

Microwave dielectric properties of scheelite ($A = \text{Ca}, \text{Sr}, \text{Ba}$) and wolframite ($A = \text{Mg}, \text{Zn}, \text{Mn}$) AMoO_4 compounds

Geun-Kyu Choi, Jeong-Ryeol Kim, Sung Hun Yoon, Kug Sun Hong*

School of Materials Science & Engineering, College of Engineering, Shinrim-dong san 56-1, Gwanack-ku, Seoul 151-742, Republic of Korea

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Abstract

The microwave dielectric properties of scheelite ($A = \text{Ca}, \text{Sr}, \text{Ba}$) and wolframite ($A = \text{Mg}, \text{Mn}, \text{Zn}$) AMoO_4 compounds and their relations with structure have been examined using a network analyzer and X-ray powder diffraction. The Mo ion polarizability has been also investigated from AMoO_4 compounds using a least square refinement technique in conjunction with the Clausius–Mosotti equation. It was found that dielectric properties such as dielectric constant, temperature coefficient of resonant frequency, and quality factor were found to be correlated with the size of A-cations and the structure of compounds. The well sintered AMoO_4 samples (>95% of theoretical density) exhibited dielectric constant of 7–11, quality factor of 37,000–90,000 GHz and temperature coefficient of resonant frequency of -57 to -87 ppm/ $^{\circ}\text{C}$, respectively. These investigations showed that AMoO_4 ceramic could be selected as a possible candidate for microwave dielectric ceramics because of its low dielectric constant and high quality factor.

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1. Introduction

According to the rapid development of mobile telecommunication system, such as mobile phones, high quality microwave dielectric ceramics are required. In the case of microwave substrate application, dielectric ceramic substrate with low permittivity and high- Q is required to minimize cross-coupling with conductors and to maintain overall high- Q circuits by lowering power dissipation. Alumina and rare-earth aluminates (such as LaAlO_3)^{1,2} are the standard materials for the manufacturing of hybrid circuits via thick or thin film technology and for high integrated multilayer circuits produced via co-firing of metallized green tapes. Although these compounds have good dielectric properties, they have some demerits such as high sintering temperature. So, the search for new materials with much better sintering and microwave dielectric properties is surely required.

Molybdenum (Mo) containing mixed oxides have been of practical interest because of their attractive luminescence property³ and possibility of negative electrode (anode) materials.⁴ The scheelite compound has been investigated as a host crystal for laser applications. Also, the dielectric constant

and the anisotropy of the dielectric constant (ϵ') of single crystals (grown by the Czochralski method) were firstly reported by Brower et al.⁵ The dielectric constant (ϵ') parallel to a -axis of CaMoO_4 single crystal was 24.0 ± 0.2 ; parallel to c -axis, 20.0 ± 0.2 at 1.59 kHz. The dielectric loss ($\tan \delta$) for both orientations of CaMoO_4 specimen was 10^{-3} or less. Unfortunately, not many works have been done on the dielectric properties for microwave application. For investigation of possibility of microwave application, we researched the divalent metal molybdate compounds. The Mo ion polarizability has been also investigated from AMoO_4 compounds using a least square refinement technique in conjunction with Clausius–Mosotti equation.

2. Experimental procedure

The powders were prepared by conventional mixed oxide method. ACO_3 [$A = \text{Ca}$ (99.99%), Sr (99.5%), Ba (99.9%)], MgO (99.9%), Mn_3O_4 (98%), ZnO (99.9%) and MoO_3 (99.9% high purity) purity powders were weighed and mixed for 24 h with stabilized zirconia media and ethanol. The mixed slurry was dried and then powders were calcined at 700–1000 $^{\circ}\text{C}$ for 2 h in air and then the calcined powder was granulated with an organic binder after ball milling for 24 h. The milled powders

* Corresponding author. Tel.: +82 2 880 8024; fax: +82 2 886 4156.
E-mail address: kshongss@plaza.snu.ac.kr (K.S. Hong).

were uniaxially pressed into disks 8 mm in diameter and 2–4 mm thick under a pressure of 1000 kg/cm². Pellets were sintered at 700–1150 °C for 2 h in air with a heating rate of 5 °C/min. The bulk density of the sintered specimens was determined by the Archimedes' method.

