\text{--}0.06\text{LiNbO}_3$  ceramics calcined at different temperatures.

Calcination temperature (°C)	Lattice parameters (Å)	
	$c$	$a$
750	4.024	3.968
800	4.026	3.971
850	4.029	3.973
900	4.025	3.967
950	4.020	3.966

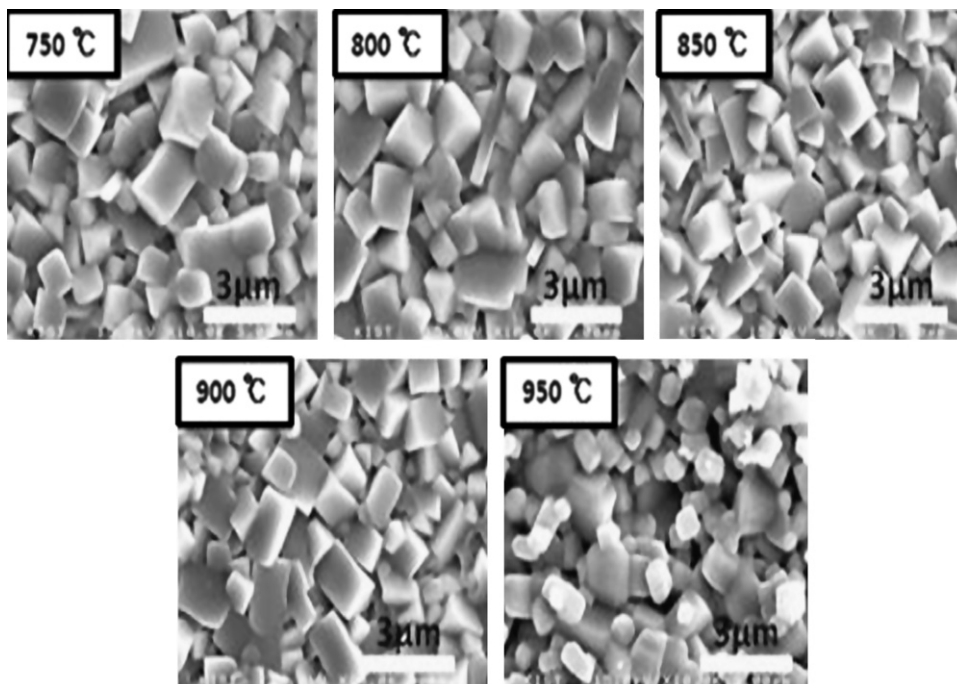


Fig. 2. SEM images of the 7 mol%  $\text{Ag}_2\text{O}$  doped  $0.94(\text{K}_{0.5}\text{Na}_{0.5})\text{NbO}_3\text{--}0.06\text{LiNbO}_3$  ceramics calcined at various temperatures.

dependence of the dielectric permittivity and loss tangent of the samples were measured employing a precision LCR Agilent 4284 A meter from 1 kHz to 1 MHz. The samples were polled under 3.5 kV/mm in a silicone oil bath at temperature of 120 °C for 30 min, and then the ceramics were cooled to room temperature under the electric field. The piezoelectric constant  $d_{33}$  was measured using  $d_{33}$  meter (YE 2730A). The temperature dependence of the dielectric constant was obtained using a programmable automatic LCR meter (FLUKE PM 6304).

### 3. Results and discussion

Fig. 1 shows the X-ray diffraction patterns of 7 mol%  $\text{Ag}_2\text{O}$  doped  $0.94(\text{K}_{0.5}\text{Na}_{0.5})\text{NbO}_3-0.06\text{LiNbO}_3$  ceramics calcined at different temperatures in the range of 750–950 °C for 2 h. When the calcination temperature is lower than 850 °C,  $\text{Ag}_2\text{O}$  doped  $0.94(\text{K}_{0.5}\text{Na}_{0.5})\text{NbO}_3-0.06\text{LiNbO}_3$  ceramics possess a single perovskite structure suggesting that  $\text{Ag}^+$  has diffused into the  $0.94(\text{K}_{0.5}\text{Na}_{0.5})\text{NbO}_3-0.06\text{LiNbO}_3$  lattices to form a homogeneous solid solution without any detectable secondary phase. However, increasing the calcination temperature upto 900 °C, it seems that the ceramics begin to lose their tetragonal structures. The lattice parameters of  $c$  and  $a$  were calculated from Bragg reflection position employing the Bragg law, and fitted by the Nelson–Riley extrapolation function [19] with the least mean square method.

