

Short communication

A nanocomposite of needle-like MnO_2 nanowires arrays sandwiched between graphene nanosheets for supercapacitors

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Abstract

A nanocomposite of needle-like MnO_2 nanowires arrays-graphene (NMNAG) is synthesized by in-situ growth of MnO_2 nanowires on the surface of graphene nanosheets (GNS), resulting in a sandwich nanoarchitecture hybrid. The preparation is a simple redox reaction between potassium permanganate (KMnO_4) and GNS, which is facile to produce in large scale at low cost. The nanocomposite exhibits high-capacitance performance (276 F g^{-1} at 0.5 A g^{-1}), excellent rate capability (62.7% capacity retention at 60 A g^{-1}), stable cycleability (no degradation after 1200 cycles) and a maximum power density of 21.6 kW kg^{-1} .

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1. Introduction

Supercapacitors (SCs), have been studied extensively in the past few decades as energy storage devices due to their higher power density than batteries, and higher energy density than conventional dielectric capacitors [1]. Currently, graphene nanosheets (GNS), as two-dimensional layers of sp^2 -bonded carbon [2], have been regarded as an ideal candidate material for supercapacitor owing to exceptional thermal stability, high surface area, and excellent conductivity. Graphene-based supercapacitors have been reported with a specific capacitance of 135 F g^{-1} in aqueous KOH [3] and 117 F g^{-1} in aqueous H_2SO_4 electrolyte [4].

However, the application of GNS is hindered because of the limited capacitance (usually $\sim 100\text{--}200 \text{ F g}^{-1}$ [5]). Previous investigations demonstrated that GNS incorporated with nanostructured pseudocapacitive materials (such as Fe_3O_4 [6], Co_3O_4 [7], MnO_2 [8,9], NiO [10]) showed a high energy density. Among them, MnO_2 is one of the most promising pseudocapacitive materials on account of low-cost and environment-friendly [11,12]. Wang's group [13] deposited MnO_2 nanoparticles on the graphene oxide sheets through a simple soft chemical route. It was found that the electrochemical performance was enhanced

by the chemical interaction between graphene oxide and MnO_2 . In addition, Yan et al. [14] synthesized a composite of graphene/ MnO_2 by microwave irradiation and the specific capacitance was up to 310 F g^{-1} . It is noted that electrochemical properties of the composite strongly depended on the morphology and crystallinity of the MnO_2 [15]. Recently, a new composite of carbon nanotubes (CNTs)/graphene, in which CNTs pillars were sandwiched between the graphene layers, has been prepared for SCs [16]. It showed a maximum specific capacitance of 385 F g^{-1} .

Herein, we design a sandwich nanoarchitecture of MnO_2 /GNS hybrid with hierarchical structure by a simple but effective strategy, which was assembled by two-dimension (2D) GNS and one-dimension (1D) free-standing needle-like MnO_2 nanowires arrays. By this means, GNS cannot only improve the electron transport for the capacitance reactions but also act as a flexible substrate, while the MnO_2 nanowires arrays can serve as high-capacity hosts and facilitate the penetration of electrolyte. This synthesis method will put forward a new route to produce MnO_2 /GNS composite with high performance in large scale at low cost.

2. Experimental

All the chemicals were of analytical grade. Graphite oxide (GO) was synthesized from natural graphite ($30 \mu\text{m}$) by a modified Hummers method [17]. GNS was prepared from the

