

Enhanced piezoelectric coefficient of HfO_2 -modified $(\text{K}_{0.44}\text{Na}_{0.56})_{0.94}\text{Li}_{0.06}(\text{Nb}_{0.94}\text{Sb}_{0.06})\text{O}_3$ lead-free ceramics

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Abstract

$(\text{K}_{0.44}\text{Na}_{0.56})_{0.94}\text{Li}_{0.06}(\text{Nb}_{0.94}\text{Sb}_{0.06})\text{O}_3-x\text{HfO}_2$ (KNLNS- $x\text{HfO}_2$) lead-free piezoceramics were prepared using the conventional solid-state method, and effects of HfO_2 content on their microstructure and electrical properties were investigated. Their phase structure is independent of the HfO_2 content ($x=0-0.008$). As the HfO_2 content increases, the grain size, the Curie temperature, and the dielectric constant decrease. The ceramic with $x=0.004$ has enhanced electrical properties, i.e., $d_{33}\sim 266$ pC/N, $k_p\sim 42.1\%$, $\epsilon_r\sim 950$, $\tan\delta\sim 0.027$, and $T_C\sim 343$ °C, showing that adding HfO_2 enhances the piezoelectric properties of KNLNS ceramics.

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Keywords: C. Piezoelectric properties; Lead-free ceramics; (K,Na)NbO₃; HfO_2

1. Introduction

$\text{Pb}(\text{Zr,Ti})\text{O}_3$ (PZT)-based piezoceramics have been widely used in some electronic devices because of their excellent piezoelectric properties and high Curie temperature (T_C) [1]. However, the PZT ceramics give rise to environmental pollution during their preparation and processing due to the volatilization of lead element which is a toxic substance [1,2]. Much attention has been paid on the ABO_3 -type lead-free piezoelectric materials in order to protect the environment [3–20], such as BaTiO_3 , $\text{Bi}_{0.5}\text{Na}_{0.5}\text{TiO}_3$, and $(\text{K}_{0.5}\text{Na}_{0.5})\text{NbO}_3$ (KNN). Among those, KNN-based lead-free piezoceramics have received considerable attention as a promising candidate for the PZT ceramics because of their excellent piezoelectric properties and high Curie temperature [2,9–22].

The pure KNN ceramic has a low piezoelectric constant (d_{33}) of $\sim 80-100$ pC/N and a high T_C of ~ 420 °C [2]. In addition, it is difficult to obtain the dense KNN ceramics because of the volatilization of the potassium element during their sintering

process [2,23]. In the past several years, some preparation techniques have been used to produce the dense KNN ceramics, including hot pressing and spark plasma sintering [2,23]. The new preparation techniques which can improve the density and electrical properties of the KNN ceramics, have not been used widely yet in view of expensive elements and complicated preparation routes [2,23]. Recently, research results have shown that the sintering as well as electrical properties except d_{33} (< 250 pC/N) of KNN ceramics can be improved by ABO_3 -type addition, for example, BaTiO_3 , SrTiO_3 , CaTiO_3 , LiSbO_3 , LiTaO_3 , LiNbO_3 , and SrTiO_3 , but their d_{33} is still poor (< 250 pC/N) [2,10,11,13,14,16]. It was reported that the addition of HfO_2 can enhance the piezoelectric properties of one piezoelectric material [22–25]. For example, Zhou et al. reported that $(\text{Ba,Ca})(\text{Ti,Hf})\text{O}_3$ ceramics have a giant d_{33} of ~ 550 pC/N [24]. Moreover, a large d_{33} of $\sim 446-680$ pC/N is also observed in $\text{Pb}(\text{Hf}_{0.445}\text{Ti}_{0.555})\text{O}_3$ – $\text{Pb}(\text{Mg}_{1/3}\text{Nb}_{2/3})\text{O}_3$ and $\text{Bi}(\text{Ni}_{1/2}\text{Hf}_{1/2})\text{O}_3$ – PbTiO_3 ceramics [25,26]. However, there were few reports with regard to the effect of HfO_2 content on the electrical properties of the KNN-based ceramics.

In this work, $(\text{K}_{0.44}\text{Na}_{0.56})_{0.94}\text{Li}_{0.06}(\text{Nb}_{0.94}\text{Sb}_{0.06})\text{O}_3-x\text{HfO}_2$ (KNLNS- $x\text{HfO}_2$) lead-free piezoceramics were prepared using the conventional solid-state method, and effects of HfO_2

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content on their microstructure and electrical properties were studied. Enhanced d_{33} is achieved in KNLNS- x HfO₂ by tuning the HfO₂ content, and the related physical mechanism was also addressed.