The X-ray powder diffraction (XRD) studies were performed with monochromatic X-ray diffractometer (XRD: Model M18XHF, Mac Science Instruments, Japan). Lattice parameters were calculated by refinement of data collected at range $20^\circ < 2\theta < 60^\circ$ with a step of 0.02, counting time of 1 s per step and Si powder was used as an internal standard. Polished and thermally etched surfaces of sintered specimens were examined using field emission scanning electron microscopy (FESEM: Model JSM6330F, Jeol, Japan). The microwave dielectric properties of sintered samples were measured at x-band frequencies (14–17 GHz) using a network analyzer (model HP8720C, Hewlett Packard, Palo Alto, CA). The quality factor ($Q \times f$) was measured by the transmission cavity method using a Cu cavity and Teflon supporter. Relative dielectric constant (ϵ_r) was measured using the post resonator method and temperature coefficient of resonant frequency (τ_f) was measured using an Invar cavity in the temperature at range 20–80 °C.

3. Results

Fig. 1 shows relative densities of AMoO_4 ($A = \text{Ca, Sr, Ba, Mg, Mn, Zn}$) samples as a function of sintering temperature. The maximum relative densities of AMoO_4 samples, which have sintered densities more than 95% of their theoretical densities were achieved over the temperature range (800–1100 °C). As shown in Fig. 1, sintering temperature of wolframite compounds was lower than that of scheelite compounds.

Fig. 2 shows the X-ray diffraction of sintered AMoO_4 compounds. The single phase structure of all compounds was confirmed by X-ray powder diffraction. As shown in Fig. 2, diffraction peaks shifted to lower angle in the same structure

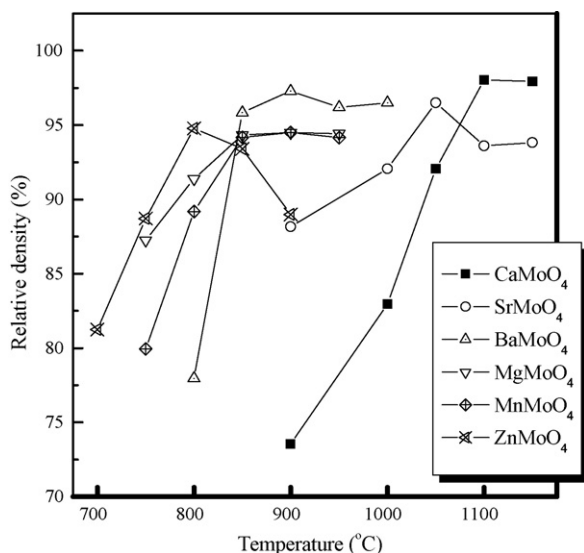


Fig. 1. The relative densities of AMoO_4 ($A = \text{Ca, Sr, Ba, Mg, Mn, Zn}$) samples as a function of sintering temperature.

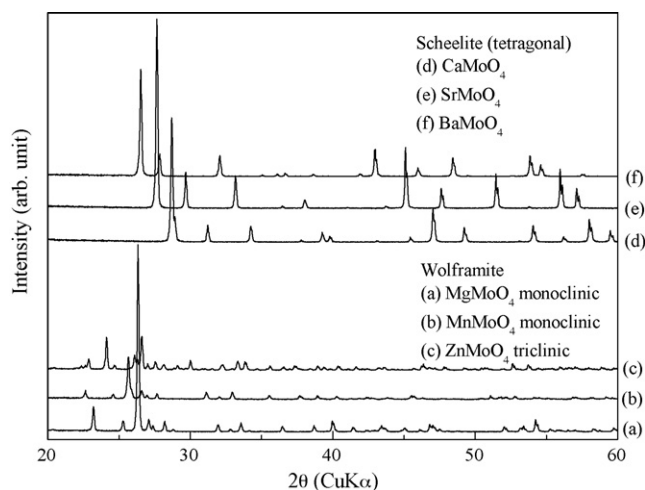


Fig. 2. XRD patterns of sintered AMoO_4 powders: (a) MgMoO_4 sintered at 900 °C for 2 h, (b) MnMoO_4 sintered at 900 °C for 2 h, (c) ZnMoO_4 sintered at 800 °C for 2 h, (d) CaMoO_4 sintered at 1100 °C for 2 h, (e) SrMoO_4 sintered at 1050 °C for 2 h, and (f) BaMoO_4 sintered at 900 °C for 2 h.

