

Influence of precursor $\text{Bi}^{3+}/\text{Fe}^{3+}$ ion concentration on hydrothermal synthesis of BiFeO_3 crystallites

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Abstract

In this study we investigated the effect of precursor $\text{Bi}^{3+}/\text{Fe}^{3+}$ ion concentration on the hydrothermal synthesis of BiFeO_3 crystallites. It is demonstrated that the phase-purity and morphology of the products is highly dependent on the metal ion concentration. Phase-pure BiFeO_3 crystals can be prepared at the $\text{Bi}^{3+}/\text{Fe}^{3+}$ ion concentration ranging from 0.025 to 0.0625 M. The samples prepared at $n(\text{Bi}^{3+}/\text{Fe}^{3+})=0.025$, 0.0375, 0.05, and 0.0625 M, are composed, respectively, of cuboid-like particles (100–200 nm), regular spherical agglomerates (30–40 μm) made up of irregular grains with size about several hundred nanometers, irregular flower-like clusters formed from irregular grains of several hundred nanometers in size, and octahedron-shaped particles (500–600 nm). These samples have a similar bandgap energy of 2.20 eV and exhibit a typical antiferromagnetic behavior at room temperature.

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1. Introduction

Bismuth ferrite (BiFeO_3) with a rhombohedrally distorted perovskite structure is one of the best-known multiferroic materials. Due to the coexistence of ferroelectric and anti-ferromagnetic orders up to quite high temperatures as well as magnetoelectric coupling between them [1–3], BiFeO_3 is regarded to be a highly promising multiferroic system for potential applications. Furthermore, BiFeO_3 is an interesting semiconductor with bandgap energy of about 2.0 eV, which exhibits a pronounced photocatalytic activity toward the degradation of various organic dyes under visible-light irradiation [4–8]. The overall properties of a material depend on numerous factors like its structure, defect, dimension, size, and morphology. For example, nanosized BiFeO_3 tends to exhibit enhanced photocatalytic activity and also present weak ferromagnetism that is absent in its bulk form [7–13]. To tailor and/

or enhance its properties, various techniques have been widely used to prepare BiFeO_3 nano/micro crystals [4–18]. Among them, the hydrothermal route offers an advantage in controlling the product morphology. The main operating parameters that have been extensively investigated include mineralizer concentration, organic additive, reaction temperature, and reaction time [5,6,11–15]. However, there has been little attention paid to the effect of $\text{Bi}^{3+}/\text{Fe}^{3+}$ ion concentration in precursor solution on the hydrothermal synthesis of BiFeO_3 crystallites. Here we undertake an investigation aimed at studying its effect on the phase-purity and morphology of prepared samples via a hydrothermal route.

2. Experimental

All raw materials and reagents used are of analytical grade without further purification. Equimolar amounts of $\text{Bi}(\text{NO}_3)_3 \cdot 5\text{H}_2\text{O}$ and $\text{Fe}(\text{NO}_3)_3 \cdot 9\text{H}_2\text{O}$ were dissolved in 20 mL of dilute nitric acid solution. To the mixture solution was then added 60 mL KOH solution with a concentration of 4 mol L^{-1} (M) drop by drop under magnetic stirring, and immediately a dark brown

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suspension solution was formed. The suspension was ultrasonically treated for 8 min and subsequently stirred vigorously with a magnetic stirrer for 30 min to create a highly uniform mixture. The resulted mixture was sealed in a Teflon-lined stainless steel autoclave of 100 mL capacity and submitted to hydrothermal treatment at 200 °C. After 6 h of reaction, the autoclave was cooled naturally to room temperature. The resulting brown precipitate was collected and washed several times with distilled water and absolute ethanol, and then dried in a thermostat drying oven at 80 °C for 12 h to obtain final BiFeO₃ product. By varying the concentration of Bi(NO₃)₃ · 5H₂O and Fe(NO₃)₃ · 9H₂O from 0.0125 to 0.075 M (final concentration in autoclave), we prepared a series of BiFeO₃ samples.

