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Short communication

Co-synthesis of ZnO/SnO₂ mixed nanowires via a single-step carbothermal reduction method

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Abstract

ZnO/SnO₂ mixed nanowires are fabricated by a single-step in-situ carbothermal reduction of mixed ZnO and SnO₂ powders with activated carbon under atmospheric pressure on a gold (Au) coated alumina substrate. Morphology and structural properties of the mixed nanowires are studied by field emission scanning electron microscopy (FESEM), X-ray diffraction (XRD) and transmission electron microscopy (TEM). FESEM investigation reveals the presence of uniform straight nanowires and branched hierarchical nanowires. The mixed nanowires have a diameter of about 15–80 nm and length ranging from two to several tens of micrometers, and thus having an aspect ratio above 1000. X-ray diffraction patterns show the crystalline nature of the ZnO/SnO₂ mixed nanowires. Elemental mapping in a TEM suggests that the ZnO and SnO₂ phases are mixed uniformly in the nanowire.

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1. Introduction

One-dimensional metal oxide semiconductor nanostructures have attracted noteworthy attention from researchers because of their electrical, chemical and optical properties [1,2] which lead to potential applications in catalysis [3], chemical and biological sensors [3,4], photodetectors [5], and lithium-ion batteries [6]. Mixed oxide nanowires with hierarchical and core-shell structures have received considerable attention as they can offer improved electrical properties, such as carrier concentration and conductivity, suitable for photovoltaic (PV) energy conversion and chemical sensing [7]. Hybrid onedimensional nanostructures of mixed metal oxide semiconductors, such as $SnO_2/W_{18}O_{49}$ [7], ZnO/SnO_2 [8–11], CeO_2/TiO_2 [12], TiO_2/ZnO [13], α -Fe₂O₃/SnO₂ [14], and α -Fe₂O₃/ZnO [15] have been synthesized by a variety of methods and they have shown better photovoltaic and chemical sensing properties compared to single oxide 1D structures. Among these oxides, ZnO ($E_g \sim 3.37 \text{ eV}$) and SnO₂ ($E_g \sim 3.6 \text{ eV}$) are wide band gap and key functional materials, which have been widely studied [16–19]. Recently, ZnO/SnO₂ mixed nanowires have been synthesized by a number of techniques, such as, vapor phase transport processes [20,21], the two-step carbon assisted thermal evaporation method [8,22], electrospinning [9,10], chemical vapor transport and condensation [23], and the hydrothermal method [20,24]. Until now, most of the ZnO/ SnO₂ mixed nanostructures have been reported to be produced by a two-step method at low pressure, where each oxide is synthesized by the carbothermal reduction method sequentially. For example, Hwang et al. [8] synthesized ZnO/SnO₂ core-shell nanowires via a two-step carbothermal reduction method with pressure inside the system maintained around 10⁻² Torr, while Lan et al. [22] produced hierarchical nanostructures at a pressure of 1 Pa.

This paper reports ZnO/SnO₂ nanowires synthesized by a single-step carbothermal reduction, which is a relatively facile method compared to the above mentioned techniques. All these methods except electrospinning and hydrothermal, are either governed by the vapour–solid (VS) mechanism or the catalyst assisted vapour-liquid-solid (VLS) mechanism. The work reported here employs the catalyst assisted VLS

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mechanism. In this process, metal nanoparticles or nanoclusters are used as nucleation seeds. These nucleation seeds determine the growth direction, interfacial energy and diameter of the 1-D metal oxide nanowires [25]. In this facile method, the morphology and properties of nanowires are generally controlled by growth parameters such as temperature, thickness of catalyst layer, rate of carrier gas flow, and distance between the source and the substrate [26]. Morphology and structural characterizations of the fabricated crystalline ZnO/SnO₂ mixed nanowires are presented and discussed.