$$\frac{C_{\cos\theta}-C_0}{C_0} = A\cos^2\theta\left(\frac{1}{\sin\theta} + \frac{1}{\theta}\right) \quad (1)$$

where,  $C_{\cos\theta}$  interplane distance calculated from the apparent Bragg peak position at  $2\theta$  and  $A$  is a fitting coefficient. The calculated lattice parameters  $c$  and  $a$  of ceramics are 4.029–4.020 and 3.973–3.966 Å, respectively. The variation of lattice parameters as a function of the calcination temperature is shown in Table 1.

Fig. 2 shows the SEM images of the specimens. The grain size became smaller gradually with the increase of temperature upto 800 °C. At 850 °C, the specimens have a uniform and dense structure compared with other specimens. However, increasing the calcination temperature further to 900 °C, they have irregular particle sizes with decreased grain sizes. Due to this decreased and irregular grain size, 7 mol%  $\text{Ag}_2\text{O}$  doped  $0.94(\text{K}_{0.5}\text{Na}_{0.5})\text{NbO}_3-0.06\text{LiNbO}_3$  ceramics have lower piezoelectric properties than those calcined at 850 °C.

The bulk density and relationship between piezoelectric constant  $d_{33}$ , electromechanical coefficient  $k_p$  of 7 mol%  $\text{Ag}_2\text{O}$  doped  $0.94(\text{K}_{0.5}\text{Na}_{0.5})\text{NbO}_3-0.06\text{LiNbO}_3$  ceramics are shown in Fig. 3. It was found that the bulk density,  $d_{33}$  and  $k_p$  increases with increasing calcination temperature, reaches a maximum value of 282 pC/N and  $k_p=37.8\%$  at 850 °C, then decreases with increasing calcination temperature. The promotion may be attributed to the increased density and enhancing the poling process. These results indicate that the ceramics calcined at 850 °C possess optimized piezoelectric property. From Figs. 1 and 3, we believe that higher calcination temperatures more than 850 °C makes the  $\text{Ag}_2\text{O}$

