

# Analysis of micro-machining characteristics of $\text{Si}_3\text{N}_4$ -hBN composites

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## Abstract

In this study, micro-machining characteristics of  $\text{Si}_3\text{N}_4$ -hBN composites were analyzed for the development of micro-pattern making methods. First,  $\text{Si}_3\text{N}_4$ -hBN composite specimens of various h-BN contents were prepared using hot-press method. *R*-Curve behavior of the composites was investigated to understand its machinability. And, micro-shaping, micro-end-milling and micro-drilling processes were performed for the machinability evaluation of the composites. In the micro-shaping, V-grooves were machined and their shapes and cutting forces were measured and analyzed. Similar processes were performed for the micro-end-milling and micro-drilling processes. The results of this study can be applied to determine optimum micro-machining conditions for  $\text{Si}_3\text{N}_4$ -hBN composites.

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**Keywords:** Machinable ceramics;  $\text{Si}_3\text{N}_4$ -hBN composites; Micro-machining; Micro-shaping; Micro-end-milling; Micro-drilling

## 1. Introduction

$\text{Si}_3\text{N}_4$  composites are known as very difficult-to-cut materials because of its high strength and hardness. Their machining mechanism is much different with metals, which can be characterized by cracking and brittle fracture due to relatively high strength and temperature resistance.<sup>1,2</sup> However,  $\text{Si}_3\text{N}_4$ -hBN composite materials are well known as one of the machinable ceramics, which can be easily machined with hard-alloy cutting tools under proper machining conditions.<sup>3–5</sup> Also, the needs for micro-parts using such materials are rapidly increasing in related application fields.

In this study, the analysis of micro-machining characteristics of  $\text{Si}_3\text{N}_4$ -hBN composites were performed to determine optimum micro-machining conditions for those applications. As a first step, the *R*-curve behavior of the  $\text{Si}_3\text{N}_4$ -hBN composites was investigated to understand its machinability. It could be observed that the *R*-curve rose slowly for this material so that the fracture toughness was low at small cracks but high at long cracks. Then, a set of micro-machining experiments, such as micro-grooving, micro-end-milling and micro-drilling, were performed to investigate the micro-machining characteristics with variable h-BN contents and different conditions. Also, cutting forces and tool wear were investigated for each experi-

mental condition. From the results, it could be investigated that the cutting forces decreased and tool life increased with the increase of h-BN contents. It can be concluded that the results of this study can be applied for the micro-parts production using  $\text{Si}_3\text{N}_4$ -hBN composites.

## 2. Composites

### 2.1. Composite processing and microstructure

The starting powders were  $\text{Si}_3\text{N}_4$  ( $\alpha$ -crystal phase >95%, average particle size, 0.17  $\mu\text{m}$ , Ube Industries, Ltd., Japan), and h-BN (>99% pure, 10  $\mu\text{m}$ , Kojundo Chemical Laboratory Ltd., Japan). The amounts of added h-BN ranged from 0 to 30 vol.%. In all specimens, 8 mol% of  $\text{Y}_2\text{O}_3$  (>99.9% pure, <0.5  $\mu\text{m}$ , Shinestu Chemical Co. Ltd., Japan) and 6 mol%  $\text{Al}_2\text{O}_3$  (>99.99% pure, 0.39  $\mu\text{m}$ , Sumitomo Chemical Co. Ltd., Japan), were added as sintering aids. The powders were mixed by wet ball-milling using  $\text{Al}_2\text{O}_3$  jar,  $\text{Si}_3\text{N}_4$  balls and ethanol for 72 h. The mixed wet powders were dried on a hot plate using a rotating stirrer to avoid gravity-induced segregation. To obtain fully dense compacts, the dried and sieved (to 16 mesh) powders were hot-pressed at 1800 °C for 2 h under a pressure of 25 MPa in a  $\text{N}_2$  atmosphere. Rectangular specimens (3 mm  $\times$  4 mm  $\times$  36 mm) were prepared by cutting, grinding, and polishing the hot-pressed compact. According to the results of XRD analysis for monolithic  $\text{Si}_3\text{N}_4$  and  $\text{Si}_3\text{N}_4$ -hBN composites, the phase transi-

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tion of  $\alpha$ - to  $\beta$ - $\text{Si}_3\text{N}_4$  was completed in all specimens, and there was no evidence for the reaction between hBN and  $\text{Si}_3\text{N}_4$  matrix.