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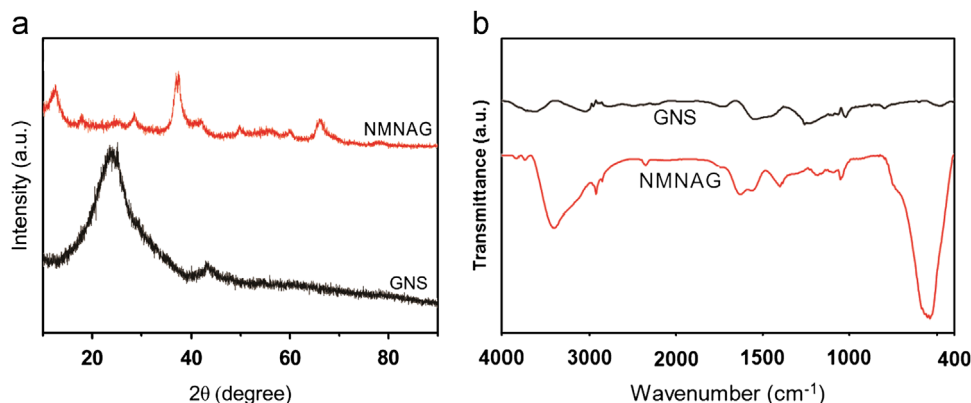
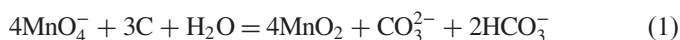


Fig. 1. (a) XRD patterns of GNS and NMNAG and (b) FTIR spectra of GNS and NMNAG.

exfoliated GO by microwave irradiation as described elsewhere [18].

The nanocomposite were synthesized through a simple but effective method. 50 mg of GNS was dispersed in deionized water (100 mL) under ultrasonication for 30 min. Then, potassium permanganate (1 g) was added and the mixture was stirred at room temperature (24 °C) for 6 h. Afterwards, 2 ml concentrated H₂SO₄ was added to the mixture drop by drop. The mixture was heated in a water bath at 80 °C for 1 h. Finally, the product was collected and washed repeatedly with distilled water, then dried at 80 °C for 24 h. The redox reaction can be described in the following:



The crystallographic structure of the materials were analyzed by X-ray diffraction system (XRD, DMAX-2500PC) equipped with Cu K α radiation ($\lambda=0.15406$ nm). Fourier transform infrared spectroscopy (FTIR) spectra were recorded on a GX spectrometer. The morphology was observed by transmission electron microscopy (TEM, LIBRA 200FE) and field-emission scanning electron microscopy (FESEM, ZEISS nova 400), respectively. Brunauer–Emmett–Teller (BET) specific surface area and pore size distribution were measured by nitrogen adsorption/de-adsorption using an automatic adsorption system (ASAP 2020M). Thermogravimetric analysis (TGA) data were collected on a thermal analysis instrument (NETZSCH STA 449C) with a heating rate of 10 °C min⁻¹ in an air flow of 50 mL min⁻¹.

The working electrode was prepared by loading the mixture of 70 wt% active materials, 20 wt% carbon black and 10 wt% polyvinylidene difluoride on a nickel foam (1 cm × 1 cm). The mass of MnO₂/GNS on the working electrode was 5.6 mg. All electrochemical measurements were done in a conventional three-electrode configuration. The three-electrode configuration consisted of Pt as the counter electrode, Ag/AgCl as the reference electrode and the above working electrode. The measurements were conducted in a 1 M Na₂SO₄ aqueous electrolyte (pH=9) at room temperature. Cyclic voltammograms (CV) were measured by Solartron (1287+1260 8 w). The galvanostatic charge/discharge and CV were performed within the voltage range from -0.1 to 0.9 V. For the cyclic voltammetric measurements, the sweep rate ranged from 10,

25, 50, 100, and 250 mV s⁻¹ and galvanostatic charge/discharge curves were measured at different current densities of 0.5, 1, 2, 5, 10, 20 and 60 A g⁻¹.

3. Results and discussion

Fig. 1(a) presents the XRD patterns of GNS and NMNAG, respectively. Typical peaks of GNS at $2\theta=25.9^\circ$ and 44.2° are observed, corresponding to the diffraction of (0 0 2) and (1 0 0), respectively. As for NMNAG, all diffraction peaks can be indexed to the α -MnO₂ phase (JCPDS no. 44-0141) and no peaks are observed corresponding to other phase of manganese oxides. The FT-IR spectra of GNS and NMNAG are presented in Fig. 1(b). The FTIR spectrum of GNS indicated that the microwave-assisted reduction of GO was relatively completed with few oxygen-containing groups [19]. In the curve of NMNAG, the broad band at 3421 cm⁻¹ can be attributed to the O–H stretching vibration of the surface water while the bands centered at 1630, 1399, and 1048 cm⁻¹ are associated with the O–H bending vibrations combined with Mn atoms, indicating the presence of bonded water. Furthermore, the presence of the bands at 540 cm⁻¹, which are assigned to the Mn–O vibrations in [MnO₆] octahedral [20], confirming the formation of α -MnO₂ phase.

