

Improved electrical and optical properties of GZO films with a thin TiO₂ buffer layer deposited by RF magnetron sputtering

Daeil Kim*

School of Materials Science and Engineering, University of Ulsan, Ulsan, Republic of Korea

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Abstract

Ga-doped ZnO (GZO) thin films were prepared by radio frequency magnetron sputtering without intentional substrate heating on bare glass and TiO₂-deposited glass substrates to investigate the effect of a thin TiO₂ buffer layer on the optical and electrical properties of the films. The thicknesses of the TiO₂ buffer layer and GZO films were kept constant at 5 and 100 nm, respectively.

As-deposited GZO/TiO₂ bi-layered films show a higher transmittance of 83.0% than that of the GZO films, and GZO/TiO₂ films show a lower resistivity ($1.03 \times 10^{-2} \Omega \text{ cm}$) than that of the GZO single layer films. In addition, the work function of the GZO film was affected by the TiO₂ buffer layer, where the GZO/TiO₂ films had a higher work-function (4.86 eV) than that of the GZO single layer films. The experimental results indicate that a 5-nm-thick TiO₂ buffer layer in the GZO/TiO₂ films results in better electrical and optical performance than conventional GZO single layer films.

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1. Introduction

Recently, there has been considerable interest in the use of Ga-doped ZnO (GZO) films as a transparent and conducting oxide (TCO) for transparent electrodes in solar cells and display devices [1,2] due to the fact that they are less expensive than conventional Sn-doped In₂O₃ (ITO) films. However, conventional GZO films have some drawbacks such as a relatively higher resistivity than the ITO films and weakness to moisture, which may deteriorate the electrical and optical performances of devices. Thus, in order to overcome these problems, transparent diffusion barrier films have been researched to simultaneously optimize the optical and electrical properties of the GZO films. Recently, Nomoto reported the effects of a ZnO buffer layer on the characteristics of transparent conducting GZO films prepared by DC magnetron sputtering [3].

In this study, GZO thin films were deposited by radio frequency (RF) magnetron sputtering on glass substrates with and without

a TiO₂ buffer layer, and then the effects of a TiO₂ buffer layer on the optical and electrical properties of the GZO films were investigated using X-ray diffraction (XRD), atomic force microscopy (AFM), scanning electron microscope (SEM), Hall effect measurements, and UV-visible spectrometry. In addition, the influence of the TiO₂ buffer layer on the work function of GZO films was evaluated using UV photoelectron spectroscopy (UPS) to evaluate GZO/TiO₂ films as a transparent anode electrode for organic light emitting diode (OLED) applications.

2. Experimental

Both GZO and TiO₂ films were deposited on glass (coming 1737) substrates without intentional substrate heating using an RF (13.56 MHz) magnetron sputtering system equipped with two cathodes. The sintered ZnO (97%)–Ga (3%) and pure TiO₂ targets were both 3 in. in diameter and 0.25 in. thick. For all depositions, the distance between the target and substrate was constant at 6 cm, and the substrate rotation speed was set to 8 rpm. The GZO/TiO₂ bi-layered films were obtained by continuously depositing each film layer without exposure of the films to the atmosphere.

*Tel.: +82 52 259 2243.

E-mail address: dkim84@ulsan.ac.kr

Substrate temperature was monitored using a K-type thermocouple in contact with the substrate, and the substrate temperature increased to 70 °C during deposition. Table 1 shows the main parameters used for deposition.

After deposition, high resolution XRD (X'pert Pro MRD, Philips) at the Korea Basic Science Institute (KBSI, Daegu center) was used to observe the thin film crystallinity. The grain size of the films was evaluated from the XRD pattern using the Scherrer formula [4].

Optical transmittance in the visible wavelength region was observed with a UV–vis spectrophotometer (Cary 100 Cone, Varian), and the glass substrates showed 92% optical transmittance in the visible wavelength range. The surface roughness investigation was performed by means of an AFM (XE-100, Park system) on $2 \times 2 \mu\text{m}^2$ sample areas under ambient conditions. The thickness of the films was measured using a surface profilometer (Dektak 3D, Veeco), and the electrical properties, such as carrier concentration and mobility, were observed with Hall effect measurements employing the van der Pauw geometry (HMS-3000, Ecopia). The performance of GZO and GZO/TiO₂ films as a TCO electrode was compared using a figure of merit. In addition, to consider the influence of a TiO₂ buffer layer on the work function of GZO films, work functions of the films were evaluated using UPS analysis.