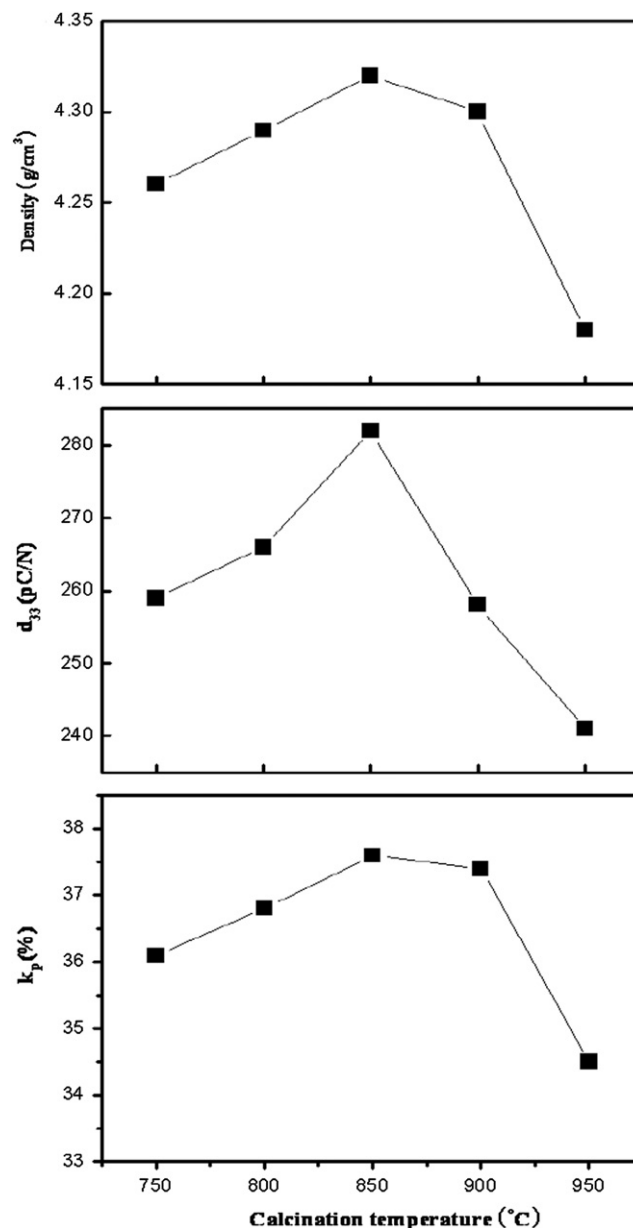


Fig. 3. Bulk density, piezoelectric constant  $d_{33}$  and electromechanical coefficient  $k_p$  of 7 mol%  $\text{Ag}_2\text{O}$  doped  $0.94(\text{K}_{0.5}\text{Na}_{0.5})\text{NbO}_3-0.06\text{LiNbO}_3$  ceramics.

doped  $0.94(\text{K}_{0.5}\text{Na}_{0.5})\text{NbO}_3-0.06\text{LiNbO}_3$  ceramics lose their uniform structure, and this worsens the piezoelectric properties.

Fig. 4 shows the frequency dependence of impedance magnitude and phase angle for  $\text{Ag}_2\text{O}$  doped  $0.94(\text{K}_{0.5}\text{Na}_{0.5})\text{NbO}_3-0.06\text{LiNbO}_3$  ceramic calcined at 850 °C. It has been known that under the ideal poling state, the phase angle will have a value of 90° in the frequency range between the anti-resonance and resonance frequencies. The observed phase angle for  $\text{Ag}_2\text{O}$  doped  $0.94(\text{K}_{0.5}\text{Na}_{0.5})\text{NbO}_3-0.06\text{LiNbO}_3$  ceramic is 75.5°, and exhibits a very squarelike resonance peak. This is in agreement with the observations for other hard piezoelectric ceramics [20].

Fig. 5 displays the polarization–electric field ( $P$ – $E$ ) hysteresis loops of 7 mol%  $\text{Ag}_2\text{O}$  doped  $0.94(\text{K}_{0.5}\text{Na}_{0.5})$

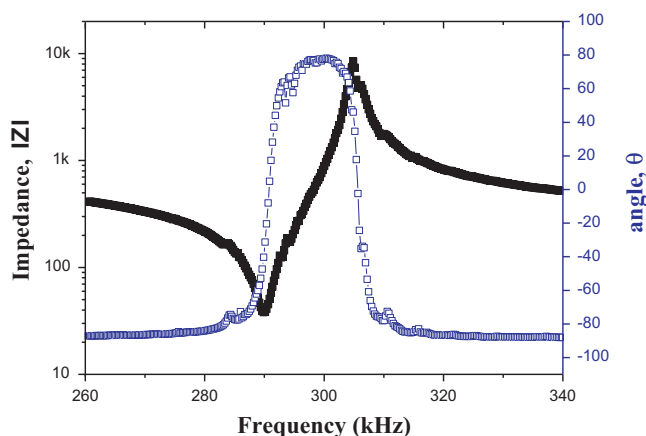


Fig. 4. Frequency dependence of impedance ( $\Omega$ ) and phase angle (deg.) for  $\text{Ag}_2\text{O}$  doped  $0.94(\text{K}_{0.5}\text{Na}_{0.5})\text{NbO}_3\text{--}0.06\text{LiNbO}_3$  ceramic calcined at  $850^\circ\text{C}$ .