2. Experimental procedure

(K_{0.44}Na_{0.56})_{0.94}Li_{0.06}(Nb_{0.94}Sb_{0.06})O₃- x HfO₂ ($x=0, 0.002, 0.004, 0.006$, and 0.008) lead-free piezoelectric ceramics were prepared using the conventional mixed oxide sintering method. In this work, raw materials were K₂CO₃ (99%), Na₂CO₃ (99.8%), Li₂CO₃ (99%), Nb₂O₅ (99.5%), Sb₂O₃ (99%), and HfO₂ (99.999%). These powders were weighed and ball milled by the zirconia ball using ethanol for 24 h, and then dried and calcined at 800 °C for 5 h. The calcined powders were mixed with 8 wt% polyvinyl alcohol (PVA) solution, and then pressed into pellets with a diameter of 10 mm and a thickness of 1.2 mm under single direction pressure of about 15 MPa. The green pellets were sintered at 1080–1095 °C for 2 h. Silver electrodes were pasted on both surfaces of the sintered samples by firing at 600 °C for 10 min, and then these samples were polarized at 30 °C in the silicon oil using an electric field of 3–4 kV/mm for 30 min.

X-ray diffraction (XRD, D8-2-Advanced, BRUKER AXS, German) was used to determine the phase structure of the sintered ceramics. Dielectric properties vs. temperatures were characterized using an LCR meter (HP4284A, Hewlett-Packard, USA) equipped with a programmable temperature box. Surface morphologies of the sintered samples were measured using the scanning electron microscopy (SEM, JSM-5610LV, JEOL, Japan). The density was measured using the Archimedes method with distilled water. The piezoelectric constant (d_{33}) was measured using a quasistatic piezoelectric d_{33} meter (ZJ-3A, Institute of Acoustics, Chinese Academy of Sciences, China) after 24 h aging, and the planar coupling coefficient (k_p) and the mechanical quality factor (Q_m) were determined using the resonance–anti-resonance technique and an impedance analyzer (Agilent 4294A). P - E hysteresis loops of the ceramics

were measured at 10 Hz using a radiant precision workstation (USA).

3. Results and discussion

The XRD patterns of KNLNS- x HfO₂ ceramics are measured at room temperature, as shown in Fig. 1(a). All the ceramics have a pure perovskite structure and there are no secondary phases, indicating the formation of stable solid solutions between KNLNS and HfO₂. That is, there is only single-phase perovskite structure without secondary phases. To characterize the evolution of the phase structure, the expanded XRD patterns are indicated in Fig. 1(b). A tetragonal phase was observed in all the ceramics regardless of the HfO₂ content [17], as confirmed by the temperature-dependent dielectric constant [Fig. 3]. It was found that the diffraction peaks shift to a lower angle as the HfO₂ content increases because of partly substitution of the Hf⁴⁺ for the Nb⁵⁺, the ionic radius of which is nearly 0.064 nm and smaller than that of Hf⁴⁺ (~0.079 nm) [27].

Surface micrographs of KNLNS- x HfO₂ ceramics with $x=0, 0.04$, and 0.08 were measured by the SEM, as shown in Fig. 2 (a)–(c). The grain size gradually decreases as the HfO₂ doping level increases. As a result, the addition of HfO₂ can adjust the microstructure and refine the grain size of KNLNS ceramics. The decrease of the grain size should be mainly assigned to different ionic radii of Nb⁵⁺ and Hf⁴⁺. It was reported that the ion with a larger ionic radius and atomic weight can easily reduce the mobility of the ions, which leads to the decrease of the grain size [28,29].