as the radius of cation increased, except ZnMoO_4 compound (the structure of ZnMoO_4 is triclinic). The structure of AMoO_4 changed from wolframite (A ionic radius ≤ 0.77 Å: Mg, Mn, Zn) to scheelite (A ionic radius ≥ 0.99 Å: Ca, Sr, Ba).⁶ The wolframite structure (space group $C12/m$, 12, monoclinic) has hexagonally close-packed oxygens with certain octahedral sites filled by A and Mo cations in an ordered way. On the contrary, the scheelite structure (space group $I4_1/a$, 88, tetragonal) has cubic-packed array of A^{2+} and $[\text{MoO}_4]^{2-}$ units with the coordination around molybdenum is quite symmetrically tetra-hedral, like for any scheelite-type compound, while the overall coordination of A site ion is eight.

The SEM micrographs of the surfaces of sintered AMoO_4 samples are shown in Fig. 3. Grain size was in range of 2–7 μm and the dense microstructures were confirmed by a SEM analysis. The dielectric properties of AMoO_4 compounds were summarized in Table 1. The samples of scheelite structure had good microwave dielectric properties. In the case of CaMoO_4 , $Q \times f$ was $\sim 90,000$ and $\epsilon_r \approx 10.8$, and τ_f about -57 ppm/°C.

4. Discussion

Table 1 shows the dielectric properties of AMoO_4 specimens. The well sintered AMoO_4 samples ($>95\%$) exhibited dielectric constant of 7–11, quality factor of 37,000–90,000 GHz and temperature coefficient of resonant frequency of -57 to -87 ppm/°C, respectively. As the size of A -site cation increased, the dielectric constant and quality factor of scheelite AMoO_4 ($A = \text{Ca, Sr, Ba}$) samples were decrease, while those of wolframite samples were increased. For investigating relationship between A -site cation and dielectric constant, we used the Shannon's suggestion⁷ and Clausius–Mosotti equation.

Shannon reported a set of 61 ion polarizabilities, which has been derived from the dielectric constants of 129 oxides and 25 fluorides using a least squares refinement technique in conjunction with the Clausius–Mosotti equation. In the case of AMoO_4