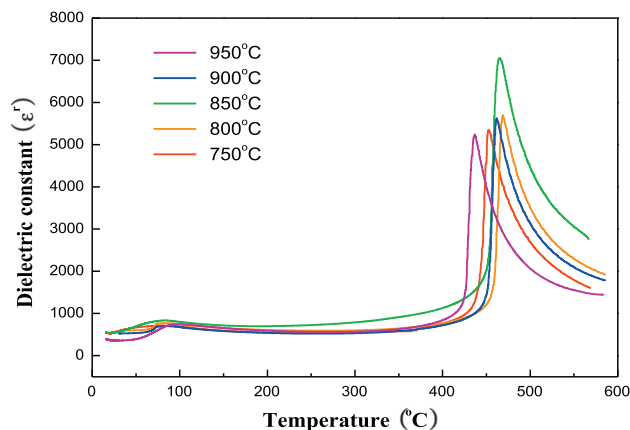


Fig. 6. Temperature-dependent relative dielectric permittivity of 7 mol%  $\text{Ag}_2\text{O}$  doped  $0.94(\text{K}_{0.5}\text{Na}_{0.5})\text{NbO}_3\text{--}0.06\text{LiNbO}_3$  ceramics calcined at different temperatures.

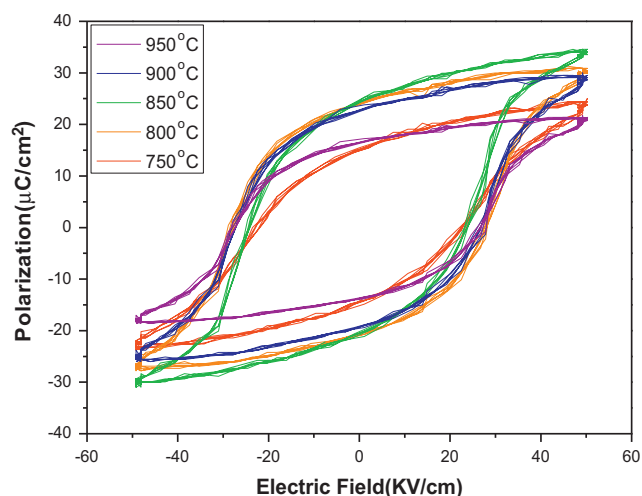


Fig. 5.  $P$ – $E$  hysteresis loops of 7 mol%  $\text{Ag}_2\text{O}$  doped  $0.94(\text{K}_{0.5}\text{Na}_{0.5})\text{NbO}_3\text{--}0.06\text{LiNbO}_3$  ceramics calcined at different temperatures.

$\text{NbO}_3\text{--}0.06\text{LiNbO}_3$  ceramics. Maximum electric field of  $50\text{ kV/cm}$  was applied to the specimens.  $P$ – $E$  hysteresis loops for the specimen, which was calcined at  $850^\circ\text{C}$ , has the highest remnant polarization of  $24.5\text{ }\mu\text{C/cm}^2$  compared with other specimens. As shown in the Fig. 5 with increasing calcination temperatures, remnant polarization increased upto  $24.5\text{ }\mu\text{C/cm}^2$  for the specimen, which calcined at  $850^\circ\text{C}$ . And then the remnant polarization began to decrease with further increasing calcination temperature. This decreased ferroelectric property was closely related with the leakage current and density of the calcined specimens.

Fig. 6 shows the dielectric constant  $\epsilon'$  as function of the temperature for 7 mol%  $\text{Ag}_2\text{O}$  doped  $0.94(\text{K}_{0.5}\text{Na}_{0.5})\text{NbO}_3\text{--}0.06\text{LiNbO}_3$  ceramics. The 7 mol%  $\text{Ag}_2\text{O}$  doped  $0.94(\text{K}_{0.5}\text{Na}_{0.5})\text{NbO}_3\text{--}0.06\text{LiNbO}_3$  ceramics showed tetragonal–cubic transition temperature ( $T_C$ ) of  $437\text{--}469^\circ\text{C}$ , which increased by decreasing the calcination temperature. As you can see in the figure, we found a weak phase transition near  $50\text{--}100^\circ\text{C}$ . We believe this phase transition temperature

is related with the orthorhombic–tetragonal phase transition. Due to this orthorhombic–tetragonal phase transition near the room temperature, we believe that the piezoelectric properties of 7 mol%  $\text{Ag}_2\text{O}$  doped  $0.94(\text{K}_{0.5}\text{Na}_{0.5})\text{NbO}_3\text{--}0.06\text{LiNbO}_3$  ceramics increased near room temperature.