The fracture surfaces and polished/etched surfaces of specimens were investigated by SEM.  $\text{Si}_3\text{N}_4$  grain size was mea-

sured from the SEM images of surfaces etched for 1 h by mixed solution of  $\text{HF} + \text{HNO}_3$  at  $80^\circ\text{C}$ . Based on the results for grain size observation, the microstructures of monolithic  $\text{Si}_3\text{N}_4$  and  $\text{Si}_3\text{N}_4$ -hBN composite revealed a needle-like fine

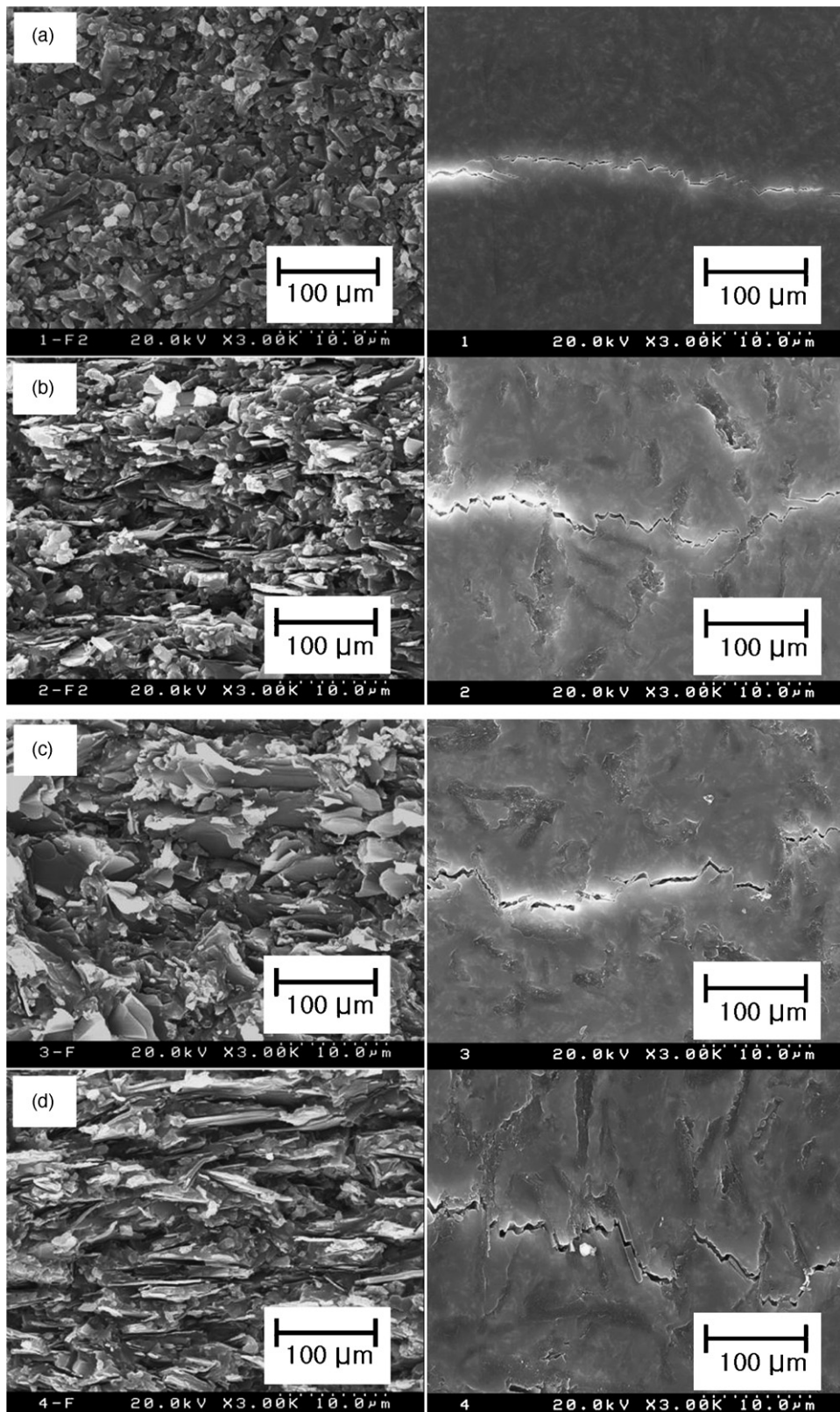


Fig. 1. SEM micrographs of fracture surfaces and crack paths by indentation of (a) 0 vol.%, (b) 20 vol.%, (c) 25 vol.% and (d) 30 vol.% of h-BN.