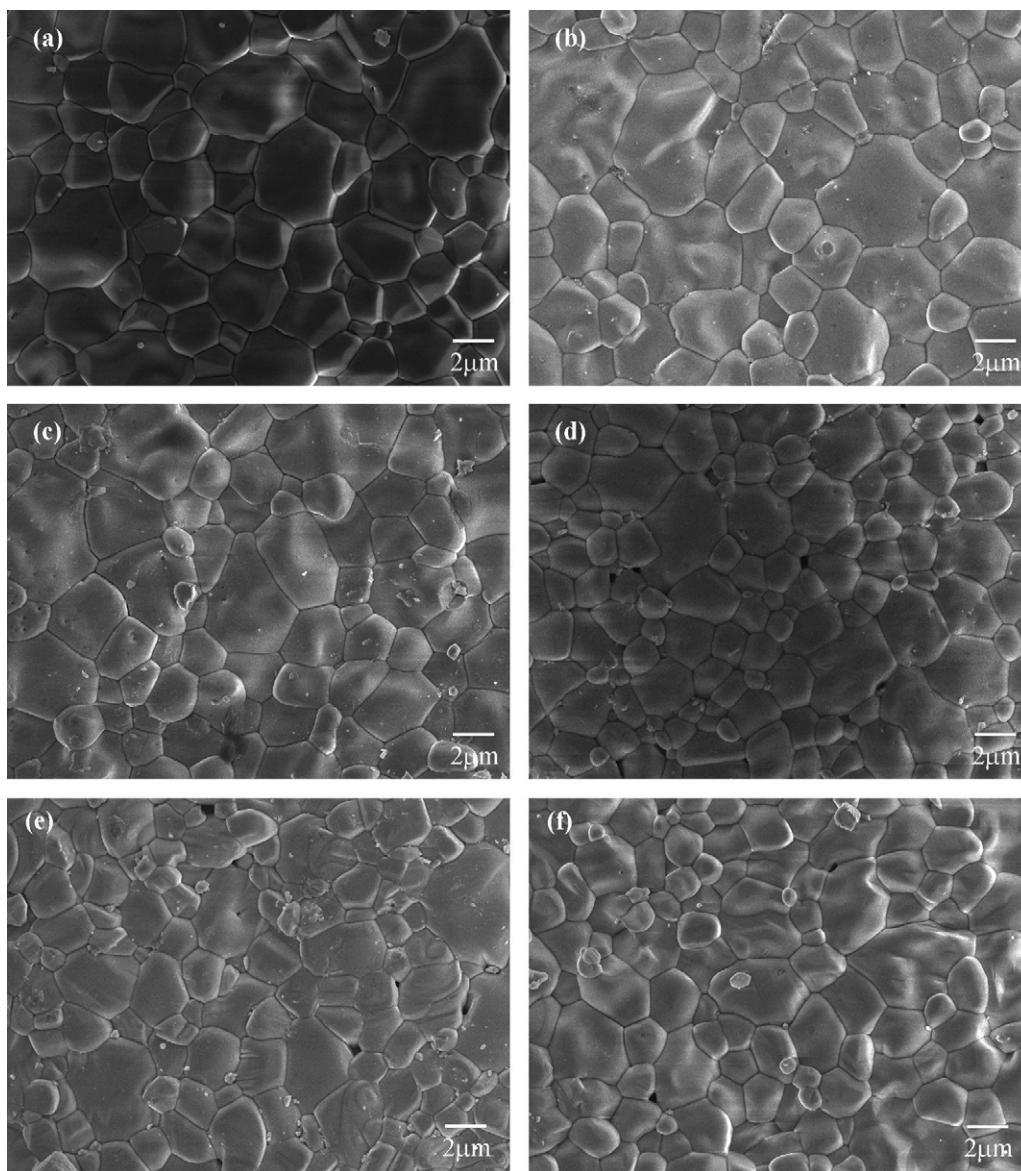


Fig. 3. SEM micrograph of the AMoO_4 specimens: (a) CaMoO_4 sintered at 1100°C for 2 h, (b) SrMoO_4 sintered at 1050°C for 2 h, (c) BaMoO_4 sintered at 900°C for 2 h, (d) MgMoO_4 sintered at 900°C for 2 h, (e) MnMoO_4 sintered at 900°C for 2 h, and (f) ZnMoO_4 sintered at 800°C for 2 h.

system, molecular polarizability can estimate from following equation.

$$\alpha(\text{AMoO}_4) = \alpha(\text{A}^{2+}) + \alpha(\text{Mo}^{6+}) + 4\alpha(\text{O}^{2-}) \quad (1)$$

where α is polarizability. The dielectric constants of AMoO_4 compounds could be calculated with polarizability from Shan-

non's suggestion and Clausius–Mosotti relation:

$$\varepsilon_{\text{cal}} = \frac{3V_m + 8\pi\alpha D}{3V_m - 4\pi\alpha D} \quad (2)$$

where V_m is molar volume. Molar volume of AMoO_4 can be calculated with the unit cell parameters determined from the diffraction pattern, and the values of molar volume are shown in

Table 1
Microwave dielectric properties of AMoO_4 compounds

Compound	Sintering temperature ($^\circ\text{C}$)	ε_r	$Q \times f$ (GHz)	τ_f (ppm/ $^\circ\text{C}$)	Porosity
CaMoO_4	1100	10.79	89700	−57	0.0195
SrMoO_4	1050	9.49	61000	−67.17	0.0350
BaMoO_4	900	9.3	37200	−79.24	0.0270
MgMoO_4	900	7.07	79100	−46.02	0.0551
MnMoO_4	900	8.55	54100	−73.91	0.0550
ZnMoO_4	800	8.67	49900	−87.49	0.0523