The morphologies of GNS and NMNAG are shown in Fig. 2. In Fig. 2(a), GNS exhibited a curled and entangled structure. The SEM image (Fig. 2(b)) clearly shows that the free-standing MnO₂ nanowires were grown on the surface of GNS. Notably, the GNS could be viewed as “the substrate” while MnO₂ nanowires could be regarded as the vertical “needles” on the surface. The TEM image (Fig. 2(c)) shows that the needle-like MnO₂ nanowires were oriented on the surface of the GNS, resulting in a sandwich nanoarchitecture. The average diameter of MnO₂ nanowires is ~ 20 nm and the length is ~ 200 nm. Fig. 2(d) shows a high-resolution TEM image of the MnO₂ nanowires. The lattice fringes give an interplanar spacing of 0.48 nm, corresponding to the (200) lattice plane of α -MnO₂ crystal, which is in good agreement with the XRD patterns.

The electrochemical performance of the nanocomposite is shown in Fig. 3. Fig. 3(a) presents CV curves of the NMNAG

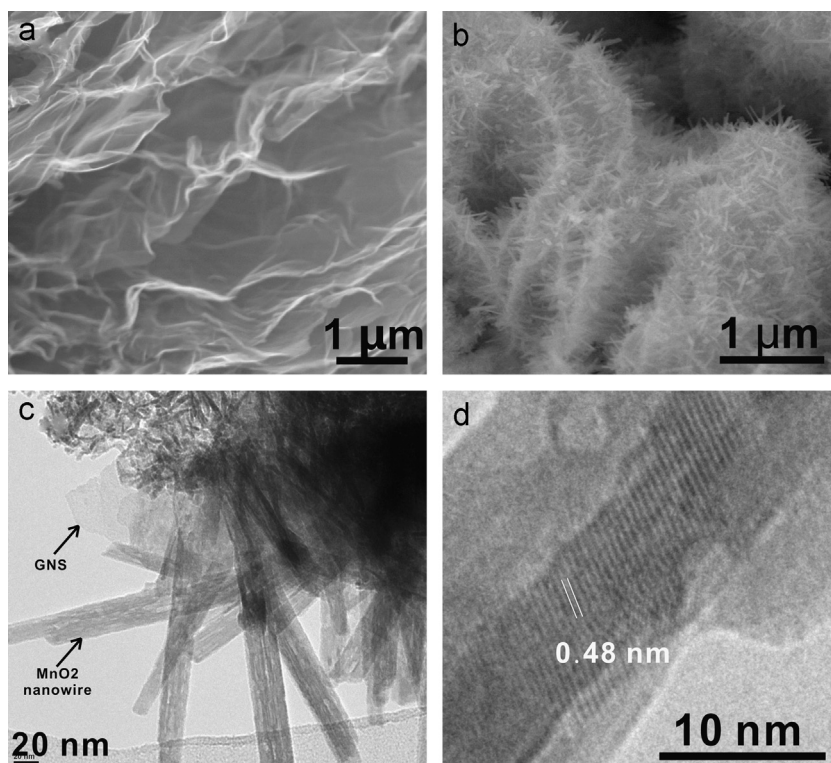


Fig. 2. SEM images of (a) GNS and (b) NMNAG, and (c) TEM images of NMNAG, (d) HRTEM image of the MnO_2 nanowire.

electrode at the scan rates of $10\text{--}250\text{ mV s}^{-1}$. The rectangular and symmetric CV curves reveal the ideal pseudocapacitive nature over the fast charge–discharge process. Moreover, with the increase of scan rate, all the CV curves do not obviously change and stay nearly rectangular, indicating the stable rate capability. This should be attributed to rapid transport of the electrolyte to the active surface of the hybrid. In comparison, the CV curves of the pristine GNS at 10 mV s^{-1} are also shown in Fig. 3(b). It is obvious that the GNS exhibits a lower specific capacitance as the CV integrated area is much smaller.