$\beta$ - $\text{Si}_3\text{N}_4$  grains. The true length of the fine  $\beta$ - $\text{Si}_3\text{N}_4$  grains was difficult to measure because the  $\text{Si}_3\text{N}_4$  grains were oriented three-dimensionally. Thus, the apparent length was measured from the two-dimensional microstructure. The length of  $\beta$ - $\text{Si}_3\text{N}_4$  slightly decreased with increasing h-BN content from about  $3.2\text{ }\mu\text{m}$  of monolith to  $2.8\text{ }\mu\text{m}$  of  $\text{Si}_3\text{N}_4$ -30% BN composite.

Fig. 1 shows the fracture surfaces, broken parallel to the hot-pressed direction, of the monolithic  $\text{Si}_3\text{N}_4$  and the  $\text{Si}_3\text{N}_4$ -hBN composites. In all specimens, evidence for pullouts of  $\text{Si}_3\text{N}_4$  grains and h-BN platelets was observed. Also, the crack paths, obtained by indentation, are shown in Fig. 1 for the same specimens. The well-polished surface was indented along the vertical direction to the hot-pressed direction. Crack paths were obtained on the polished section by indentation with a Vickers hardness tester under a load of 196 N. The crack paths in all specimens were sinusoidal due to grain pullout and bridging of grains during crack propagation. Grain bridging enhances fracture toughness resulting in crack resistance behavior (*R*-curve behavior) as fracture toughness increases with crack growth. It is notable that crack paths for  $\text{Si}_3\text{N}_4$ -hBN composites were sinusoidal in larger scales compared with the crack path for  $\text{Si}_3\text{N}_4$  monolith. Grain bridging at a larger scale leads to more effective toughening.<sup>6</sup>

Based on these microstructural observations, grain bridging and pullout could be possible toughening mechanisms for the monolithic  $\text{Si}_3\text{N}_4$  as well as  $\text{Si}_3\text{N}_4$ -hBN composite. In general, hot-pressed  $\text{Si}_3\text{N}_4$ -hBN composites exhibit strong texture of BN grains oriented with the *c*-axis parallel to hot-pressing direction. This anisotropic texture may enhance fracture toughness further of the composites by more effective crack bridging and pull-out of grains oriented vertically to the crack propagating direction.

## 2.2. *R*-Curve behavior analysis for machinability evaluation

For the *R*-curve characterization, indentation under loads from 4.9 to 196 N was employed on the polished surfaces, and 4-point flexural strength was determined. *R*-Curves were obtained from the relationships of the indentation load, indentation crack length and the strength after indentation following the Krause's procedure.<sup>7</sup> Fig. 2 presents the indentation strength as a function of indentation load, and *R*-curves were obtained from the data. For the monolithic  $\text{Si}_3\text{N}_4$ , fracture toughness rises rapidly at the relatively short cracks and saturates to a plateau quickly. On the other hand, the  $\text{Si}_3\text{N}_4$ -hBN composites show slowly rising *R*-curves. At small crack lengths cracks should form and grow easily along the layer-structured BN particles resulting in very low fracture toughness. As cracks grow, grain bridging becomes effective to enhance fracture toughness. The addition of BN particles resulted in the larger scale of grain bridging as shown in Fig. 1, and the plateau fracture toughness was even higher than the monolithic  $\text{Si}_3\text{N}_4$ . This slowly rising *R*-curve behavior could be very important for the machinability of these composites. BN particles in the composites decrease significantly the fracture toughness of the composite at small crack