To study the effect of the HfO₂ content on their phase transitions as a function of the temperature, the curves of the temperature-dependent dielectric properties are shown in Fig. 3 (a)–(c). One phase transition was observed in these ceramics, which corresponds to the tetragonal–cubic phase transition (T_C) [13]. T_C slowly decreases as the HfO₂ content gradually increases, showing that HfO₂ slightly affects T_C of KNLNS ceramics. A similar phenomenon was demonstrated in

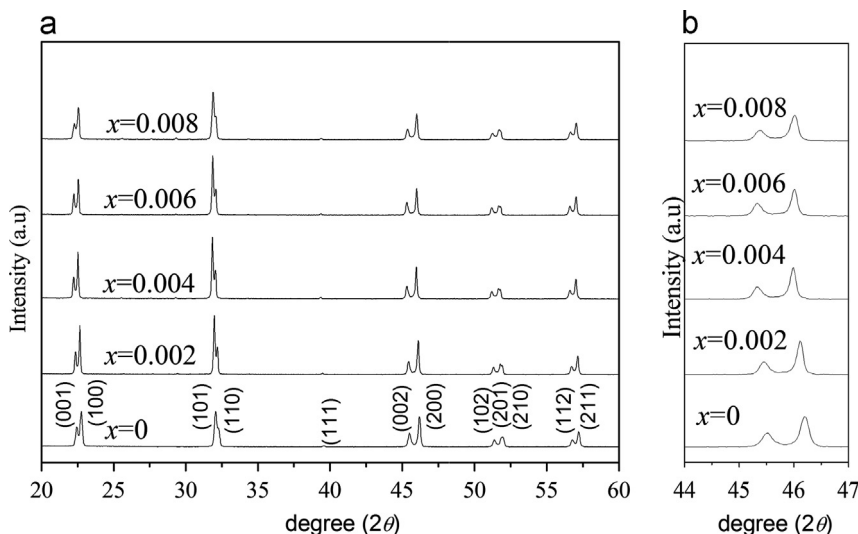


Fig. 1. XRD patterns of KNLNS- x HfO₂ ceramics: (a) 20–60° and (b) 44–47°.

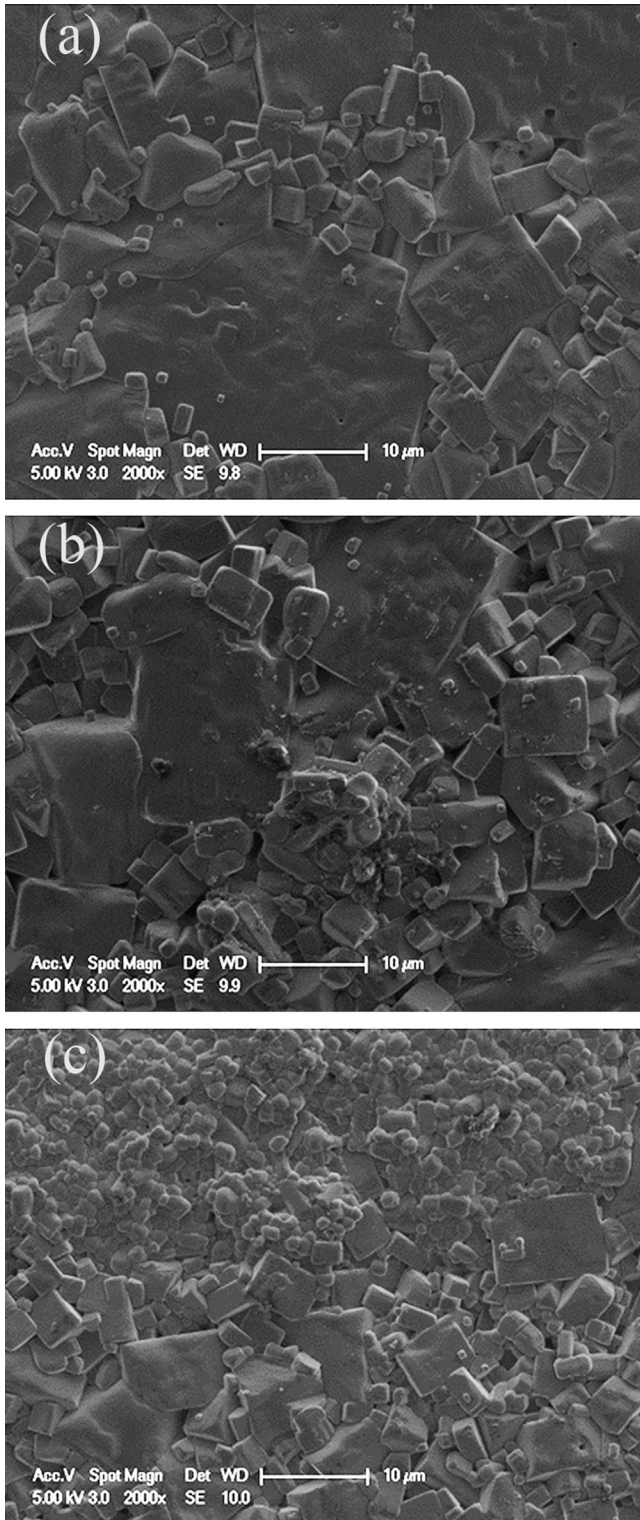


Fig. 2. SEM surface micrographs of KNLNS- x HfO₂ ceramics with (a) $x=0$, (b) $x=0.004$, and (c) $x=0.008$.

