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### Short communication

# LiCa<sub>3</sub>ZnV<sub>3</sub>O<sub>12</sub>: A novel low-firing, high Q microwave dielectric ceramic

Congxue Su, Liang Fang\*, Zhenhai Wei, Xiaojun Kuang, Hui Zhang

State Key Laboratory Breeding Base of Nonferrous Metals and Specific Materials Processing, Key Laboratory of Nonferrous Materials and New Processing Technology, Ministry of Education, College of Materials Science and Engineering, Guilin University of Technology, Guilin 541004, China

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#### Abstract

A novel low-firing microwave dielectric ceramic LiCa<sub>3</sub>ZnV<sub>3</sub>O<sub>12</sub> with low dielectric loss was synthesized by the solid-state reaction method. The phase composition, microstructure, and sintering behavior were investigated. The LiCa<sub>3</sub>ZnV<sub>3</sub>O<sub>12</sub> ceramic could be well densified at around 900 °C and exhibited favorable microwave dielectric properties with a low relative permittivity ( $\varepsilon_r$ ) of 11.5, a fairly high quality factor ( $Q_u f$ ) of 81,100 GHz, and a negative temperature coefficient of resonant frequency ( $\tau_f$ ) approximately -72 ppml°C. The LiCa<sub>3</sub>ZnV<sub>3</sub>O<sub>12</sub> ceramic can be co-fired with Ag electrode, which makes it a promising ceramic for low temperature co-fired ceramic (LTCC) applications as substrate. Furthermore, 0.8LiCa<sub>3</sub>ZnV<sub>3</sub>O<sub>12</sub>-0.2CaTiO<sub>3</sub> ceramic sintered at 925 °C for 4 h exhibited improved microwave dielectric properties of  $\varepsilon_r$ =13.2,  $Q_u f$ =59,600 GHz, and  $\tau_f$ =1.3 ppml°C.

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# 1. Introduction

The ever increasing demand for the low permittivity  $(\varepsilon_r)$ , high quality factor  $(Q_uf)$  dielectrics in microelectronic industry constantly accelerates the search for novel microwave dielectrics for substrate, packaging, and antenna applications [1,2]. In general, the advanced substrate materials should have low relative permittivity to increase signal speed, a high quality factor to increase the frequency selectivity, and a near-zero temperature coefficient of resonant frequency  $(\tau_f)$  to ensure the stability of the frequency against temperature changes. Several ceramics such as silicates and aluminates [3–6] have been developed and regarded as low  $\varepsilon_r$  and high  $Q_uf$  microwave dielectric ceramics. However, their high sintering temperatures ( $\geq 1300$  °C) limit their application in low-temperature co-fired ceramic (LTCC)-based devices. The microwave dielectric materials used in the fabrication of LTCC devices have to be

\*Corresponding author. Tel./fax: +86 7735896290.

E-mail address: fanglianggl001@yahoo.cn (L. Fang).

sintered below 950 °C for being co-fired with Ag electrode, of which the melting point is approximately 961 °C.

To reduce the sintering temperature of dielectric ceramics, adding sintering aids, such as low-melting glass or oxides, is a common method. However, this usually degrade seriously the microwave dielectric properties of the dielectric materials as the glass or oxides will produce a large amount of amorphous phase in the materials. Therefore, new materials with a low  $\varepsilon_r$ , a high  $Q_u f$ , and a near zero  $\tau_f$ , as well as the ability to be sintered at < 950 °C without requiring glass or oxide additives should be developed for application as the substrate of the LTCC devices.

Recently, many  $V_2O_5$ -rich microwave dielectric ceramics with intrinsic low sintering temperatures have attracted much attention, such as  $Mg_2V_2O_7$  [7],  $R_2V_2O_7$  (R=Ba, Sr, and Ca) [8],  $LiCa_3MgV_3O_{12}$  [9],  $Ca_5A_4(VO_4)_6$  ( $A^{2+}=Mg$ , Zn) [10], etc. These ceramics not only could be well sintered below 950 °C, but also exhibit a good combination of microwave dielectric characteristics:  $\varepsilon_r = 9 \sim 12.1$ ,  $Q_u f = 15,200 \sim 74,700$  GHz, and  $\tau_f = -26 \sim 83$  ppm/°C. Furthermore, the cubic garnet structure of  $LiCa_3MgV_3O_{12}$  was advantageous in terms

of low bulk density (for lightness of the electronic module) and good compatibility with Ag electrode [9].