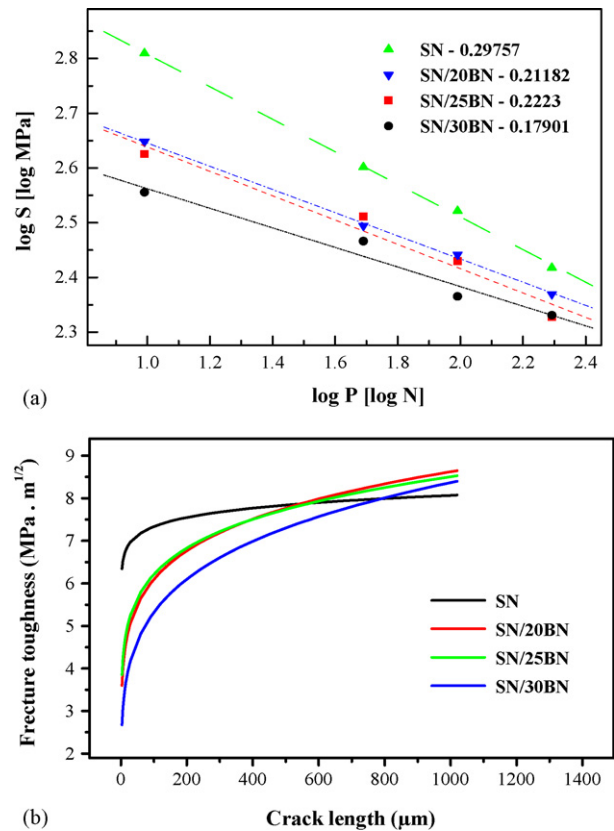


Fig. 2. (a) Indentation load vs. strength and (b) *R*-curves for monolithic  $\text{Si}_3\text{N}_4$  and  $\text{Si}_3\text{N}_4$ -hBN composites containing 20 vol.%, 25 vol.% and 30 vol.% of h-BN.

lengths, which renders machinability. On the other hand, high fracture toughness at large crack lengths owing to the large-scale grain bridging minimize catastrophic failure or machining damage.

## 3. Experiments and results

### 3.1. System setup for micro-machining processes

A micro-machining stage used for this study is illustrated in Fig. 3. The stage has three-directional moving slides; X- and Y-axis with  $200\text{ mm} \times 200\text{ mm}$  strokes and 5 nm resolution, Z-axis with 100 mm stroke and 40 nm resolution. Also, it has two rotating spindles; an air-bearing spindle for milling and drilling processes with maximum 70,000 rpm, a crossing spindle for turning process with 0–6000 rpm range. A tool dynamometer is implemented on the XY table to measure cutting forces during the micro-machining processes. Measured cutting force signals are sent to data acquisition system through charge amplifier to analyze the micro-machining process characteristics. Fig. 4 illustrates possible micro-machining processes using the micro-machining stage.<sup>8–11</sup>

### 3.2. Machinability evaluation in micro-shaping process

First, micro-shaping experiments were performed to form micro-V-grooves using the shaping tools with  $90^\circ$  inclination



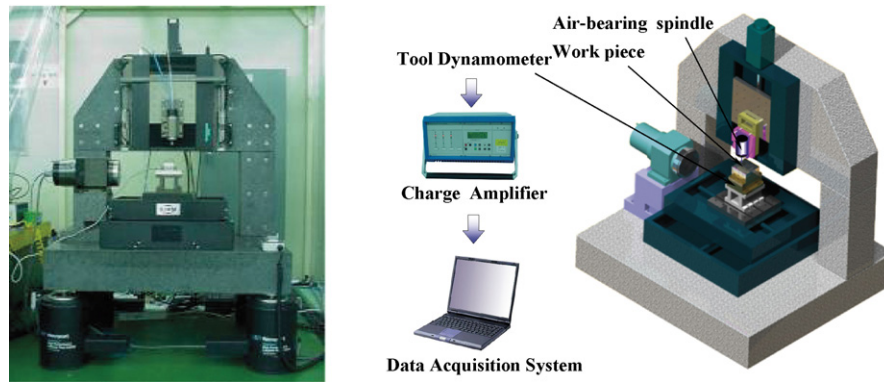


Fig. 3. Micro-machining stage for the experiments.