3. Results and discussion

Fig. 1 shows the XRD patterns of the GZO single layer and GZO/TiO₂ bi-layer films prepared without substrate heating. Although neither film showed a diffraction peak of ZnO (0 0 2), GZO/TiO₂ bi-layer films show a larger grain size of 10.21 nm than that (8.9 nm) of the GZO single layer films. In a previous study, Kim reported similar results, namely, that in ITO/Au bi-layered films, the crystallization of the upper ITO film is promoted by an Au buffer layer without intentional substrate heating [5].

Surface morphology of TCO films is an important factor in determining the optical and electrical properties [6]. Fig. 2 shows AFM images of GZO films prepared on bare glass substrates and TiO₂ deposited on glass substrates. The root mean square (RMS) roughness of the GZO film (2.2 nm) is higher than that of the GZO/TiO₂ film (1.1 nm). From the AFM images, one can conclude that TiO₂ buffer layers may enhance the flatness of the GZO/TiO₂ films. During buffer layer deposition, TiO₂ film grows preferably in a sunken region on the substrate. Thus, the GZO/TiO₂ film has the more

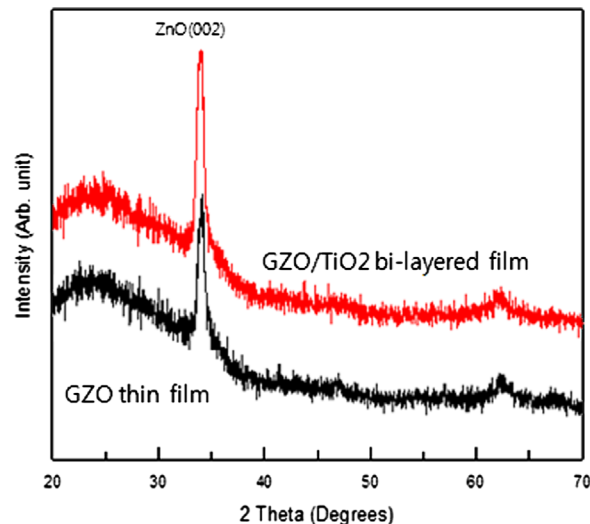


Fig. 1. The XRD patterns of the GZO and GZO/TiO₂ bi-layer films.

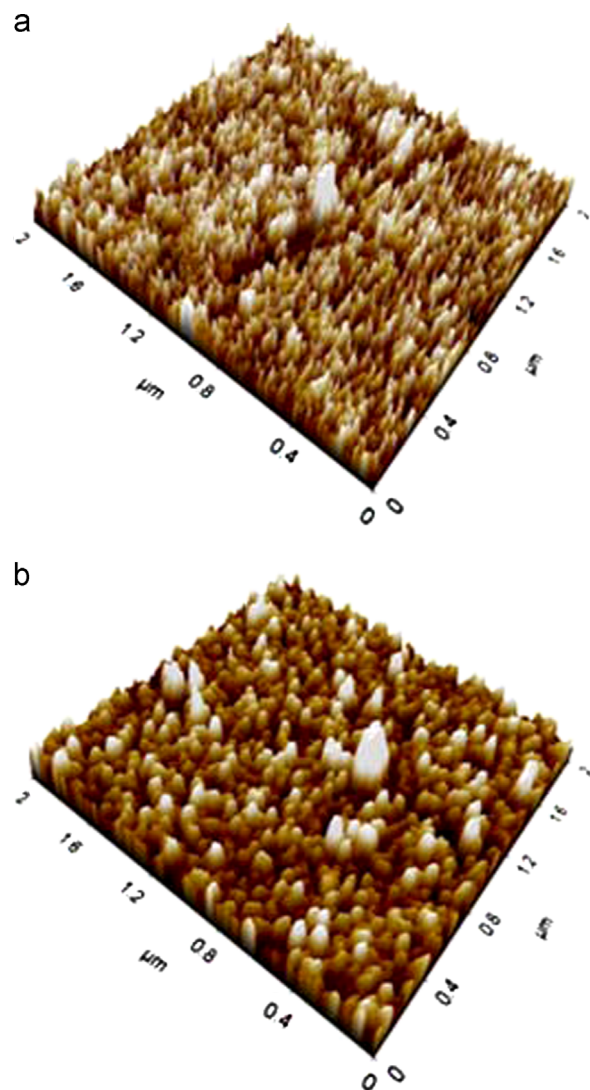


Fig. 2. The RMS roughness of the GZO and GZO/TiO₂ Bi-layer films. (a)GZO, 2.83 nm, (b) GZO/TiO₂, 1.53 nm.

Table 1
Deposition conditions of GZO and TiO₂ thin films.