The phase purity of the products was checked by means of X-ray powder diffraction (XRD) with Cu K α radiation. The particle morphology was investigated by a field-emission scanning electron microscope (SEM). The ultraviolet (UV)–visible diffuse reflectance spectra were measured using a UV–visible spectrophotometer with an integrating sphere attachment. A vibrating sample magnetometer (VSM) was used to measure the magnetic hysteresis loops at room temperature.

3. Results and discussion

Fig. 1 shows the XRD patterns of BiFeO₃ samples prepared at different Bi³⁺/Fe³⁺ ion concentrations in precursor solution. Phase-pure BiFeO₃ samples with a rhombohedral *R3m* structure (PDF card no. 74-2016) are seen to be formed at the metal ion concentrations ranging from 0.025 to 0.0625 M. However, when the metal ion concentration is raised up to 0.075 M or reduced down to 0.0125 M, the prepared samples present a second Bi₂Fe₄O₉ phase. The XRD results suggest that appropriate Bi³⁺/Fe³⁺ ion concentrations in precursor solution are necessary in producing phase-pure BiFeO₃ crystals via the present hydrothermal synthesis route.

Fig. 2 shows the SEM images of the as-prepared pure BiFeO₃ samples, revealing that the morphology of BiFeO₃ crystals is highly dependent on the precursor Bi³⁺/Fe³⁺ ion concentration. The sample prepared at $n(\text{Bi}^{3+}/\text{Fe}^{3+})=0.025$ M is mainly composed of cuboid-like particles in the size range of 100–200 nm

(Fig. 2(a)). When $n(\text{Bi}^{3+}/\text{Fe}^{3+})=0.0375$ M, the resulted sample presents regular spherical agglomerates of 30–40 μm in diameter (Fig. 2(b)). The higher-magnification image inserted in Fig. 2(b) reveals that the spherical agglomerates are made up of irregular grains with a size of several hundred nanometers. From Fig. 2(c), one can see that the sample prepared at $n(\text{Bi}^{3+}/\text{Fe}^{3+})=0.05$ M consists of irregular flower-like clusters formed from irregular grains of several hundred nanometers in size. The SEM image shown in Fig. 2(d) reveals the synthesis of octahedron-shaped particles with size ranging from 500 to 600 nm when $n(\text{Bi}^{3+}/\text{Fe}^{3+})=0.0625$ M.

The crystallization process of BiFeO₃ can be simply described as follows. First, the starting materials, Bi(NO₃)₃ · 5H₂O and Fe(NO₃)₃ · 9H₂O, were dissolved in dilute nitric acid to form a uniform Bi³⁺/Fe³⁺ ion solution. Then, KOH was added to the solution, which reacted with Bi³⁺ and Fe³⁺ ions to produce amorphous Bi(OH)₃ and Fe(OH)₃ precipitates (Eqs. (1) and (2)). Finally, Bi(OH)₃ and Fe(OH)₃ reacted under the hydrothermal condition to yield BiFeO₃ particles (Eq. (3)). The hydrothermal formation of BiFeO₃ particles can be explained by the “dissolution–crystallization” mechanism [19]. According to this mechanism, the reactants underwent an attack of the mineralizer, KOH, to dissolve and form ion groups. Then BiFeO₃ particles were formed by nucleation, precipitation, dehydration and growth, in which the crystal growth only occurred in the region of supersaturated fluid. When varying the Bi³⁺/Fe³⁺ ion concentration in precursor solution, the nucleation and growth process of BiFeO₃ particles could be changed, consequently resulting in morphologically different products. However, when the metal ion concentration was too high or too low, the requirement of supersaturation and stoichiometry for the crystallization of BiFeO₃ particles could be locally destroyed. Another possible reaction as indicated in Eq. (4) could take place, leading to the formation of the second Bi₂Fe₄O₉ phase in final product.

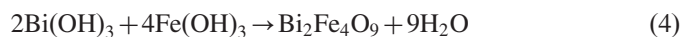
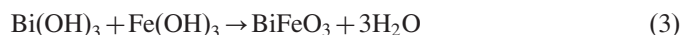


Fig. 3(a) shows the UV–visible diffuse reflectance spectra of BiFeO₃ samples. In order to highlight the optical absorption features, the corresponding differential spectra are derived and shown in Fig. 3(b), where the peak wavelengths are characterized to be the absorption edges of the samples. It is seen that these samples have similar absorption peaks. According to the theoretical results [20], the absorption edge at 564 nm is attributed to the electron transition from valence band to conduction band, from which the bandgap energy (E_g) of the BiFeO₃ crystallites is obtained to be 2.20 eV.

