

Electrochemical behaviour and degradation of (Ni,M)/YSZ cermet electrodes (M = Co,Cu,Fe) for high temperature applications of solid electrolytes

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Abstract

Homogeneous powders of alternative (Ni,M)/YSZ cermets were prepared by combustion synthesis to obtain homogeneous mixtures of submicrocrystalline powders. These powders were used to prepare symmetrical cells cermet/YSZ/cermet, with porous cermet layers and a dense electrolyte. The cermets were reduced in H₂, and tested as potential fuel electrodes for high temperature electrochemical applications. The best performance was found for Ni_{1-x}Co_x/YSZ cermets but small fractions of Cu might also play a positive effect. Changes in capacitance suggest that this corresponds to cermets with enhanced microstructures. Lower firing conditions are probably needed to avoid the degradation of cermets containing Cu or Fe.

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1. Introduction

The performance of Ni/YSZ cermet anodes is mostly dependent on microstructural features^{1–3} which might affect the phase connectivity and triple phase contacts. The triple boundary length plays a major role on the electrochemical oxidation of H₂,^{3–7} and may also explain the minimum of polarisation resistance of Ni/YSZ cermets for about 40 vol.% of Ni.⁵ However, a fraction of coarse YSZ particles may contribute to anchor the cermet microstructure yielding improved performance.⁶

One might also consider cermets with alternative compositions to modify the metal/YSZ contact angle, and thus enhance the triple contacts. For example, improved percolation was attained for Ni/YSZ-based cermets on adding small fractions of titania.⁸ One may also seek other potential advantages such as changes in catalytic properties of the metallic phase, enhanced creep resistance, better oxidation resistance or better

thermal expansion compatibility by changing the metallic components of the cermets.

Additions of Cu should improve the oxidation resistance of the cermets. However the relatively low melting temperatures of this metal and of the corresponding oxides do not allow high temperature processing and are also likely to cause faster microstructural ageing at typical cell working temperatures. Ni–Co and Ni–Fe alloys are less resistant to oxidation but should be less prone to difficulties of processing or fast ageing.

2. Experimental procedure

Cermet precursors were prepared by exothermic reaction of nitrates with urea.⁹ These reactions are fast and yield very homogeneous mixtures of sub-microcrystalline powders, usually in the form of irregular agglomerates.⁹ This contributes to prevent excessive shrinkage on firing at temperatures up to 1450 °C, except possibly for cermets containing Cu (Fig. 1).

Symmetrical cermet/electrolyte/cermet cells were prepared from those powders by co-pressing and co-firing at 1450 °C, and reducing in H₂ at 800 °C, yielding porous cermet layers bonded to a dense YSZ electrolyte

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layer. Cu-containing cermetes were prepared both as single layer $\text{Ni}_{1-x}\text{Cu}_x + \text{YSZ}$ cermetes, and bilayered cermetes $(\text{Ni} + \text{YSZ})/(\text{Ni}_{1-x}\text{Cu}_x + \text{YSZ})$ to hinder diffusion of Cu into the electrolyte.

An Autolab spectrometer (ECO Chimie B.V.) was used to characterise the symmetrical cermet/YSZ/

cermet cells in wet H_2 or wet $\text{N}_2 + \text{H}_2$ atmospheres, obtained by bubbling through water at temperatures in the range 20–56 °C. The oxygen partial pressure was monitored by a YSZ sensor, and these readings were used to evaluate the water vapour partial pressure in the atmosphere, on assuming equilibrium in the gas phase. The impedance spectra obtained under these conditions are suitable to separate the main contributions of the polarisation resistance.

3. Results and discussion

Fig. 2 shows typical impedance spectra obtained for cermet/YSZ/cermet cells. The polarisation resistance comprises a contribution in the low frequency range (1–6 Hz), and a contribution at higher frequencies. This contribution (R_{Hf}) is temperature dependent (Fig. 3), and is almost independent of the fraction of H_2 in the atmosphere (Figs. 4 and 5). On the contrary, the low frequency contribution is strongly dependent on the gas composition (Fig. 6) and is almost insensitive to changes in temperature. These trends thus resemble the main contributions reported for optimised Ni + YSZ cermetes.¹ One might thus assume that the contribution R_{Hf} is mainly dependent on the cermet microstructure, as reported by those authors.