HfO₂-modified piezoelectric materials [28]. In addition, the ceramic with $x=0.004$ has a lower dielectric loss ($\tan\delta$) because of a denser microstructure.

Fig. 4 shows the dielectric behavior of KNLNS- x HfO₂ ceramics as a function of HfO₂ content, measured at room temperature. It was found that $\tan\delta$ and ϵ_r gradually decrease

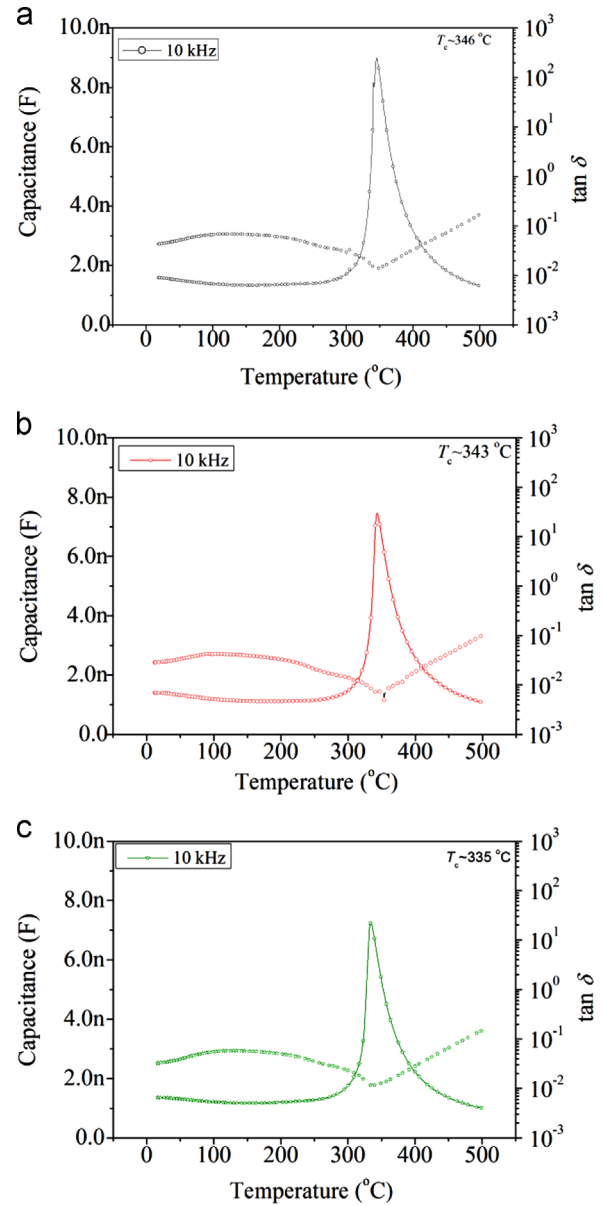
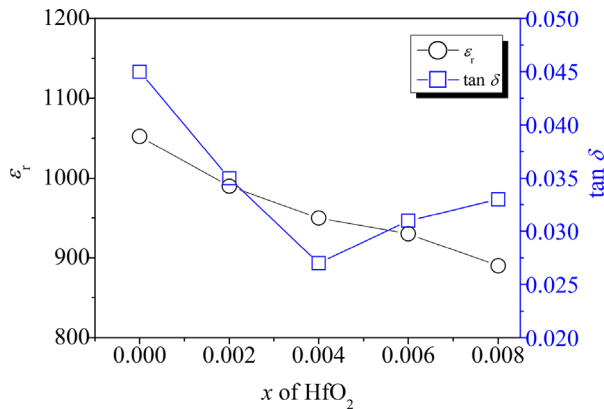
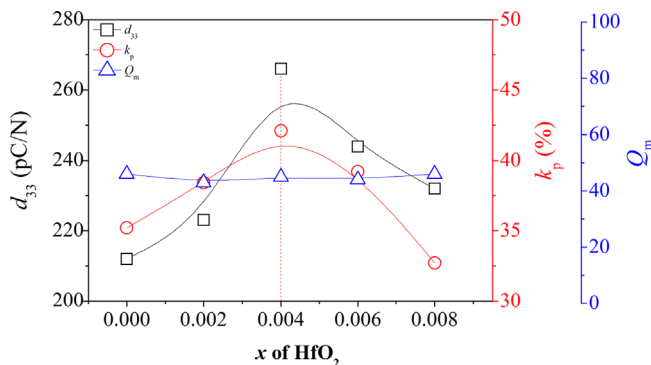
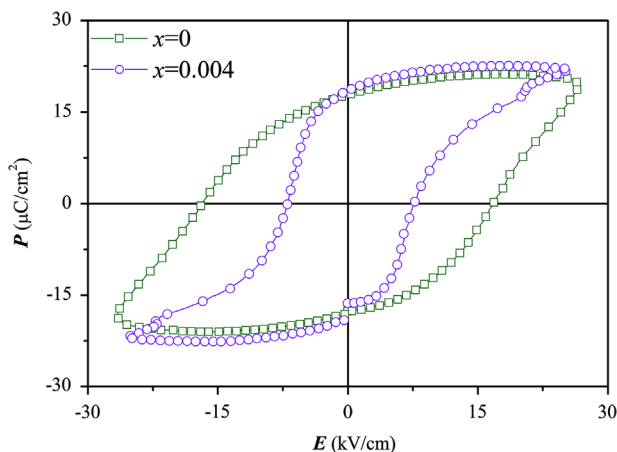