#### 4. Conclusion

The 7 mol%  $\text{Ag}_2\text{O}$  doped  $0.94(\text{K}_{0.5}\text{Na}_{0.5})\text{NbO}_3\text{--}0.06\text{LiNbO}_3$  ceramics were prepared by a mixed-oxide method under various calcinations temperatures. Increasing calcination temperatures indicated that the ceramics have crystallized and possess tetragonal properties. The calcination temperature of  $850^\circ\text{C}$  was an optimized temperature for densification, piezoelectric, and dielectric properties. The density  $\rho$ , piezoelectric constant, electromechanical coefficient  $k_p$  and dielectric permittivity obtained were  $4.32\text{ g/cm}^3$ ,  $282\text{ pC/N}$ ,  $37.8\%$  and  $652$ , respectively. This means that 7 mol%  $\text{Ag}_2\text{O}$  doped  $0.94(\text{K}_{0.5}\text{Na}_{0.5})\text{NbO}_3\text{--}0.06\text{LiNbO}_3$  ceramics can be used as an alternative for lead-free piezoelectric applications.

#### Acknowledgments

This research was supported by the Basic Science Research Program through the National Research Foundation of Korea (NRF), which was funded by the Ministry of Education, Science and Technology (Grant no. 2010-0011536) and the National Research Foundation of Korea (NRF) grant funded by the Korea Government (MEST; No. 2011-0029625).

#### References

- [1] K. Shiratsuyu, K. Hayashi, A. Ando, Y. Sakabe, Piezoelectric characterization of low-temperature-fired  $\text{Pb}(\text{Zr,Ti})\text{O}_3\text{--}\text{Pb}(\text{Ni,Nb})\text{O}_3$  ceramics, Japanese Journal of Applied Physics 39 (2000) 5609–5612.