	GZO	TiO ₂
Base pressure (Pa)	1.5×10^{-4}	1.3×10^{-4}
Deposition pressure (Pa)	1.8×10^{-2}	1.3×10^{-1}
Power density (W/cm ²)	RF, 3.5	RF, 4.0
Deposition rate (nm/min)	10	1
Ar/O ₂ gas flow rate	5/0.03	20

flat surface than that of the GZO single layer films. Recently, J. Park reported that a Ni interlayer in ITO/Ni/ITO multilayer films also promotes the flatness of the upper ITO films [7].

Table 2 shows the influence of the TiO₂ buffer layer on the electrical properties of the films. The GZO/TiO₂ films have a lower resistivity of $1.03 \times 10^{-2} \Omega \text{ cm}$ than that of the GZO

Table 2

The comparison of electrical properties of the films.

Films	Concentration ($10^{20}/\text{cm}^3$)	Mobility (cm/S)	Resistivity ($\Omega \text{ cm}$)
GZO	4.4	7.2	1.9×10^{-2}
GZO/TiO ₂	6.9	9.1	1.0×10^{-2}

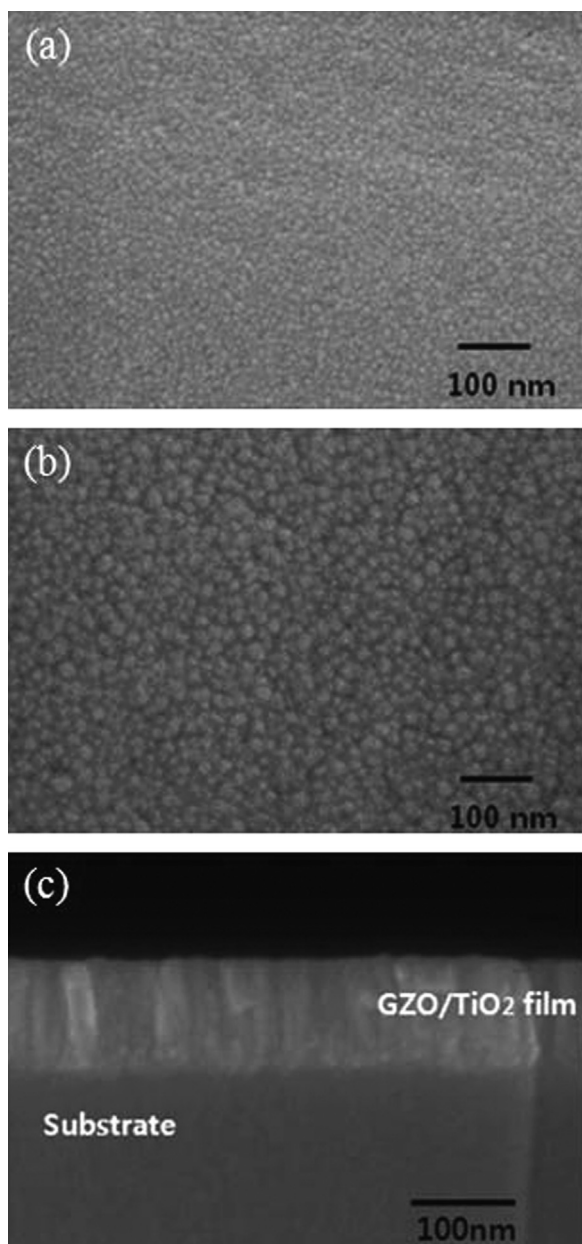


Fig. 3. SEM image of the GZO and GZO/TiO₂ bi-layer films. (a) GZO, (b) GZO/TiO₂ film, (c) cross-sectional image of GZO/TiO₂ film.

single layer film due to increases in both carrier concentration and mobility.

Fig. 3 shows the surface and cross-section images of as deposited GZO and GZO/TiO₂ films. The GZO/TiO₂ films show the larger grain size than that of GZO single layer films. Since increased crystallization of GZO/TiO₂ films results in a lower density of the grain boundary which acts as a trap for charge carrier, it can be concluded that the electrical resistivity of GZO/TiO₂ films is decreased with carrier concentration as shown Table 2.

Fig. 4 shows the optical transmittance for GZO and GZO/TiO₂ films. For the GZO film, the average transmittance in the visible range is about 82.5%, and the transmittance of ITO/TiO₂ films is about 83.0%. Table 3 provides a comparison of optical and electrical properties of the films. The GZO/TiO₂ films had a lower sheet resistance than that of the GZO single layer films.