Fig. 4 show the magnetic hysteresis loops of BiFeO₃ samples measured at room temperature. The magnetization for all samples is seen to change nearly linearly with applied magnetic field, implying a typical antiferromagnetic behavior.

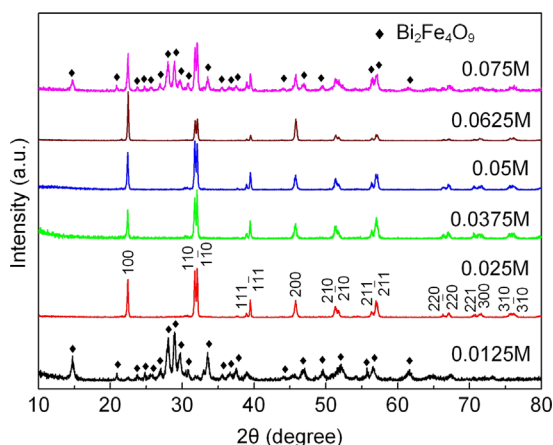


Fig. 1. XRD patterns of BiFeO₃ samples prepared at different Bi³⁺/Fe³⁺ ion concentrations in precursor solution ranging from 0.0125 to 0.075 M.

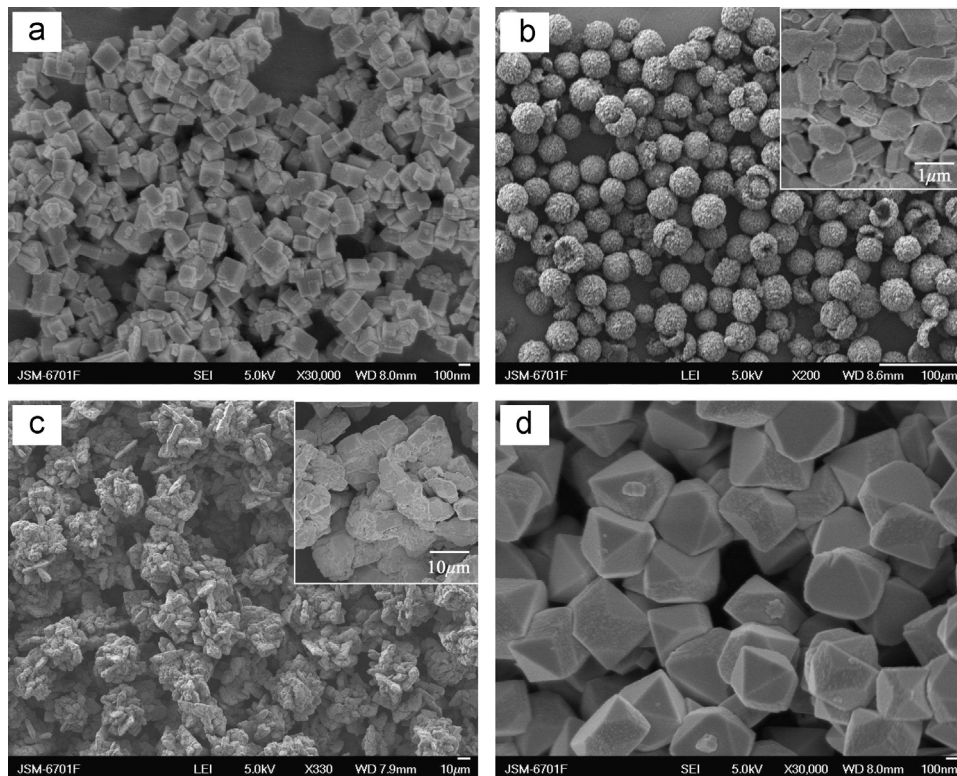


Fig. 2. SEM images of BiFeO₃ samples prepared at different Bi³⁺/Fe³⁺ ion concentrations in precursor solution. (a) 0.025 M, (b) 0.0375 M, (c) 0.05 M, and (d) 0.0625 M.