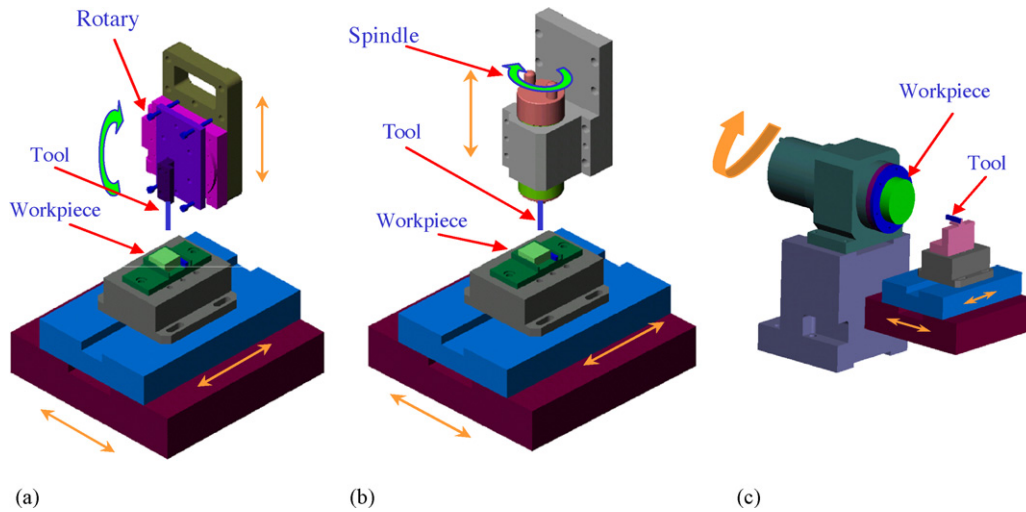


Fig. 4. Possible machining processes using the micro-stage: (a) shaping, (b) milling, drilling and (c) turning.

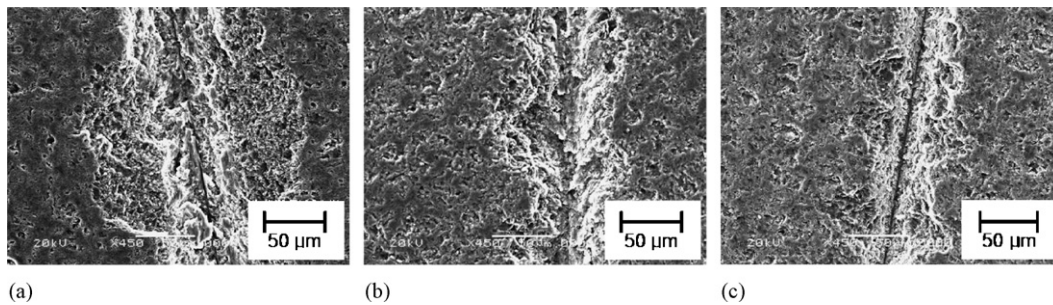


Fig. 5. SEM photographs of machined grooves: (a) 10 vol.% BN, (b) 20 vol.% BN and (c) 30 vol.% BN.

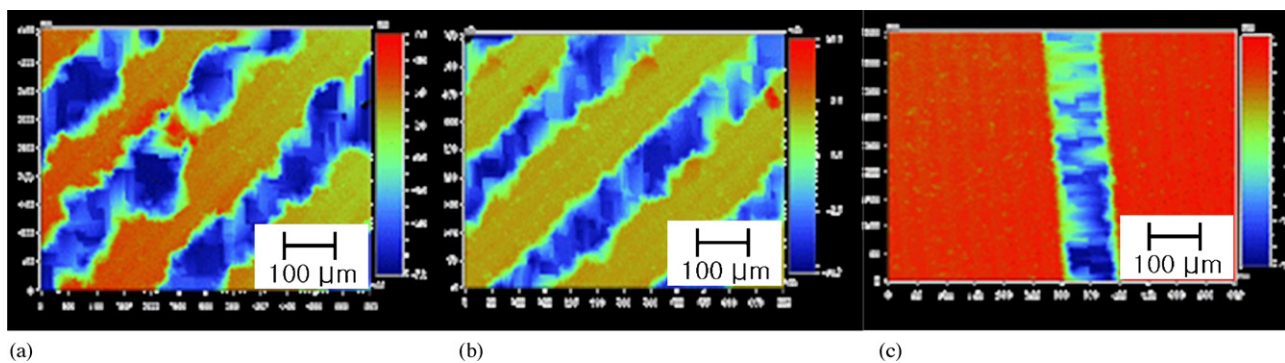


Fig. 6. Measured 3D groove shapes using surface profiler: (a) 10 vol.% BN, (b) 20 vol.% BN and (c) 30 vol.% BN.

Table 1  
Micro-V-groove shaping conditions

	Feed: 25	Feed: 50	Feed: 100	Feed: 25 (step)
h-BN (10 vol.%)	Depth: 30 $\mu\text{m}$	Depth: 30 $\mu\text{m}$	Depth: 30 $\mu\text{m}$	Depth: 10 $\mu\text{m}$ for each step
h-BN (20 vol.%)				
h-BN (30 vol.%)				

Feed = mm/min.

angle. Imposed shaping conditions for prepared composite specimen are shown in Table 1. Fig. 5 shows the SEM photographs of machined V-grooves using micro-shaping methods for various BN contents when federate = 5 mm/min. Fig. 6 shows measured 3D shapes of grooves using non-contact type surface profiler. From the figures, it could be observed that sharper and cleaner groove edges could be obtained as the h-BN contents in the composites increased. Such phenomenon was due to much more brittle fractures occur during shaping processes as the h-BN contents decrease. Fig. 7 shows measured cutting force variations according to h-BN contents. It could be seen that cutting forces decreased as the h-BN contents increased. Such results correspond to the *R*-curve behavior analysis results as shown in Fig. 2.