Partial substitution of Ni with Co lowers the contribution R_{Hf} (Fig. 3) whereas other $(\text{Ni}, \text{M}) + \text{YSZ}$ cermetes with $\text{M} = \text{Fe}$ or Cu tend to yield poorer results, except for the bilayered $(\text{Ni} + \text{YSZ})/(\text{Ni}_{0.96}\text{Cu}_{0.04} + \text{YSZ})$ cermet. The results in Figs. 4 and 5 show that improved behaviour usually correspond to higher capacitance C_{Hf} and vice-versa, thus suggesting that this is related to changes in triple phase boundary, as reported for Ni patterns onto YSZ substrates.^{4,6} However, doping with Cu is likely to cause some degradation of the electrolyte (Fig. 2) possibly due to diffusion of Cu. This may also be the case for cermetes containing Fe. Further work is thus needed to understand the effects of these additives

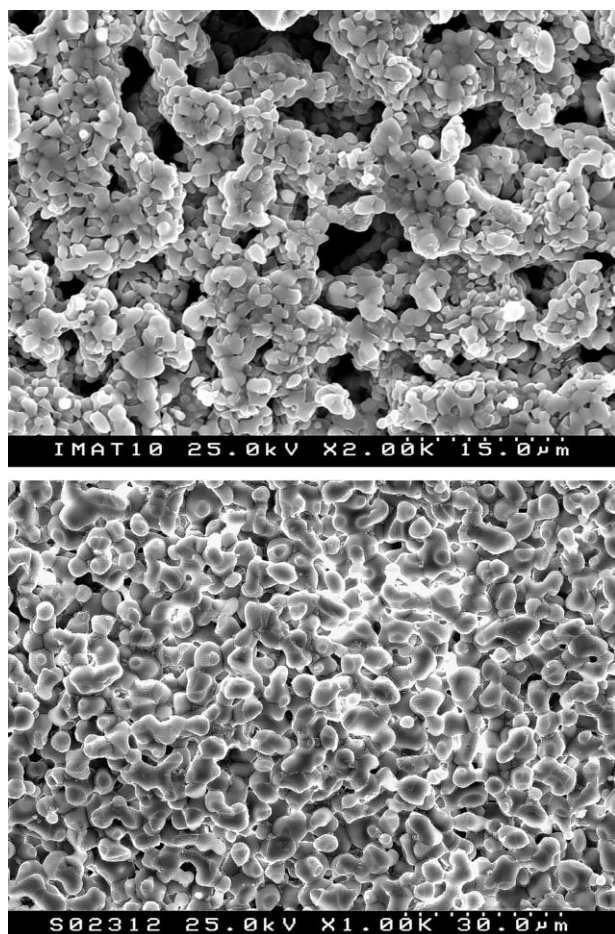


Fig. 1. SEM microstructures of $\text{Ni}_{0.5}\text{Co}_{0.5} + \text{YSZ}$ and $\text{Ni}_{0.96}\text{Cu}_{0.04} + \text{YSZ}$ cermetes fired at 1450 °C.