In our previous work,  $V_2O_5$ -rich garnet compound  $LiCa_3MgV_3O_{12}$  was investigated and regarded as an attractive promising candidate for LTCC substrate [9]. The Shannon's effective ionic radii of  $Zn^{2+}$  (0.74 Å) is very similar to that of  $Mg^{2+}$  (0.72 Å) [11], so it is worthwhile to investigate the Zn analog ceramic. In the present work, a new low-firing and low loss microwave dielectric ceramic  $LiCa_3ZnV_3O_{12}$  (LCZV) was prepared, and its sintering behavior, microwave dielectric properties and chemical compatibility with silver were reported. Tuning the temperature coefficients of resonant frequency ( $\tau_f$ ) by the addition of  $CaTiO_3$  with a large positive  $\tau_f$  value (approximately +800 ppm/°C) was also carried out [12].

# 2. Experimental procedure

LiCa $_3$ ZnV $_3$ O $_{12}$  ceramics were prepared by conventional mixed oxide route from the high-purity oxide powders of Li $_2$ CO $_3$ , CaCO $_3$ , ZnO, and V $_2$ O $_5$  (>99%, Guo-Yao Co. Ltd., Shanghai, China). Stoichiometric amounts of the powders were weighed and milled in alcohol medium using zirconia balls for 4 h. The mixtures were dried and calcined at 750 °C for 4 h. The calcined powders were ground well and mixed with 5 wt% of polyvinyl alcohol solution as the binder. The powders were then uniaxially pressed into cylindrical disks of 12 mm diameter and 7 mm height under a pressure of 200 MPa. The samples were heated to 600 °C for 4 h to remove the organic binder and then sintered at 870  $\sim$ 930 °C for 4 h.

The phase compositions of sintered samples were determined using an X-ray diffractometer (XRD) ( $CuK\alpha_1$ , 1.54059 Å, Model X'Pert PRO, PANalytical, Almelo, The Netherlands). The bulk densities of sintered ceramics were measured by Archimedes' method. The microstructure observation of the samples was performed using field-emission scanning electron microscope (FE-SEM, Model S4800, Hitachi, Japan). Composition analysis was performed using energy-dispersive spectroscopy (EDS, IE 350; INCA, Oxford, U.K.). The microwave dielectric properties were measured using a network analyzer (Model N5230A, Agilent Co., Palo Alto, CA) and a temperature chamber (Delta 9039, Delta Design, San Diego, CA). The temperature coefficient of resonant frequency ( $\tau_f$ ) was measured in the temperature range from 25 °C to 85 °C. The  $\tau_f$  value was calculated as follows:

$$\tau_f = \frac{f_{85} - f_{25}}{60f_{25}} \tag{1}$$

where  $f_{85}$  and  $f_{25}$  are the resonant frequencies at 85 °C and 25 °C, respectively.

# 3. Results and discussion

The X-ray diffraction pattern as well as the FE-SEM image for the surface of LCZV ceramic sintered at 900  $^{\circ}$ C for 4 h are given in Fig. 1. Comparison of our experimental data with those in literatures [13] shows that the LCZV adopts the same garnet-structure as LiCa<sub>3</sub>MgV<sub>3</sub>O<sub>12</sub> (PDF files 00-024-1212).

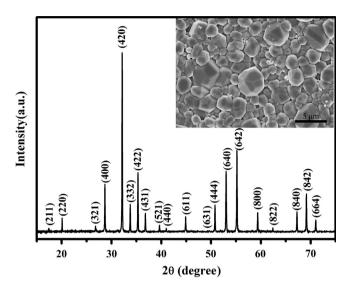


Fig. 1. X-ray diffraction result and FE-SEM image for the surface of a LiCa<sub>3</sub>ZnV<sub>3</sub>O<sub>12</sub> ceramic sintered at 900  $^{\circ}C$  for 4 h.

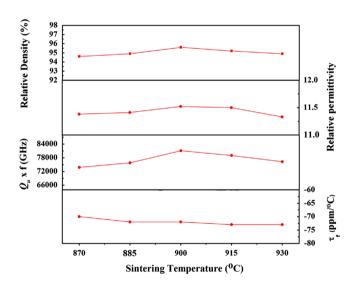


Fig. 2. The relative density, relative permittivity, quality factor, and temperature coefficient of the resonant frequency of LiCa<sub>3</sub>ZnV<sub>3</sub>O<sub>12</sub> ceramic as a function of sintering temperature.