### 3.3. Machinability evaluation in micro-end-milling process

As a next experiment, machinability of the composite was evaluated for micro-end-milling process. Micro-end-milling

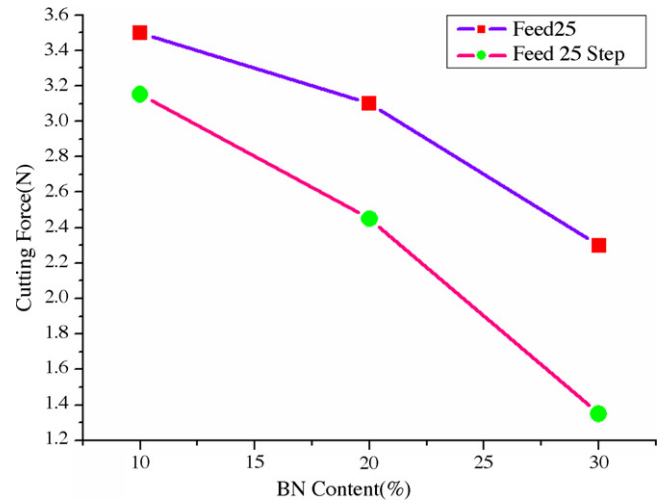


Fig. 7. Cutting forces variations according to BN contents.

method was applied to form micro-barriers with flat-type end mills. Machined barrier widths are 80 and 1000  $\mu\text{m}$ . Fig. 8 shows the SEM photographs of the machined micro-barriers. It could be observed that generally clear barriers were formed. However, when h-BN contents were low, brittle fractures resulted in collapse or crack of the barrier boundaries as shown in Fig. 9(a). It was observed that serious tool wear occurred when machining low vol.% of h-BN composite (5%) as shown in Fig. 10.