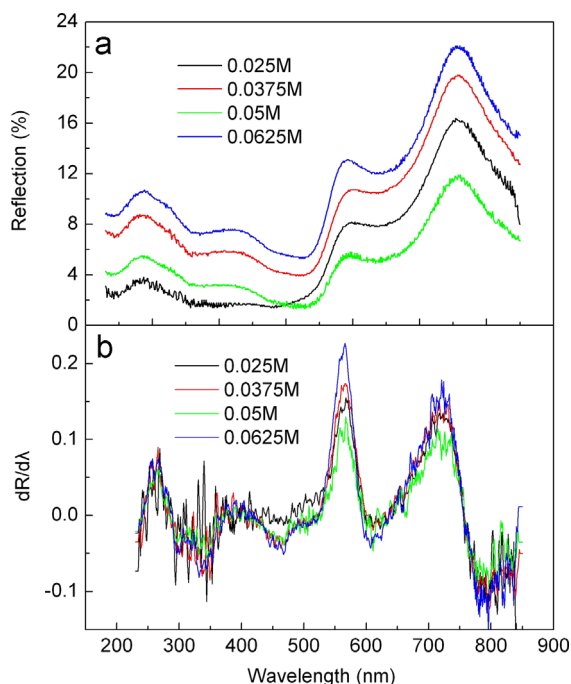


Fig. 3. (a) UV–visible diffuse reflectance spectra of BiFeO₃ samples and (b) the corresponding first derivative of the diffuse reflectance spectra.

It should be noted that weak ferromagnetism is generally observed for nanosized BiFeO₃ and exhibits an increasing trend with reducing the grain size [10]. This is because that the long-range helical antiferromagnetic order of BiFeO₃ (with a

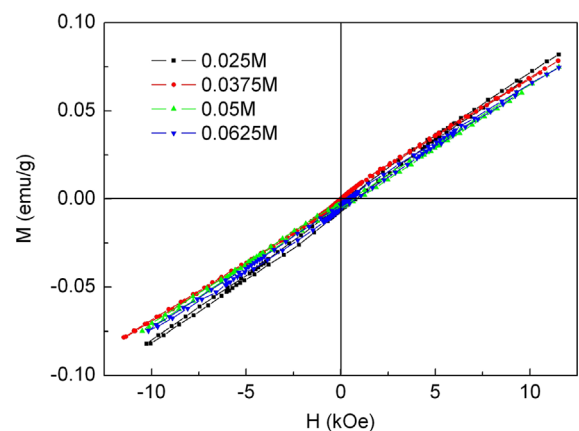


Fig. 4. Magnetic hysteresis loops of BiFeO₃ samples measured at room temperature.

period of ~ 62 nm [21]) is destroyed at nanoscale, making the spin compensation incomplete. The absence of macroscopic magnetization for our BiFeO₃ samples indicates that they crystallize in relatively large-sized microcrystals.

4. Conclusions

A hydrothermal route was used to synthesize BiFeO₃ crystallites, where the effect of Bi³⁺/Fe³⁺ ion concentration in precursor solution on the phase-purity and morphology of

the products was investigated. XRD analysis indicates that phase-pure BiFeO₃ crystals can be prepared at a Bi³⁺/Fe³⁺ ion concentration of 0.025–0.0625 M. SEM observation demonstrates that the samples prepared at $n(\text{Bi}^{3+}/\text{Fe}^{3+})=0.025$ and 0.0625 M present cuboid-like particles (100–200 nm) and octahedron-shaped particles (500–600 nm), respectively. When $n(\text{Bi}^{3+}/\text{Fe}^{3+})=0.0375$ and 0.05 M, the resulted samples are composed, respectively, of regular spherical agglomerates (30–40 μm) and irregular flower-like clusters, both of which are made up of irregular grains with size about several hundred nanometers. UV–visible diffuse reflectance spectra show that these samples have a similar bandgap energy of 2.20 eV. Magnetic hysteresis loop measurement reveals a typical antiferromagnetic behavior for these samples at room temperature.

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