Table 2
Molar volume, observed and calculated permittivity of AMoO₄ compounds

Compound	V_m	ϵ_{obs}	ϵ_{cal}	Δ (%)
CaMoO ₄	78.064	11.11	11.45	−3.113
SrMoO ₄	87.575	9.99	9.73	2.584
BaMoO ₄	99.9	9.68	9.67	0.038
MgMoO ₄	80.363	7.65	6.79	11.245
MnMoO ₄	85.913	9.26	7.39	20.113
ZnMoO ₄	86.667	9.35	6.47	30.827

Table 2. The dielectric constant in Table 1 was polycrystalline data, even though well sintered AMoO₄ samples have densities more than 95% of theoretical density. So, the dielectric constant in Table 2 (ϵ_{obs}) was corrected by following equation.

$$\epsilon_{\text{obs}} = \epsilon_{\text{exp}}(1 + 1.5P) \quad (3)$$

where P is porosity and ϵ_{exp} is measured dielectric constant.⁸ Unfortunately, Mo ion polarizability was not reported by Shannon. We investigated Mo ion polarizability from AMoO₄ compounds using a least square refinement technique in conjunction with the Clausius–Mosotti equation. Ion polarizabilities, α_D (ion), can be used as a set of refinable parameters in a least squares procedure which minimizes the function.

$$\sum_{i=1}^M \omega_i (\alpha_{\text{obs}} - \alpha_{\text{cal}})^2 \quad (4)$$

where i varies over the number of measurements of α_{obs} for variety of compounds and $\omega_i = \sigma_i$, and σ_i is the estimated % error in the experimental dielectric constant. The results of least squares refinement were evaluated by examining the weighted variance of fit (F):

$$F = \frac{(\sum_i \omega_i [(\alpha_{\text{obs}})_i - (\alpha_{\text{cal}})_i]^2)}{\sum_i \omega_i} \quad (5)$$

The first results from refinements of $\alpha_D(\text{Mo})$ made using experimental dielectric constant data from six AMoO₄ compound were $\alpha_D(\text{Mo})=4$ in $F=0.748\%$. This results in a poor agreement between observed and calculated dielectric constant. But second results from three scheelite compound, $\alpha_D(\text{Mo})=3.28$ in $F=0.013\%$, shows good agreement with deviations in observed and calculated dielectric constant in scheelite system ($\Delta\% < 3.1\%$), as shown in Table 2. This phenomenon was explained by following structural consideration. In wolframite structure system, the overall coordination around molybdenum is six, giving rise to ribbons of edge-sharing distorted octahedral, which form a polymeric anionic structure with composition $[(\text{Mo}_2\text{O}_8)^{4-}]_n$. Similarly, the coordination at A site ion is also six and gives rise to parallel ribbons of sharing distorted octahedral. In fact, the wolframite structure can be taken as a slightly distorted superstructure of the $\alpha\text{-PbO}_2$ -type structure, where the octahedral are occupied alternatively by the two different cations.⁹ In wolframite MgMoO₄ structure system, Mo ions have two site positions. The one Mo position has two molybdenum–oxygen distances being definitely longer (3.088 \AA $2\times$) than the other four (1.731 \AA $2\times$, 1.801 \AA $2\times$) around Mo

atom, the other Mo position has longer (3.425 \AA $2\times$) distance and four shorter (1.730 \AA $2\times$, 1.752 \AA , 1.845 \AA) distance.¹⁰ So that it seems reasonable to interpret this structure on the basis of a distorted tetrahedral coordination around molybdenum, as well as MnMoO₄ structure system.⁶ On the contrary, the coordination around molybdenum in scheelite structure system is quite symmetrically tetrahedral (Mo–O: 1.755 \AA $4\times$ in CaMoO₄ system). Shannon explained the deviations of ϵ_{cal} from ϵ_{obs} generally reflect one or more of the following: inaccurate dielectric constants, ionic or electronic conductivity, the presence of H₂O or CO₂ in channels, the presence of “rattling” or “compressed” cations having unusually large or small cation polarizabilities, respectively, or the presence of dipolar impurities. The presence of rattling cations group, shorted by Shannon, shows dielectric deviation ($\Delta\% = 13\text{--}34\%$), similarly in our wolframite system, in Table 2. This effect was described by Dunitz and Orgel¹¹ as a progressive, loosening of the central cation at the center of its surrounding octahedron to off-center displacements characteristic of ferroelectric and antiferroelectric substances, as the size of the central cation in an octahedron decreased. The polyhedron can adjust its configuration to adapt to the small cation by movement of the cation and or anions resulting in polyhedral distortion.