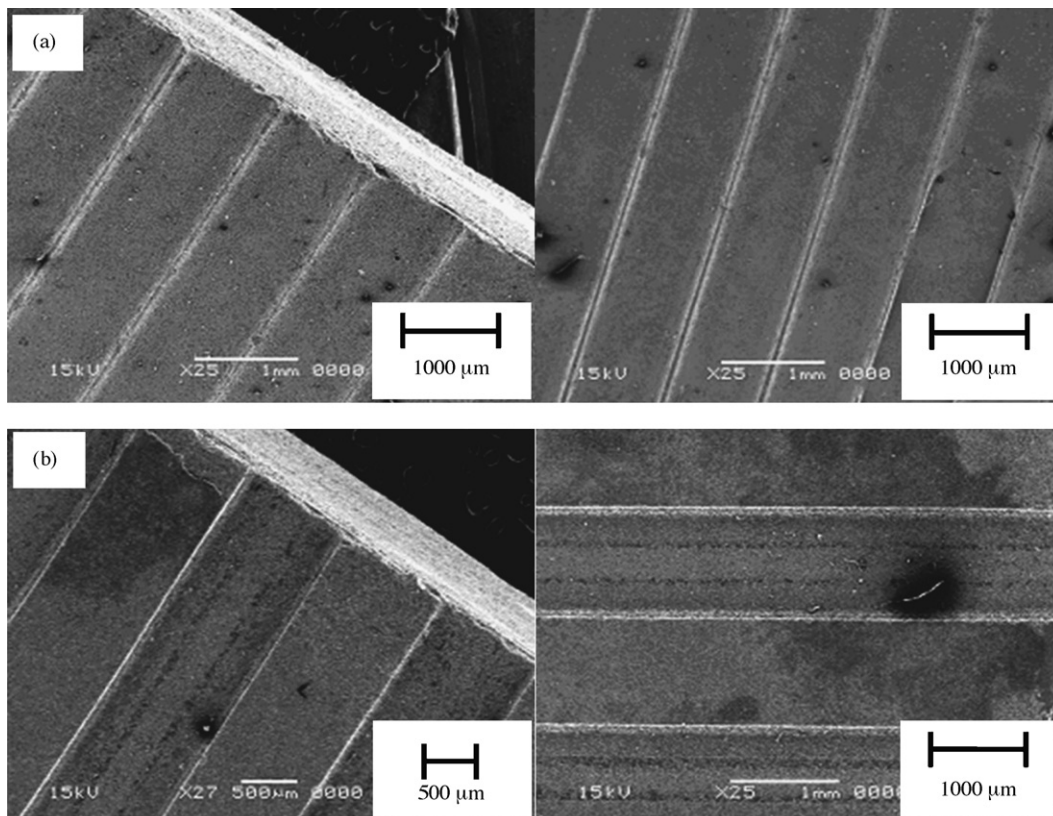


Fig. 8. Micro-barriers machined using micro-end-milling method: (a) barrier width: 80  $\mu\text{m}$  and (b) barrier width: 1000  $\mu\text{m}$ .



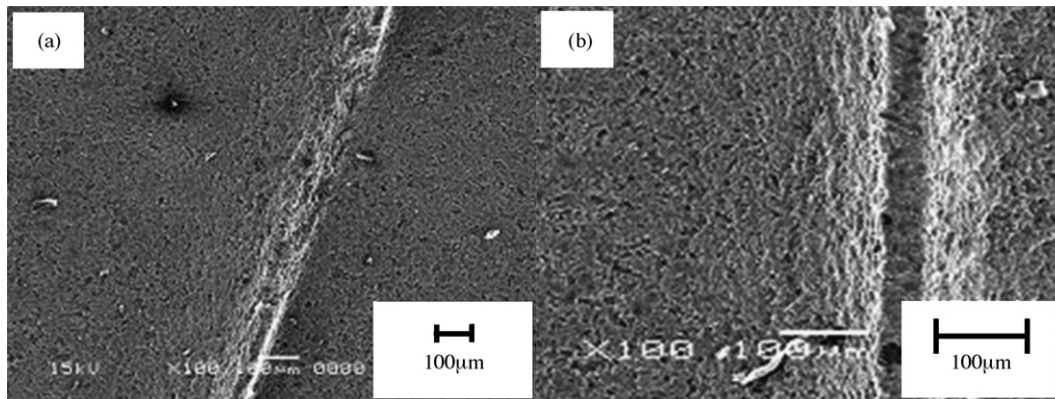


Fig. 9. SEM photographs of machined surfaces (barrier width: 80 µm): (a) 10 vol.% BN and (b) 20 vol.% BN.

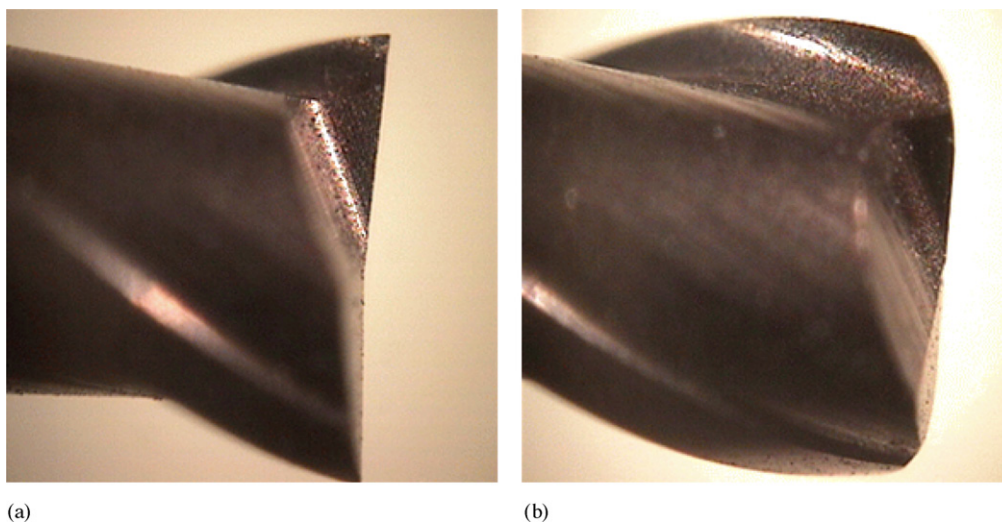


Fig. 10. Tool wear by micro-machining of machinable ceramics (5 vol.% of h-BN): (a) before and (b) after.