- [2] H. Ouchi, K. Nagano, S. Hayakawa, Piezoelectric properties of  $\text{Pb}(\text{Mg}_{1/3}\text{Nb}_{2/3})\text{O}_3\text{--PbTiO}_3\text{--PbZrO}_3$  solid solution ceramics, *Journal of the American Ceramic Society* 48 (1965) 630–635.
- [3] S.T. Chung, K. Nagata, H. Igarashi, Piezoelectric and dielectric properties of  $\text{Pb}(\text{Ni,Nb})\text{O}_3\text{--Pb}(\text{Zn,Nb})\text{O}_3\text{--PbTiO}_3$  system ceramics, *Ferroelectrics* 94 (1989) 243–247.
- [4] S.W. Choi, R.T.R. Shrout, S.J. Jang, A.S. Bhalla, Dielectric and pyroelectric properties in the  $\text{Pb}(\text{Mg}_{1/3}\text{Nb}_{2/3})\text{O}_3\text{--PbTiO}_3$  system, *Ferroelectrics* 100 (1989) 29–38.
- [5] M. Iwata, R. Aoyagi, M. Masaki, I. Suzuki, N. Yasuda, Y. Ishibashi, Polarization reversals with  $90^\circ$  domain walls in  $\text{Bi}_4\text{Ti}_3\text{O}_{12}$ , *Journal of the Korean Physical Society* 51 (2007) 740–745.
- [6] H. Birol, D. Damjanovic, N. Setter, Preparation and characterization of  $(\text{K}_{0.5}\text{Na}_{0.5})\text{NbO}_3$  ceramics, *Journal of the European Ceramic Society* 26 (2006) 861–866.
- [7] Y. Saito, H. Takkao, T. Tani, T. Nonoyama, K. Takatori, T. Homma, T. Nagaya, M. Nakamura, Lead-free piezoceramics, *Nature* 432 (2004) 84–87.
- [8] E. Ringgaard, T. Wurlitzer, *Journal of the European Ceramic Society* 25 (2005) 2701–2706.
- [9] R.E. Jaeger, L. Egerton, Hot pressing of potassium–sodium niobates, *Journal of the American Ceramic Society* 45 (1962) 209–213.
- [10] B. Zhang, J.F. Ki, K. Wang, H.L. Zhang, Compositional dependence of piezoelectric properties in  $\text{Na}_x\text{K}_{1-x}\text{NbO}_3$  lead-free ceramics prepared by spark plasma sintering, *Journal of the American Ceramic Society* 89 (2006) 1605–1609.
- [11] Y.P. Guo, K.I. Kakimoto, H. Ohsato,  $\text{Na}_{0.5}\text{K}_{0.5}\text{NbO}_3\text{--LiTiO}_3$  lead-free piezoelectric ceramics, *Materials Letters* 59 (2005) 241–244.
- [12] S.H. Moon, Y.S. Ham, Y.H. Lee, S.M. Nam, J.H. Koh, S.J. Jeong, M.S. Kim, The ferroelectric properties on  $\text{Li}_2\text{O}$  excess  $(\text{Na}_{0.51}\text{K}_{0.47}\text{Li}_{0.02})(\text{Nb}_{0.8}\text{Ta}_{0.2})\text{O}_3$  piezoelectric properties, *Journal of the Korean Physical Society* 56 (2010) 399–404.
- [13] M.R. Yang, C.S. Hong, C.C. Tsai, S.Y. Chu, Effect of sintering temperature on the piezoelectric and ferroelectric characteristics of  $\text{CuO}$  doped  $0.95(\text{Na}_{0.5}\text{K}_{0.5})\text{NbO}_3\text{--}0.05\text{LiTiO}_3$  ceramics, *Journal of Alloys and Compounds* 488 (2009) 169–173.
- [14] S.H. Park, Ch.W. Ahn, S. Nahm, J.S. Song, Microstructure and piezoelectric properties of  $\text{ZnO}$ -added  $(\text{Na}_{0.5}\text{K}_{0.5})\text{NbO}_3$  ceramics, *Japanese Journal of Applied Physics* 43 (2004) 1072–1074.
- [15] D. Lin, K.W. Kwok, H.L.W. Chan, Effects of  $\text{MnO}_2$  on the microstructure and electrical properties of  $0.94(\text{K}_{0.5}\text{Na}_{0.5})\text{--}0.06\text{Ba}(\text{Zr}_{0.5}\text{Ti}_{0.5})\text{O}_3$  lead-free ceramics, *Materials Chemistry and Physics* 109 (2008) 455–458.
- [16] Y. Guo, K. Kakimoto, H. Ohsato, Phase transitional behaviour and piezoelectric properties of  $(\text{Na}_{0.5}\text{K}_{0.5})\text{NbO}_3\text{--LiNbO}_3$  ceramics, *Applied Physics Letters* 85 (2004) 4121–4123.
- [17] J.H. Koh, S.I. Khartsev, A. Grishin, Ferroelectric silver niobate–tantarate thin films, *Applied Physics Letters* 77 (2000) 4416–4418.
- [18] M.S. Chae, J.H. Koh, Effects of  $\text{Ag}_2\text{O}$  dopants on the piezoelectric properties of  $0.94(\text{K}_{0.5}\text{Na}_{0.5})\text{NbO}_3\text{--}0.06\text{LiNbO}_3$  ceramics, *Journal of the Korean Physical Society* 60 (2012) 280–283.
- [19] J.H. Koh, S.J. Jeong, M.S. Ha, S.J. Song, Aging of piezoelectric properties in  $\text{Pb}(\text{MgNb})\text{O}_3\text{--Pb}(\text{ZrTi})\text{O}_3$  multilayer ceramic actuators, *Journal of Applied Physics* 96 (2004) 544–548.
- [20] J. Hao, Z. Xu, R. Chu, Y. Zhang, G. Li, Q. Yin, Effect of  $\text{MnO}_2$  on phase structure, microstructure and electrical properties of  $(\text{K}_{0.5}\text{Na}_{0.5})_{0.94}\text{Li}_{0.06}\text{NbO}_3$  lead-free ceramics, *Materials Chemistry and Physics* 118 (2009) 229–233.