All the peaks could be well indexed using the cubic garnet cell and there was no evidence of any secondary phases present. The lattice parameters of LCZV were calculated as a=12.439 Å, and its theoretical density calculated by the lattice parameters was  $3.71 \text{ g/cm}^3$ . Homogenous microstructures with little pores can be observed for the LCZV ceramic sintered at  $900 \,^{\circ}\text{C}$  for 4 h. The grain sizes of the ceramic were between 1 and 5  $\mu\text{m}$ .

Fig. 2 shows the variations of relative density,  $\varepsilon_r$ ,  $Q_uf$ , and  $\tau_f$  of LCZV ceramics as a function of the sintering temperature. The relative density of LCZV ceramic increased with increasing sintering temperature, reached a maximum value (95.6%) at 900 °C, and then decreased slightly. As the sintering temperature increases,  $\varepsilon_r$  increases with a trend similar to their density since the density and  $\varepsilon_r$  of ceramic are associated with the elimination of the pores [14,15]. With increasing sintering

temperature, the  $Q_u f$  value of LCZV ceramics reaches the maximum with a value of 81,100 GHz (at 10.902 GHz) and decreases thereafter. Many factors are believed to affect the microwave dielectric loss and these can be divided into two parts: the intrinsic loss and the extrinsic loss [16]. Intrinsic loss is mainly caused by lattice vibration modes whereas extrinsic loss is dominated by second phase, oxygen vacancies, grain sizes, and densification or porosity. The densification plays an important role in controlling the dielectric losses. The improvements in quality factor of the ceramics can be attributed to the increase in densification. On the other hand, the deterioration of  $Q_{uf}$  might be due to the decreased densification caused by the evaporation of lithium at elevated temperatures [17,18]. The  $\tau_f$  values of LCZV ceramics do not change remarkably with increasing sintering temperature and remain stable at approximately  $-72 \text{ ppm/}^{\circ}\text{C}$ .

LiCa<sub>3</sub>ZnV<sub>3</sub>O<sub>12</sub> ceramic developed in the present study has a low sintering temperature (900 °C), a good combination of microwave dielectric properties ( $\varepsilon_r$ =11.5,  $Q_u f$ =81,100 GHz), low bulk density (<4 g/cm<sup>3</sup>) and low cost (cheap raw materials). However, the temperature stability of resonant frequency is poor  $(\tau_f = -72 \text{ ppm/}^{\circ}\text{C})$  which precludes its use as substrate material for the practical applications. The value of  $\tau_f$  can be tuned nearly to zero by adding ceramics with a large positive  $\tau_f$  value like CaTiO<sub>3</sub> ( $\tau_f = +800 \text{ ppm/}^{\circ}\text{C}$ ) [12]. In this paper, the microwave dielectric properties and XRD patterns of (1-x)LiCa<sub>3</sub>ZnV<sub>3</sub>O<sub>12</sub>-xCaTiO<sub>3</sub>  $(0 \le x \le 0.2)$  ceramics sintered at 925 °C for 4 h are showed in Table 1 and Fig. 3. The  $\varepsilon_r$ values increased from 11.3 to 13.2 and the  $\tau_f$  values increased from -72 to 1.3 ppm/°C with increasing x values from 0 to 0.2. All the sintered ceramics exhibited high  $Q_u f$  values. The 0.8LiCa<sub>3</sub>ZnV<sub>3</sub>O<sub>12</sub>–0.2CaTiO<sub>3</sub> ceramic sintered at 925 °C exhibits favorable microwave dielectric properties with  $\varepsilon_r = 13.2$ ,  $Q_u f = 59,600$  GHz (at 9.985 GHz), and  $\tau_f = 1.3$ ppm/°C. EDS analysis was carried out to identify the second phase. Fig. 4 shows the result from energy-dispersive spectroscopy of second phase in 0.8LiCa<sub>3</sub>ZnV<sub>3</sub>O<sub>12</sub>-0.2CaTiO<sub>3</sub> ceramic sintered at 925 °C. Ca and Ti element are detected at a ratio of approximately 1:1, which confirms CaTiO3 as the secondary phase.