Fig. 4 shows relation between the temperature coefficient of resonant frequency ($-\tau_f$) and unit-cell volume of AMoO₄. Generally temperature coefficient of resonant frequency (τ_f) is represented by the thermal expansion coefficient (α_l) and the temperature coefficient of permittivity (τ_ϵ), as follows:

$$\tau_f = -\alpha_l - \frac{1}{2} \tau_\epsilon \quad (6)$$

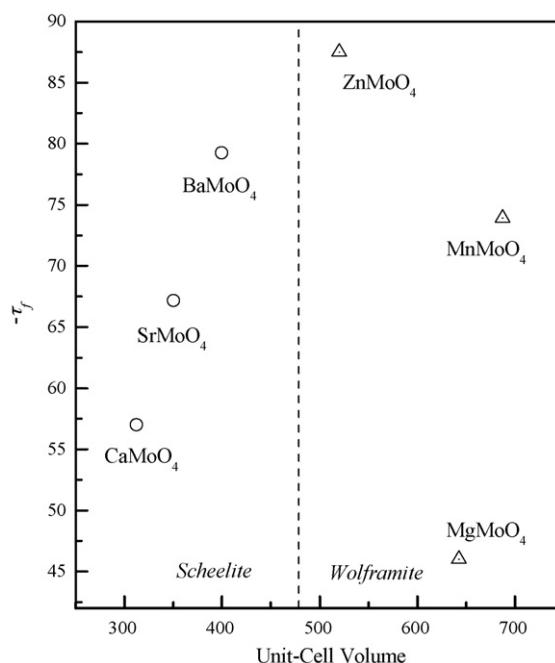


Fig. 4. Temperature coefficient of resonant frequency of AMoO₄ compounds as a function of unit-cell volume.

The thermal expansion coefficient of ceramic is generally in the range of 10 ppm/°C, therefore the sign and magnitude of τ_f depends on τ_ε . From Clausius–Mosotti equation, τ_ε can be derived a following expression.

$$\tau_\varepsilon = \frac{1}{\varepsilon} \left(\frac{\partial \varepsilon}{\partial T} \right) = \frac{(\varepsilon - 1)(\varepsilon + 2)}{\varepsilon} (A + B + C)$$

$$\left(A = -\frac{1}{3V} \left(\frac{\partial V}{\partial T} \right)_P, \quad B = \frac{1}{3\alpha_m} \left(\frac{\partial \alpha_m}{\partial V} \right)_T \left(\frac{\partial V}{\partial T} \right)_P, \right.$$

$$\left. C = \frac{1}{3\alpha_m} \left(\frac{\partial \alpha_m}{\partial T} \right)_V \right) \quad (7)$$

Bosman and Havinga suggested that the terms A and B , which correlate to volume expansion, have very similar in magnitude but opposite in sign, therefore the resulting effect on τ_ε of terms A and B is generally ignored.^{12–14} Lee et al. suggested that the term C represents the restoring force acting on the ions and restoring force depends on the shape of potential well, therefore this term correlated with structure and lattice parameter.¹³ Since the unit-cell volume has strong correlation with lattice energy, τ_ε is proportional to the relative magnitude of unit-cell volume. As can be shown in Fig. 4, the magnitude of $-\tau_f$ increased as the unit-cell volume increased in the same structure of AMoO_4 .