Fig. 3(c) illustrates the charge–discharge curves of the NMNAG at different current density. The specific capacitance is calculated according to the discharge curves. At a low current density of 0.5 A g^{-1} , the specific capacitance is up to 276 F g^{-1} , which is remarkably higher than the pristine GNS (81 F g^{-1}). The results are in good accordance with the CV evaluation. Fig. 3(d) shows the specific capacitance of GNS and NMNAG at different current densities. It is noted that the NMNAG electrode exhibits not only a higher specific capacitance, but also a better stability at high current density compared to the bare GNS. As presented in Fig. 3(d), a capacitance of 173 F g^{-1} ($\sim 62.7\%$ retention) was retained at the high current density of 60 A g^{-1} . To the best of our knowledge, this capability is superior to or comparable with the previous literatures relative to graphene/ MnO_2 composites [8,9,12–14]. The mass loading of MnO_2 in the NMNAG was 98.5% by weight according to the TGA curve, indicating that

the MnO_2 almost contributes to the whole capacitance of the NMNAG.

Fig. 4(a) displays the capacitance versus cycle number for NMNAG at 5 A g^{-1} . A slight increase of the initial capacitance can be clearly observed after 1200 cycles due to the activation effect, demonstrating the excellent electrochemical stability. This result has also been confirmed by comparing the charge–discharge curves (inset of Fig. 4a) of 1st cycle and 1200th cycle. Fig. 4(b) shows the Ragone plot of the prepared electrodes. The energy density and power density (based on active materials on both the electrodes) are derived from the charge–discharge curves at different current densities. The energy density reaches 38.3 Wh kg^{-1} at 250 W kg^{-1} power density and still remains 24 Wh kg^{-1} at 21.6 kW kg^{-1} power density. This performance is significantly better than those of recently reported MnO_2 –graphene [21,22], NiO–graphene [23], and RuO_2 –graphene [24] composites.

The high rate capability and excellent cycling stability of the needle-like MnO_2 nanowires arrays/graphene nanocomposite would provide a big forward for high-performance supercapacitors. In the 3D nanoarchitecture, the graphene conductive network improves the electron conductivity of MnO_2 , facilitating the rapid charge transport. The needle-like nanowires arrays can enlarge the interface area between MnO_2 and electrolyte, thus improving the pseudocapacitance occurring at the interface. And the ordered space between the free-standing MnO_2 nanowires can act as a buffer to alleviate the volume expansion of the MnO_2 during the charge/discharge process.