2. Experimental procedure

The synthesis of ZnO/SnO₂ mixed nanowires in this work involved a carbothermal reduction process. Here activated carbon powder acts as a reducing agent for the metal oxide powders. Commercial ZnO (350 nm, 99% purity, Sigma-Aldrich), SnO_2 (~325 mesh, 99.9% trace metals basis, Sigma-Aldrich) and activated carbon (R&M Chemicals, UK) powders with a molar ratio of 9:1:10 were ground together by ball milling for 8 h to get a homogeneous mixture of nanoparticles with an average particle size of 150 nm. A thin Au layer of 40 Å thickness was deposited on an alumina substrate by a sputter coater. A quartz boat loaded with milled powder mixture was placed in the center of a horizontal tube furnace (Lindberg Blue M: TF55035COMA1). The substrate was located at 9.5 cm away from the source powder. Then Ar (purity 99.99%) was introduced as a carrier gas with a constant flow rate of 25 sccm throughout the process. The temperature of the furnace was increased from room temperature to 900 °C at a rate of 30 °C min⁻¹. This temperature was maintained for a growth time of 2 h and subsequently cooled down to room temperature. A light gray layer was observed on the alumina substrate after being cooled down to room temperature. The control experiments were conducted with two samples: (1) with activated carbon and no Au layer and (2) with Au layer and no activated carbon.

The as-prepared samples were collected for phase identification by X-ray diffraction (XRD) using a PANalytical Empyrean utilizing CuK α radiation (1.54059 Å). The morphology of the synthesized nanowires was determined by a field emission scanning electron microscopy (FESEM, Auriga Zeiss Ultra-60). The morphological features and elemental distribution were characterized by a transmission electron microscopy (TEM, FEI Tecnai F-20 microscopy) in the STEM mode.

3. Results and discussion

Fig. 1 illustrates the XRD pattern of nanowires, indicating the presence of crystalline würtzite structure for ZnO (JCPDS no. 79-0208) and tetragonal cassiterite structure for SnO₂ (JCPDS no. 77-0449). In addition, the diffraction lines resulting from the substrate are also observed. Additionally, some peaks of ternary compound of face centered cubic spinel Zn₂SnO₄ (JCPDS no. 24-1470) are also detected. This result indicates that a part of SnO₂ and ZnO react to form Zn₂SnO₄. The presence of sharp and strong X-ray diffraction peaks

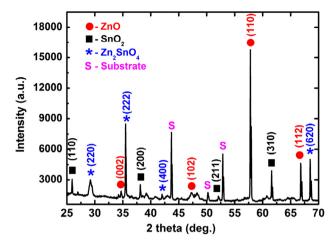


Fig. 1. XRD patterns of ZnO/SnO2 mixed nanowires.

suggests that the ZnO/SnO_2 mixed nanowires are well crystallized.

FESEM micrographs of the samples prepared at 900 °C for 2 h are shown in Fig. 2. Two control experiments were performed to find the effect of Au layer and activated carbon on nanowire growth. In Fig. 2(a), it is seen that no nanowires are observed in the absence of Au on alumina substrate with the presence of activated carbon. Similarly, no nanowires are present on Au/alumina substrate in the absence of activated carbon as it can be seen in Fig. 2(b). Here, spheroidal shaped Au particles are seen that are formed by dewetting of the Au layer. But, in the presence of both carbon and Au, uniform and well distributed nanowires can be seen on Au/alumina substrate (Fig. 2(c)). The diameter of the nanowires is in the range of 15-80 nm with a length of 16-50 µm. The inset of Fig. 2(c) shows the enlarged image of the tip of a nanowire. The presence of the Au cap at the tip of the nanowire is a strong evidence that the growth of nanowires is by the vaporliquid-solid (VLS) assisted mechanism [27]. It is believed that at 900 °C, the activated carbon dissociates ZnO and SnO₂ powders into Zn and Sn vapors, which condense on the Au/ alumina substrate at lower temperature. The Sn and Zn metal vapors react with Au layer and form an alloy droplet on the alumina substrate. This alloy droplet becomes super saturated after receiving more metal vapors that initiates the nanowire growth. Therefore, the presence of activated carbon and Au are essential for the growth of nanowires, where the activated carbon acts as a reducing agent producing the metal vapors and the Au acts as a catalyst to grow the nanowires.