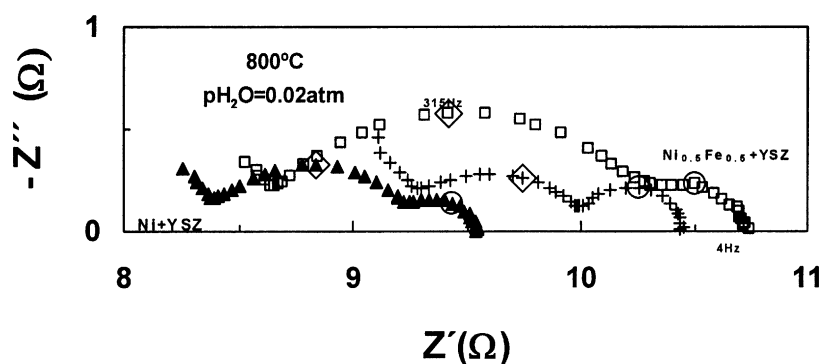


Fig. 2. Impedance spectra obtained for symmetrical cermet/YSZ/cermet cells with single layers cermetes Ni + YSZ (\blacktriangle) and $\text{Ni}_{0.5}\text{Fe}_{0.5} + \text{YSZ}$ (\square), and double layer cermet $(\text{Ni} + \text{YSZ})/(\text{Ni}_{0.96}\text{Cu}_{0.04} + \text{YSZ})$ ($+$) at 800 °C, in H_2 with $\text{pH}_2\text{O} \approx 0.02$ atm.

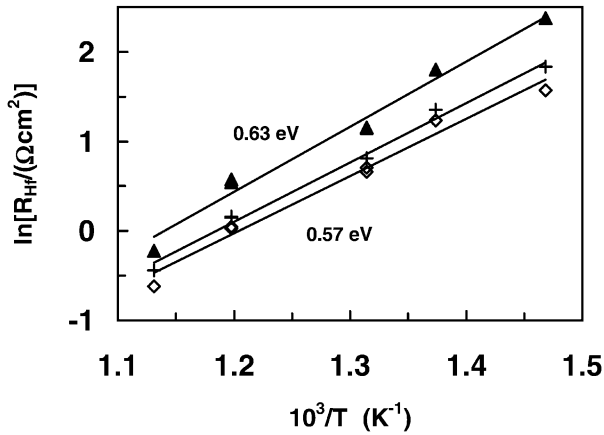


Fig. 3. Temperature dependence of the polarisation resistance in H_2 with $pH_2O=0.04\text{--}0.05$ atm obtained for cermets Ni+YSZ (▲); $Ni_{0.7}Co_{0.3}+YSZ$ (+); and $Ni_{0.5}Co_{0.5}+YSZ$ (◇).

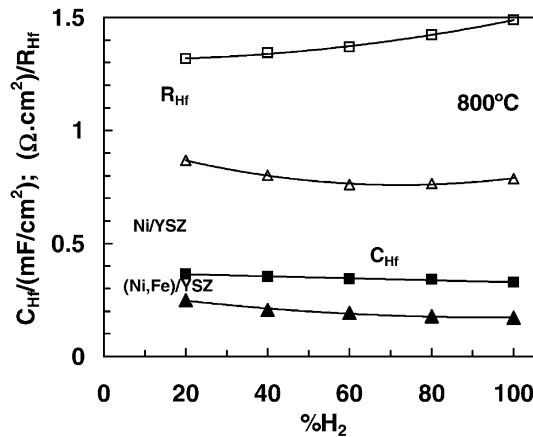


Fig. 4. Effects of H_2 content on R_{Hf} (open symbols) and C_{Hf} (closed symbols) for single layer cermets Ni+YSZ (■, □) and $Ni_{0.5}Fe_{0.5}+YSZ$ (▲, △), at 800 °C, and under relatively dry atmospheres ($pH_2O \approx 0.02$ atm).

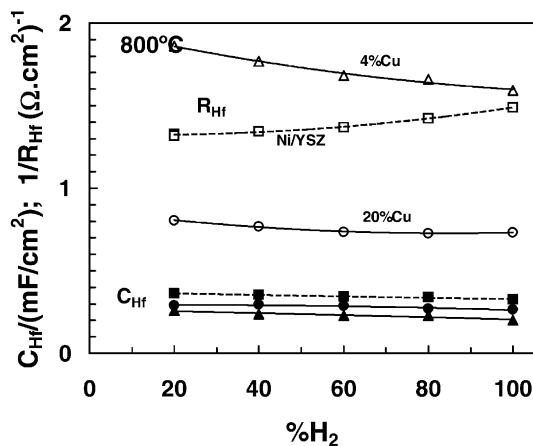


Fig. 5. Effects of H_2 content on R_{Hf} (open symbols) and C_{Hf} (closed symbols) for Ni+YSZ (■, □) and double layer cermets $(Ni+YSZ)/(Ni_{1-x}Cu_x+YSZ)$, $x=0.04$ (▲, △) and 0.2 (●, ○), at 800 °C, and under relatively dry atmospheres ($pH_2O \approx 0.02$ atm).

and to attempt to optimise the composition and firing conditions of the cermets.