5. Conclusion

In our study, well sintered AMoO_4 samples (>95% of theoretical densities) exhibited dielectric constant of 7–11, quality factor of 37,000–90,000 GHz and temperature coefficient of resonant frequency of -57 to -87 ppm/°C, respectively. CaMoO_4 sample exhibited $Q \times f$ value of $\sim 90,000$ GHz, dielectric constant (ε_r) = 10.79 and $\tau_f = -57$ ppm/°C. This results show that AMoO_4 can be selected as a suitable candidate for microwave dielectric material because of its low dielectric constant and high- Q , such as alumina.

The results from refinement of $\alpha_D(\text{Mo})$ made using experimental dielectric constant data of AMoO_4 compounds show good agreement with deviations in observed and calculated dielectric constant in scheelite system ($\Delta\% < 3.1\%$, $F = 0.013\%$, $\alpha_D(\text{Mo}) = 3.28$). Otherwise, poor agreement in wolframite system ($\Delta\% \approx 8\text{--}30\%$) is well interpreted by rattling cation in oxygen octahedron.

It was found that temperature coefficient of resonant frequency was correlated with structure and unit-cell volume. The magnitude of τ_f was increased as the unit-cell volume increased in the same structure of AMoO_4 . In scheelite AMoO_4 system, $-\tau_f$ shows linearity for unit-cell volume.

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References

- Penn, S. J., Alford, N. M., Templeton, A., Wang, X., Xu, M., Reece, M. et al., Effect of porosity and grain size on the microwave dielectric properties of sintered alumina. *J. Am. Ceram. Soc.*, 1997, **80**, 1885.
- Cho, S. Y., Kim, I. T. and Hong, K. S., Microwave dielectric properties and application of rare-earth aluminates. *J. Mater. Res.*, 1999, **14**, 114.
- Johnson, L. F., Boyd, G. D., Nassau, K. and Soden, R. R., Continuous operation of a solid-state optical maser. *Phys. Rev.*, 1962, **126**(4), 1406.
- Sharma, N., Shaju, K. M., Subba Rao, G. V., Chowdari, B. V. R., Dong, Z. L. and White, T. J., Carbon-coated nanophase CaMoO_4 as anode material for Li ion batteries. *Chem. Mater.*, 2004, **16**, 504.
- Brower, W. S. and Fang, P. H., Dielectric constants of PbMoO_4 and CaMoO_4 . *Phys. Rev.*, 1966, **149**, 646.
- Daturi, M., Busca, G., Borel, M. M., Leclaire, A. and Piaggio, P., Vibrational and XRD study of the system $\text{CdWO}_4\text{--CdMoO}_4$. *J. Phys. Chem. B.*, 1977, **101**, 4358.
- Shannon, R. D., Dielectric polarizabilities of ions in oxides and fluorides. *J. Appl. Phys.*, 1993, **73**(1), 348.
- Kingery, W. D., Bowen, H. K. and Uhlmann, D. R., *Introduction to Ceramics*. Wiley, New York, 1986.
- Hyde, B. G. and Andersson, S., *Inorganic Crystal Structure*. Wiley, New York, 1989.
- Bakakin, V. V., Klevtsova, R. F. and Gaponenko, L. A., Crystal structure of magnesium molybdate MgMoO_4 an example of modified closest packing with two types of tetrahedra. *Kristallografiya*, 1982, **27**, 38.
- Dunitz, J. D. and Orgel, L. E., Stereochemistry of ionic solids. *Adv. Inorg. Chem. Radiochem.*, 1960, **2**, 1.
- Colla, E. L., Reaney, I. M. and Setter, N., Effect of structural changes in complex perovskites on the temperature coefficient of the relative permittivity. *J. Appl. Phys.*, 1993, **74**, 3414.
- Lee, H. J., Hong, K. S. and Kim, S. J., Dielectric properties of MnNb_2O_6 compounds. *Mater. Res. Bull.*, 1997, **32**, 847.
- Cho, S. Y. and Hong, K. S., Contribution of structure to temperature dependence of resonant frequency in the $(1 \times x)\text{La}(\text{Zn}_{1/2}\text{Ti}_{1/2})\text{O}_3 + x\text{ATiO}_3$ ($A = \text{Ca, Sr}$) system. *J. Am. Ceram. Soc.*, 2001, **84**, 753.