### 3.4. Machinability evaluation in micro-drilling process

Finally, machinability of the composite was evaluated in micro-drilling process. For the experimental works, a micro-drilling system was used, which consists of a high-speed air spindle (max. 60,000 rpm). The drilling process was divided into a certain number of steps and a drill was fed into the composite repetitively until reach the desired depth. Such method

allows to avoid micro-drill fracture problems and to easily discharge chips and heats. Drills of 100 µm diameter were used for the experiment, and thrust forces were measured using tool dynamometer under several conditions. The measured results are shown in Fig. 11. Fig. 12 shows the cutting force variations for different conditions. From the figure, it could be observed that the thrust force decreases as spindle rpm increases.

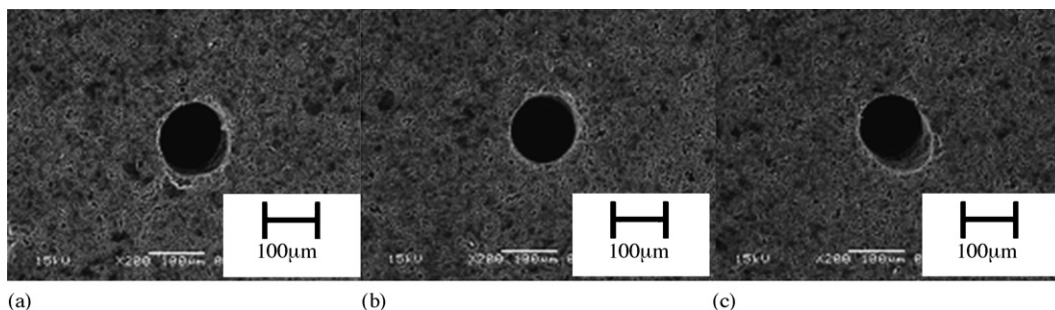


Fig. 11. Drilled micro-holes using 100 µm drill for 30 vol.% BN: (a) 40,000 rpm, (b) 50,000 rpm and (c) 60,000 rpm.

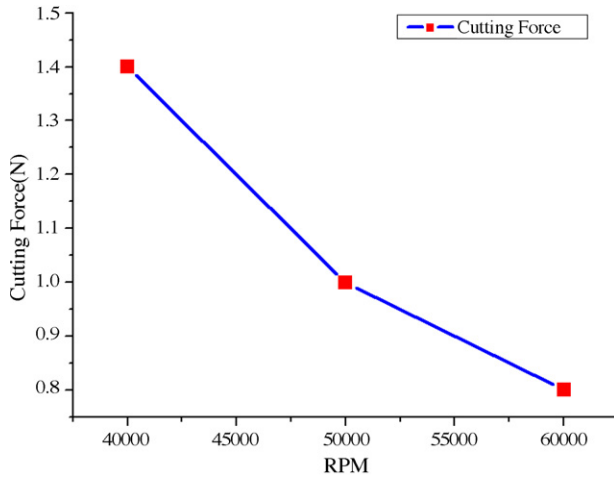


Fig. 12. Cutting force variations according to spindle rpm.

#### 4. Conclusions

In this study, *R*-curve behavior was analyzed for the machinability evaluation of the  $\text{Si}_3\text{N}_4$ -hBN composites. Higher hBN content in the composites resulted in more slowly rising *R*-curves, which could enhance machinability. Micro-machining processes were performed to evaluate the machinability, and the results can be summarized as follows:

- (1) In the micro-shaping process of the  $\text{Si}_3\text{N}_4$ -hBN composites, higher contents of h-BN and smaller depth of cut can result in better V-grooves.
- (2) In the micro-grooving process, good results could be obtained for smaller depth of cut and feed rate. However, serious tool wear was observed for the composites with less than 5% of h-BN contents. Such tool damage made the machining process harder regardless of pressing methods to make the composites.
- (3) In the micro-barrier making process using micro-end-milling process, good results could be obtained for 20% h-BN composites. However, rough detachment of particles was observed at the cutting-in region of the tool due to brittle fracture.

fracture. Machinability and machined shapes were very poor for lower h-BN contents in the composites.

- (4) In the micro-drilling process, thrust forces decreased as spindle rpm increases.
- (5) Such results can be applied to determine the optimum h-BN contents and matching micro-machining process conditions for the  $\text{Si}_3\text{N}_4$ -BN composites to make micro-patterns and shapes.

#### Acknowledgement

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