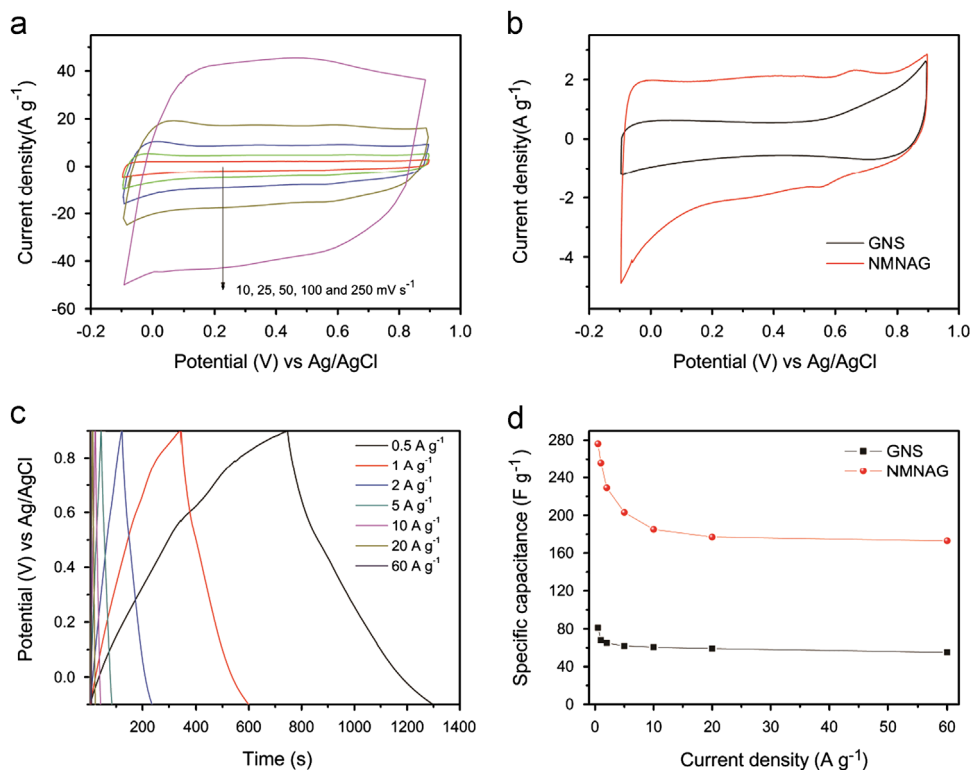


Fig. 3. (a) Cyclic voltammograms obtained from NMNAG electrode at different sweep rates. (b) CV curves of GNS and NMNAG at scan rate of 10 mV s⁻¹. (c) charge–discharge curves of NMNAG at different density. (d) The specific capacitance as a function of different current density of the NMNAG and the pristine GNS. (For interpretation of the references to color in this figure legend, the reader is referred to the web version of this article.)

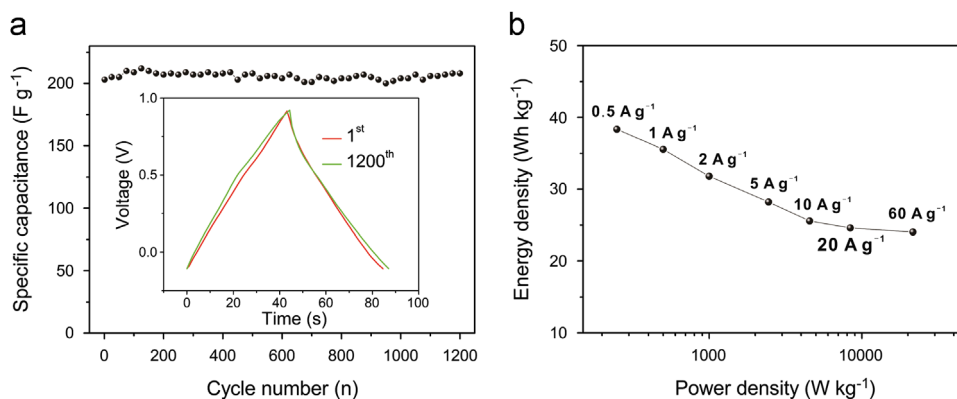


Fig. 4. (a) Specific capacitance as a function of cycle number at 5 A g⁻¹ of the NMNAG electrodes. Insets show 1st cycle and 1200th cycle of the NMNAG electrodes. (b) Ragone plot (power density vs. energy density) of the NMNAG electrodes. The energy densities and power densities were derived from the charge–discharge curves at different current densities.

4. Conclusions

In summary, we have prepared the nanoarchitecture of MnO₂/GNS composite with hierarchical structure by a simple but effective method, in which free-standing needle-like MnO₂ nanowires arrays were sandwiched between graphene nanosheets. The integration of GNS and the free-standing needle-like MnO₂ nanowires endows the hybrid with good capacitive performance. The intriguing nanoarchitecture combining the free-standing MnO₂ nanowires arrays with the flexible, large

surface area and highly conductive GNS will offer great promise in high-power energy storage device applications.

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