To evaluate the chemical compatibility of LCZV ceramic with silver electrode, LCZV was cofired with 20% Ag powders at 900 °C for 2 h and the product was analyzed to detect interactions between the low-fired samples and electrodes. The X-ray powder diffraction pattern and the backscattered electron image of LCZV cofired with Ag at 900 °C are presented in Fig. 5. As the XRD pattern does not show the formation of any

Table 1 Bulk densities and microwave dielectric properties of  $(1-x)\text{LiCa}_3\text{ZnV}_3\text{O}_{12}-x\text{CaTiO}_3$  ( $0 \le x \le 0.2$ ) ceramics sintered at 925 °C for 4 h.

x value	Bulk density (g/cm <sup>3</sup> )	$\varepsilon_r$	$Q_u f$ (GHz)	$\tau_f \text{ (ppm/°C)}$
0	3.51	11.3	72,400	-72
0.05	3.49	11.6	70,800	-69
0.1	3.45	12.3	67,200	-46
0.2	3.39	13.2	59,600	+1.3

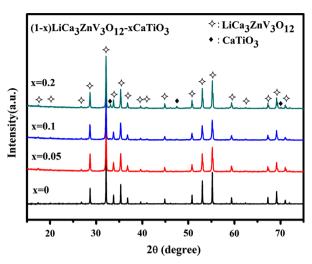
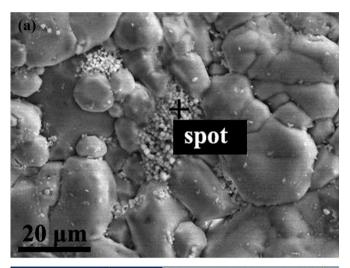


Fig. 3. X-ray diffraction patterns of (1-x)LiCa<sub>3</sub>ZnV<sub>3</sub>O<sub>12</sub>-xCaTiO<sub>3</sub>  $(0 \le x \le 0.2)$  ceramics sintered at 925 °C for 4 h.



(b)	Element	Weight%	Atomic%			
C 2	ОК	41.83	66.57			
P	Ca K	24.97	15.86			
1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1	Ti K	30.41	16.17			
(a)	VK	2.79	1.40			
	•					
2 4	6	8 1	0 12			
Full Scale 4155 cts Cursor: 0.000 keV						

Fig. 4. FE-SEM image of  $0.8\text{LiCa}_3\text{ZnV}_3\text{O}_{12}$ – $0.2\text{CaTiO}_3$  ceramic sintered at 925 °C, and result of energy-dispersive spectroscopy of second phase in 0.8LCZV–0.2CTO.

other phase except for LCZV and Ag, and the backscattered electron image of LCZV/Ag sample cofired at 900 °C for 2 h reveals no interaction to form new phases after firing, then it is considered that the LCZV compound has an excellent chemical compatibility with silver.

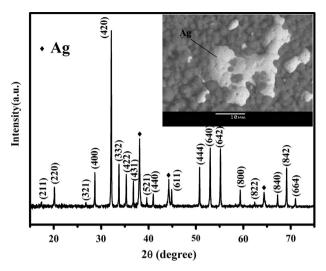


Fig. 5. XRD patterns of a LiCa $_3$ ZnV $_3$ O $_{12}$  ceramic mixed with 20 wt% Ag sintered at 900 °C for 2 h. Inset shows a backscattered electron image of LiCa $_3$ ZnV $_3$ O $_{12}$  sample cofired with Ag.

### 4. Conclusions

A novel low-firing vanadate garnet LiCa<sub>3</sub>ZnV<sub>3</sub>O<sub>12</sub> ceramic was investigated as promising low-loss dielectric ceramic for microwave applications. LCZV ceramic can be densified at a relatively low sintering temperature of 900 °C and exhibited good microwave dielectric properties of  $\varepsilon_r$ =11.5,  $Q_u f$ =81,100 GHz, and  $\tau_f$ =-72 ppm/°C. By adding CaTiO<sub>3</sub> into LCZV, the microwave properties were improved. 0.8LiCa<sub>3</sub>ZnV<sub>3</sub>O<sub>12</sub>-0.2CaTiO<sub>3</sub> ceramic sintered at 925 °C for 4 h exhibited improved properties of  $\varepsilon_r$ =13.2,  $Q_u f$ =59,600 GHz, and  $\tau_f$ =1.3 ppm/°C. The backscattered electron image and XRD analysis confirmed that the ceramic did not react with silver powders, indicating that LCZV might be an attractive promising candidate for LTCC application.

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