The dependence of the low frequency contribution of the polarisation resistance on the gas composition has been reported by many authors for Ni electrodes or Ni-based cermets.^{1,10–13} Mizusaki and co-authors⁴ proposed a Butler–Volmer equation to explain the decrease in polarisation resistance with increasing partial pressures of H_2 and water vapour. A mechanism based on dissociative adsorption or diffusion on the surface of Ni should yield linear relations $1/R_{Lf} \propto pH_2O$, and $1/R_{Lf} \propto (pH_2)^{1/2}$. Jiang and Badwal¹¹ proposed a similar mechanism by taking into account that the temperature dependence of R_{Lf} is weak. On the contrary, Primdhal and Mogensen¹⁴ proposed a model based on the assumption that gas flow limitations prevail. In this case, the resistance R_{Lf} should vary as $(1/pH_2O + 1/pH_2)$, the capacitance shows the opposite trend, $C_g \propto (1/pH_2O + 1/pH_2)^{-1}$, and the relaxation frequency should be independent of the gas composition. The actual results show that R_{Lf} drops almost linearly with temperature, whereas C_g usually shows a weaker dependence. The dependence of relaxation frequency on gas composition (Table 1) is also a

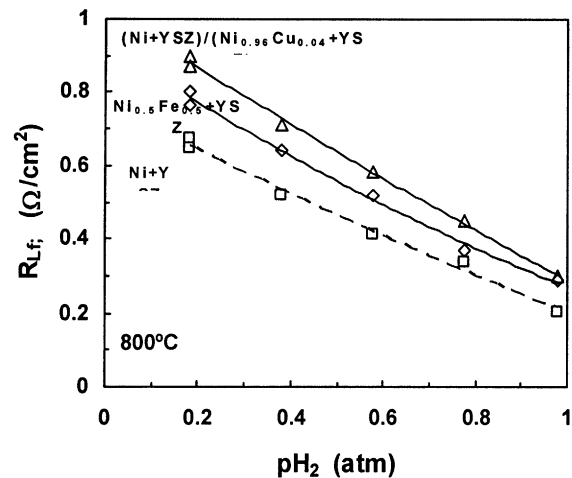


Fig. 6. Effects of H_2 content on the values of R_{Lf} obtained for cermets Ni+YSZ (□), $Ni_{0.5}Fe_{0.5}+YSZ$ (◇), and $(Ni+YSZ)/(Ni_{0.96}Cu_{0.04}+YSZ)$ (△), at 800 °C and under relatively dry atmospheres ($pH_2O \approx 0.02$ atm).

Table 1

Relevant parameters of the low frequency contribution obtained for a cermet electrode $(Ni+YSZ)/(Ni_{0.96}Cu_{0.04}+YSZ)$ with area 1.46 cm^2 , at 700 °C and under $pH_2O=0.02$ atm

pH_2	R_{Lf}	C_{Lf}	f_{Lf} (Hz)
0.18	1.14	0.109	1.3
0.38	0.952	0.103	1.6
0.58	0.777	0.107	1.9
0.78	0.577	0.11	2.5
0.98	0.364	0.118	3.7

confirmation that gas transport is unlikely to control the overall process.

4. Conclusions

Alternative (Ni,M) + YSZ cermet with M=Co, Fe, or Cu can be prepared from powders obtained by a combustion synthesis method, which yields uniform distributions of the cermet components. Cermet/YSZ/cermet cells were also prepared from those powders. The main features of the electrode behaviour obtained with such symmetrical cells in atmospheres containing H₂ and H₂O are close to results reported for optimised Ni/YSZ cermets in identical conditions, except possibly for the magnitude of the polarisation resistance. One of its contributions is still mainly dependent on temperature and probably also on microstructural features, and the low frequency contribution is mainly dependent on the gas composition. Changes in cermet composition effect the microstructural contribution, and may yield a decrease in polarisation resistance, especially for partial substitution of Ni by Co, or doping with copper.

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