Different types of nanostructures are observed under FESEM at different locations on the Au/alumina substrate, which are illustrated in Fig. 3(a) and (b). Fig. 3(a) shows a comb-like branch structure of ZnO/SnO₂ nanowires. In this structure, the diameter of the backbone nanowire is 80 nm and the branched nanowires show a uniform diameter of 30 nm (Inset of Fig. 3(a)). Fig. 3(b) illustrates the morphology of ZnO/SnO₂ hierarchical nanostructure. More importantly, the secondary branches from backbone nanowires are noteworthy (Inset of Fig. 3(b)). The end of these secondary branches has Au tip which indicates that VLS mechanism is operative

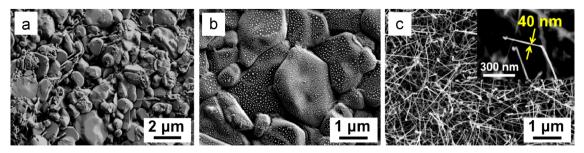


Fig. 2. FESEM images of the nanostructures of samples prepared at 900 °C for 2 h grown on (a) alumina substrate in the presence of activated carbon, (b) Au/alumina substrate in the absence of activated carbon and (c) Au/alumina substrate in the presence of activated carbon.

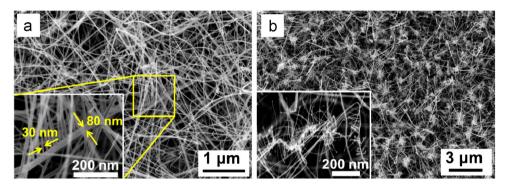


Fig. 3. FESEM micrographs of ZnO/SnO₂ mixed nanowires: (a) comb-like and (b) hierarchical nanostructures.

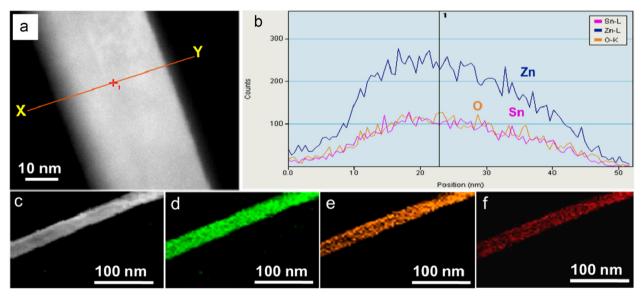


Fig. 4. (a) Magnified STEM image of ZnO/SnO_2 nanowire, (b) cross-sectional compositional line profiles of Zn, Sn and O, (c) EDX elemental mapping of ZnO/SnO_2 nanowire demonstrating the spatial distribution of (d) Zn, (e) Sn, and (f) O.

here. Au plays an important role of catalysts to grow both the branch and hierarchical nanowires. This kind of hierarchical nanostructures have been reported by various fabrication methods [20,22]. In our study, hierarchical nanostructures consist of trunk and branches, which are composed of mixed ZnO/SnO₂ material. However, a controlled synthesis of these kinds of nanostructures is still a challenge. It has been suggested that the backbone nanowire grew quickly with the Au catalyst particle at the tip, while smaller alloy droplets

would be formed on the backbone nanowire surface which leads to the hierarchical nanowires in different location [28]. The growth of hierarchical nanowires is triggered, when the critical value of temperature and vapor supersaturation ratio is achieved. In our case, we cannot clearly state which factor mostly influences the hierarchical nanowires. Further study is underway to clarify this issue.

Fig. 4(a) and (b) shows concentration line profiles of Zn, Sn and O across a nanowire. The profile suggests that both Sn and

Zn are homogeneously distributed within the nanowire and Zn is richer than Sn. Additionally, Fig. 4(d)–(f) demonstrates uniform spatial distribution of Zn, Sn, and O in EDX elemental mapping shown in Fig. 4(c). This result clearly indicates that ZnO and SnO₂ are mixed throughout the nanowires that are synthesized by the single-step carbothermal reduction method. This is very different than what is reported in literature, where they grow sequentially on top of each other. This clearly represents a simple process to grow mixed nanowires with homogeneous composition with potential applications in chemical sensing and other devices.

4. Conclusions

In summary, this study verifies that a large number of ZnO/SnO₂ mixed nanowires per unit area can be grown on Au/alumina substrate by employing a single-step carbothermal reduction method. The role of the activated carbon and Au are systematically studied. According to TEM results, mixed ZnO/SnO₂ phases are confirmed to be uniformly distributed throughout the nanowire. The catalyst assisted growth of nanowire on alumina substrate constitutes to be a promising method to grow nanowires over large area. The unique morphology and homogeneous composition of ZnO/SnO₂ mixed nanowires may lead to their potential applications.

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