Fig. 3. Temperature-dependent dielectric properties in KNLNS- x HfO₂ ceramics with (a) $x=0$, (b) $x=0.004$, and (c) $x=0.008$.

with an increase in the HfO₂ content[29], which reveals that ϵ_r and $\tan\delta$ are sensitive to the HfO₂ content. It was reported that the grain size strongly affects the ϵ_r of functional materials [28]. Cai et al. reported that smaller grain size results in a lower ϵ_r because the ϵ_r of the grain is larger than that of the grain boundary [27]. As a result, the addition of HfO₂ leads to the enhanced dielectric behavior of KNLNS- x HfO₂ ceramics.

Fig. 5 plots the piezoelectric properties of KNLNS- x HfO₂ ceramics, measured at room temperature. d_{33} increases as the HfO₂ content increases, reaches a maximum value of ~ 266 pC/N at $x=0.004$, and decreases as the HfO₂ content further increases. Similar to the change of d_{33} , k_p also exhibits a similar trend, reaching a maximum value (i.e., $\sim 42.1\%$). However, Q_m remains almost unchanged with the variation of the HfO₂ content ($x=0-0.008$). As a result, the enhanced piezoelectric behavior of

Fig. 4. Dielectric properties of KNLNS- x HfO₂ ceramics.Fig. 5. Piezoelectric properties of KNLNS- x HfO₂ ceramics.Fig. 6. Ferroelectric properties of KNLNS- x HfO₂ ceramics.

this work could be attributed to the addition of optimum HfO₂ content, together with a denser microstructure.

Fig. 6 shows the polarization–electric field (P – E) hysteresis loops of the ceramics with $x=0$ and 0.004 , measured at 10 Hz and room temperature. All the ceramics show a typical saturated P – E loop, while an obvious difference is also observed. It was found that a “fat” P – E loop is shown in the ceramic with $x=0$, while the ceramic with $x=0.004$ has a slimmer and more saturated P – E loop. In addition, the ceramic with $x=0.004$ has a much smaller E_C value than that of the pure one, showing that the domain of the ceramic with $x=0.004$ is more easily rotated

under a lower electric field. A similar result was observed in Hf-modified BaTiO₃ ceramics [27].

4. Conclusions

The conventional solid-state method was used to prepare the KNLNS- x HfO₂ lead-free piezoceramics, and their microstructure and piezoelectric properties are very sensitive to the HfO₂ content. All the ceramics have a tetragonal phase, which is independent of the small content of HfO₂ addition investigated in this study. The addition of HfO₂ results in a reduction in the grain size, while their T_C value remains almost unchanged. The ceramic with $x=0.004$ has an enhanced piezoelectric behavior (i.e., $d_{33} \sim 266$ pC/N and $k_p \sim 42.1\%$). As a result, the piezoelectric properties of KNN-based ceramics could be improved by adding HfO₂.

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