The figure of merit (FOM) is an important index for evaluating the performance of transparent conducting oxide (TCO) films [8]. FOM is defined as $\text{FOM} = T^{10}/R_s$, where T is the optical transmittance and R_s is the sheet resistance. The FOM reached a maximum of $1.4 \times 10^{-4} \Omega^{-1}$ for the GZO/TiO₂ films, which is higher than the $7.5 \times 10^{-5} \Omega^{-1}$ for the GZO single layer films prepared in this study. Since a higher FOM value indicates better quality TCO films, it is supposed that the GZO film with a 5-nm-thick TiO₂ buffer layer will likely perform better in TCO applications than GZO single layer films.

The high work function of TCO films, which is close to the value of the highest occupied molecular orbital (HOMO) of the organic layer, allows hole injection from TCO to the organic layer of OLED, which results in a decrease in the turn-on voltage of the OLED. However, the work function of conventional ITO films is lower than the HOMO of the organic layer of OLEDs. Thus, several techniques have been developed to increase the work function of ITO [9,10].

Fig. 5 shows the kinetic energy cut-off spectra obtained from the GZO/TiO₂ films. This allowed the determination of the work function values directly from the spectra by fitting straight lines into their kinetic energy cut-off and determining the intersection with the baseline of the spectra. Table 4 shows the compared work functions of conventional ITO, GZO and

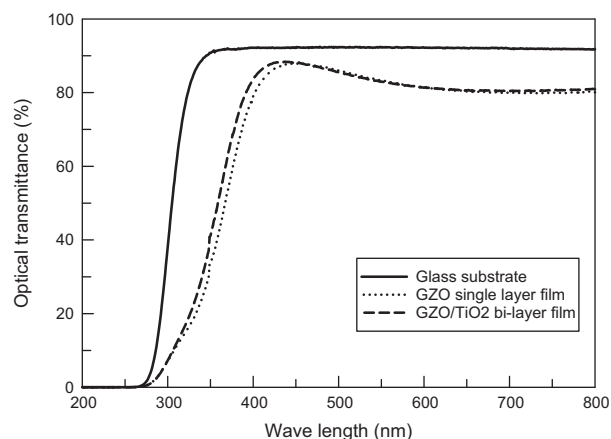


Fig. 4. The optical transmittance of the GZO and GZO/TiO₂ bi-layer films.

Table 3
The comparison of figure of merit (FOM, Ω^{-1}).

Films	Sheet resistance (Ω/\square)	Transmittance (%)	FOM (Ω^{-1})
GZO	1930	82.5	7.5×10^{-5}
GZO/TiO ₂	1037	83.0	1.4×10^{-4}

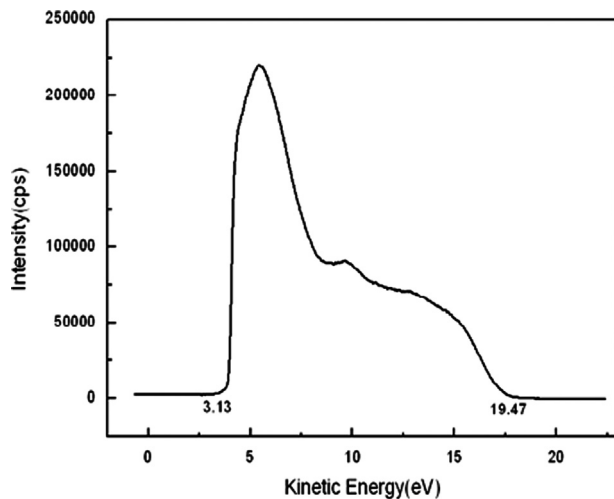


Fig. 5. The kinetic energy cut-off spectra obtained from the GZO/TiO₂ bi-layer films.

Table 4
Comparison of the work function of the ITO, GZO and GZO/TiO₂ films.

TCO films	Work function (eV)	Reference
ITO	4.43	[10]
GZO	4.47	This study
GZO/TiO ₂	4.86	This study

GZO/TiO₂ films. The GZO/TiO₂ films show a higher work function of 4.86 eV. Thus, adding a TiO₂ buffer layer is one useful method to increase the work function of GZO films.

4. Conclusions

Both GZO single layer and GZO/TiO₂ bi-layered films were prepared by RF magnetron sputtering on glass substrates. The structural, optical and electrical properties of the GZO films were dependent on the TiO₂ buffer layer.

From AFM observations, it is apparent that TiO₂ buffer films enhance the flatness of the GZO/TiO₂ films. The figure of merit for the GZO/TiO₂ bi-layered films reached a maximum value of $1.47 \times 10^{-4} \Omega^{-1}$, which was greater than that of the GZO single layer films.

Also, GZO/TiO₂ bi-layered films show a higher work function than that of the GZO single layer films. These results indicate that the 5-nm-thick TiO₂ buffer layer in the GZO/TiO₂ films results in better performance than GZO single layer films.

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